

2024 European School on Magnetism

Magnetocaloric effect

Kelly Morrison



Overview

1. Context

2. Maxwell Relations

How can thermodynamics help us?

3. Measurement Techniques

How can you measure the magnetocaloric effect?

4. Materials

What benchmark values might we be comparing to?

5. Universal Curves

What can universality tell us about a material system?

6. Material Engineering

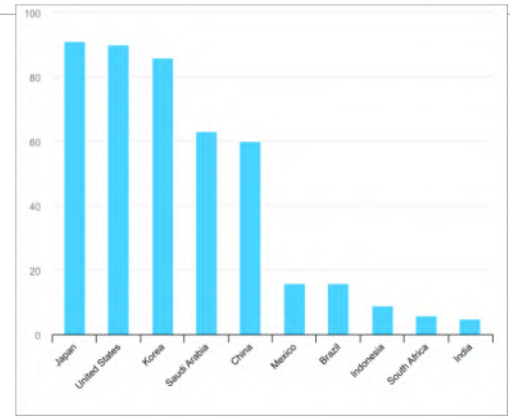
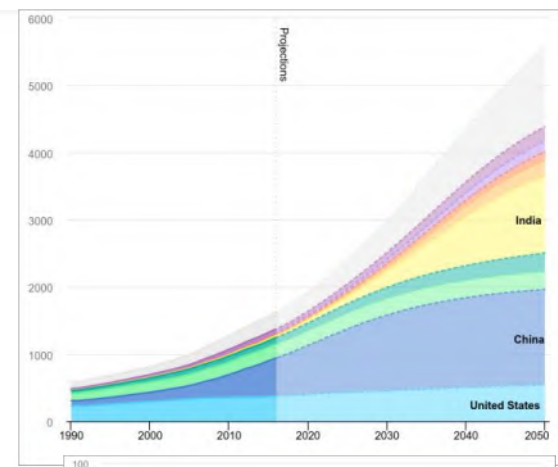
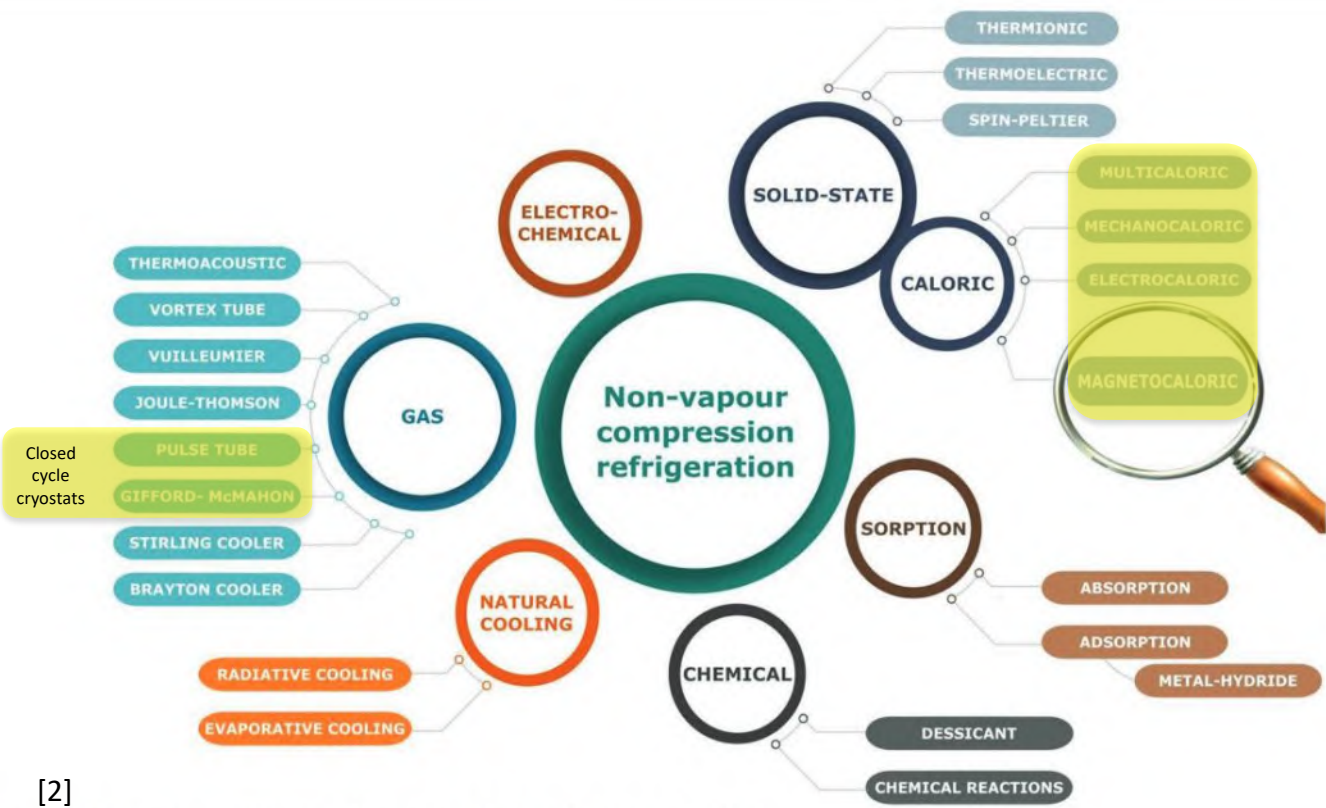
Beyond entropy change, what material properties should we be looking to improve?

7. Future Outlook

Rising Temperatures

Air conditioning and refrigeration accounts for 20% of electricity used in buildings today^[1]

Refrigerants account for 7.8% total greenhouse gas emission



[2]

Figure 1. Non-vapor-compression refrigeration and air conditioning technologies.

Want to know more?: See ESM 2021, Oliver Gutfleisch *Magneto (and multi-)caloric materials for efficient refrigeration*

[1] International Energy Agency, The Future of Cooling, (2018), <http://www.iea.org/reports/the-future-of-cooling>

[2] A. Kitanovski, *Adv. Energy Materials* **10** 1903741 (2020) 3 of 41

Magnetic Cooling

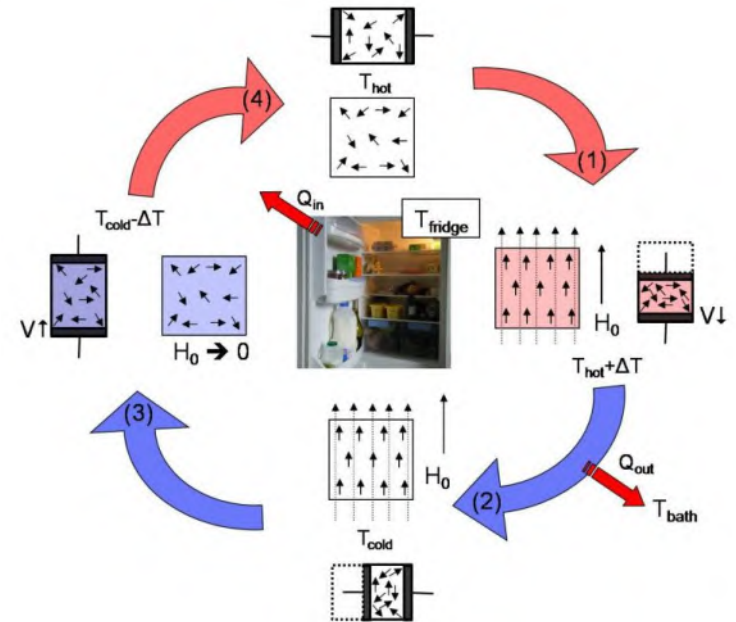
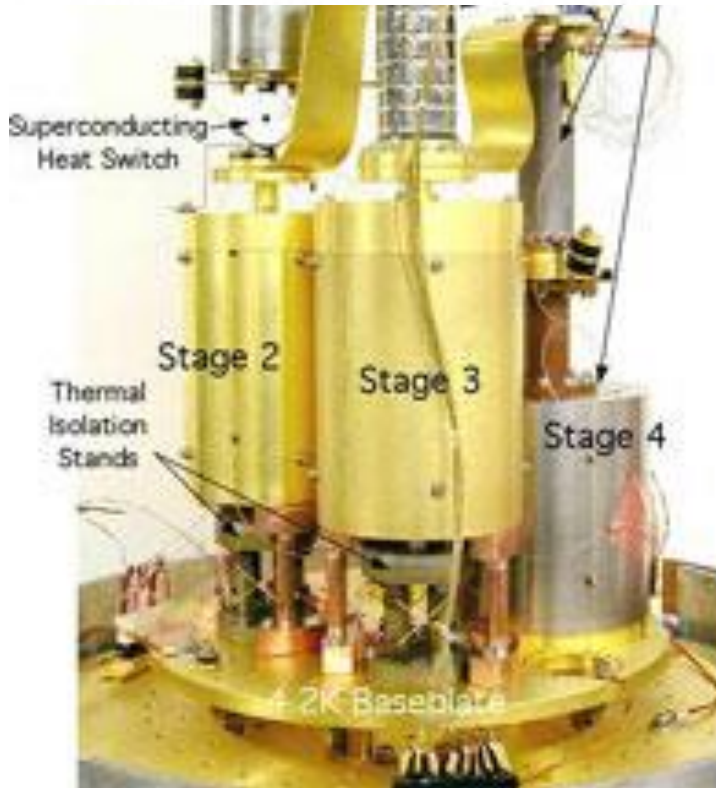
[CONTRIBUTION FROM THE CHEMICAL LABORATORY OF THE UNIVERSITY OF CALIFORNIA]

A THERMODYNAMIC TREATMENT OF CERTAIN MAGNETIC EFFECTS. A PROPOSED METHOD OF PRODUCING TEMPERATURES CONSIDERABLY BELOW 1° ABSOLUTE

By W. F. GIAUQUE

RECEIVED DECEMBER 14, 1926

PUBLISHED AUGUST 5, 1927



“Photograph of technology demonstration of a Continuous Adiabatic Demagnetization Refrigerator developed at NASA’s Goddard Space Flight Center.”

[Figure taken from

<http://ixo.gsfc.nasa.gov/technology/xms.html>]

Magnetic Cooling

VOLUME 78, NUMBER 23

PHYSICAL REVIEW LETTERS

9 JUNE 1997

Giant Magnetocaloric Effect in $\text{Gd}_5(\text{Si}_2\text{Ge}_2)$

V. K. Pecharsky and K. A. Gschneidner, Jr.

Ames Laboratory and Department of Materials Science and Engineering, Iowa State University, Ames, Iowa 50011-3020

(Received 22 November 1996)

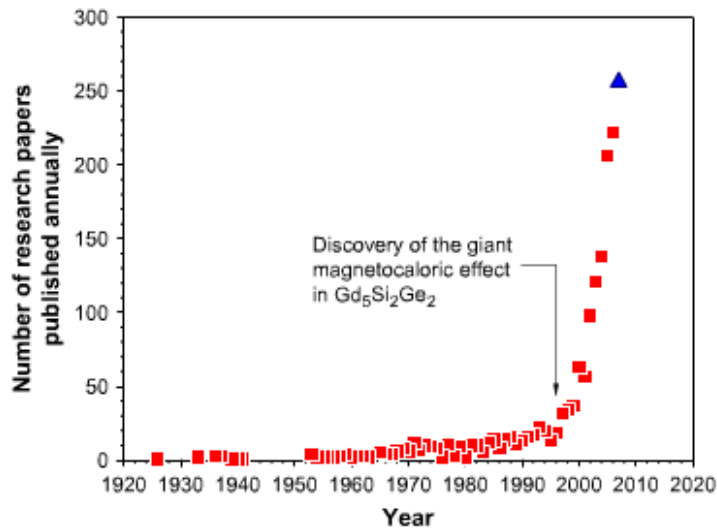
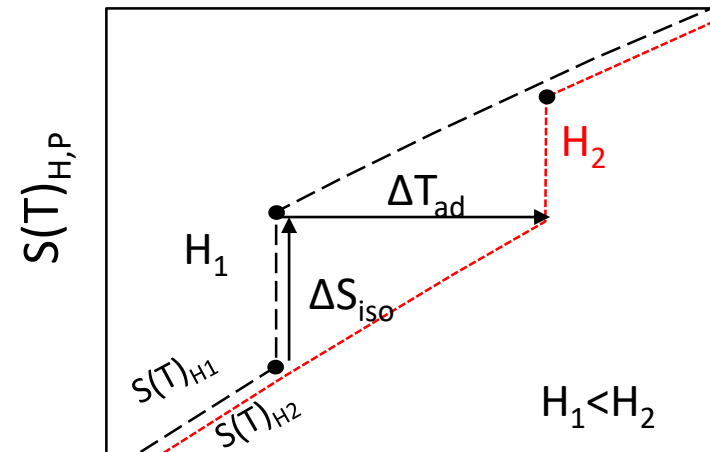


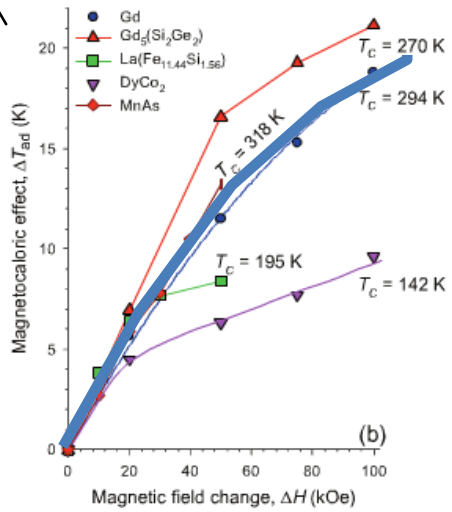
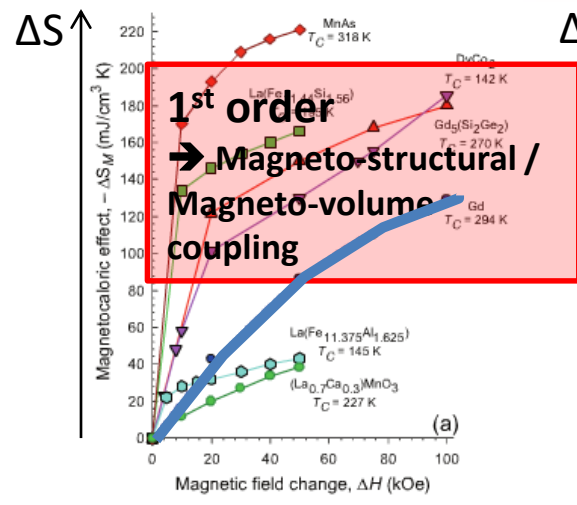
Fig. 1 - The number of research papers published annually over the past 80 years containing the word "magnetocaloric" in the title, abstract, or among the keywords. The values for 2007 (triangle) are based on the number of papers abstracted during the first three-fourths of the year.



$$\Delta T_{ad}(T, \Delta H) \approx -\frac{T\Delta S(T)}{C(T)}$$

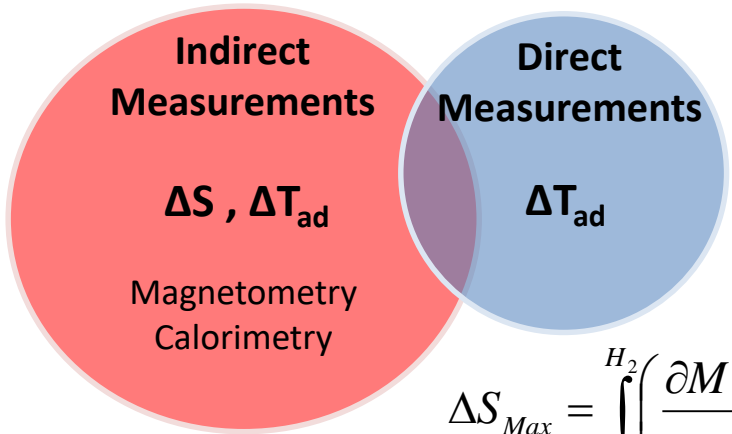
Refrigerant capacity $\sim \Delta S \cdot \Delta T_{ad}$

Research Aims

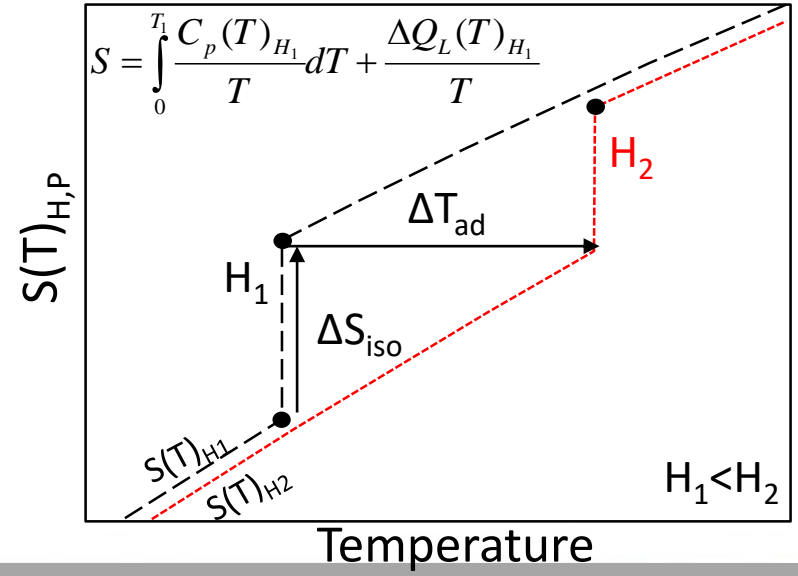


- ΔS , ΔT_{ad}
- H_c , ΔH
- Cost
- Manufacturability
- Tunability

2011 – Cambridge prototype attained $T_{span} \sim 35$ K with La(Fe,Si,Co)₁₃ plates.



