

# Magnetic Imaging Techniques

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# Types of Magnetism

*Ferromagnetism*

*Helimagnetism*

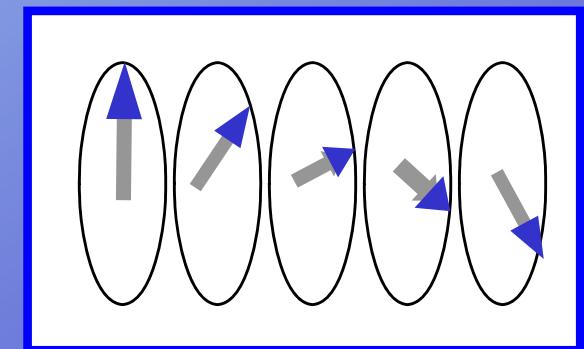
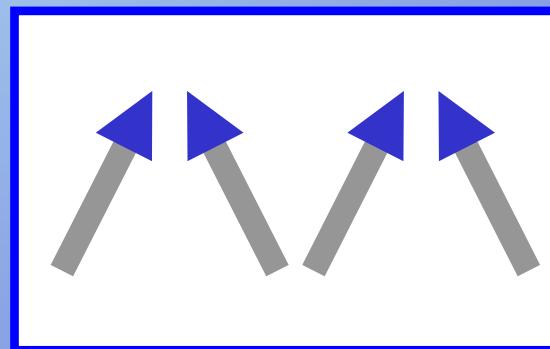
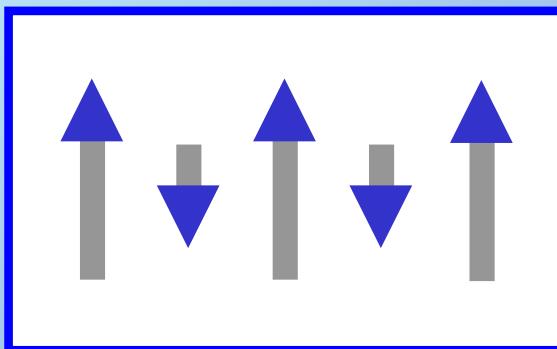
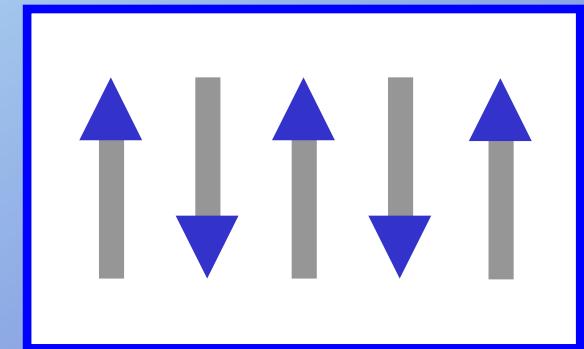
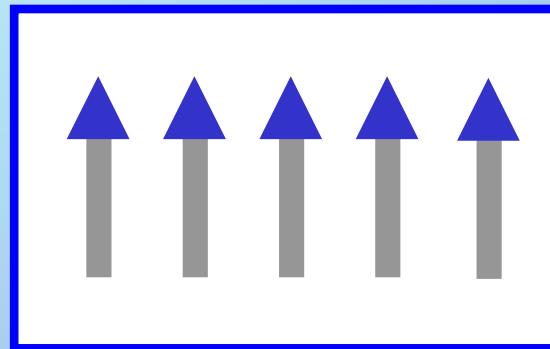
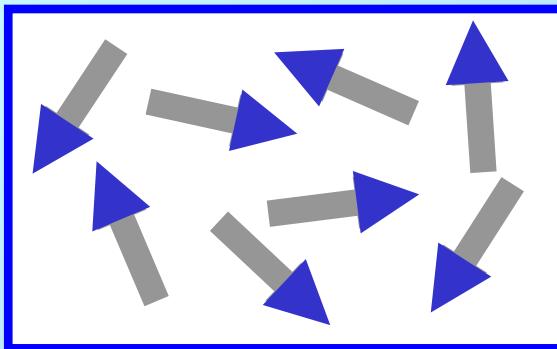
*Antiferromagnetism*

*Ferrimagnetism*

*Spin-Canted Magnetism*

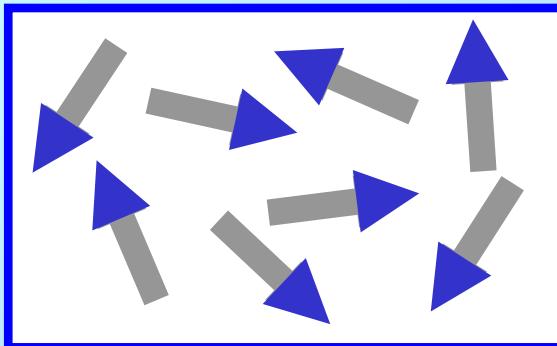
*Paramagnetism*

# Types of Magnetism

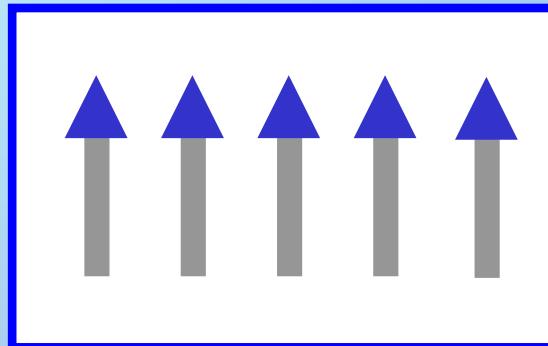


# Types of Magnetism

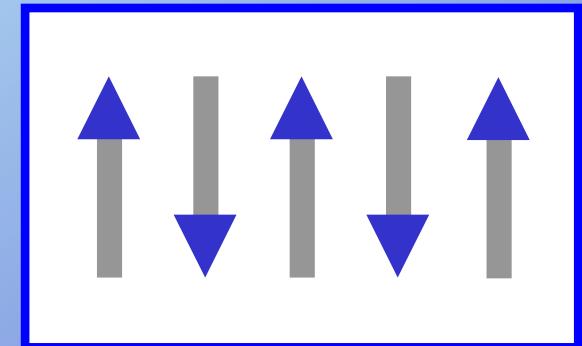
*Paramagnetism*



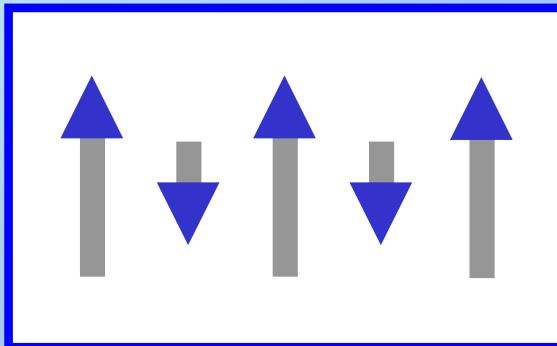
*Ferromagnetism*



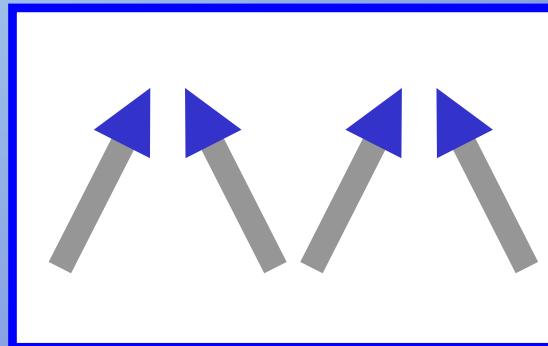
*Antiferromagnetism*



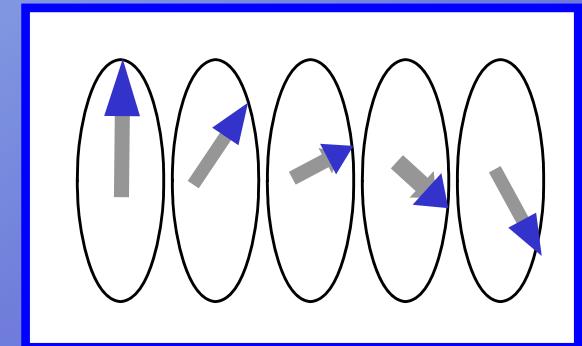
*Ferrimagnetism*



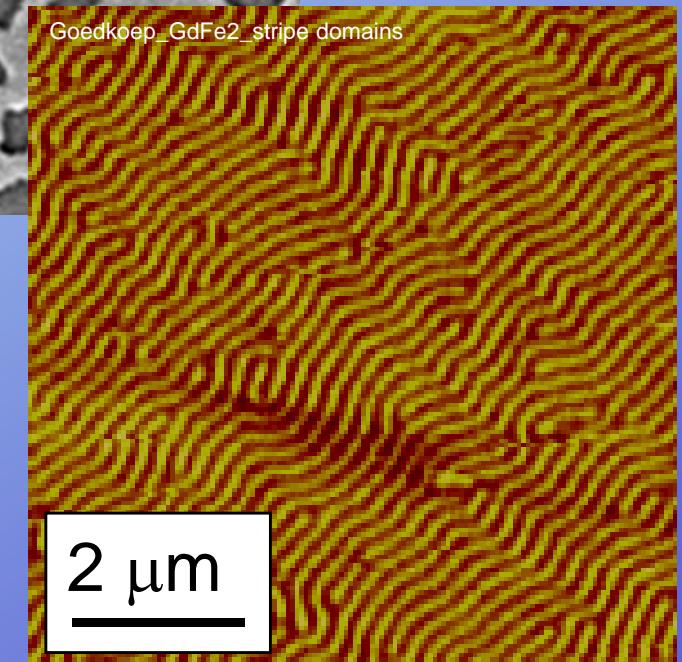
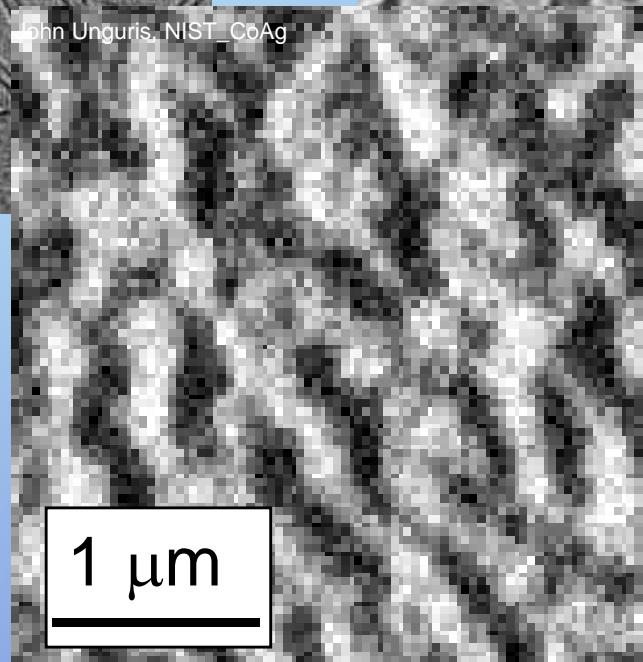
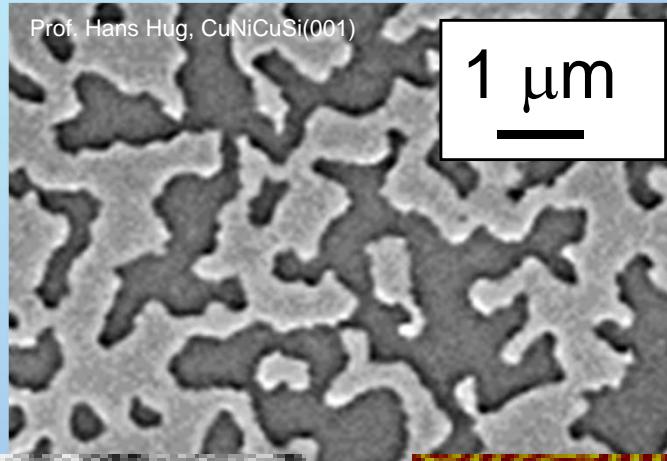
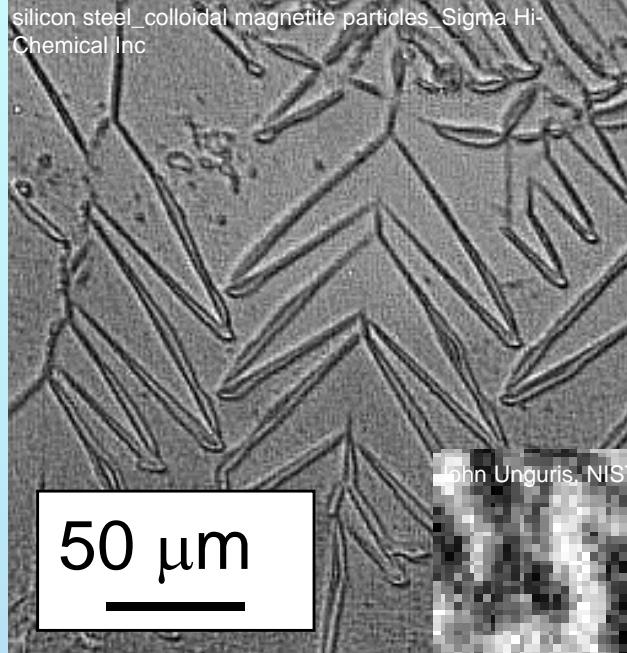
*Spin-Canted*



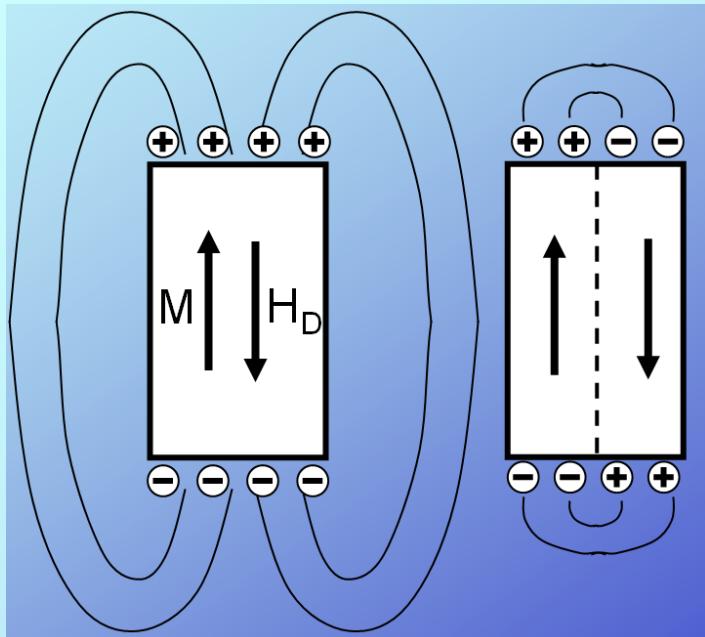
*Helimagnetism*



# Magnetic Domains ?



# Imaging Techniques

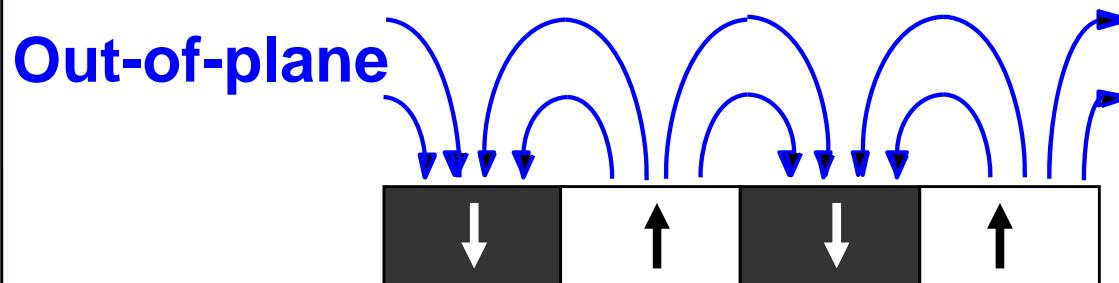


- Magnetic flux density (or induction):  

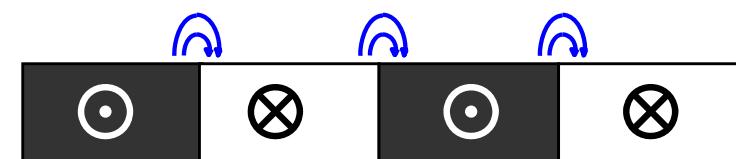
$$\mathbf{B} = \mu_0 \mathbf{H} + \mathbf{J}$$

$$\mathbf{H} = \text{magnetic field (external + stray field)}$$

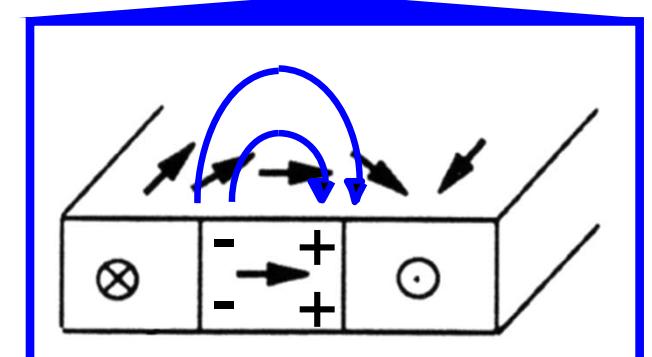
$$\mathbf{J} = \text{magnetic polarization (magnetization } \mu_0 \mathbf{M} \\ = \mathbf{q}/\mathbf{V} \text{ magnetic moment / volume)}$$
- Maxwell's equation  $\text{div} \mathbf{B} = 0$ , therefore:  
 $\text{div} \mathbf{H} = -\text{div} \mathbf{M} \rightarrow$  any divergence of  $\mathbf{M}$  creates a stray field  $\mathbf{H}$  (zero external field)



**In-plane**



*Imaging techniques provide information on stray field, magnetization (magnitude and direction), magnetic induction (includes stray field and magnetization) at material surface or through a thin film*

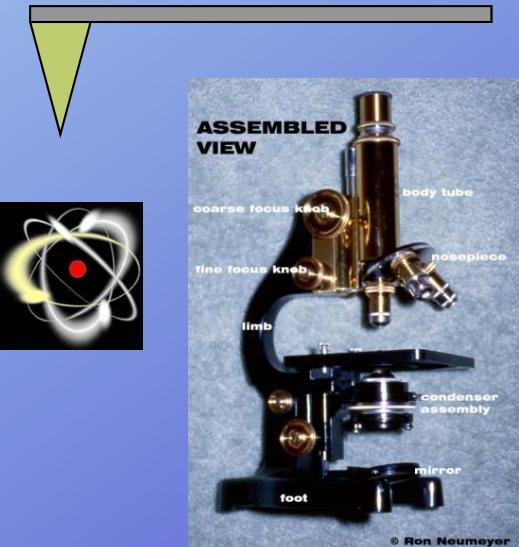


# Imaging Techniques

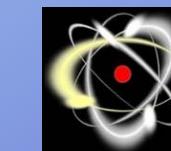
Bitter Technique



Scanning Probe Microscopy



Electron Microscopy



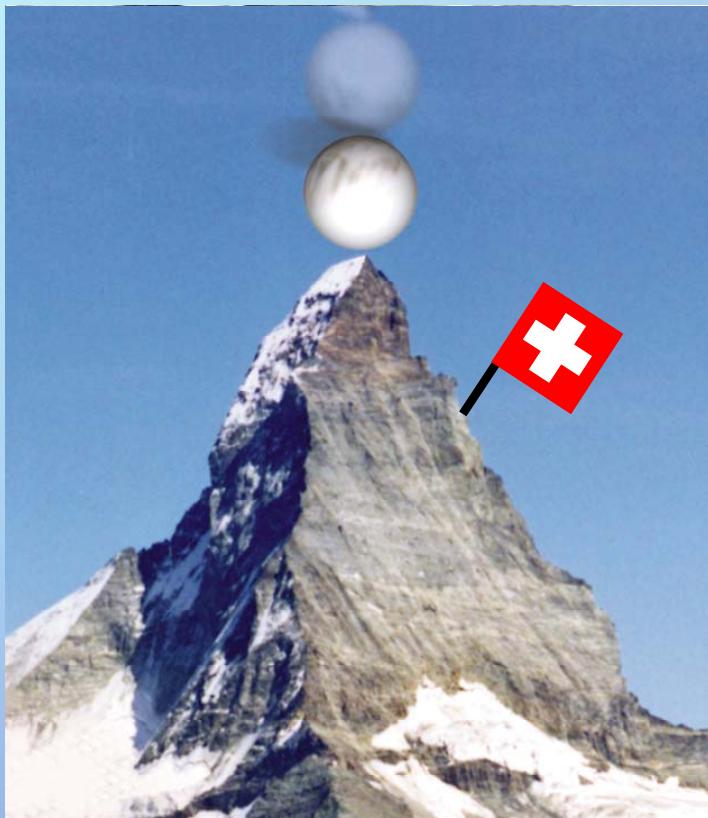
Kerr Microscopy



X-Ray Microscopy Techniques

# Scanning Probe Microscopy

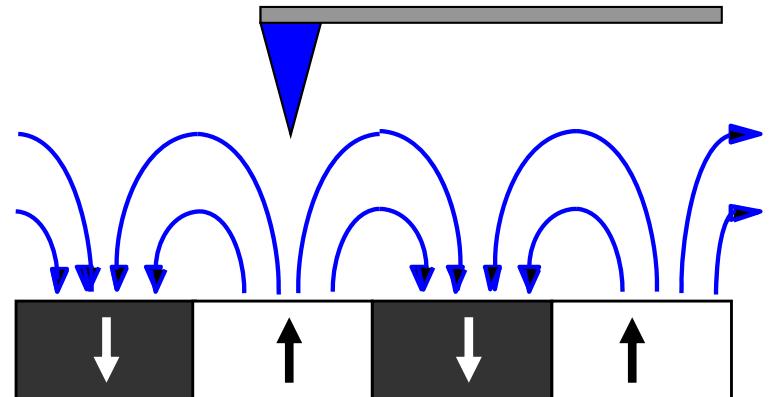
# Scanning Probe Microscopy



*If an atom was as  
large as a ping-pong  
ball...*

*...the tip would have  
the size of the  
Matterhorn!*

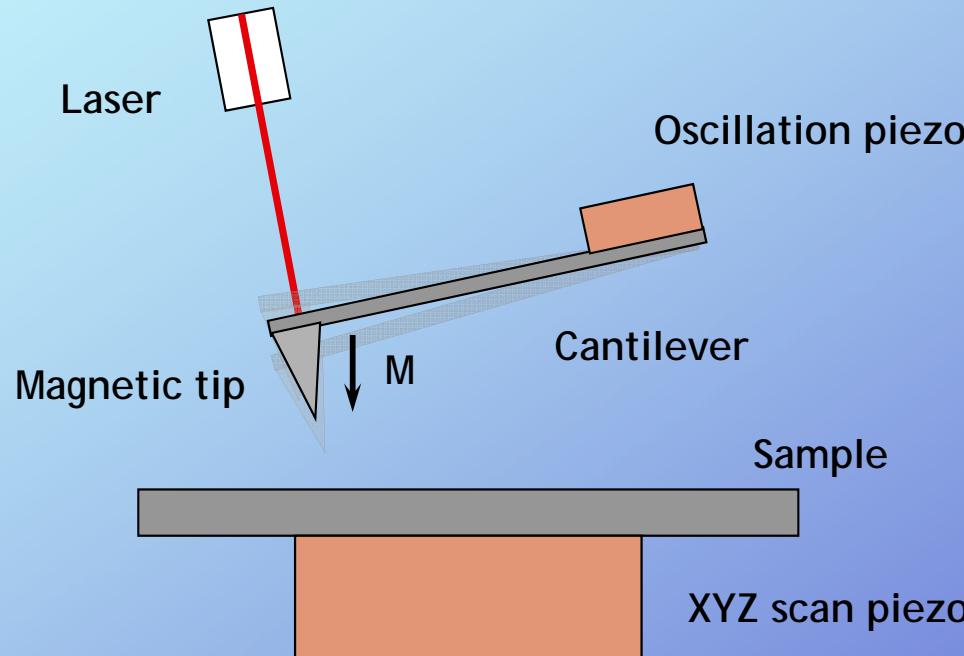
# Magnetic Force Microscopy



- Stray field interaction between film and magnetic tip
- Forces are on the order of  $10^{-10}$  N
- Employ a cantilever (CL) "spring"
- The force sensing carried out in two ways:
  - Static force sensing: CL brought near to surface and bends down or up (interaction attractive or repulsive). But CL might "snap" onto the surface.
  - Dynamic force sensing: CL is oscillated at a certain frequency, typically at its resonance frequency or a bit off (5%)

# Non Contact Dynamic MFM

Schematic drawing:



- The cantilever is oscillated with a fixed amplitude (nm).
- The sample is scanned by the XYZ scan piezo.
- The measurement signal is the *frequency shift* (shift of cantilever's resonance frequency), measured with a quadrant diode detector or laser interferometer.