$$\Delta S_{Max} = \int_{H_1}^{H_2} \left(\frac{\partial M(T, H)}{\partial T} \right)_H dH$$



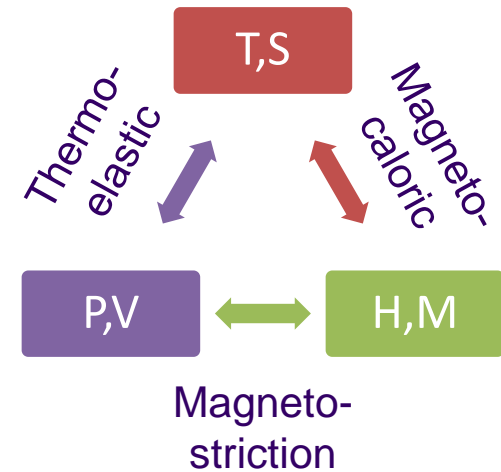
Maxwell Relations

$$dG = -SdT - MdH + VdP$$

$$\left(\frac{\partial S}{\partial H}\right)_{T,P} = \left(\frac{\partial M}{\partial T}\right)_{H,P}$$

$$\left(\frac{\partial S}{\partial P}\right)_{T,H} = -\left(\frac{\partial V}{\partial T}\right)_{H,P}$$

$$\left(\frac{\partial M}{\partial P}\right)_{H,T} = -\left(\frac{\partial V}{\partial H}\right)_{T,P}$$



Equation of state:

$$\frac{H}{M} = A + BM^2 + CM^4$$

Maxwell Relations

$$dU = TdS + HdM - PdV$$

$$dG = -SdT - MdH + VdP$$

$$\left(\frac{\partial G}{\partial T}\right)_{H,P} = -S \Rightarrow \frac{\partial}{\partial H} \left(\frac{\partial G}{\partial T}\right)_{H,P} = -\left(\frac{\partial S}{\partial H}\right)_{T,P}$$

$$\left(\frac{\partial G}{\partial H}\right)_{T,P} = -M \Rightarrow \frac{\partial}{\partial T} \left(\frac{\partial G}{\partial H}\right)_{T,P} = -\left(\frac{\partial M}{\partial T}\right)_{H,P}$$

$$\left(\frac{\partial G}{\partial P}\right)_{T,H} = V$$

$$\frac{\partial}{\partial H} \left(\frac{\partial G}{\partial T}\right) = \frac{\partial}{\partial T} \left(\frac{\partial G}{\partial H}\right)$$

$$\therefore -\left(\frac{\partial S}{\partial H}\right)_{T,P} = -\left(\frac{\partial M}{\partial T}\right)_{H,P}$$

$$\Delta S_{Max} = \int_{H_1}^{H_2} \left(\frac{\partial M(T,H)}{\partial T}\right)_H dH$$

Similarly

$$-\left(\frac{\partial S}{\partial P}\right)_{H,T} = \left(\frac{\partial V}{\partial T}\right)_{H,P}$$

$$-\left(\frac{\partial M}{\partial P}\right)_{H,P} = \left(\frac{\partial V}{\partial H}\right)_{T,P}$$

Magnetometry: Potential problems

Same sample measured 2 different ways.
Why so different?

Magnetometry - See ESM 2024, Stuart Cavill.
Magnetic measurement instruments and techniques

$$\Delta S_{Max} = \int_{H_1}^{H_2} \left(\frac{\partial M(T, H)}{\partial T} \right)_H dH$$

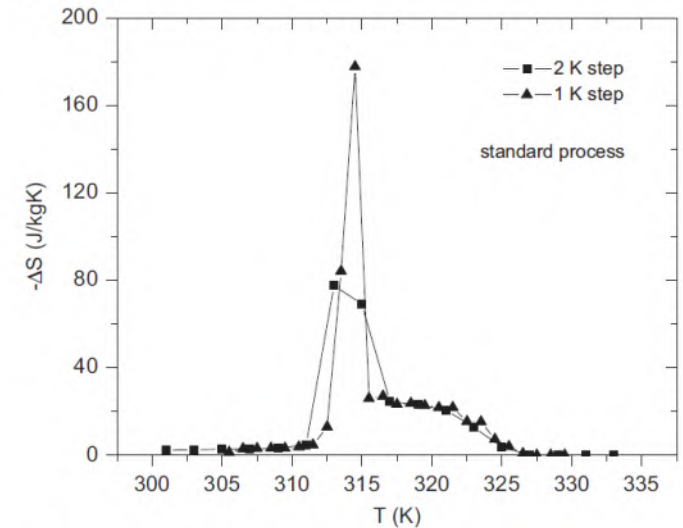
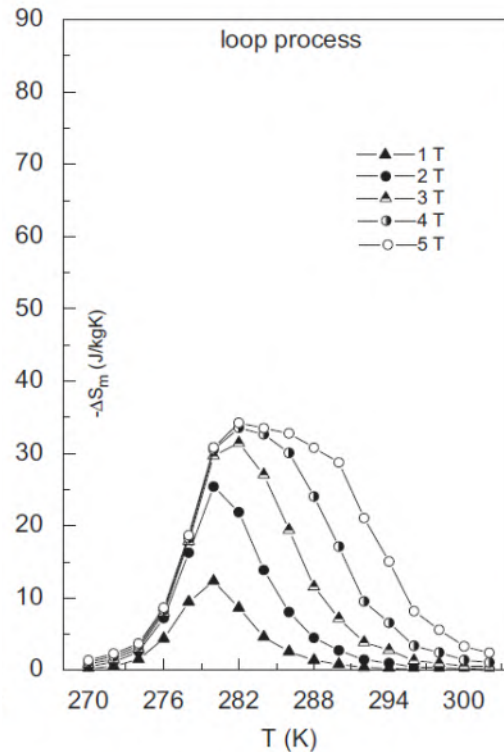
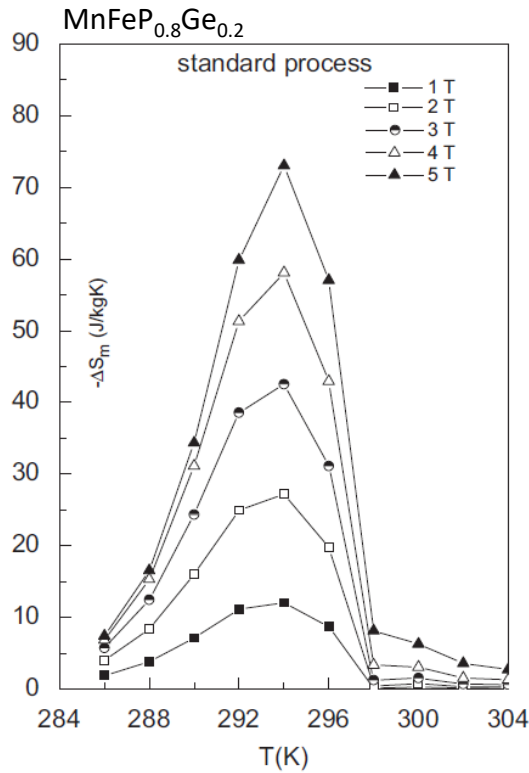


Fig. 9. (a) Magnetization isotherms measured according to the standard process. (b) Entropy change as a function of temperature for Mn_{0.99}Cu_{0.01}As for 1 and 2 K steps for a 5 T field change.

Magnetometry: Field History

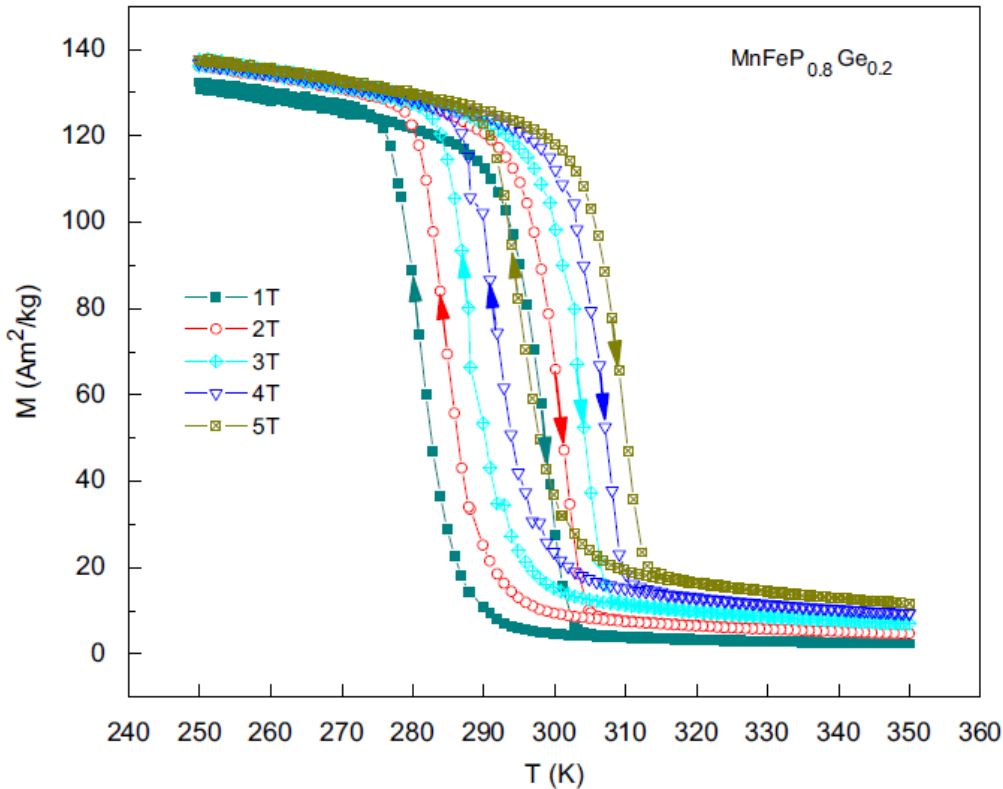


Fig. 1. Temperature dependence of the magnetization of $\text{MnFeP}_{0.8}\text{Ge}_{0.2}$ measured in constant fields of 1, 2, 3, 4 and 5 T with temperature increasing and decreasing in a step of 2 K, the arrows indicate the warming and cooling processes.

$$\Delta S_{\text{Max}} = \int_{H_1}^{H_2} \left(\frac{\partial M(T, H)}{\partial T} \right)_H dH$$

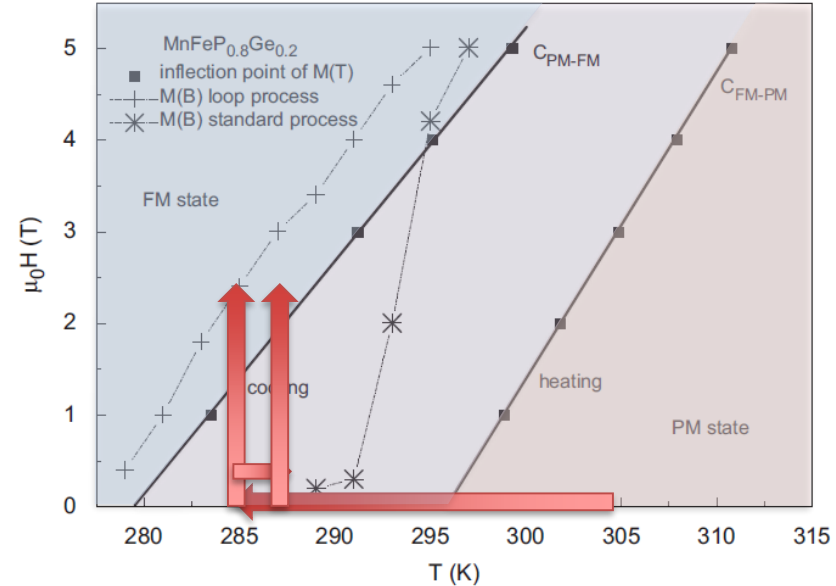
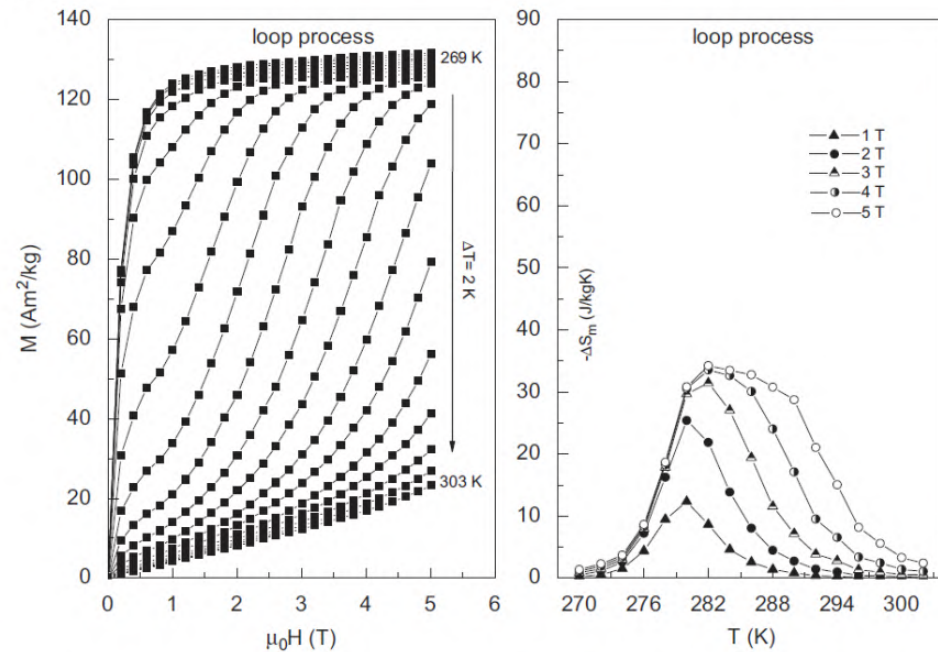
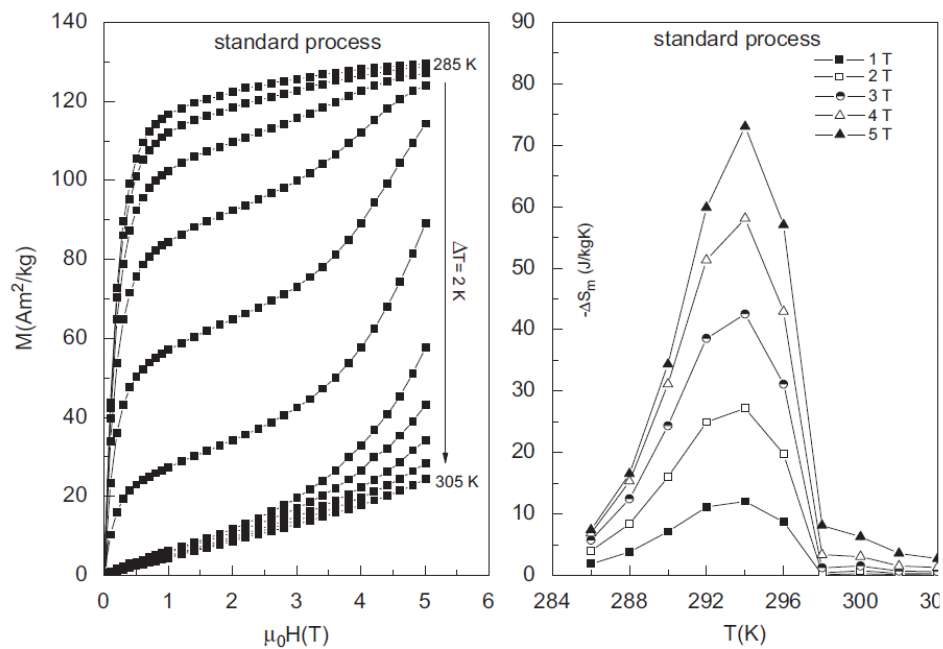
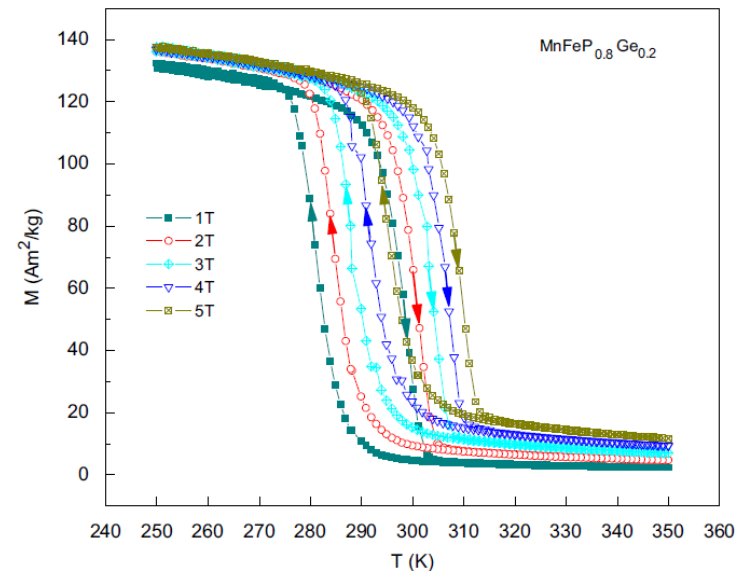


Fig. 2. Magnetic phase diagram of the compound $\text{MnFeP}_{0.8}\text{Ge}_{0.2}$ as derived from isofield magnetization measurements shown in Fig. 1 ■ (other symbols derived from M(B) see Figs. 3(a) and 4(a)).

Magnetometry

Constructing isofield $M(T)$ from isothermal $M(H)$ data requires consideration of field history where there is thermal/field hysteresis.