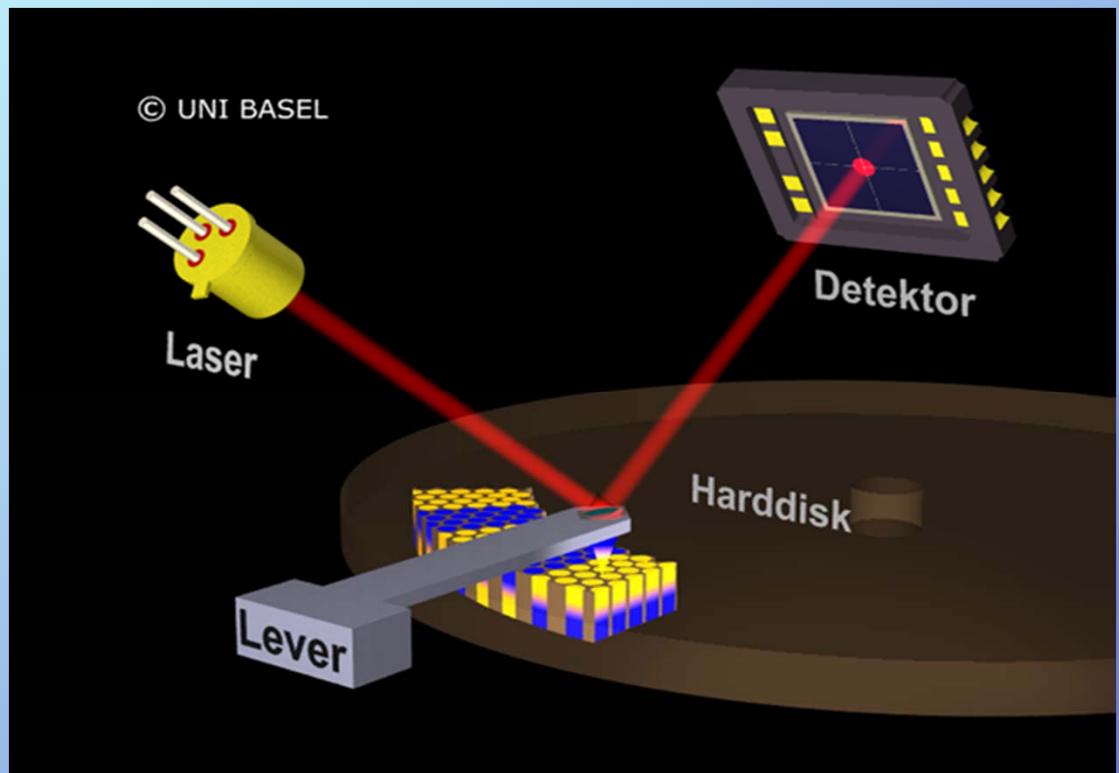
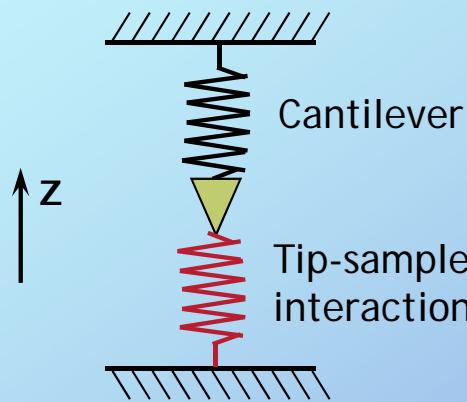
# SFM Dynamic Operation Modes

Measurement Signal:

Massless spring system →  
harmonic oscillator

Freq.  $f_0 = \frac{1}{2\pi} \sqrt{\frac{c_L}{m}}$

spring constant  
mass



Linear part of Taylor Expansion

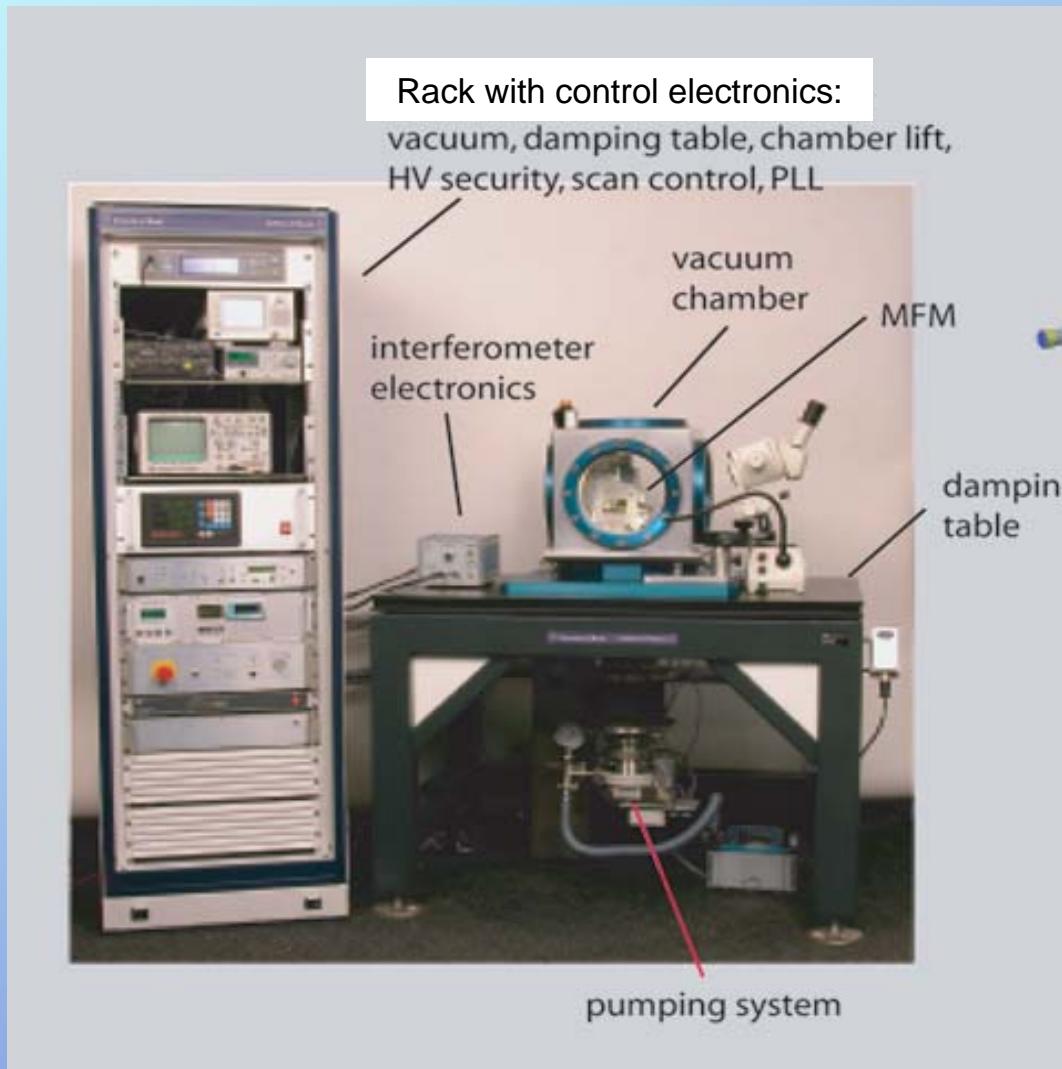
$$f_P = \frac{1}{2\pi} \sqrt{\frac{1}{m} \left( c_L - \frac{\partial F}{\partial z} \right)}$$

$$\delta f \equiv f_P - f_0 \approx -\frac{f_0}{2c_L} \frac{\partial F}{\partial z}$$

$\frac{\partial F}{\partial z}$  interaction force  
gradient

- Attractive forces → lower resonance frequency → -ve freq. shift
- Repulsive forces → higher resonance frequency → +ve freq. shift

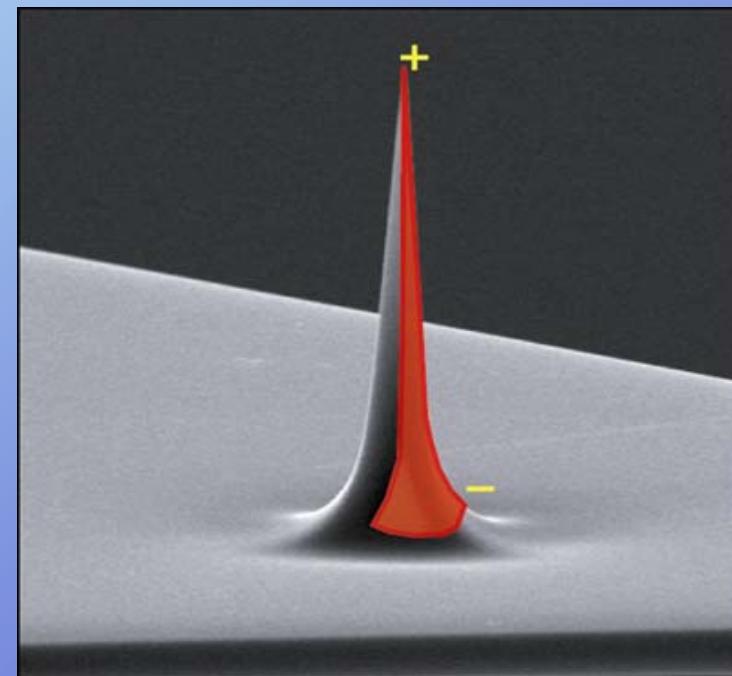
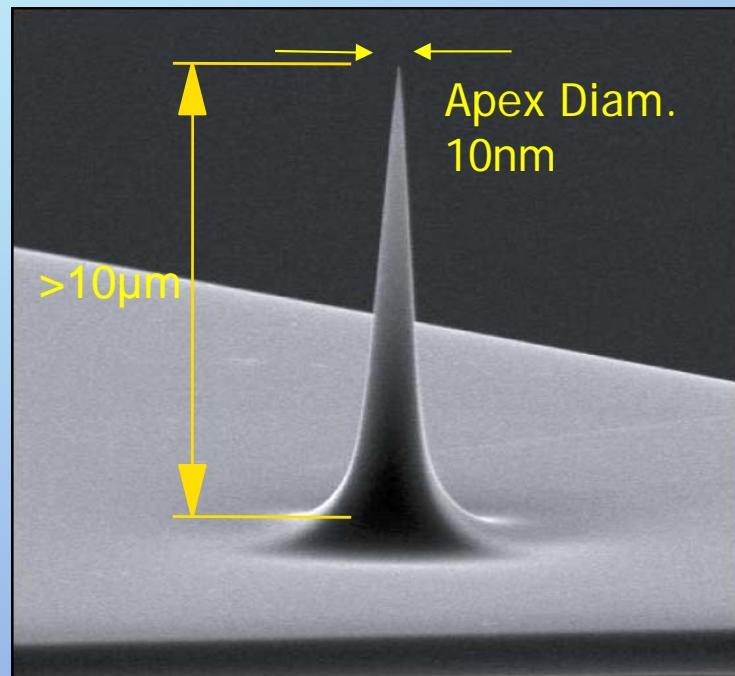
# The High Resolution MFM



Peter Kappenberger, Hans Hug

# High Resolution MFM: Requirements

- High aspect ratio tip with small apex diameter, small cone angle.
- Ultra-thin & smooth ferromagnetic coating (3-6nm).
- High measurement sensitivity limited only by thermal noise of cantilever due to gas molecules hitting cantilever. Therefore need to go to vacuum (results in a large Q, i.e. a low damping).



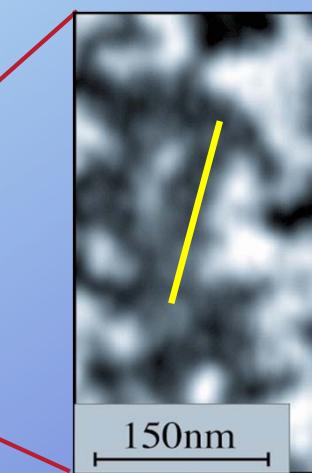
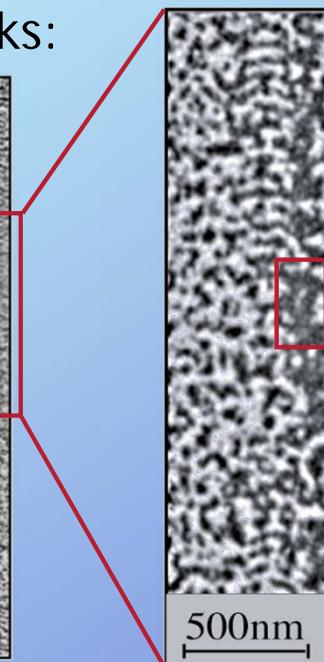
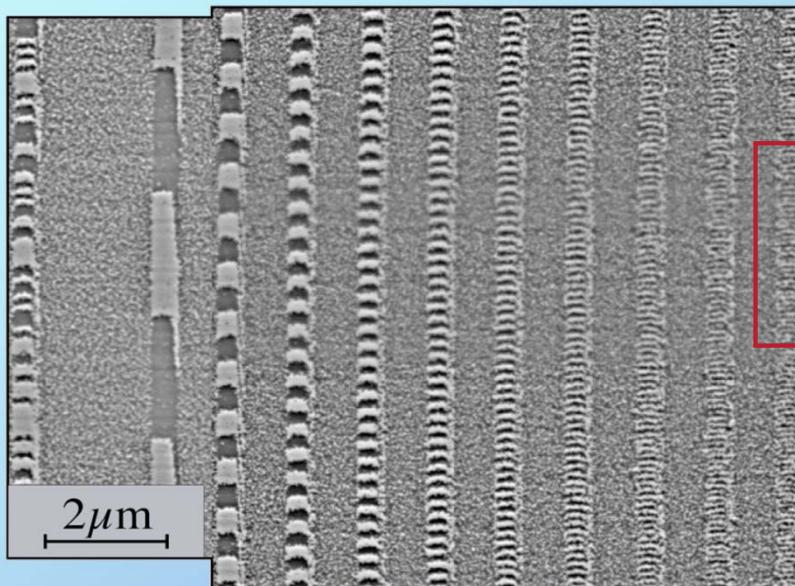
T.R. Abrecht, J. Appl. Phys. 69(2), p668 (1994)

Peter Kappenberger, Hans Hug

# High Resolution MFM: Example

High resolution MFM images of hard disk media (sample by Seagate Research):