Magnetometry: Round Robin

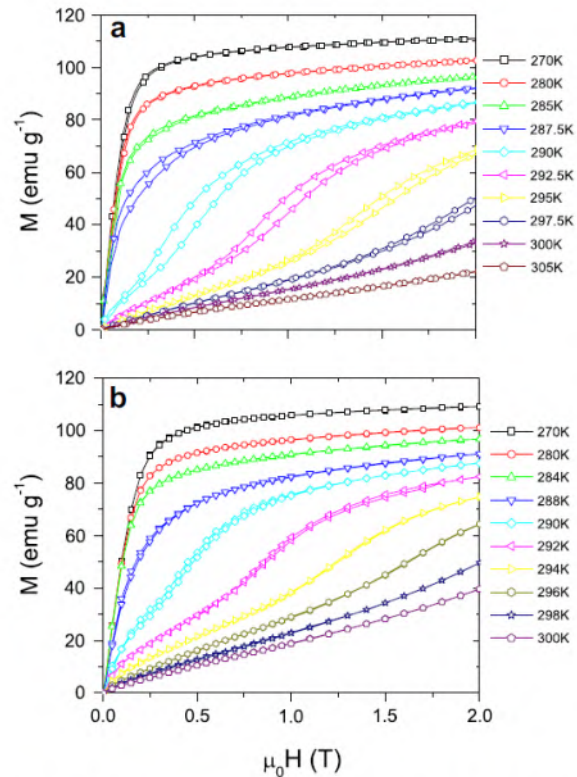


Fig. 3 – Magnetisation versus applied field for the MCP1011 sample (colour online). Data taken at (a) Imperial College (IC) using an Oxford Instruments vibrating sample magnetometer, (saturation magnetisation 112 emu.g^{-1} at 270 K and 2 T) and (b) IFW using a PPMS SQUID magnetometer (saturation magnetisation 110 emu.g^{-1} at 270 K and 2 T).

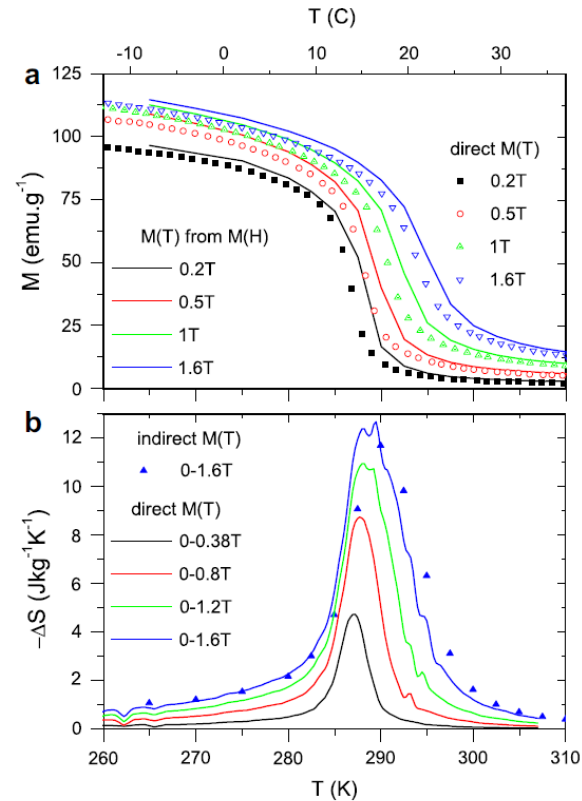
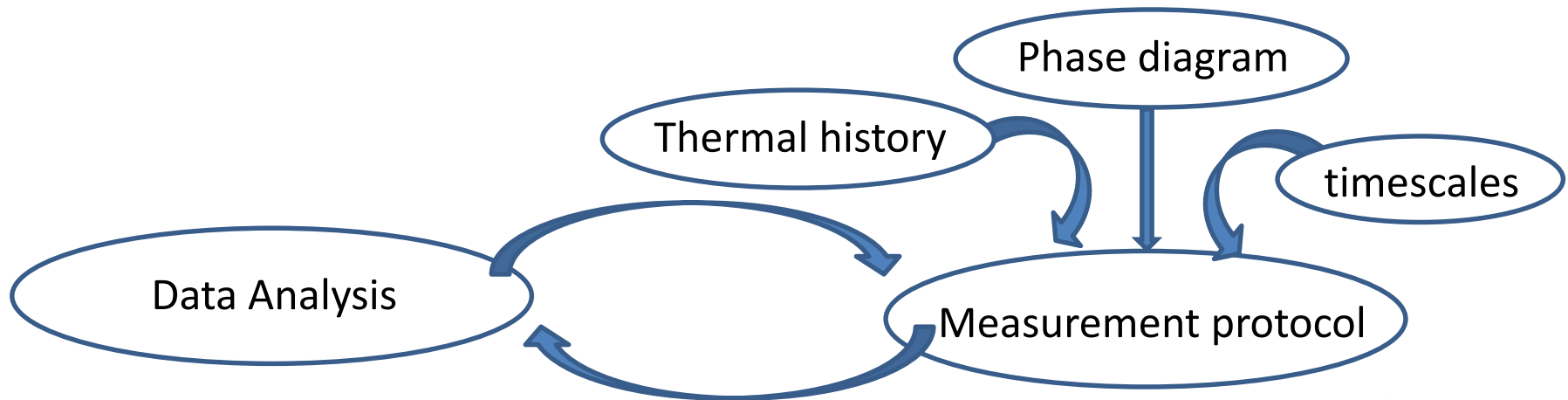


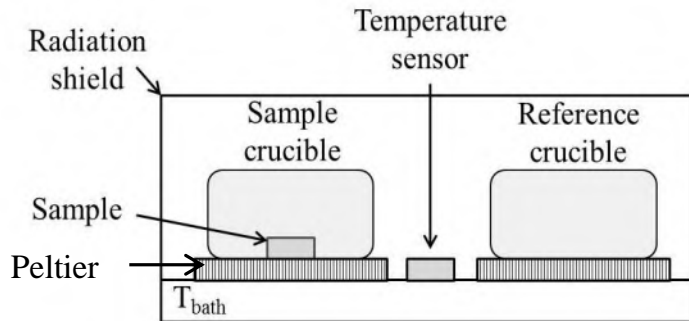
Fig. 4 – Entropy change determined from magnetisation measurements (colour online). (a) $M(T)$ curves measured directly (VAC) and extracted from $M(H)$ curves (IC/IFW). (b) The entropy change ΔS calculated from the $M(T)$ curves shown in (a).

Calorimetry: Potential Problems

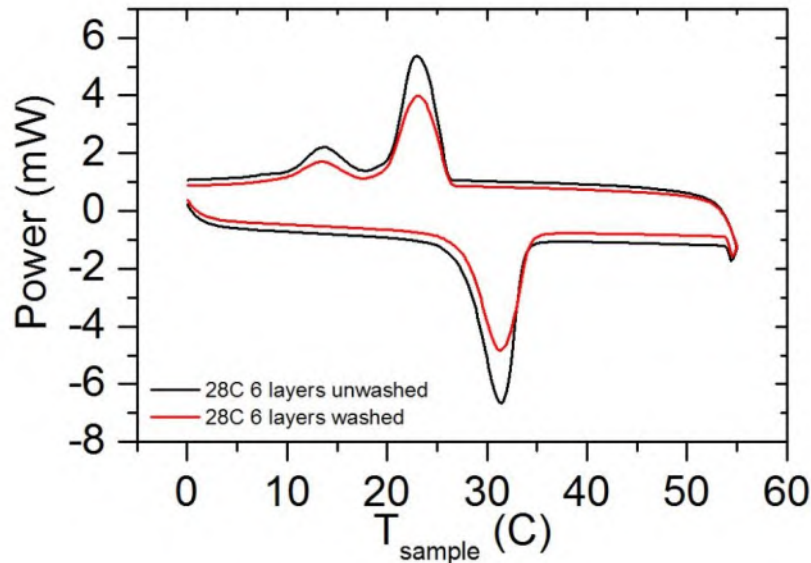
Phase diagram:	Effect of temperature/pressure
Thermal history:	“Isothermal” measurements
Timescales:	Driving frequency Thermal conductivity Equipment limitations
Data analysis:	Limitations of model used



Differential Scanning Calorimetry (DSC)



Heat flow measured.
Temperature scanned.



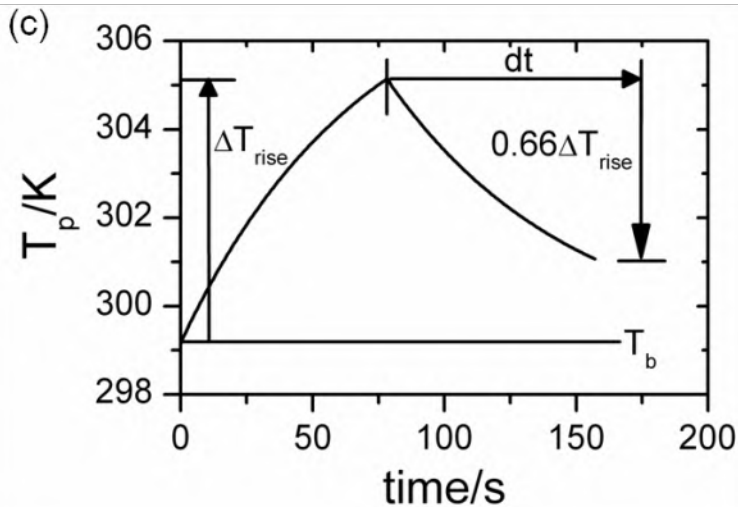
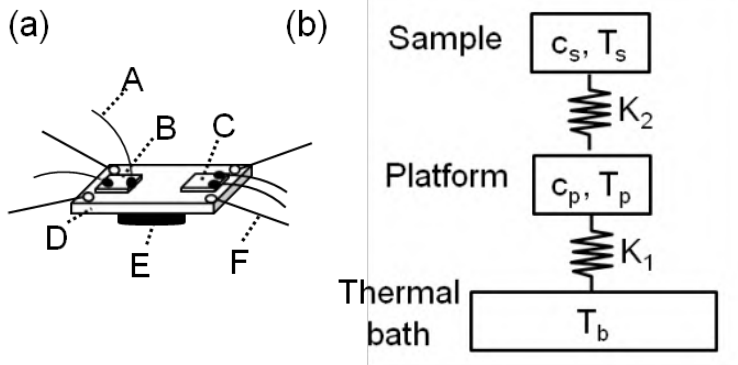
Power supplied to sample
(to keep it at same temperature as reference)

$$C_p = \frac{P}{\dot{T}m}$$

Heat capacity \rightarrow C_p \leftarrow Heating rate \leftarrow \dot{T} \leftarrow mass \leftarrow m

P is the power supplied to the sample.

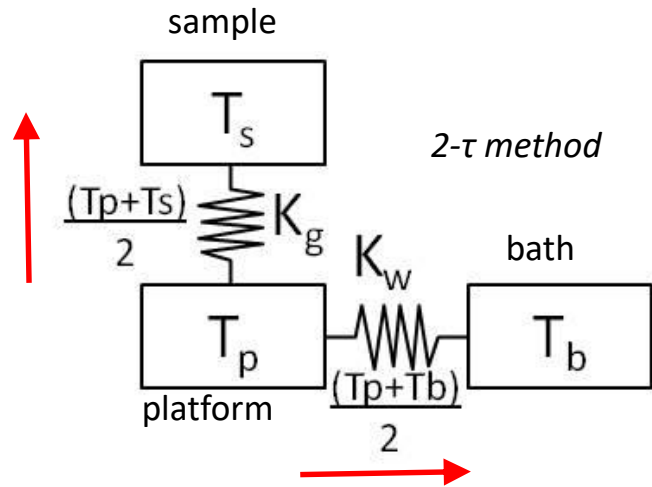
Relaxation Calorimetry



e.g. Quantum Design PPMS with heat capacity option

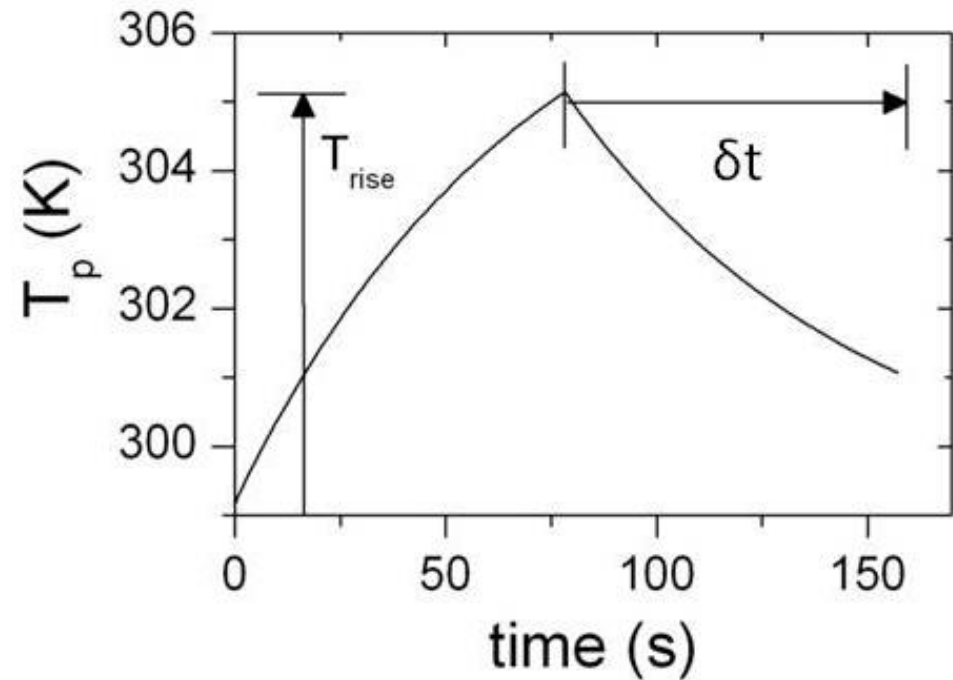


The PPMS heat capacity option: Curve fitting method^[1]



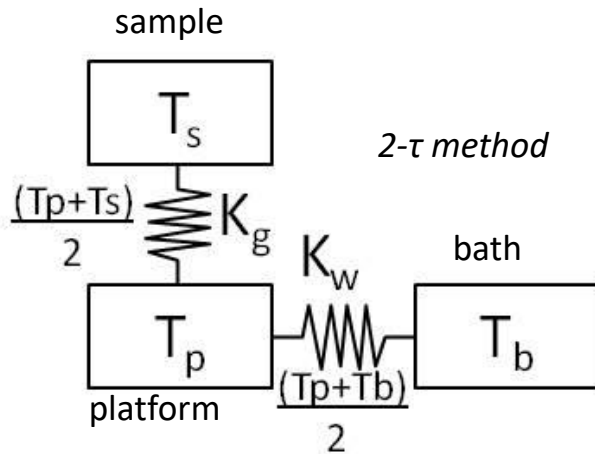
One relaxation curve gives you:

1. 1 data point averaged over T_{rise}
2. K_w
3. K_g



Sharp changes in heat capacity, or latent heat can result in poor measurement

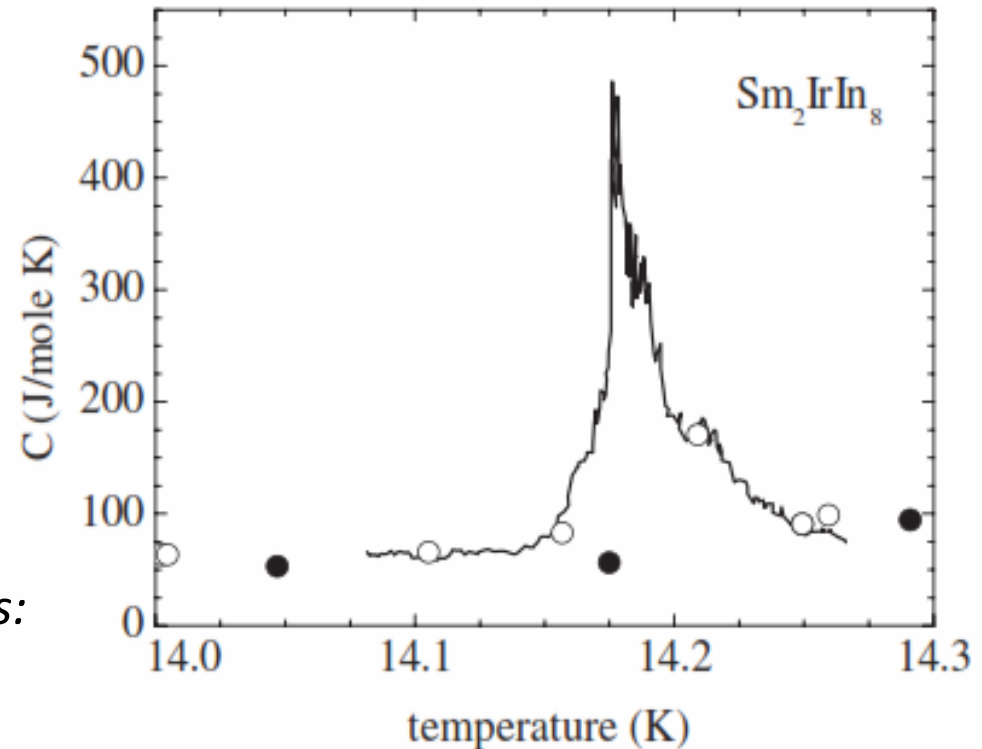
The PPMS heat capacity option: Curve fitting method^[1]



Evaluate same relaxation curve at every point using K_w, K_g and following equations:

$$C_s = \left[-C_p \frac{dT_p}{dt} + K_w [T_p(t) - T_b] + P(t) \right] \times \frac{dt}{dT_s}$$

Example^[1]



$$T_s = \frac{C_p \frac{dT_p}{dt} + K_w [T_p - T_b] - P(t)}{K_g} + T_p(t)$$

^[1]J.C. Lashley *et al.*, *Cryogenics* **43** 369–378 (2003)

H. Suzuki *et al.*, *Cryogenics* **50** 693–699 (2010)

Measurement techniques: round robin study

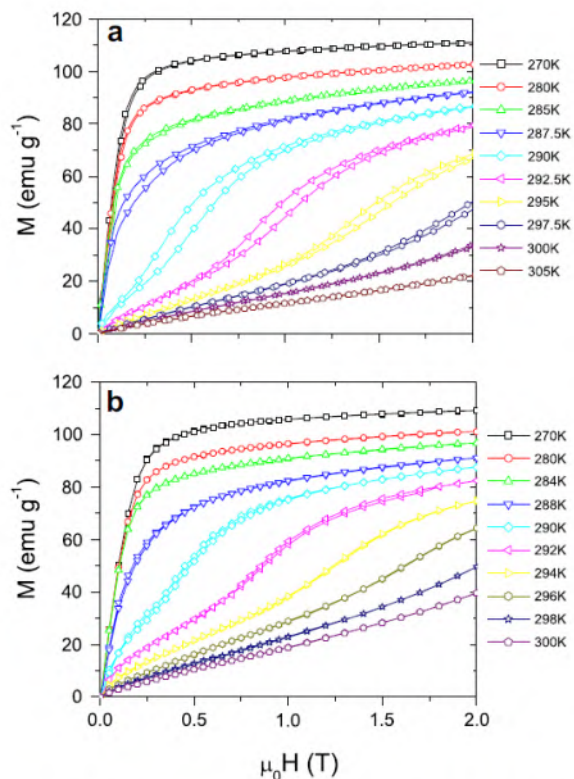


Fig. 3 – Magnetisation versus applied field for the MCP1011 sample (colour online). Data taken at (a) Imperial College (IC) using an Oxford Instruments vibrating sample magnetometer, (saturation magnetisation 112 emu.g^{-1} at 270 K and 2 T) and (b) IFW using a PPMS SQUID magnetometer (saturation magnetisation 110 emu.g^{-1} at 270 K and 2 T).