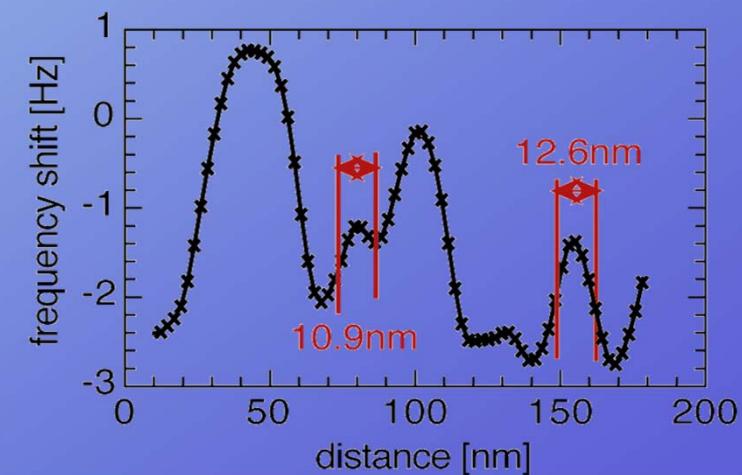
Bits organised in circumferential tracks:

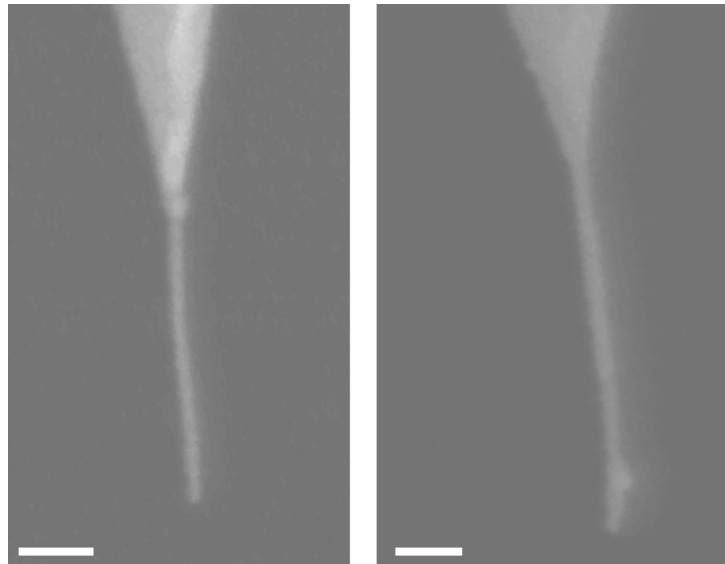


Increasing density / decreasing bit size

hr-MFM images important in HD research:

- detect defects in the bits
- measure media properties



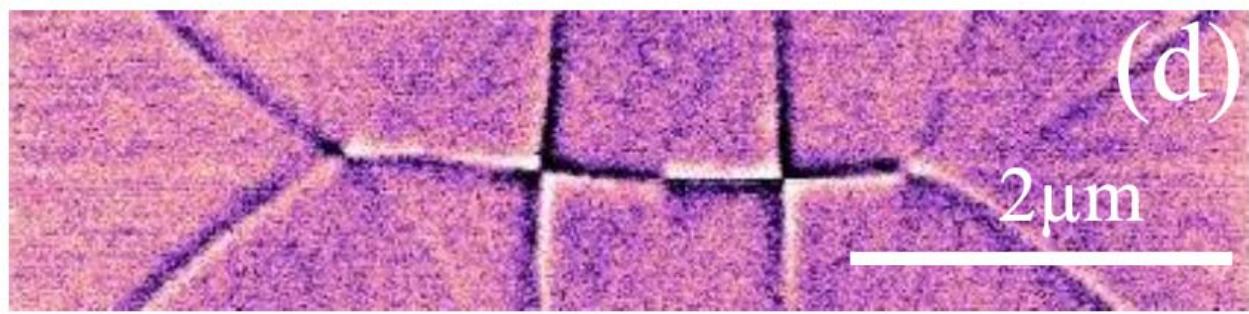
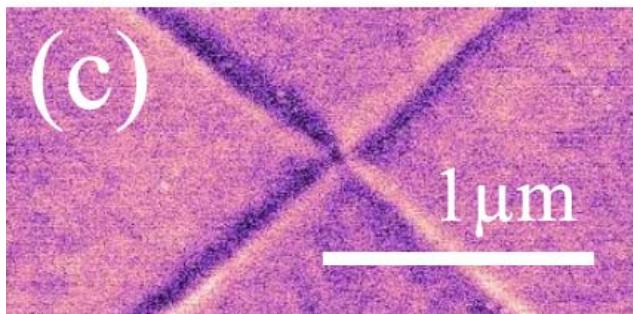
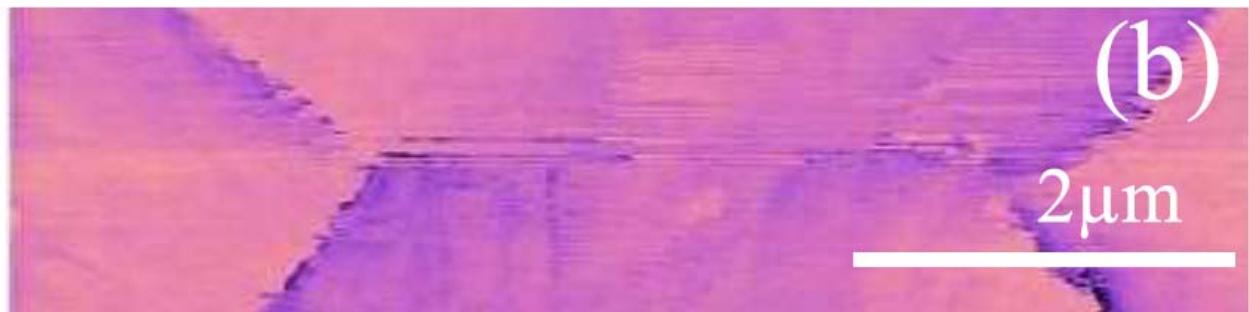
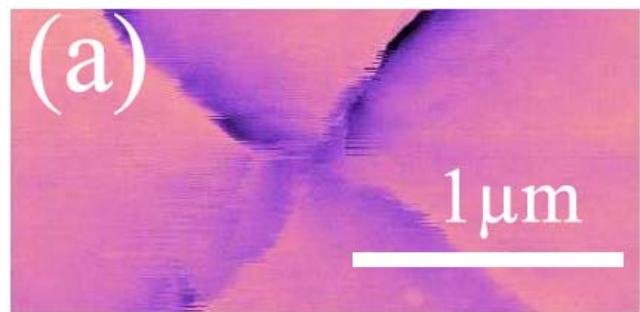


Y. Lisunova, J. Heidler, I. Levkivskyi,  
I. Gaponenko, A. Weber, Ch. Caillier,  
L.J. Heyderman, M. Kläui and P. Paruch  
Nanotechnology Accepted (2013)

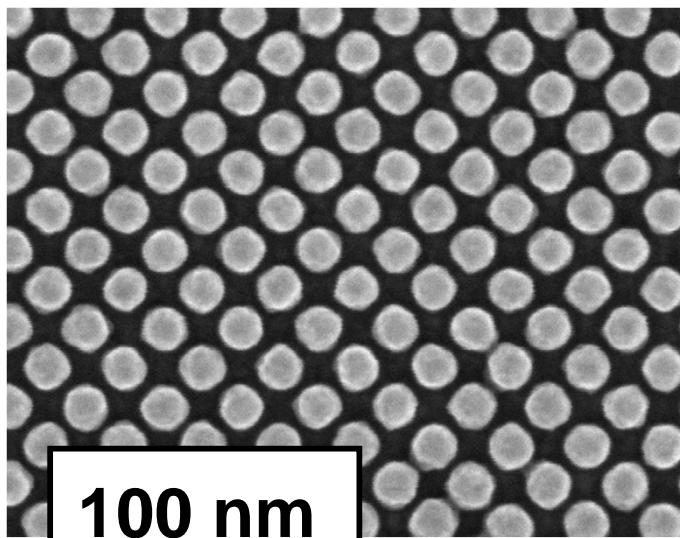
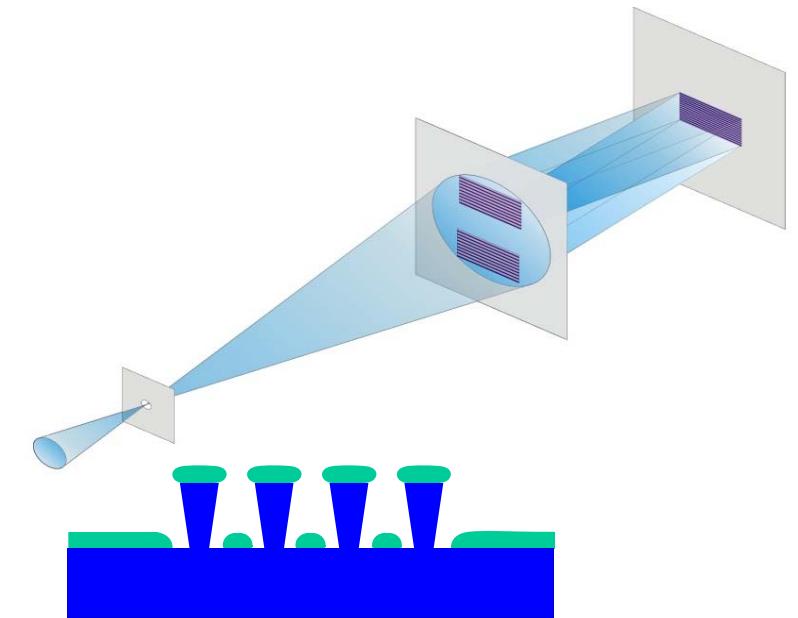
**High resolution with no vacuum!**

Scale Bar: 100 nm

Commercial  
CNT



# Magnetic Islands with EUV-IL



SWISS LIGHT SOURCE  
SLS

EULITHA

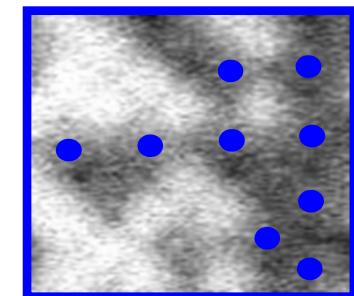
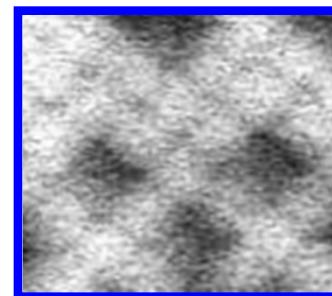
## Co/Pd Multilayer on $\text{SiO}_x$ Pillars

Period = 50 nm  $\rightarrow$  263 Gbit/in<sup>2</sup>

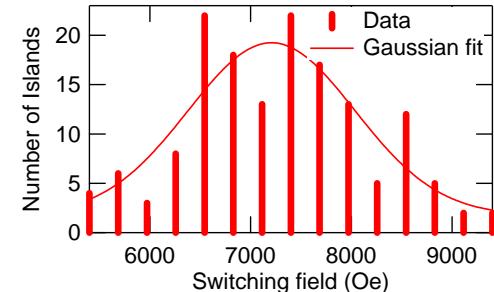
Diameter = 28.4 nm,  $\sigma = 5\%$

MFM  
measurements  
Switched Islands:

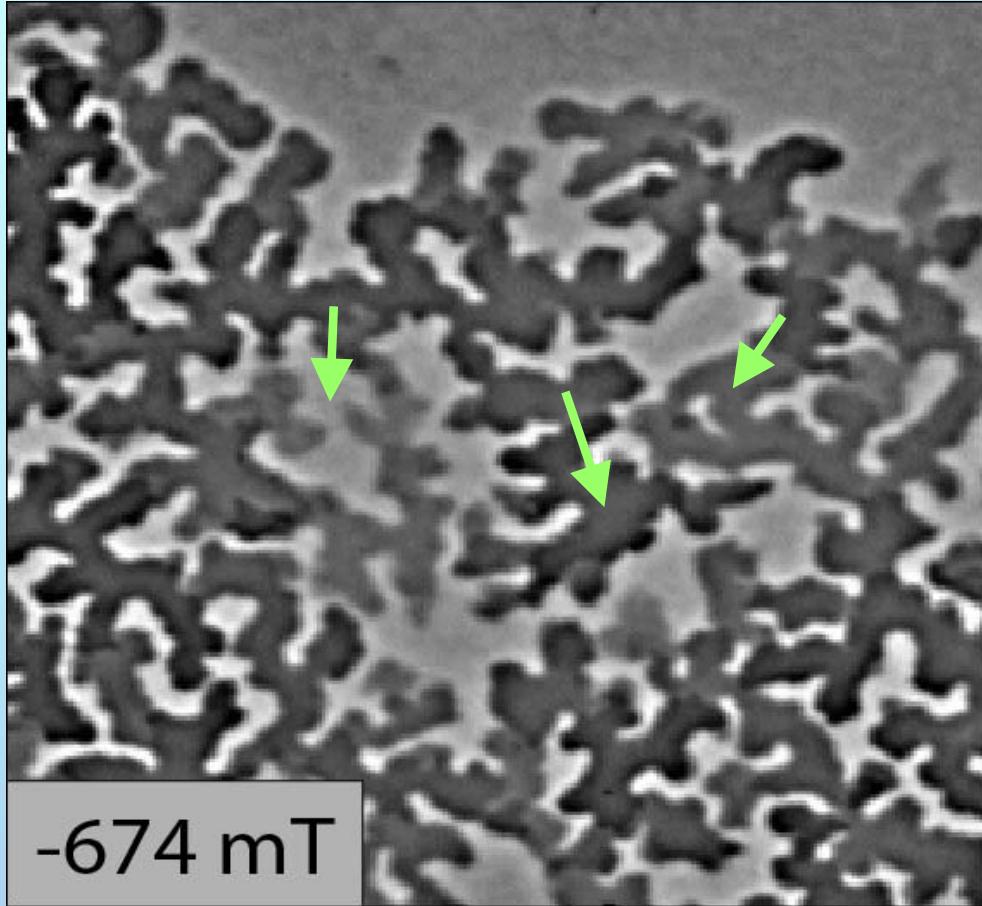
H = -6.5 kOe      -7 kOe



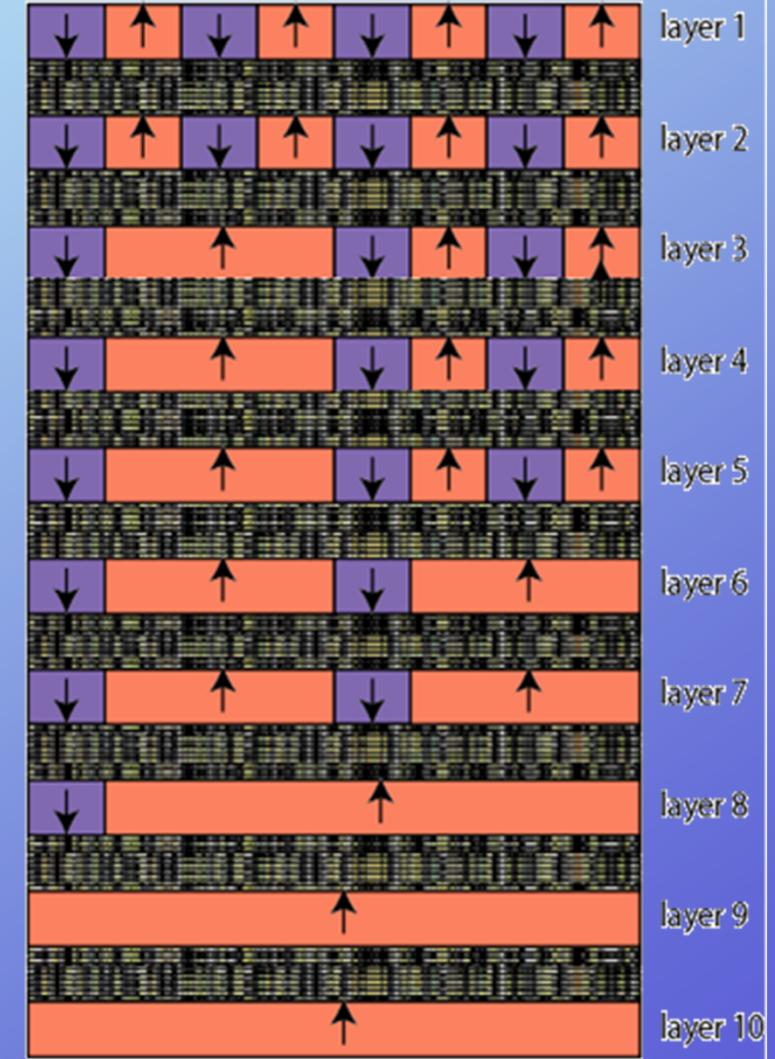
Mean switching field: 7200 Oe  
SFD ( $\sigma/\text{mean}$ ) = 11.5 %



# CoPt Multilayer in Hysteresis Loop

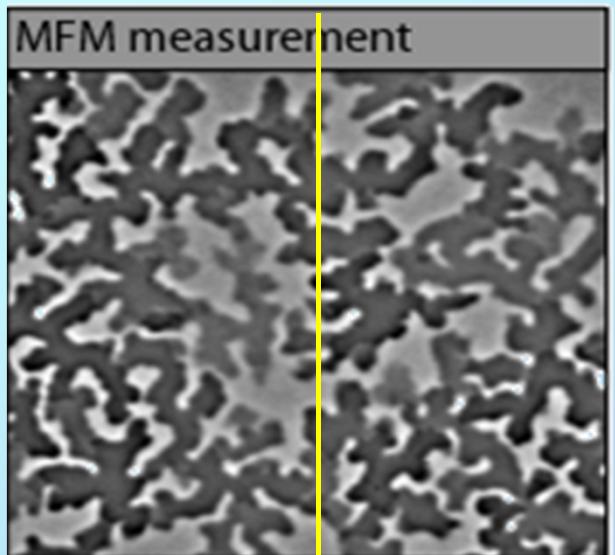


- Magnetization of layers from top to bottom.
- Which grey level corresponds to which layer?
- Take a calibrated tip and compare simulation with image.



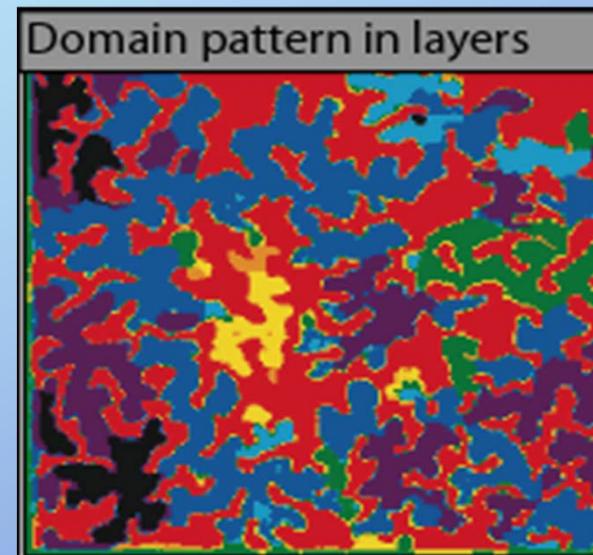
# Simulation of MFM Images

1

 $\Delta f$  [Hz]6.0  
0  
-6.0  
-12.0

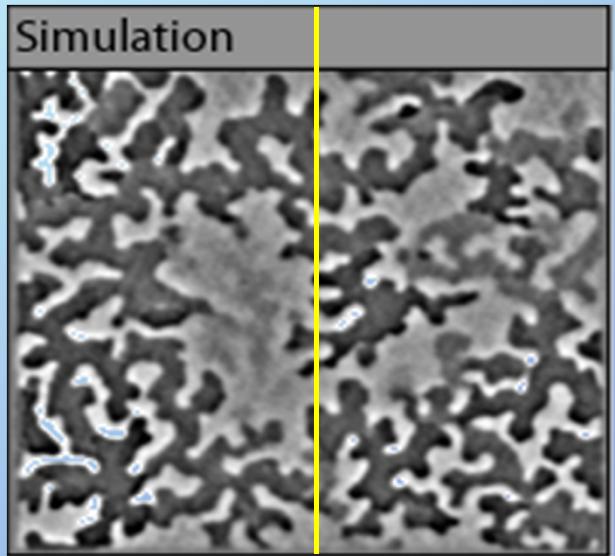
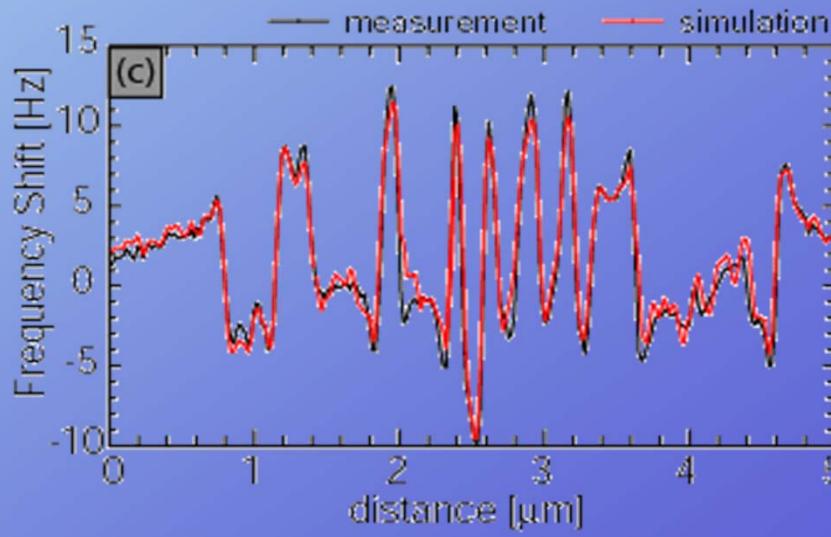
2

Guess



Colours:  
domains  
in different  
layers

3

 $\Delta f$  [Hz]6.0  
0  
-6.0  
-12.0

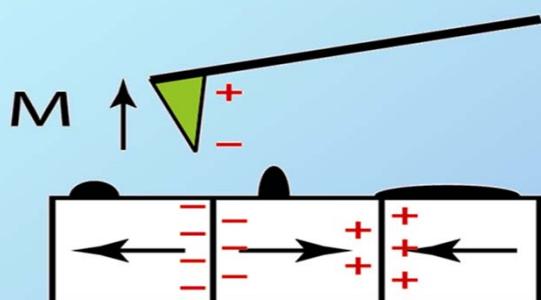
Peter Kappenberger, Hans Hug

## Topography Separation

Magnetic contrast inverts when rotating the tip magnetization by 180°.  
Topographic contrast remains (van der Waals is always attractive).

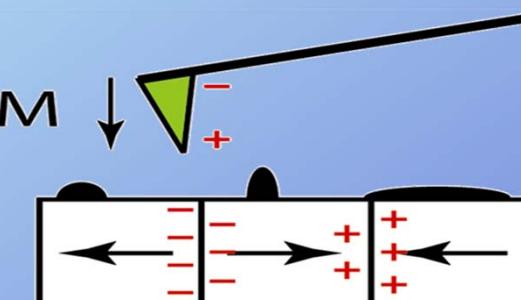
→ Acquire two images with opposite tip magnetization states.

A



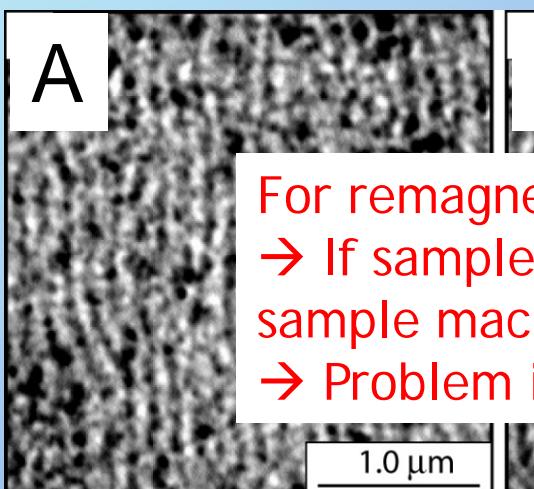
Tip magnetization up

B

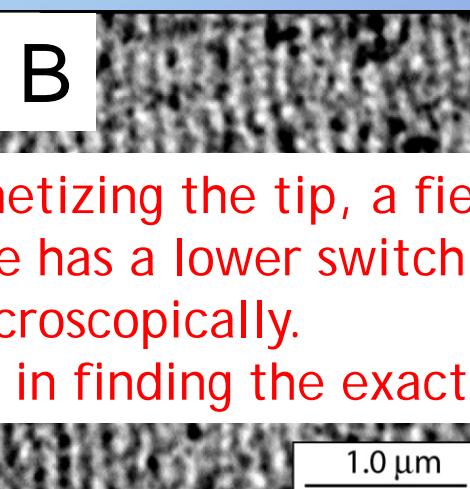


Tip magnetization down

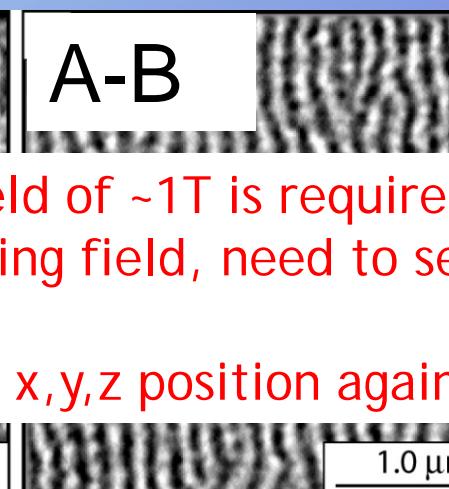
A



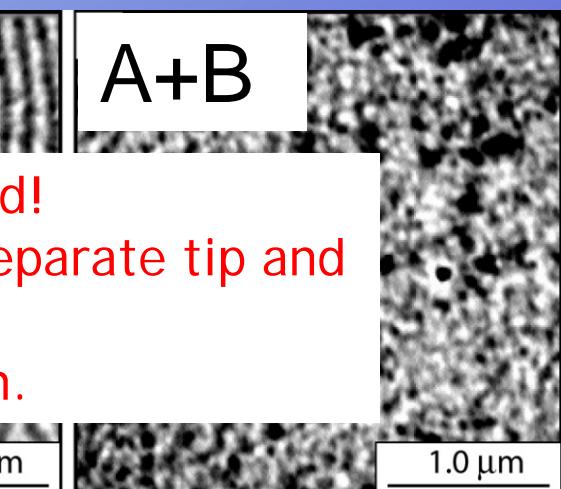
B



A-B



A+B



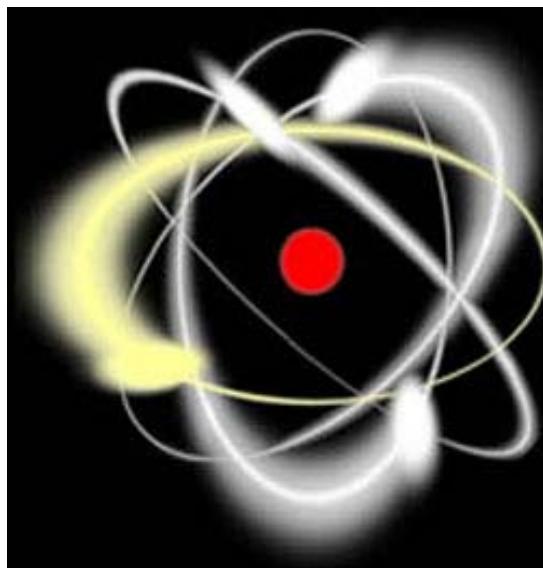
For remagnetizing the tip, a field of ~1T is required!