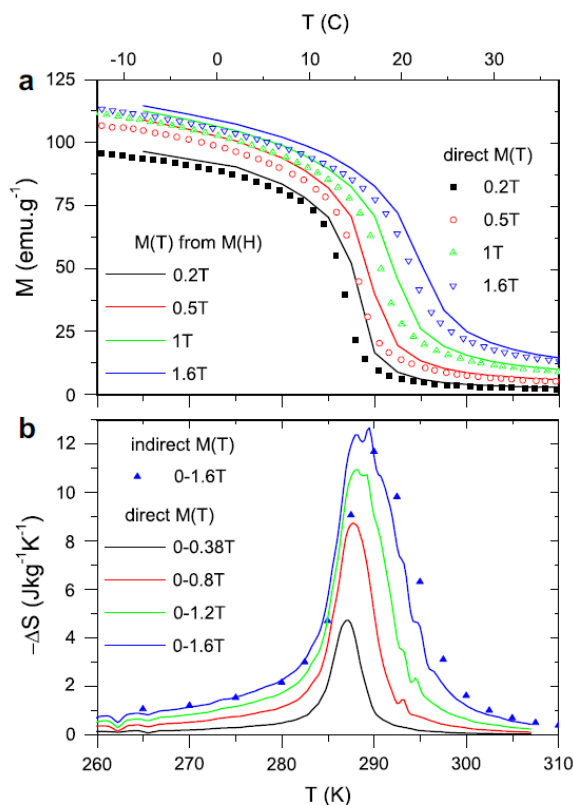


Fig. 4 – Entropy change determined from magnetisation measurements (colour online). (a) $M(T)$ curves measured directly (VAC) and extracted from $M(H)$ curves (IC/IFW). (b) The entropy change ΔS calculated from the $M(T)$ curves shown in (a).

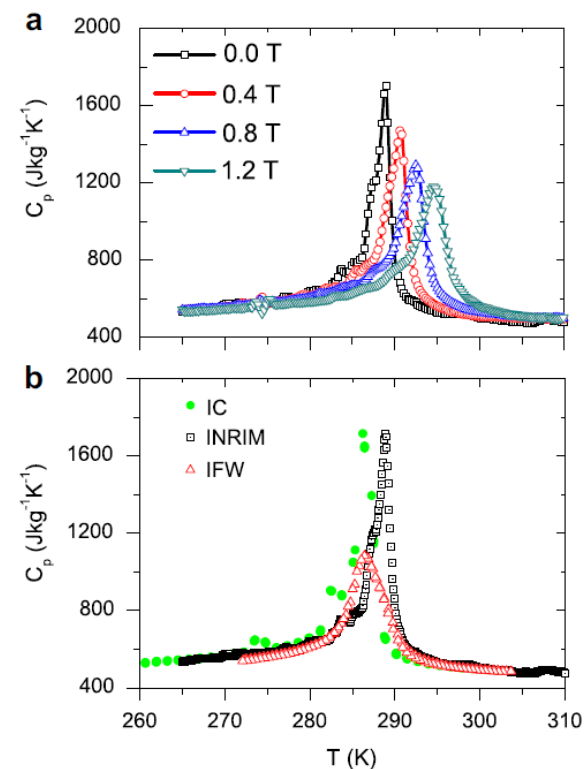


Fig. 5 – Heat capacity data for the MCP1011 sample (colour online). (a) INRIM data taken in various fixed external fields, and (b) heat capacity in zero field: a comparison of heat capacity measured in three laboratory environments using different methods.

Want to know more?: See ESM 2019, Vittorio Basso.
Indirect techniques: calorimetry, dilatometry, transport

Giant Magnetocaloric Effect in $Gd_5(Ge_xSi_{1-x})_4$

ADVANCED
MATERIALS

$Gd_5(Si_xGe_{1-x})_4$: An Extremum Material**

By Vitalij K. Pecharsky* and
Karl A. Gschneidner Jr.

The large shear displacements of atomic layers in $Gd_5(Si_xGe_{1-x})_4$ materials, coupled with the change of crystallographic symmetry and magnetic order, characterizes these transformations as magnetic–martensitic, which are extremely rare. The start and the end of the magnetic–martensitic transitions depends strongly on the direction of change (i.e., increasing or decreasing) of either or both the temperature and magnetic field. These profound bonding, structural, electronic, and magnetic changes, which occur in the $Gd_5(Si_xGe_{1-x})_4$ system, bring about some extreme changes of the materials' behavior resulting in a rich variety of unusually powerful magneto-responsive properties, such as the giant magnetocaloric effect, colossal magnetostriction, and giant magnetoresistance.

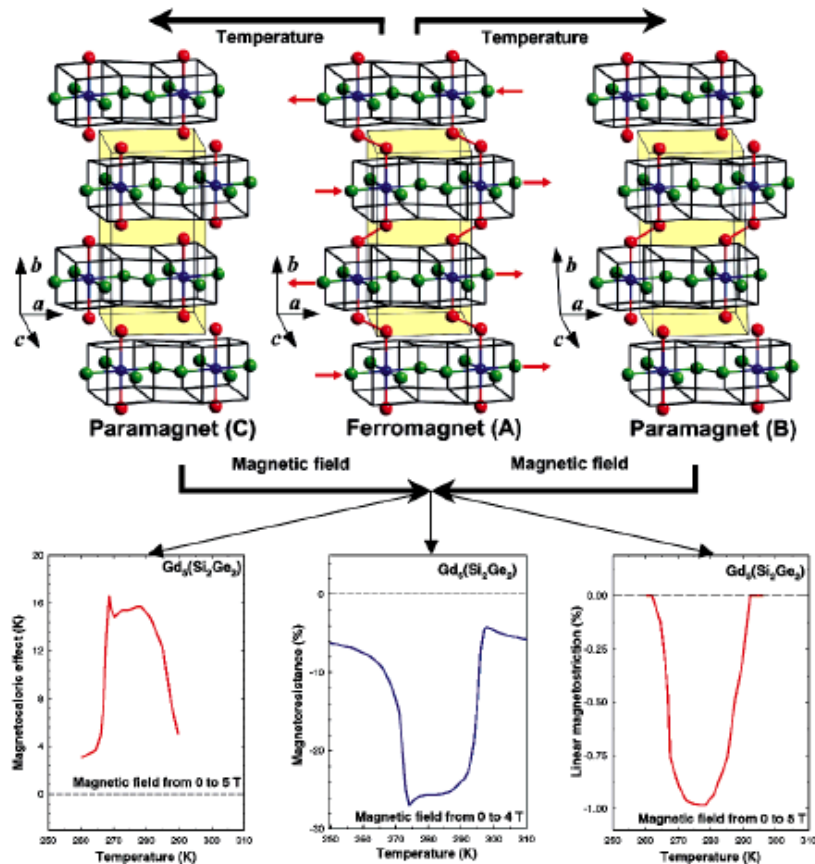
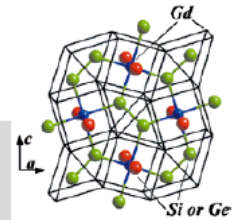
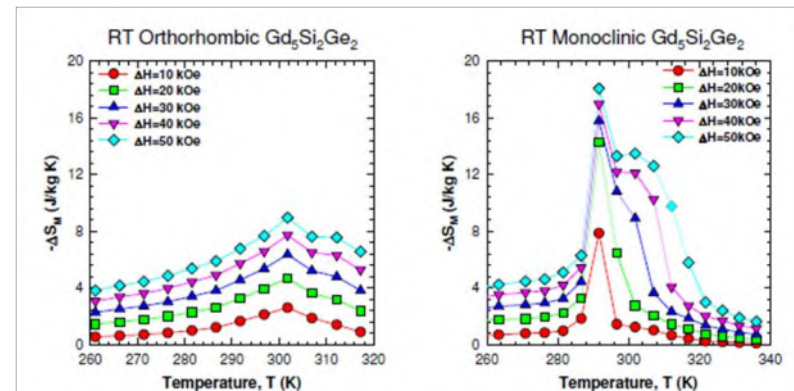


Fig. 3. Correlation between the magnetic responses of the $Gd_5(Si_xGe_{1-x})_4$ materials and their crystal structures for $0 \leq x \leq 0.5$. At low temperatures the compounds are ferromagnetic (A) with all slabs (light blue) connected via the Si(Ge)–Si(Ge) covalent bonds. Depending on the composition, the materials become paramagnetic with either one-half (B) or none (C) of the slabs connected above the Curie temperatures as shown by long horizontal arrows at the top of the figure. The transitions from state A to B or A to C are coupled with shear movement of the slabs by -0.8 (A \rightarrow B) or -1.1 Å (A \rightarrow C) as indicated by red horizontal arrows. When a magnetic field is applied above Curie temperature, the reverse magnetic–martensitic transitions occur (B \rightarrow A or C \rightarrow A), as shown by long horizontal arrows in the middle) resulting in the giant magnetocaloric effect, giant magnetoresistance, and colossal magnetostriction. These are shown at the bottom of the figure for the $Gd_5(Si_2Ge_2)$ composition (i.e., for $x = 0.5$). The unit cells of the three crystallographic modifications existing in the $Gd_5(Si_xGe_{1-x})_4$ system are highlighted in yellow.

Paramagnetic - Monoclinic
Ferromagnetic - Orthorhombic

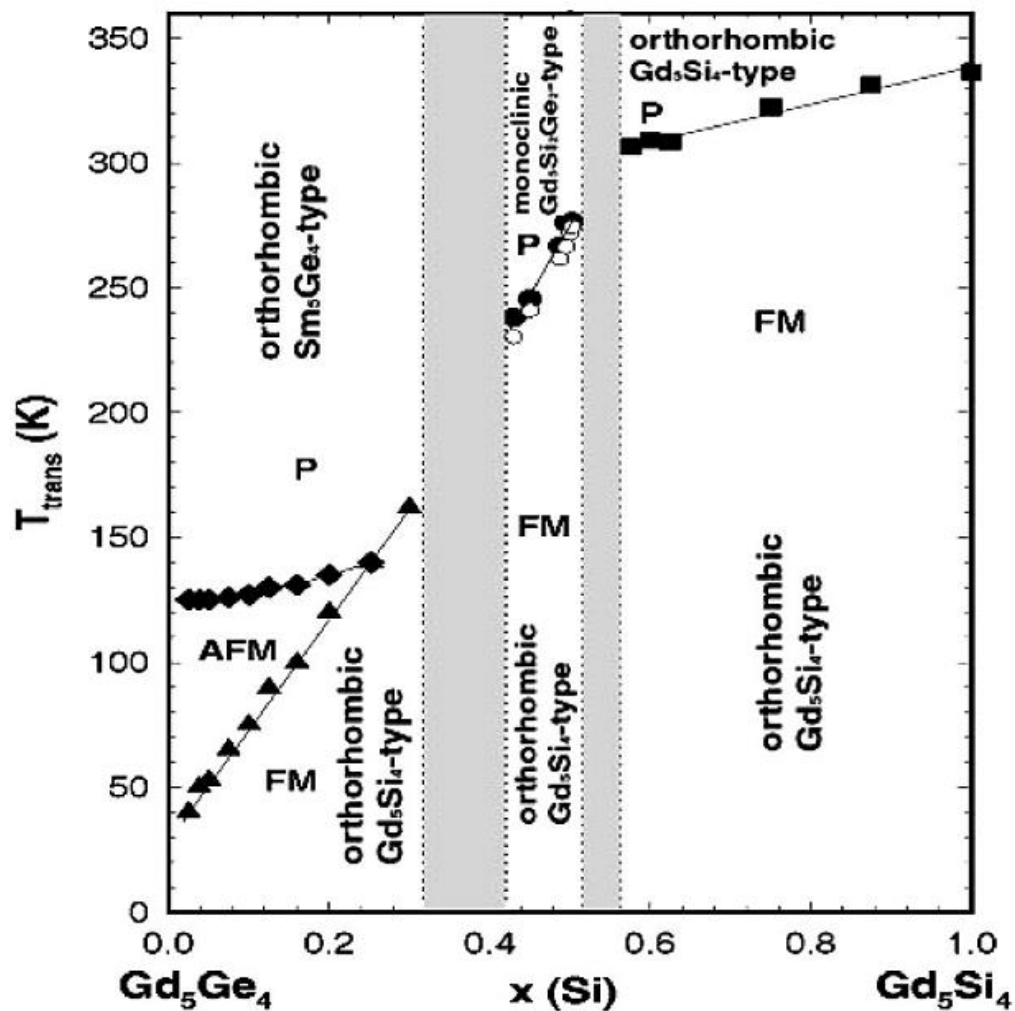
Adv. Mater. 2001, 13, No. 9, May 3



Same alloy, different heat treatments to obtain the two structures.

Courtesy of K.A. Gschneidner

Magneto-structural Coupling



>20 K AFM-FM

<20K Magnetic glass

$T_N \sim 130$ K



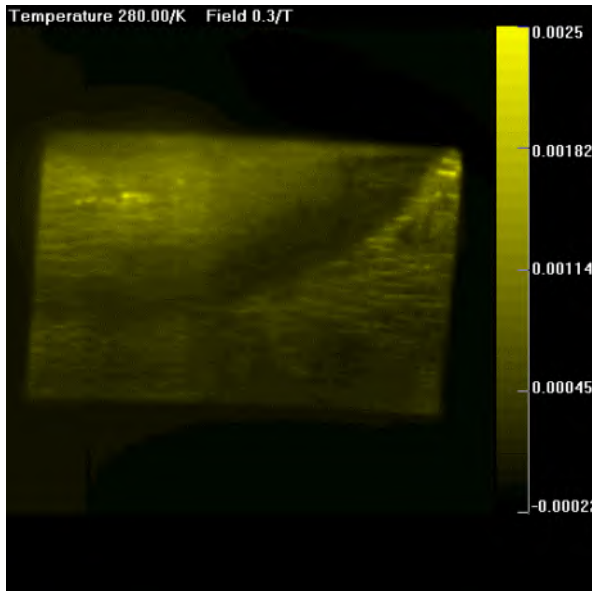
$T_C \sim 275$ K:

FM-PM +

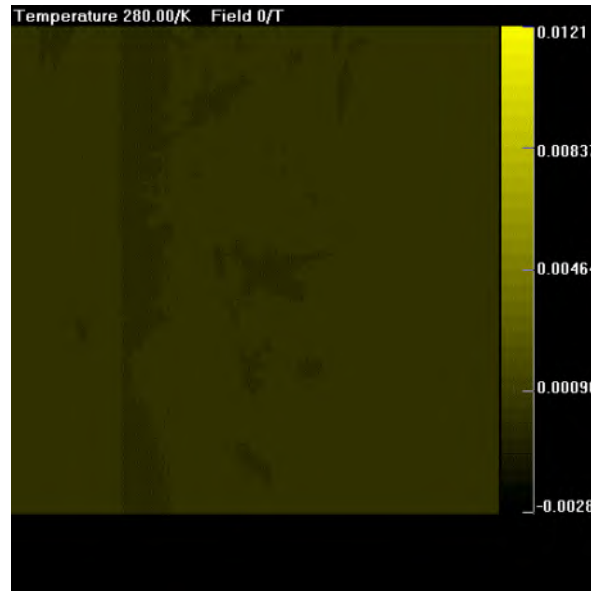
orthorhombic-monoclinic

Gd₅Ge₂Si₂ Single Crystal

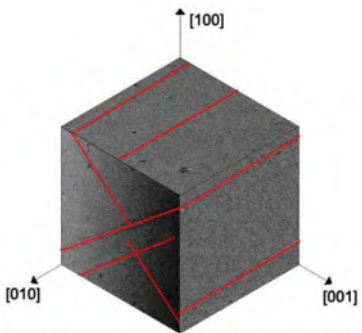
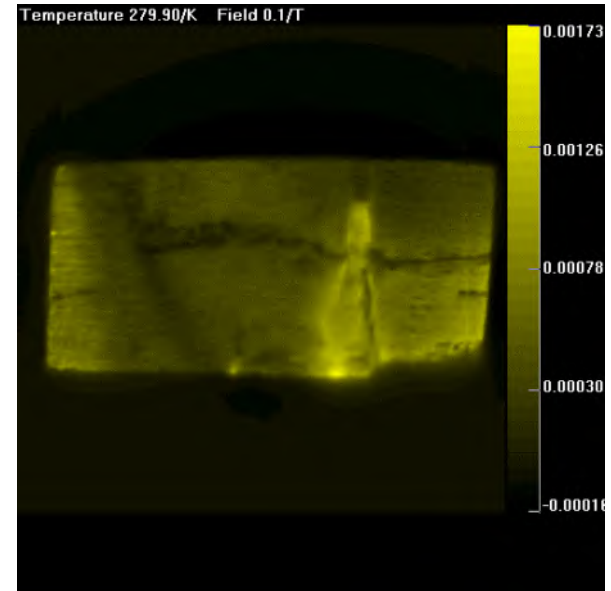
a axis



b axis



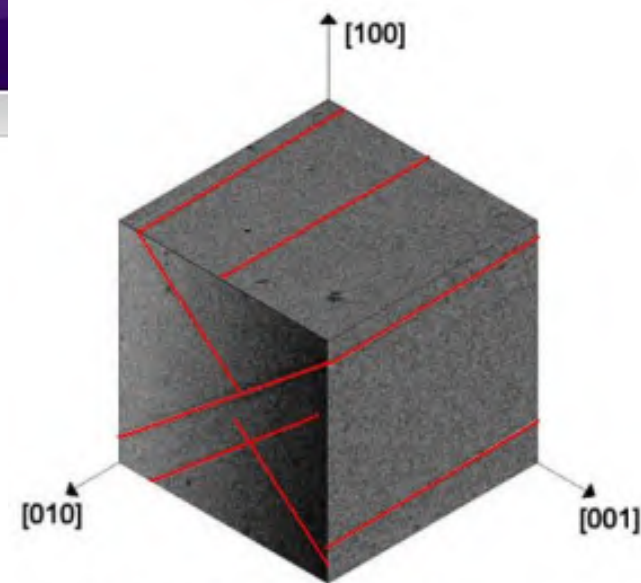
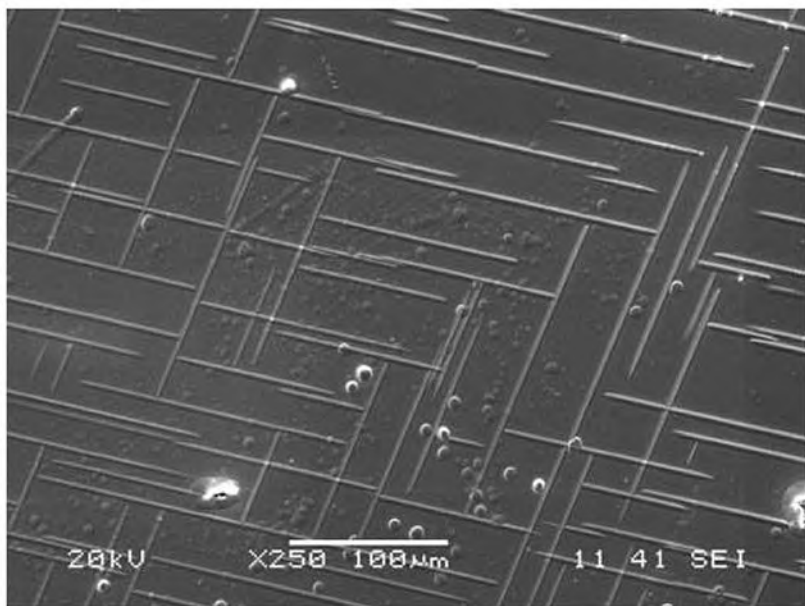
c axis



Digitally rendered 3-D image
of a single crystal Gd₅Si₂Ge₂

Gd₅Ge₂Si₂ Single crystal

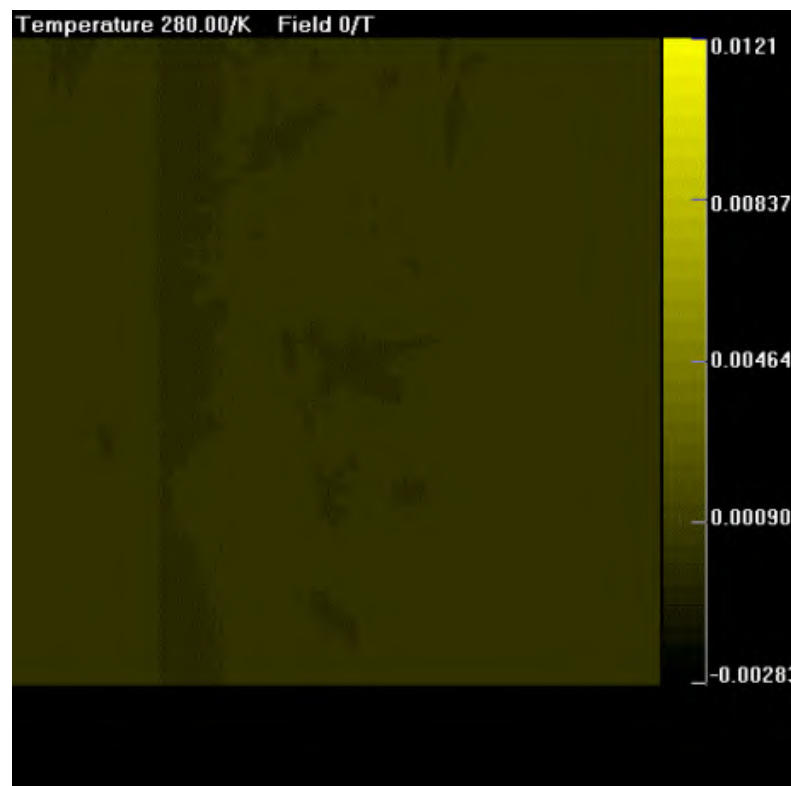
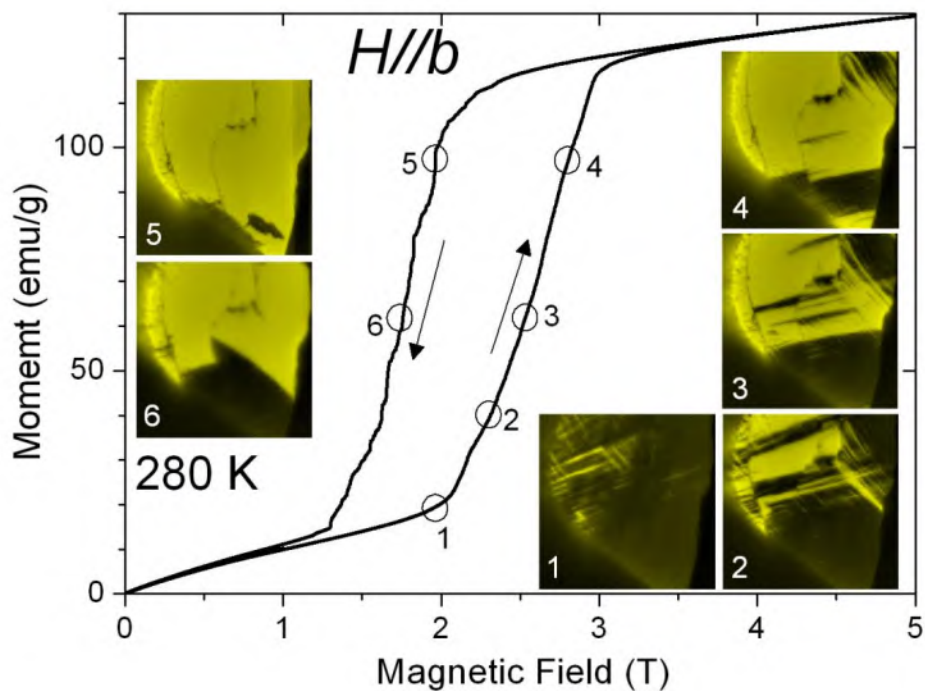
- Platelets of Gd₅(Ge_xSi_{1-x})₃
- Criss-cross pattern in [010] b-face
- Stripe pattern in [100] a- and [001] c-face



Digitally rendered 3-D image of a single crystal Gd₅Si₂Ge₂

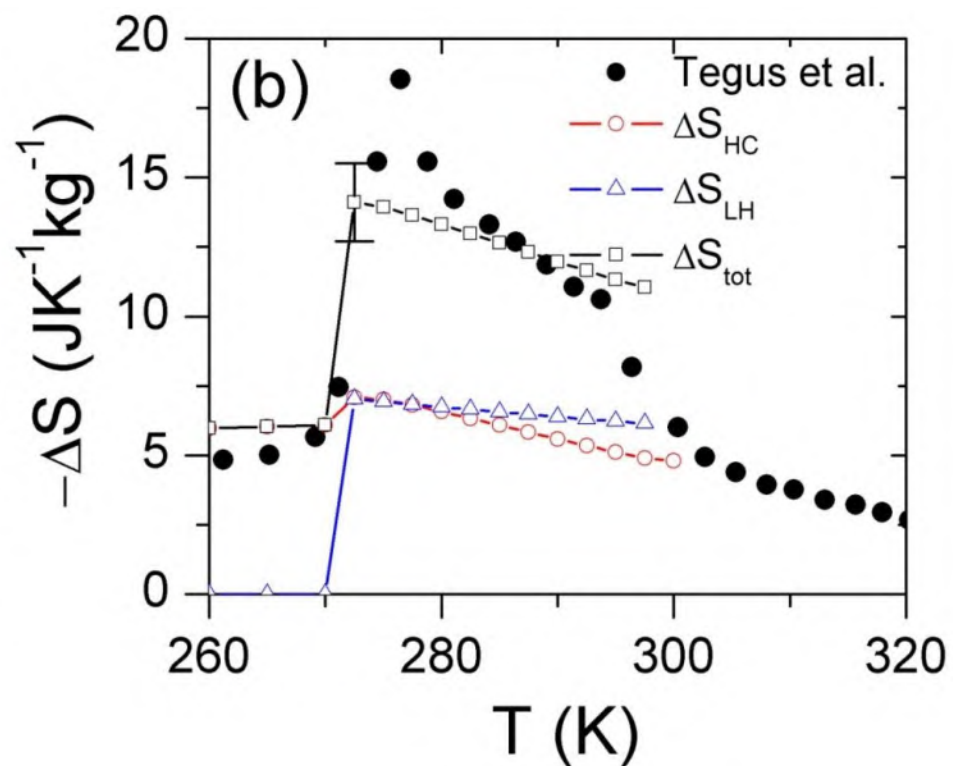
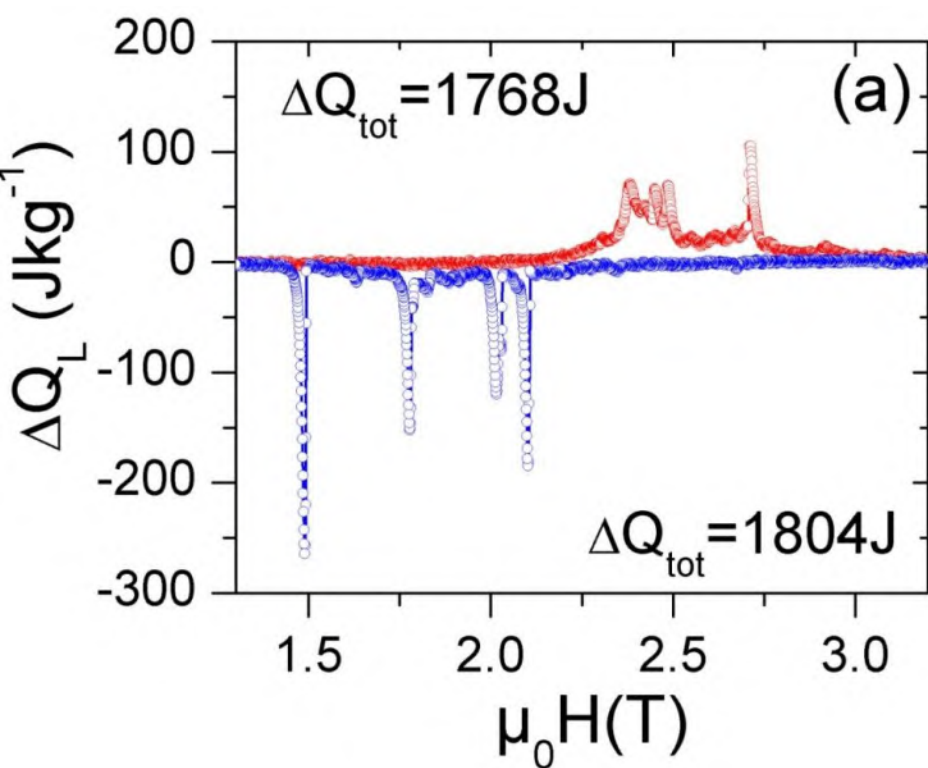
How does presence of platelets affect AFM-FM transition?

Gd₅Ge₂Si₂ Single Crystal



Presence of PM platelets seed the phase transition.

Gd₅Ge₂Si₂ Microcalorimetry



Microcalorimetry: $\Delta S_{\text{LH}} = 1/2 \Delta S_{\text{total}}$

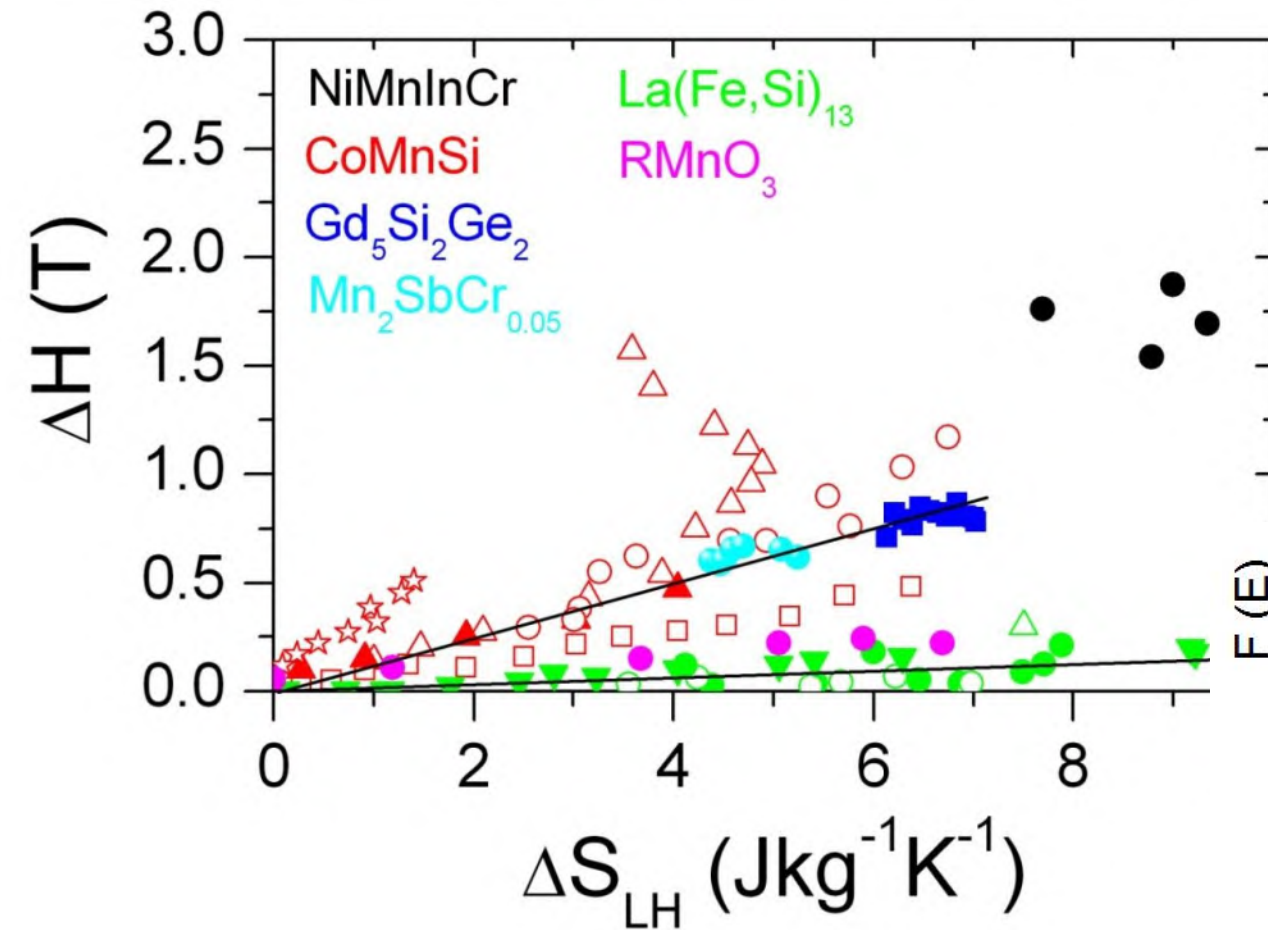
(Bulk methods estimate ΔS_{str} as somewhere between 40 and 60% of ΔS_{total} [2],[3])

[1] K. Morrison *et al.*, MRS Conference proceedings, **1310** 47-53 (2011).

[2] G.J. Liu *et al.*, *Appl. Phys. Lett.* **88**, 212505 (2006).