- If sample has a lower switching field, need to separate tip and sample macroscopically.
- Problem in finding the exact x,y,z position again.

## MFM Summary

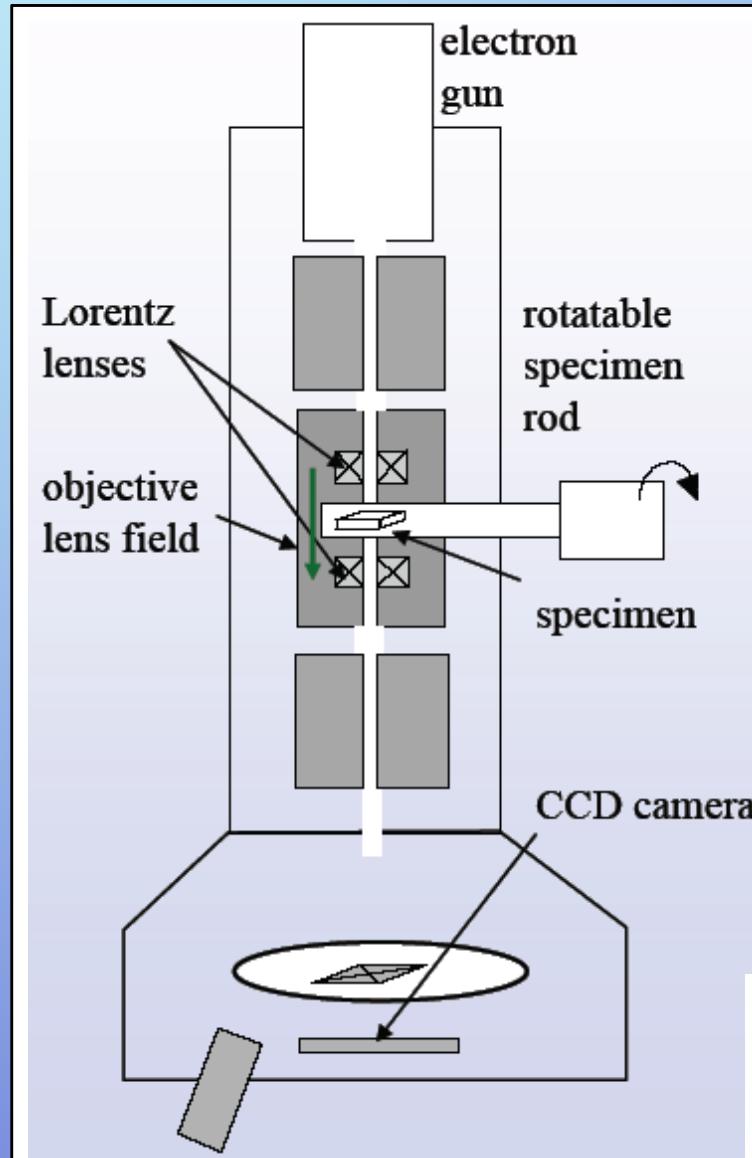
- Maps the magnetic stray field (or derivative)
- Resolution: typically 20-30 nm, high resolution: 10 nm
- Non-destructive
- Especially sensitive to z-component of stray field: ideal for perpendicular anisotropy materials (e.g.magnetic media) but can also image domain walls in in-plane samples
- Requires virtually no sample preparation
- Surface should be relatively flat
- Tip quality is critical
- Influence of tip: difficult to measure magnetically soft samples

# Electron Microscopy



# Transmission Electron Microscopy

# Transmission Electron Microscope



**Lenses**

**Condenser**  
illuminate

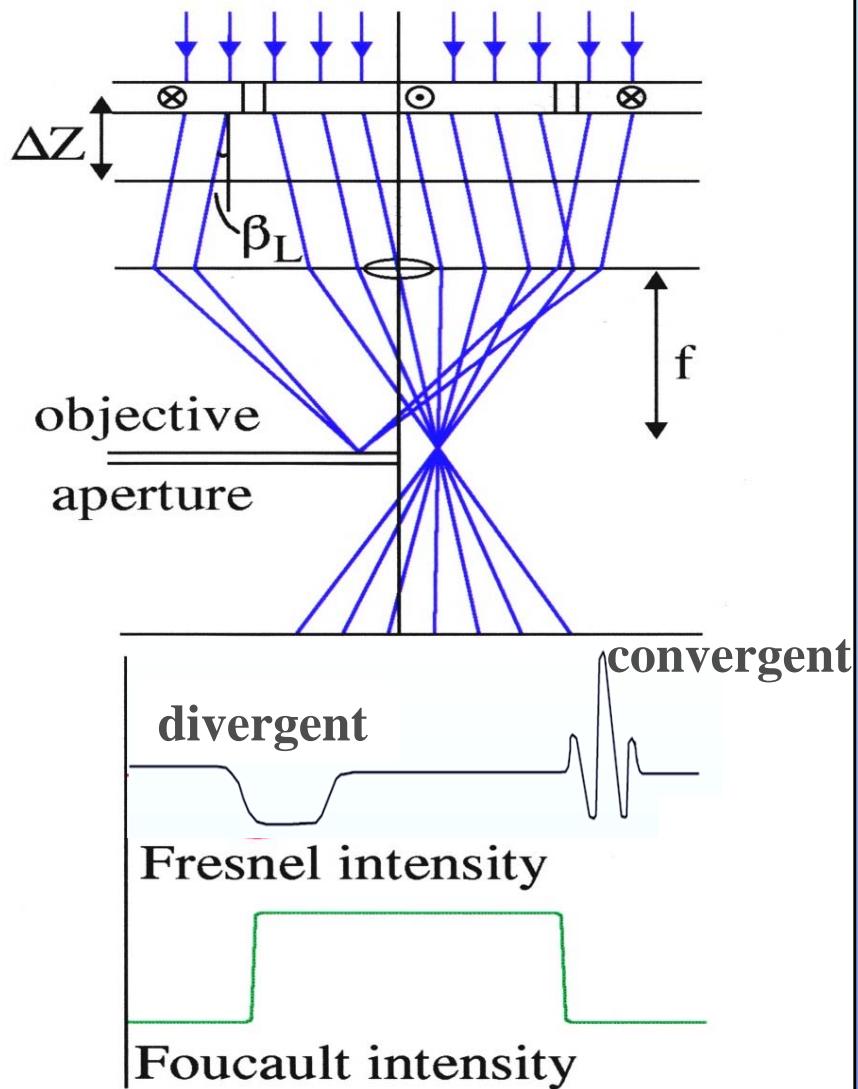
**Objective**  
focus

**Projector**  
image



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# Lorentz Microscopy Classical Theory



- Electron voltage 100 - 200 keV
- Particle representation, corresponding to classical beam optics, the electrons are deflected by the Lorentz force:
$$\mathbf{F} = q(\mathbf{v} \times \mathbf{B})$$

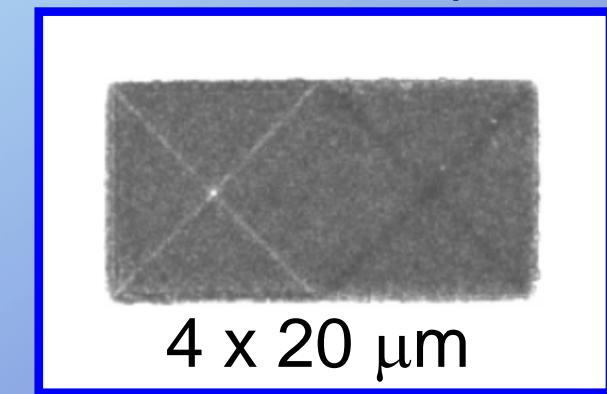
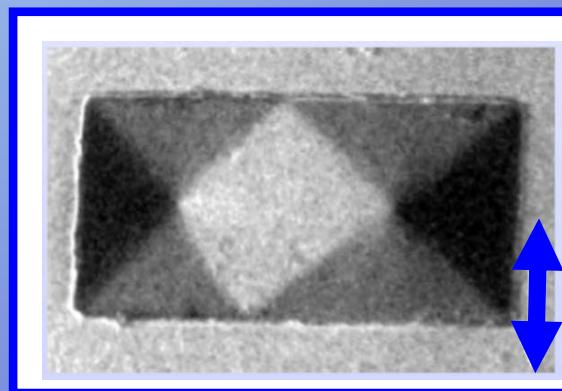
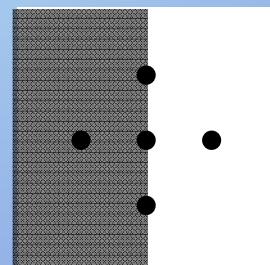
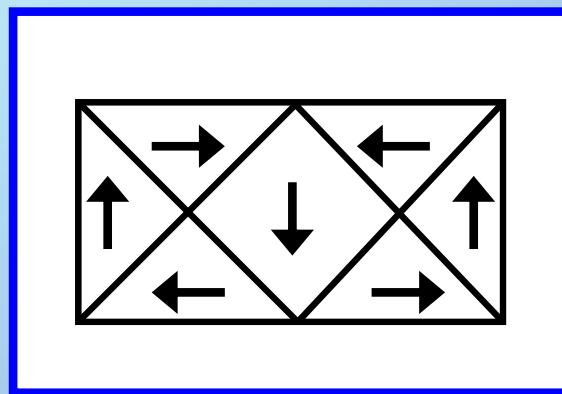
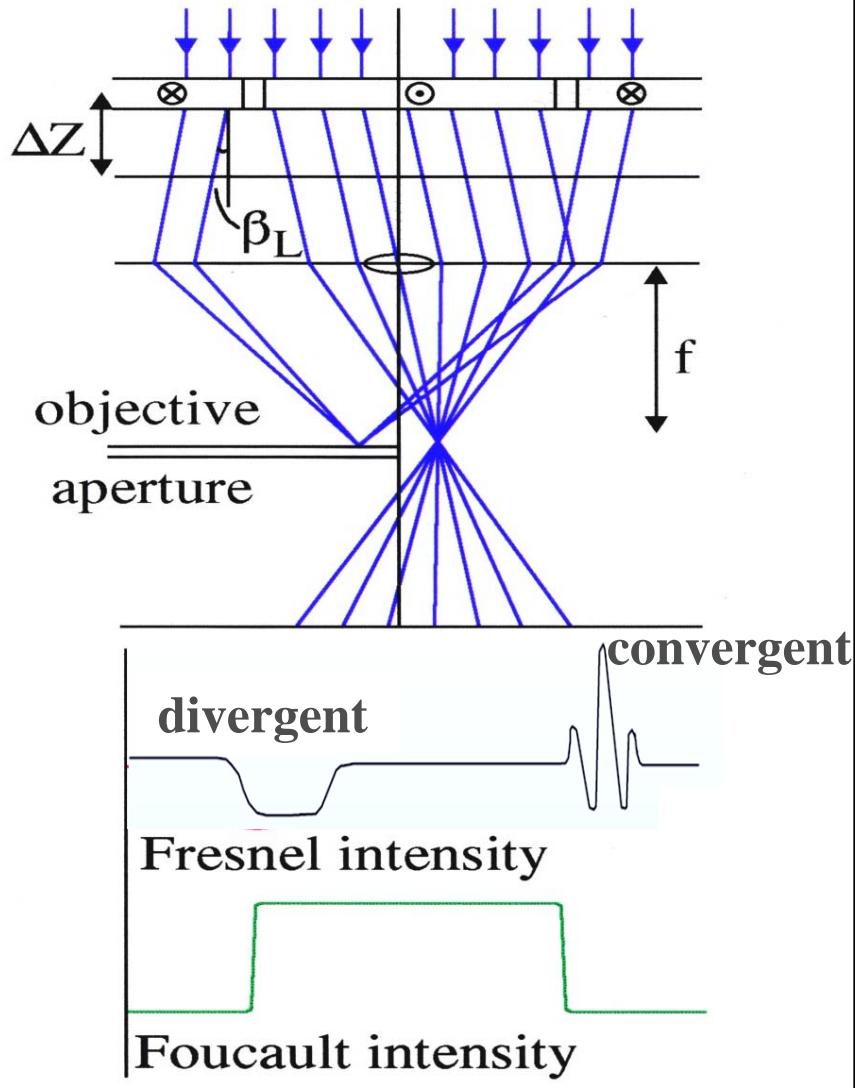
$q$  = electron charge  
 $v$  = electron velocity  
 $B$  = magnetic flux density
- With deflection angle:  $\beta = (e \lambda B t) / h$
- Only components of  $B$  perpendicular to the electron beam are effective
- Includes stray fields outside sample

# Fresnel and Foucault Microscopy

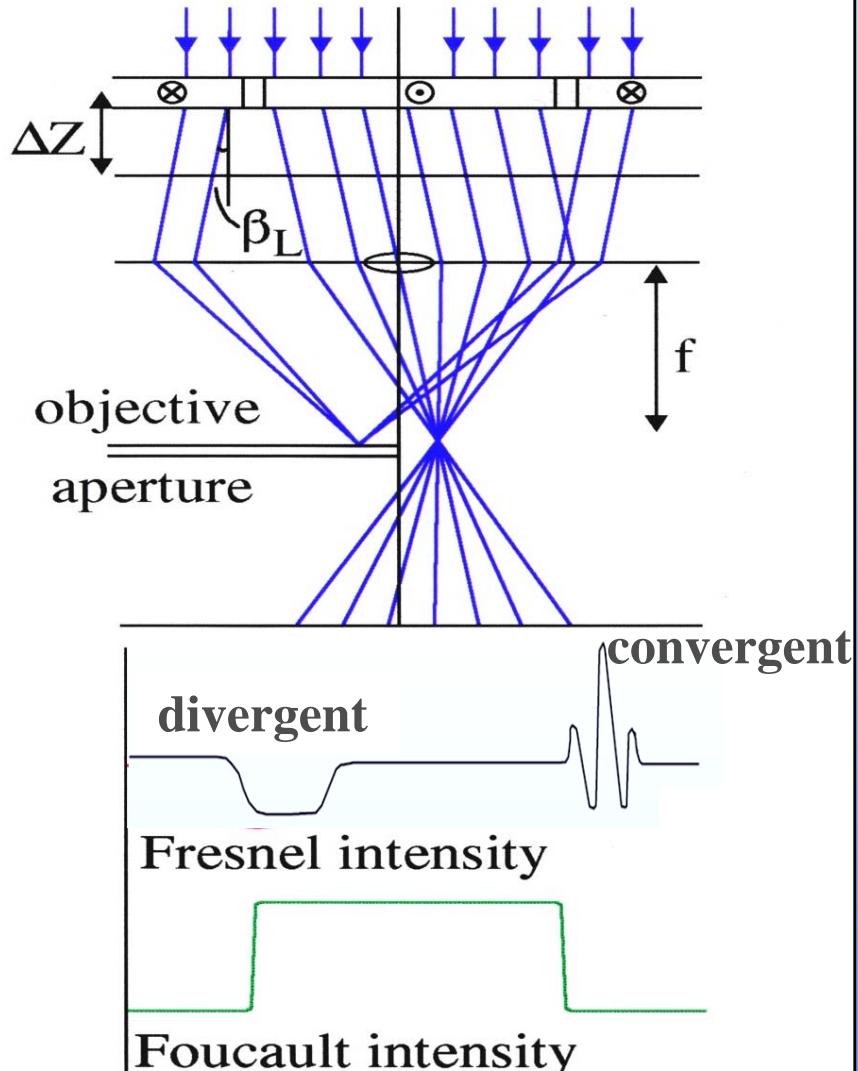


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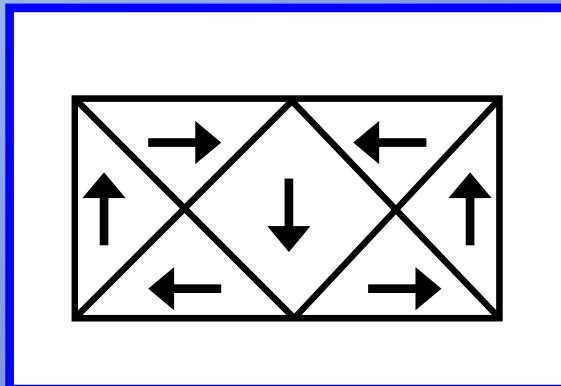
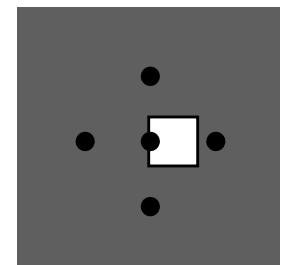
John Chapman



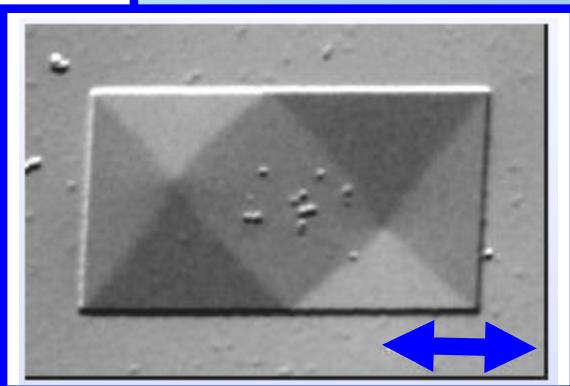
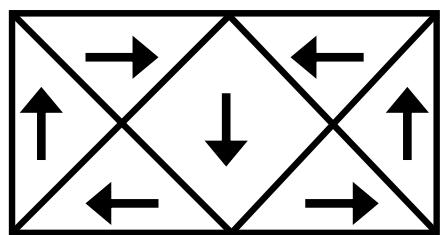
# Coherent Foucault



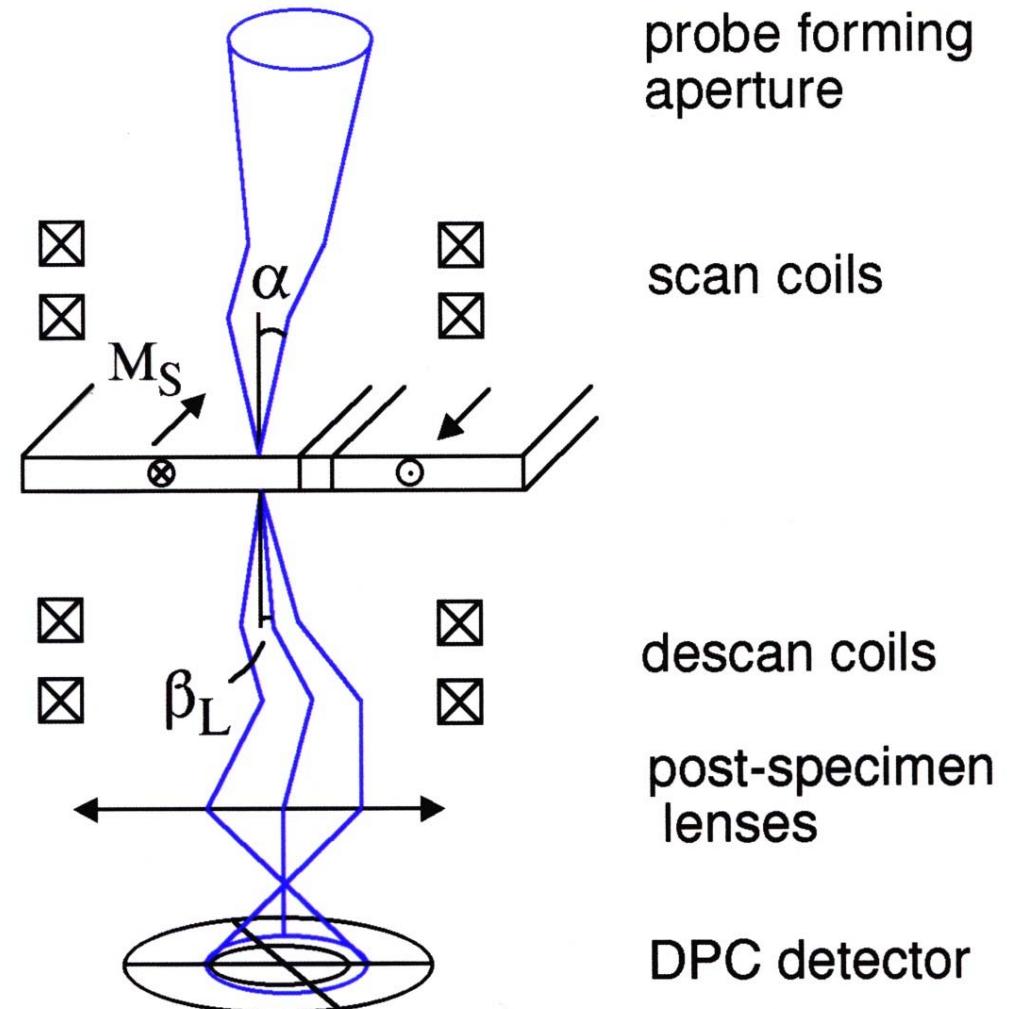
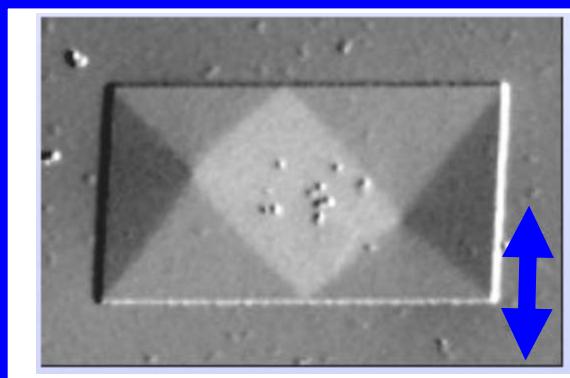
- Field Emission Gun source: small source angle
- Electron transparent phase shifting aperture: holes in SiN membrane, 50 nm thick, 200 keV, gives  $\pi$  phase shift of all magnetic spots
- Interference fringes parallel to induction direction with periodicity  $h/eBt$
- In-line Holography !



# Differential Phase Contrast



4 x 20  $\mu\text{m}$



probe forming  
aperture

scan coils

descan coils

post-specimen  
lenses

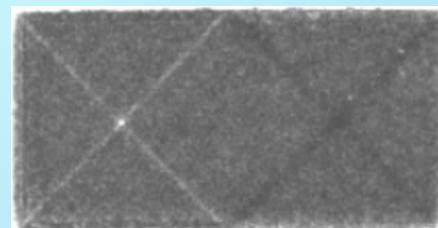
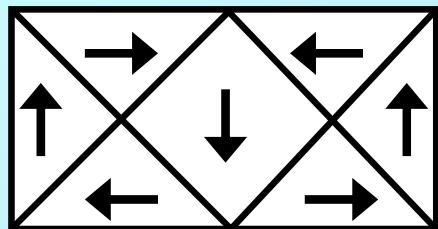
DPC detector