[3] V.K. Pecharsky *et al.*, *J. Magn. And Magn. Mater.* **321**, 3541-3547 (2009).

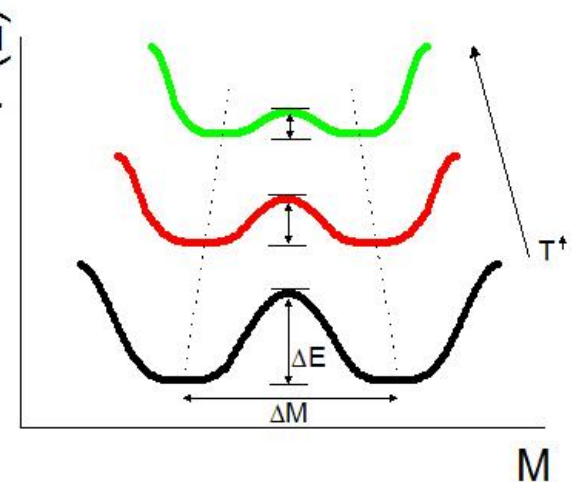
Approaching the Critical Point



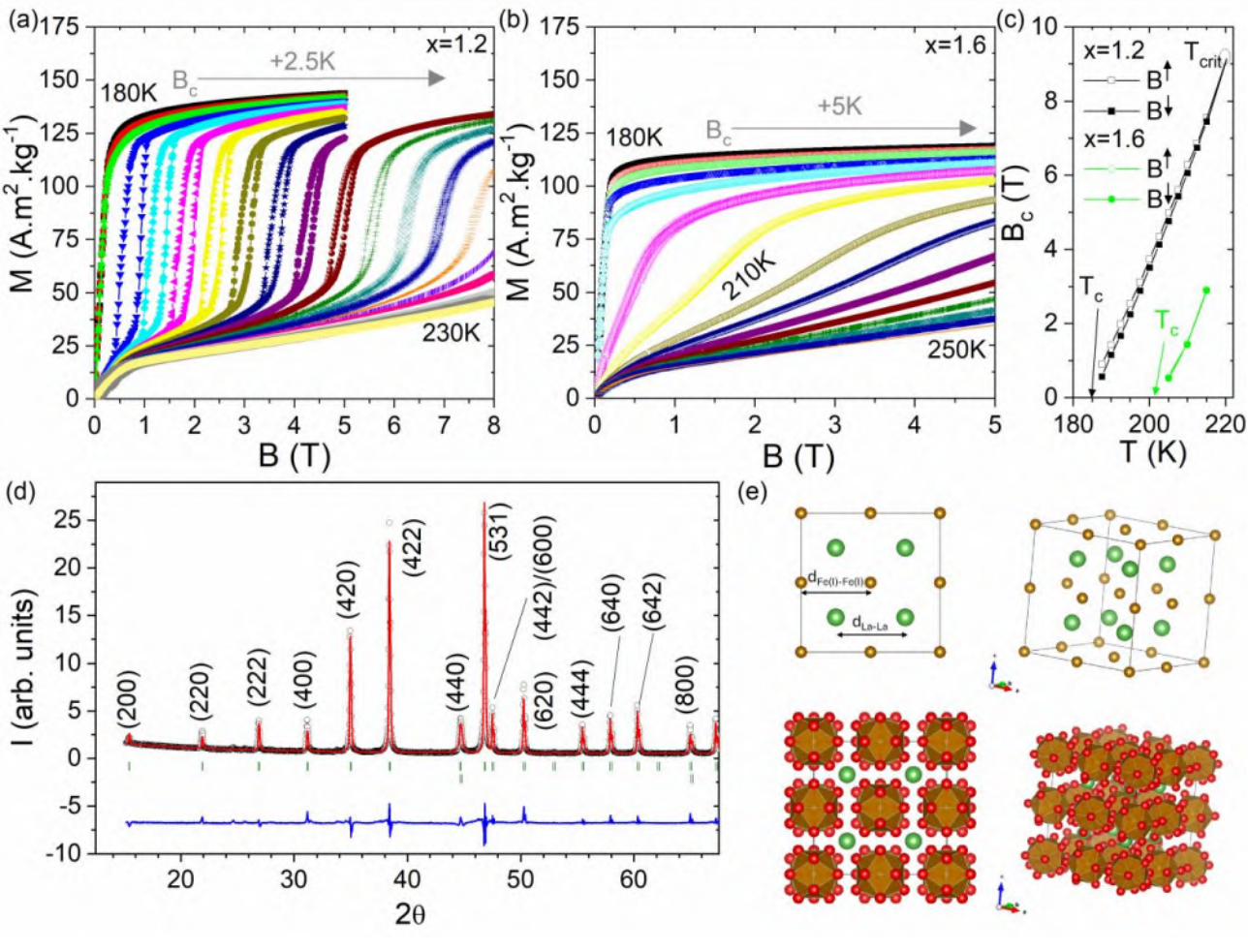
Aim:

Maximise ΔS
whilst limiting ΔH

Coupling to a structural phase transition significantly enhances the entropy change, but at what cost?



LaFe_{13-x}Si_x

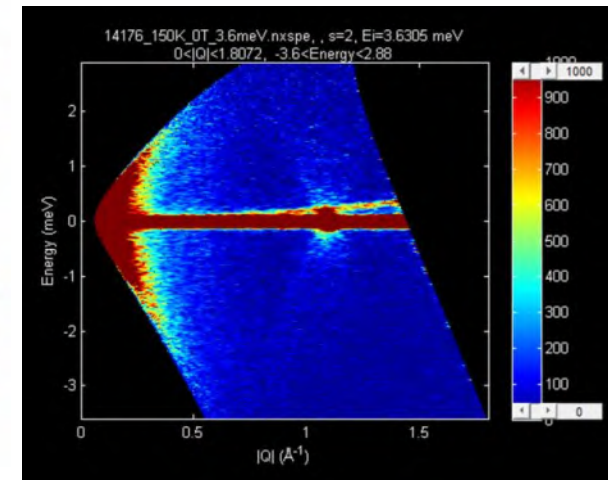


FOPT with tricritical point:

- $x = 1.2$ as a function of T, B

SOPT:

- $x = 1.6$ as a function of T



Want to know more on materials?

See ESM 2013, Karl Sandeman.

Magneto-caloric materials

The tri-critical point

This type of behavior, where three critical lines come together at a point, is rather exceptional in nature (we shall mention some other examples below) and we therefore propose a special name: **tricritical point**.

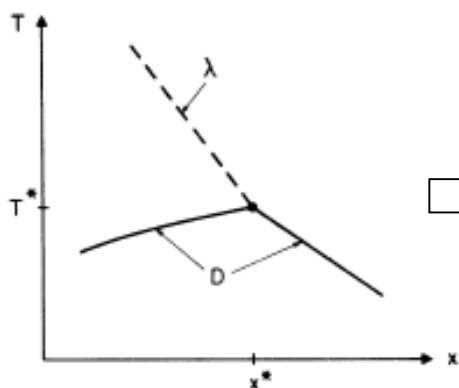


FIG. 1. Phase diagram (schematic) for He³-He⁴ mixtures near the critical mixing point. The two-fluid coexistence curve is labeled D and the dashed curve is the line of lambda transitions.

$$\eta, \Delta = \mu_3 - \mu_4$$

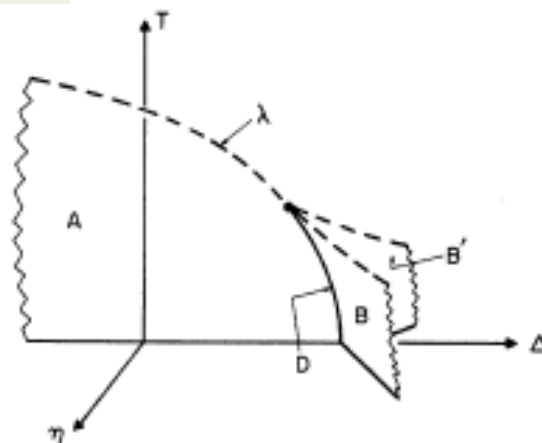


FIG. 2. Phase diagram (schematic) for He³-He⁴ mixtures in $T\Delta\eta$ space. Note that only the $T\Delta$ plane with $\eta=0$ is experimentally accessible.

Universal Scaling

$$\Delta S_{Max} = \int_0^{H_2} \left(\frac{\partial M(T, H)}{\partial T} \right)_H dH$$

PHYSICAL REVIEW B 81, 224424 (2010)

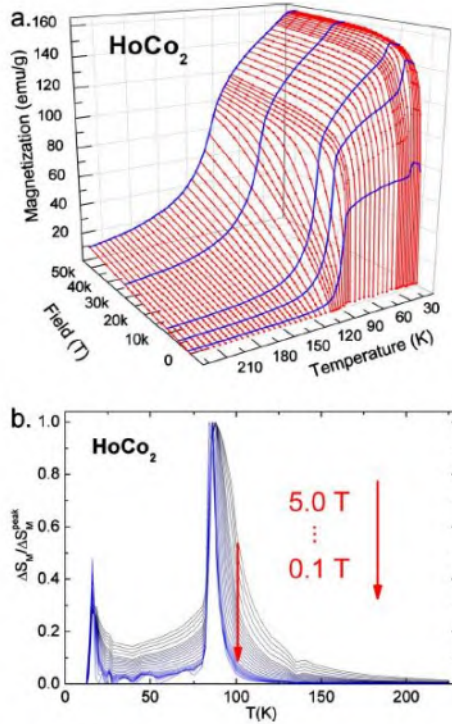


FIG. 1. (Color online) (a) Magnetization measurements as function of field for different temperatures for HoCo₂. The values of applied field during the measurement were 0, 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.7, 0.9, 1.0, 1.2, 1.4, 1.6, 1.8, 2.0, 2.2, 2.4, 2.6, 2.8, 3.0, 3.5, 4.0, 4.5, and 5.0 T. (b) Normalized entropy change versus temperature for different applied fields for HoCo₂.

$$h = \Delta S_{Max} / \Delta S_{Max}^{Peak}$$

$$\theta = \begin{cases} -\frac{T - T_c}{T_{r1} - T_c} & T \leq T_c \\ \frac{T - T_c}{T_{r2} - T_c} & T > T_c \end{cases}$$

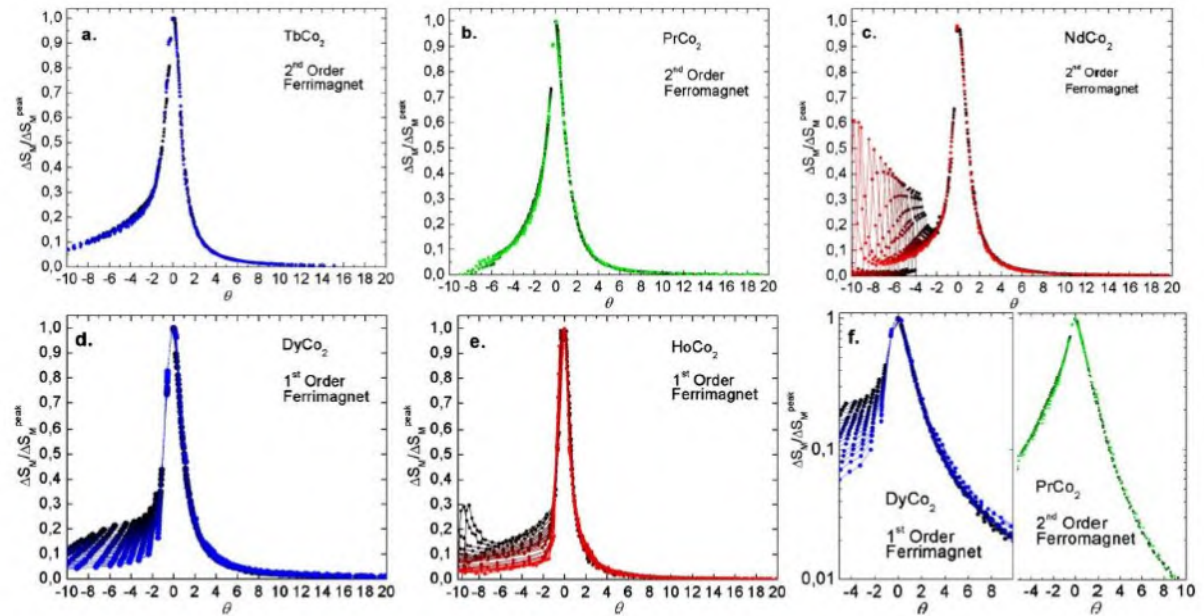
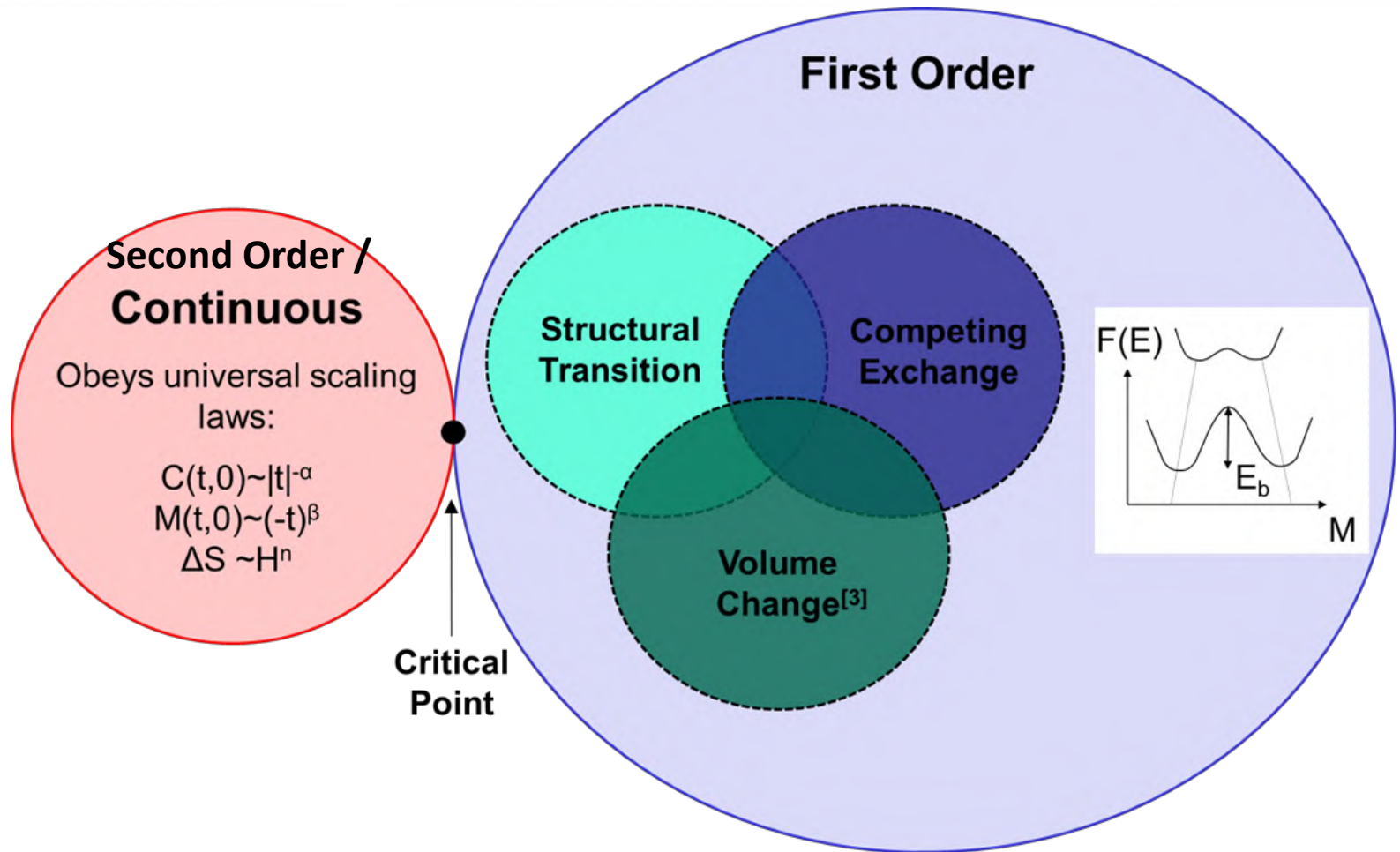


FIG. 2. (Color online) Normalized entropy change as a function of the rescaled temperature θ for the cobalt Laves phases studied in this work. A universal curve for the second-order phase transitions of TbCo₂ (panel a), PrCo₂ (panel b), and NdCo₂ (panel c) is demonstrated while a breakdown of the universal curve for the first order phase transitions of DyCo₂ (panel d) and HoCo₂ (panel e) can be observed. The panel f shows a comparison of the rescaled curves for PrCo₂ and DyCo₂ (vertical axis in logarithmic scale).

Universal Scaling



The Banerjee Criterion

Landau Free Energy Expansion

$$\frac{H}{M} = A + BM^2 + CM^4 + DM^6 + \dots$$

Banerjee Criterion:
(Landau only)

$$B < 0$$

Shimizu Criterion:
(Landau+spin fluctuations)

$$\frac{3}{16} < \frac{AC}{B^2} < \frac{9}{20}$$

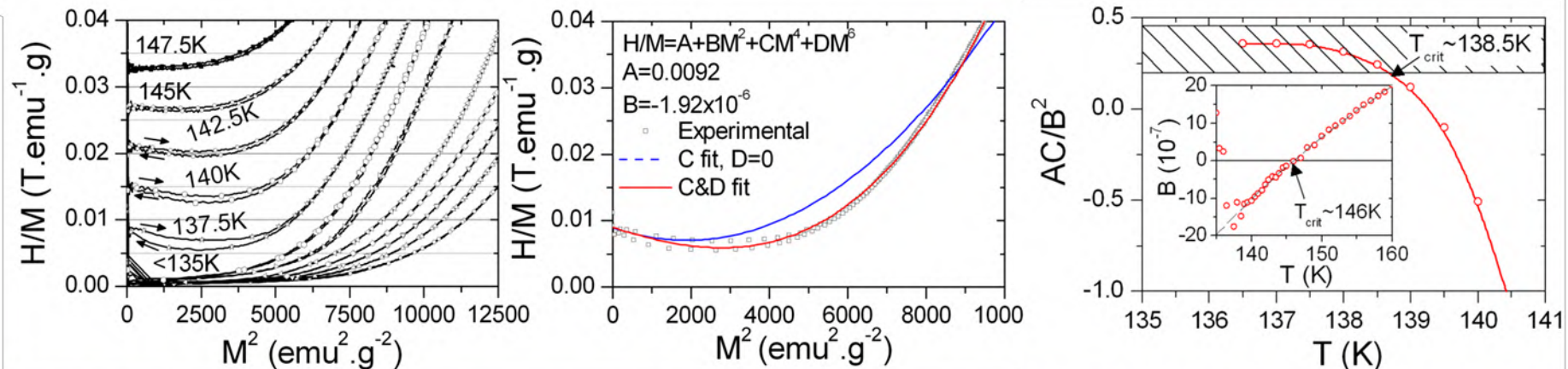
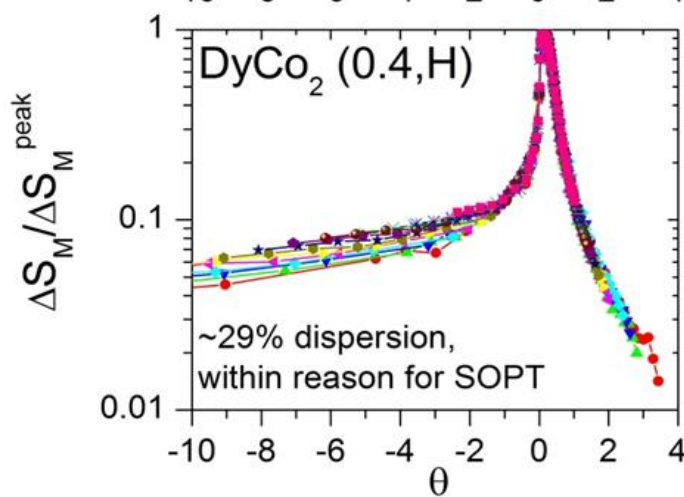
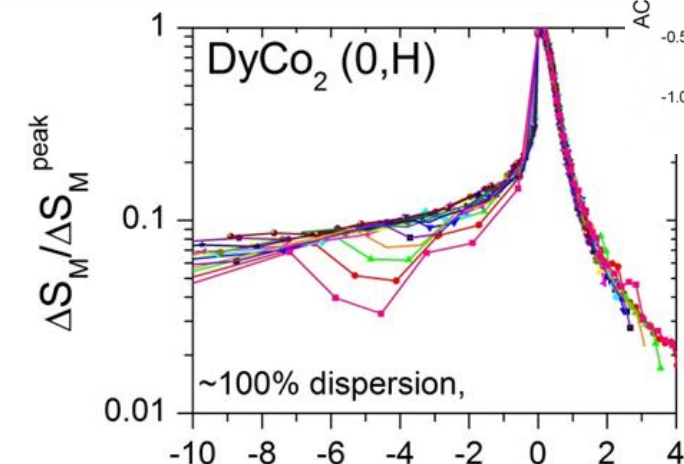
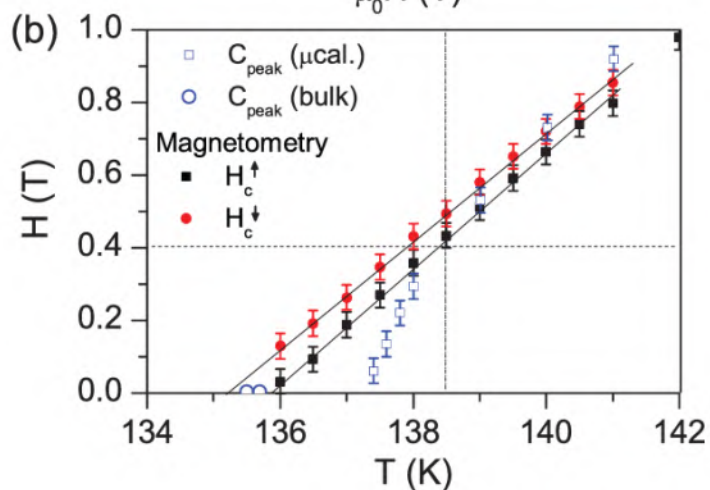
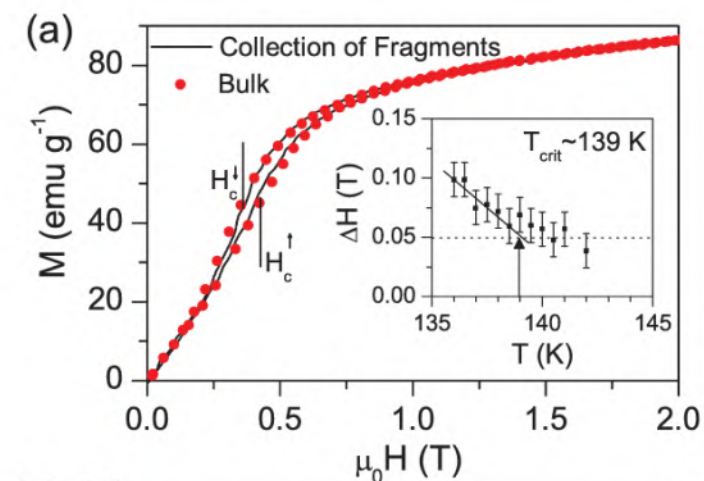


Figure 1 – Results of Landau fitting to isothermal magnetisation data. Left: Arrott plots of magnetisation data for 2.5 K increments from below T_c (~135.5 K) to 147.5 K. Centre: Example of Landau fitting at 137.5 K. Right: Temperature variation of the quantity AC/B^2 where A , B , and C are parameters extracted from Landau fitting. Inset shows B as a function of temperature. Note that the Shimizu criterion indicates $T_{crit} = 138.5$ K whereas the Banerjee criterion indicates $T_{crit} = 146$ K.

Universal Scaling



Thermal Engineering: Next Challenge?

➤ ΔS , ΔT_{ad}

Several potential material systems

e.g. $\text{La}(\text{Fe},\text{Si})_{13}$ $\text{MnFe}(\text{P},\text{Si})$

➤ H_c , ΔH

➤ Cost

Engineering problem unique to each system

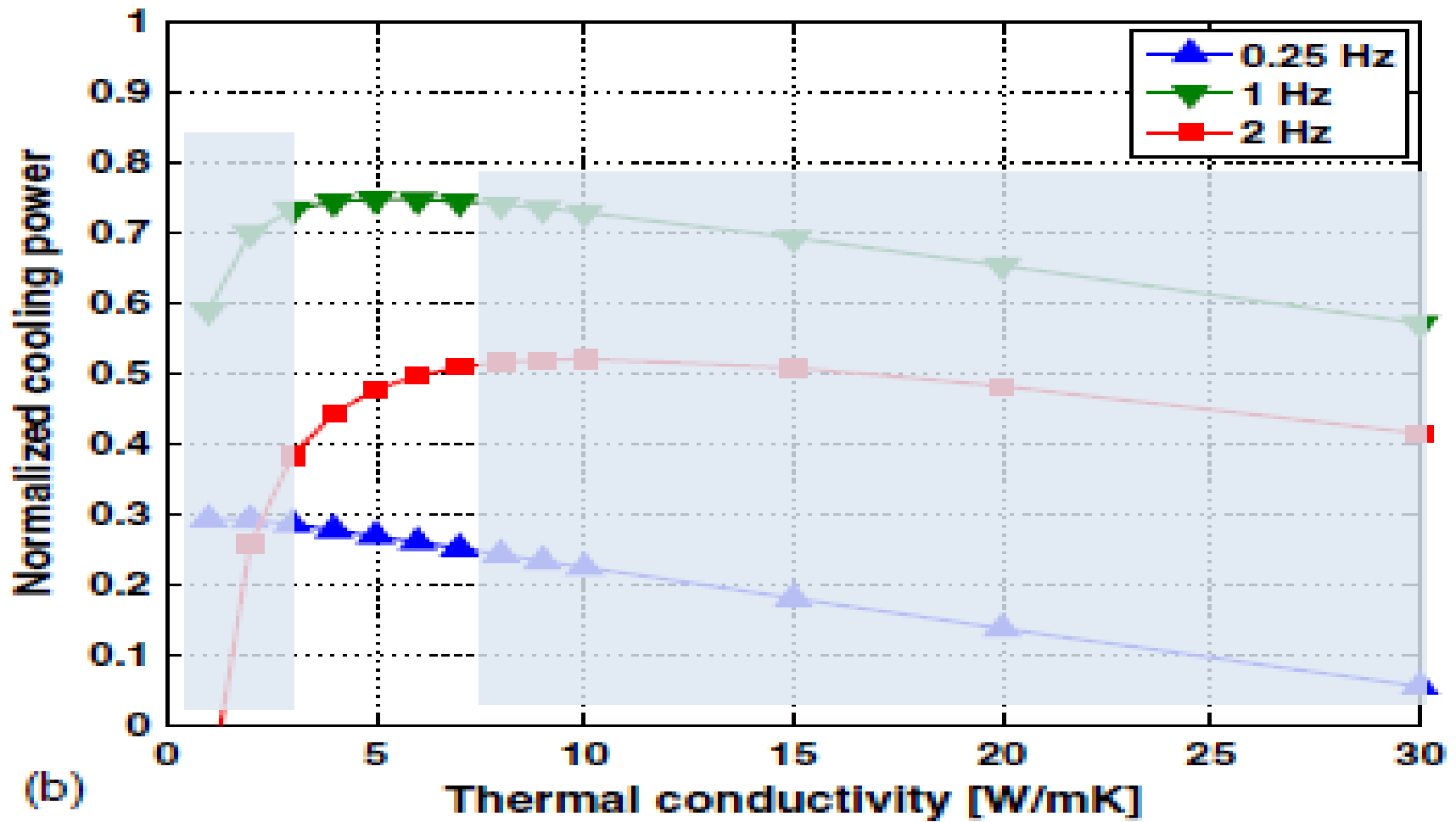
➤ Manufacturability

➤ Tunability

Tailor T_c , κ^1 to optimise the working performance

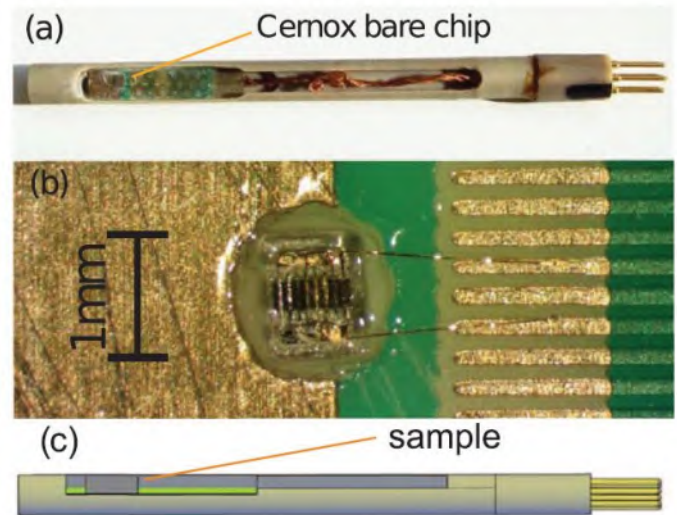
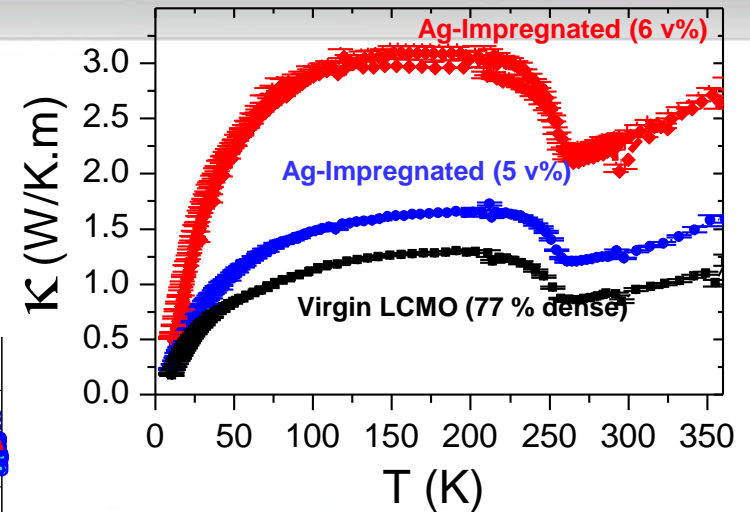
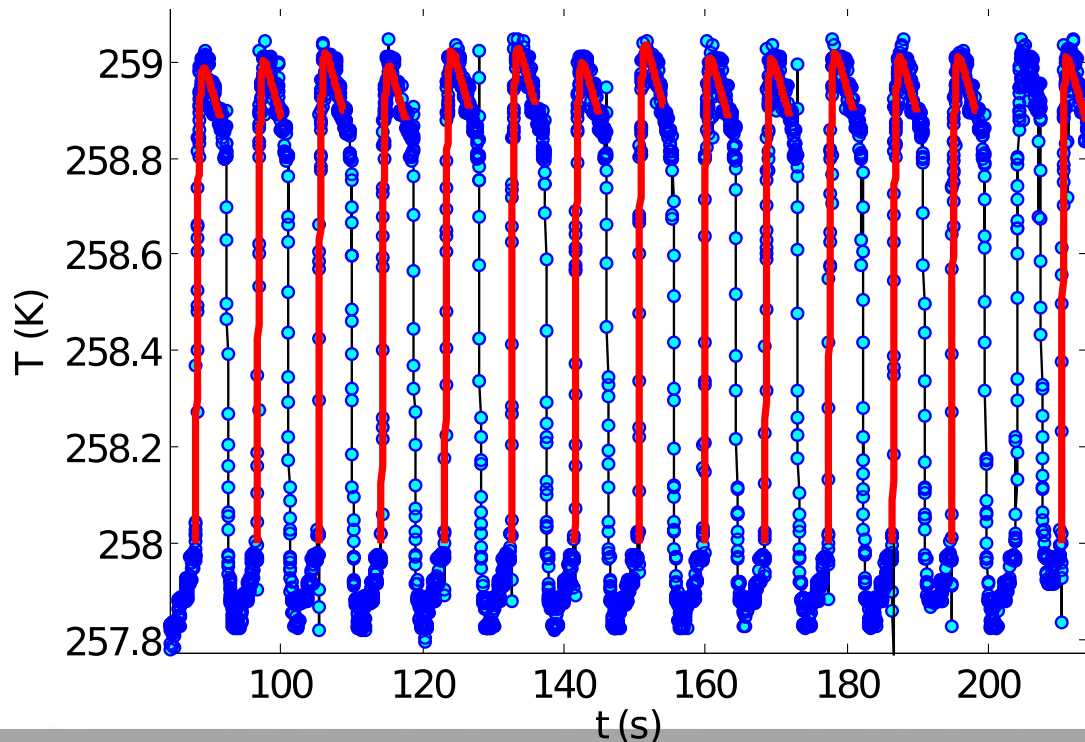
Want to know more?: See ESM 2021, Oliver Gutfleisch
Magneto (and multi-)caloric materials for efficient refrigeration

Thermal Engineering



Thermal Engineering

Optimise κ Improve cycling stability



J. Lyubina *et al.*, *Adv. En. Mater.*, **2**, 1323-1327 (2012)

J.A. Turcaud *et al.*, *Scripta Mater.*, **68**, 510-513 (2013)

G. Porcari *et al.*, *Rev Sci Instrum.* **84**, 073907 (2013)



TCCbuilder:

an open-source tool for the analysis of thermal control elements and thermal control circuits

LIBRARY OF MATERIALS AND THERMAL CONTROL ELEMENTS

Temperature- and field-dependent properties:
 $\rho(T, \text{field})$, $c_p(T, \text{field})$, $k(T, \text{field})$

GRAPHICAL USER INTERFACE



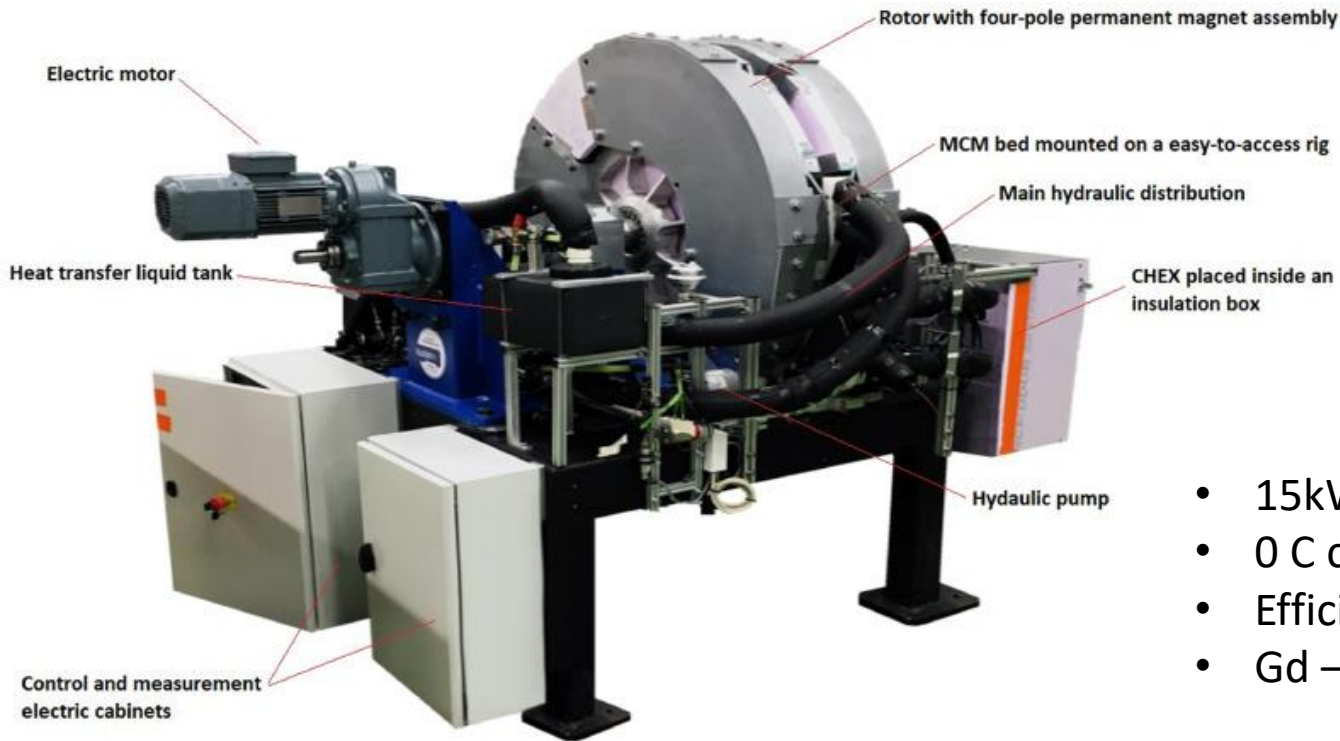
Online workshop
on TCCbuilder usage

Thursday, September 5th 2024
2pm CEST

(Registration is required, but free of charge)

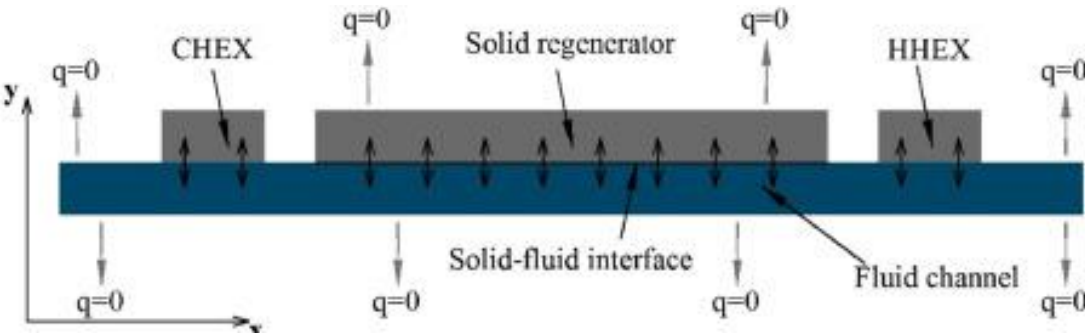
Registration by 1st September

Applications – Ubibblue (now Magnoric)



- 15kW
- 0 C cold temperature
- Efficiency 60% Carnot COP
- Gd – Gd-Er segments to tailor T_c

Int. J. Refrig. **122**, 256-265 (2021)



Applications - MagnoTherm



ESM 2021, Oliver Gutfleisch
Magneto (and multi-)caloric materials for efficient refrigeration



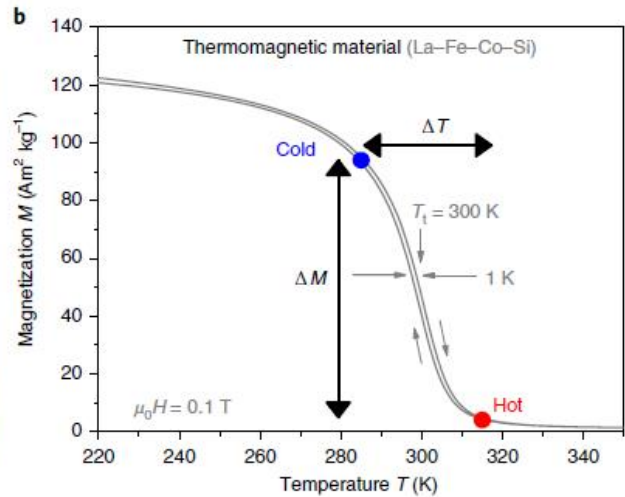
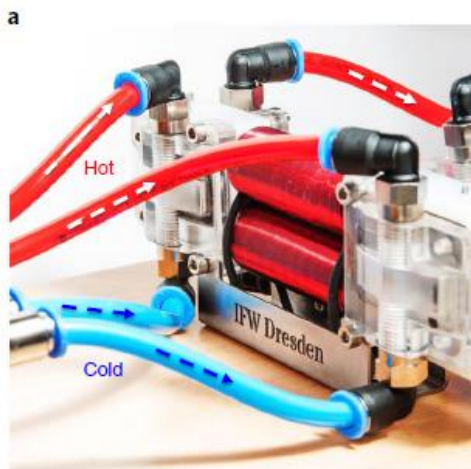
Polaris: 80 L beverage cooler

Double door: 1000L capacity
commercial refrigerators



Manual (handcrank) fridge

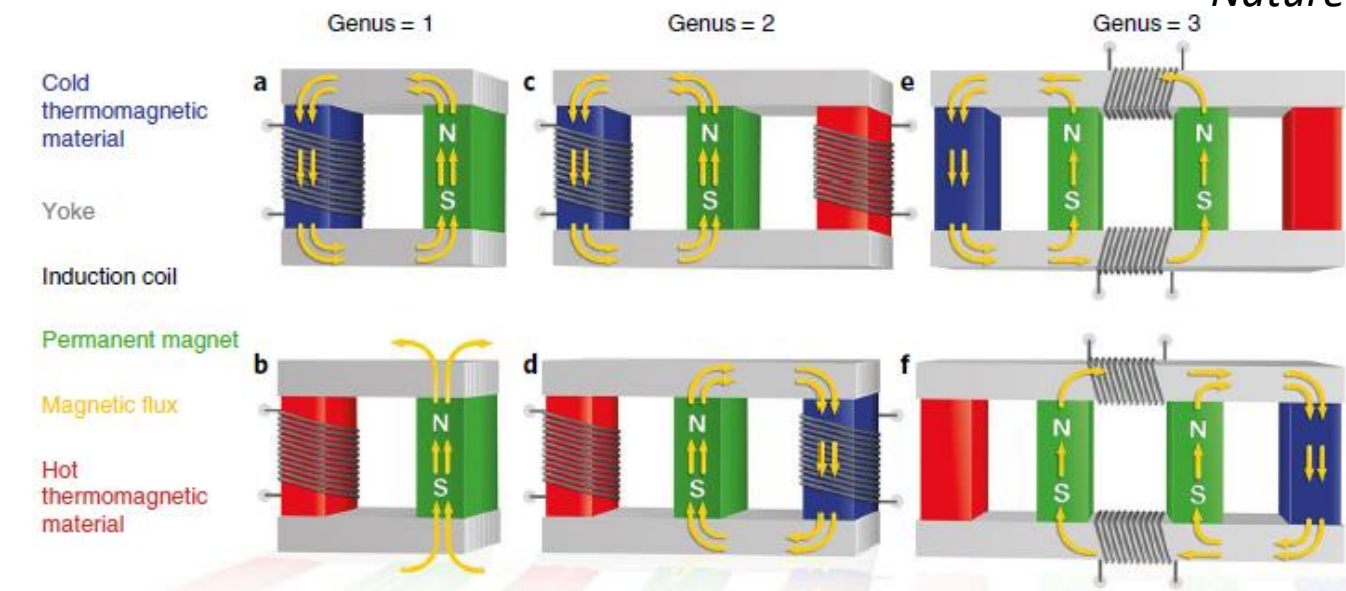
Applications: Energy Harvesting



Low grade heat
 Optimum frequency 1 Hz
 Efficiency $1.7 \times 10^{-3} \%$ of η_{Carnot}

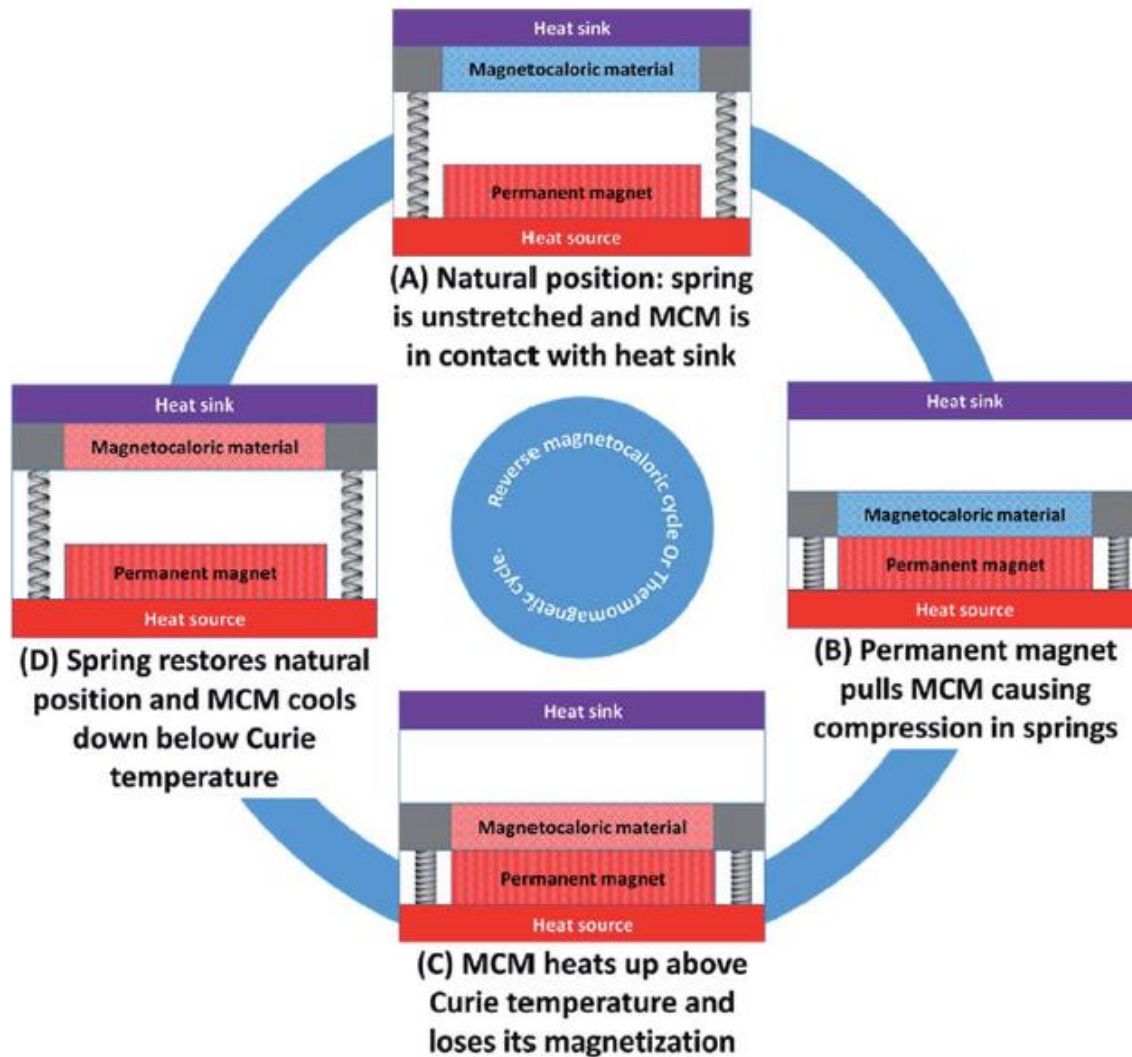
A. Kitanovski, *Advanced Energy Materials* **10**, 1903741 (2020)

A. Waske *et al.*, *Nature Energy* **4**, 68-74 (2019)



HEAT4ENERGY
 EU project
 launched in
2024

Applications: Energy Harvesting

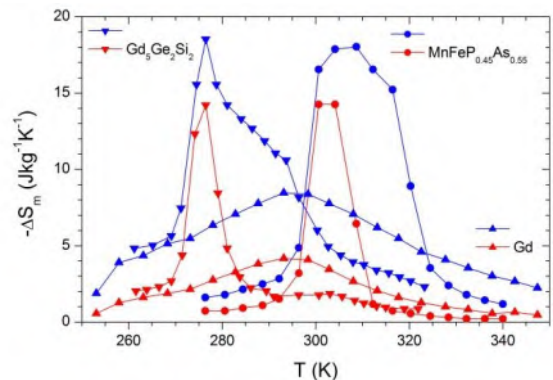


M. Ujihara, G.P. Carman, and D.G. Lee, *APL* **91**, 093508 (2007)

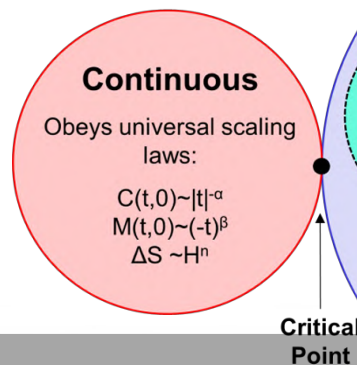
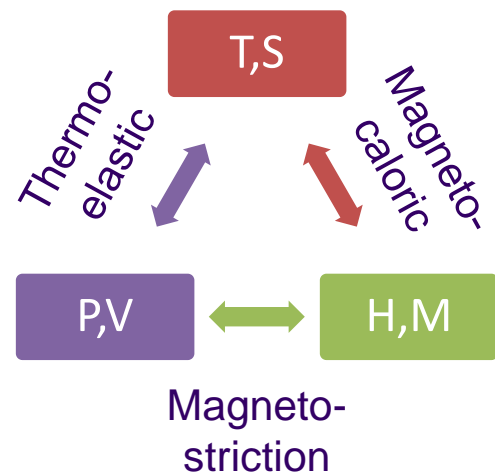
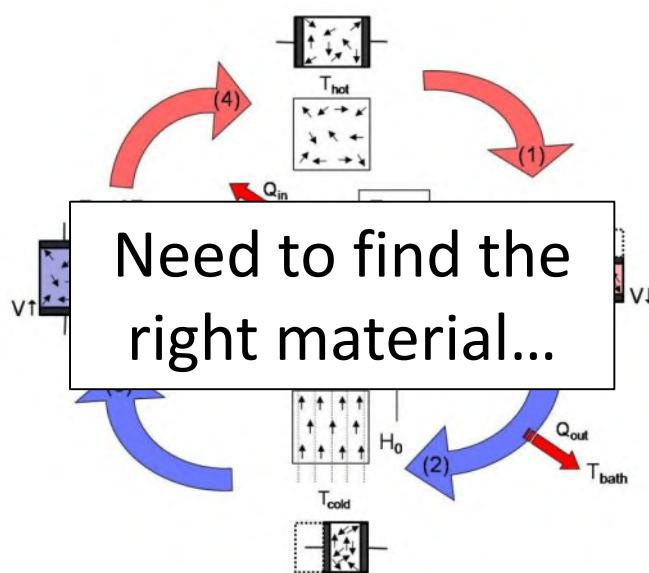
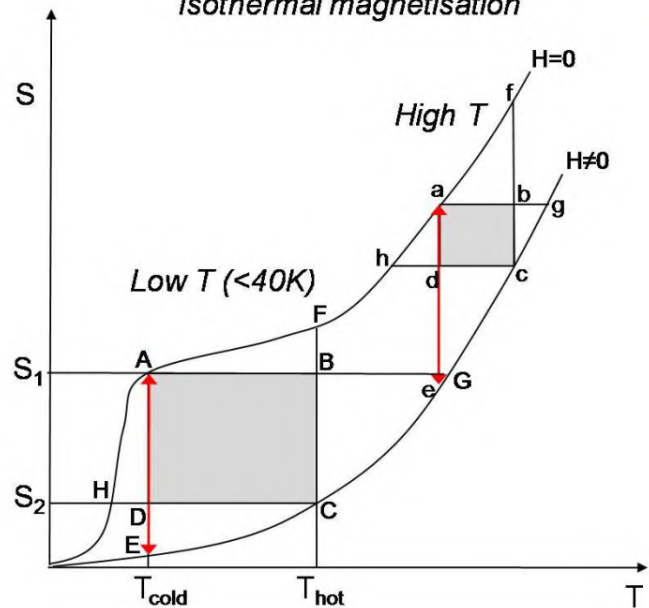
- 1.85-3.61 mW/cm² for $\Delta T=50K$.
- Predicted as high as 36.1 mW/cm² if thermal contacts improved

R. A. Kishore and S. Priya, *Sustainable Energy and Fuels*, **1**, 1899 (2017)

Magnetocaloric Effect (Magnetic Refrigeration / Energy Harvesting)



Isothermal magnetisation



Any Questions?

Imperial College

A. Berenov, J. Turcaud,
M. Bratko, K.G. Sandeman,
Z. Gercsi, J. Lyubina,
E. Lovell, D.A. Caplin,
L.F. Cohen

Ames Laboratory

V.K. Pecharsky,
K.A. Gschneidner,
Y. Mudryk

Vacuumschmelze

A. Barcza

IFW Dresden

J.D. Moore

TU Darmstadt

O. Gutfleisch, K. P. Skokov

University of Parma

G. Porcari

Loughborough University

J. Betuoras, A. Caruana

 Engineering and Physical Sciences
Research Council

www.sseec.eu

 **SSEEC**

Solid State Energy Efficient Cooling