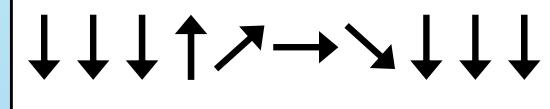
UNIVERSITY  
of  
GLASGOW

John Chapman

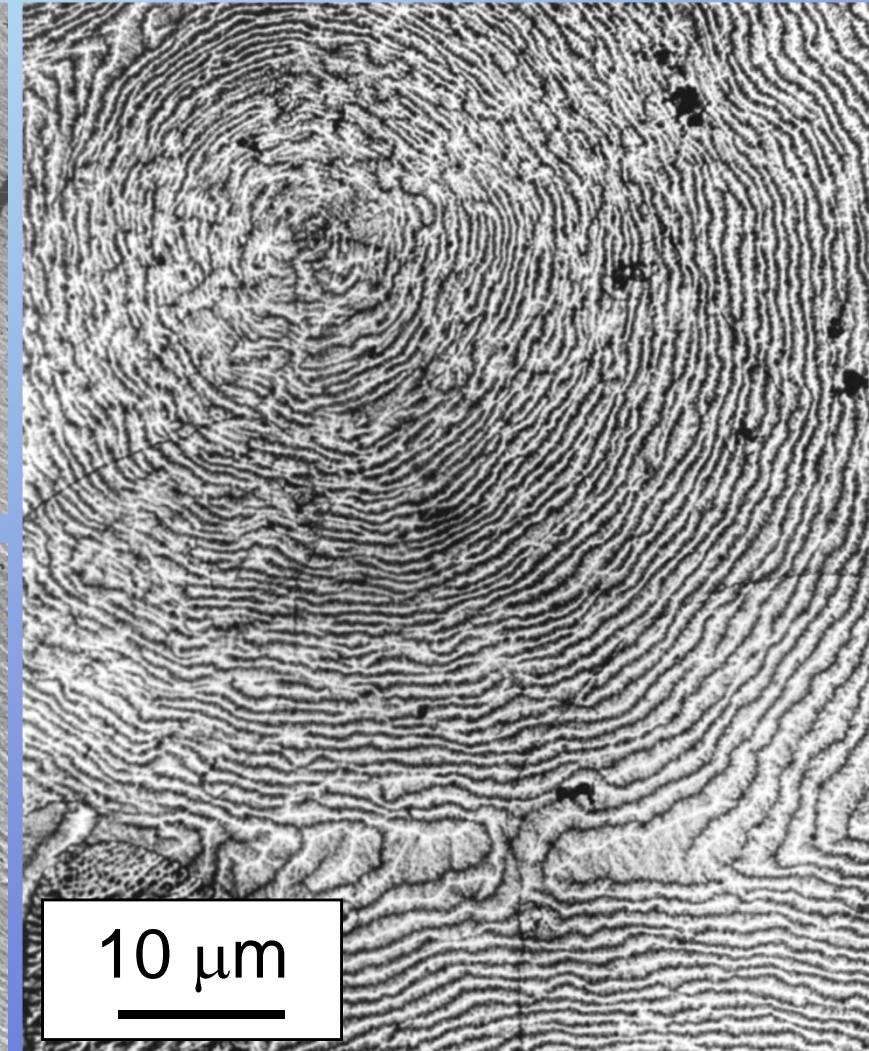
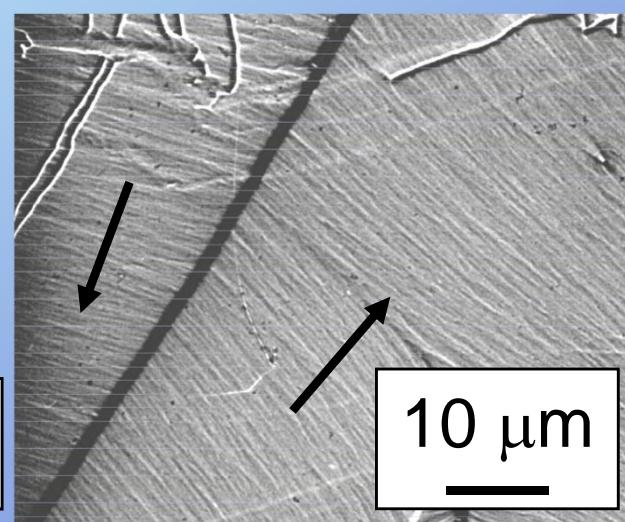
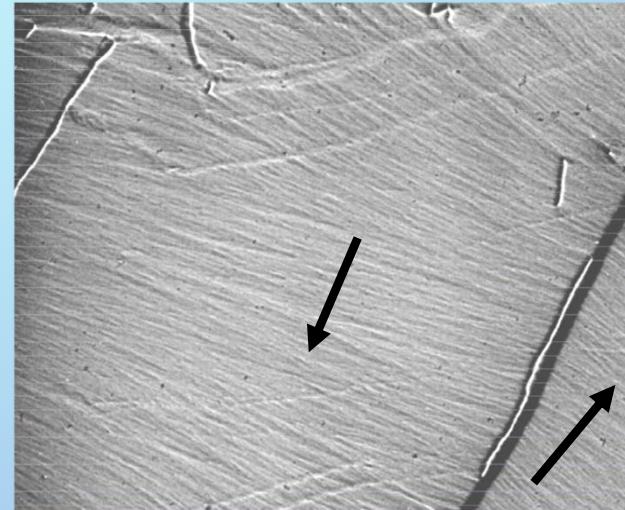
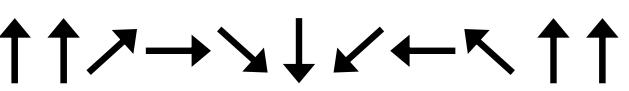
# Fresnel Microscopy of Different Domain Walls



90° Wall

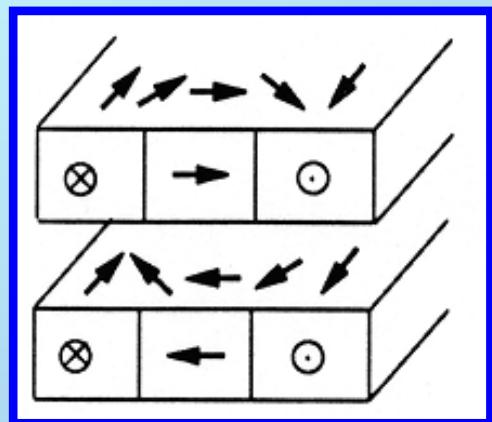


180° Wall

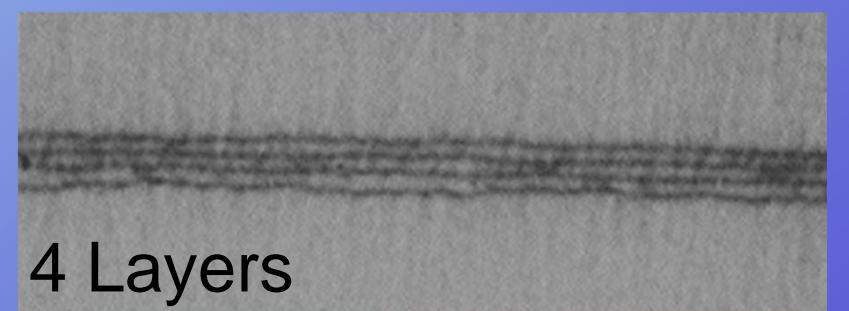
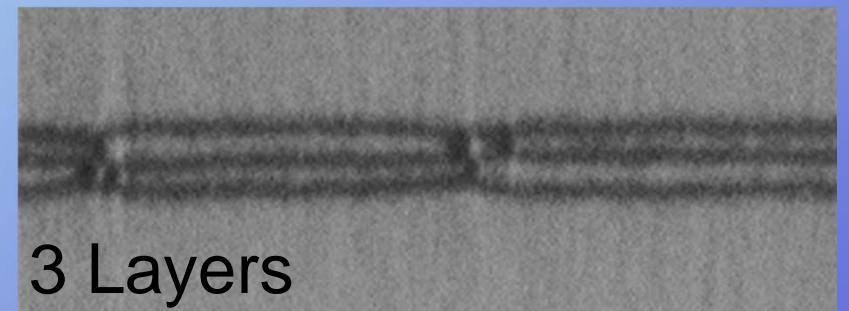
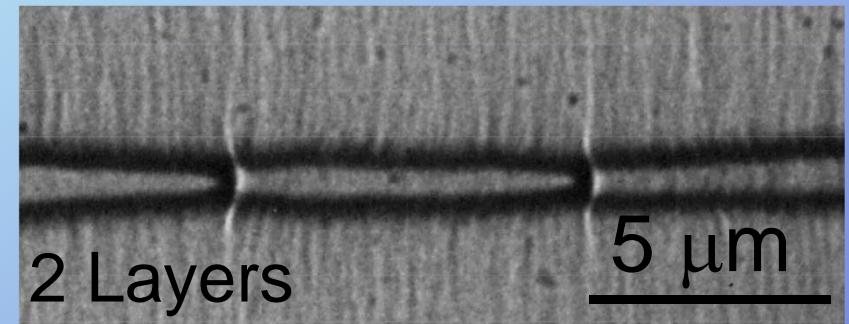
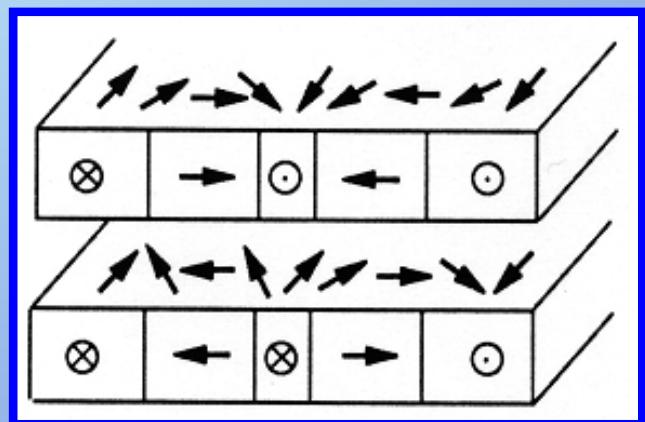


# Fresnel Microscopy: Domain Walls in Multilayers

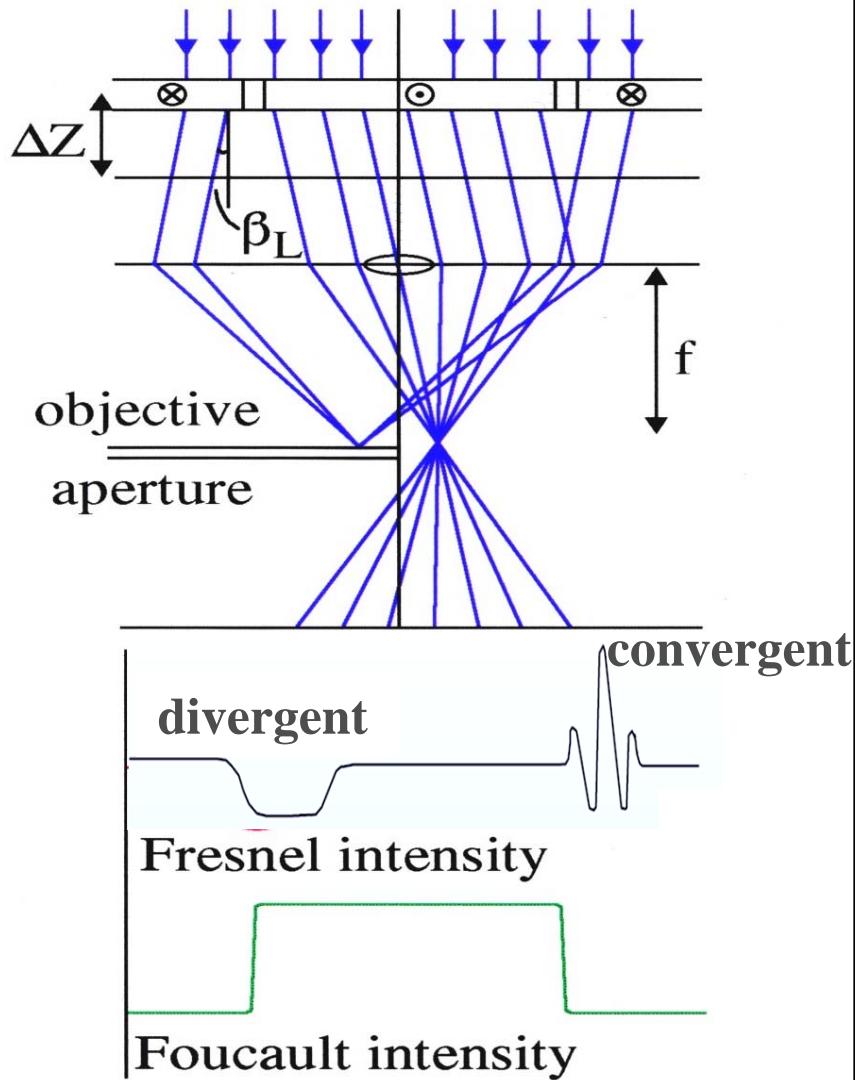
Single wall



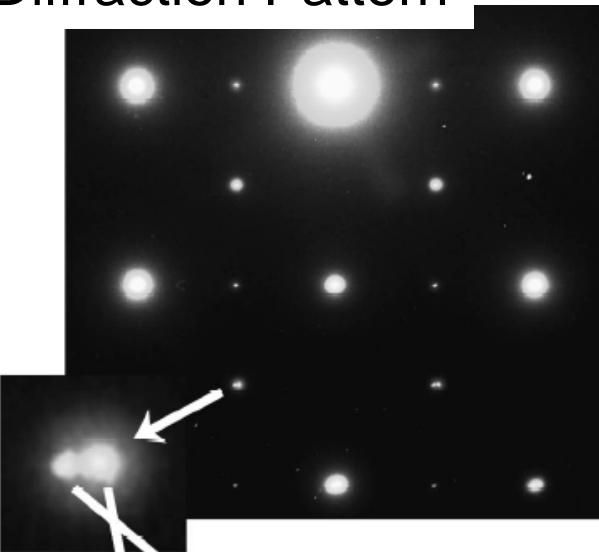
Twin wall



# Microstructure and Magnetic Information

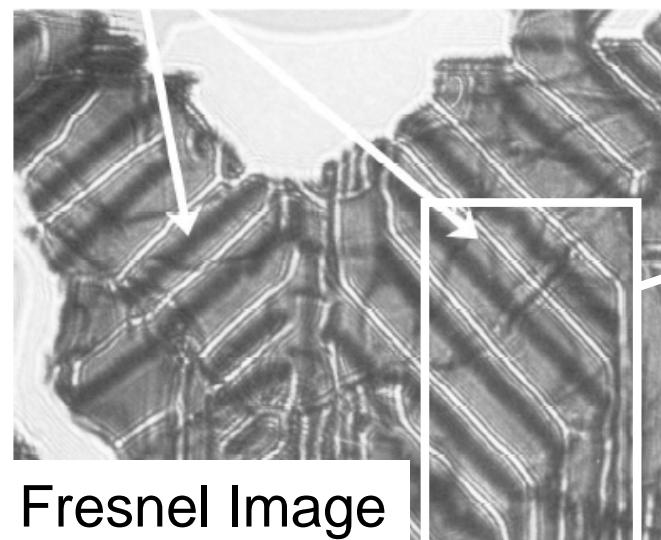


Diffraction Pattern

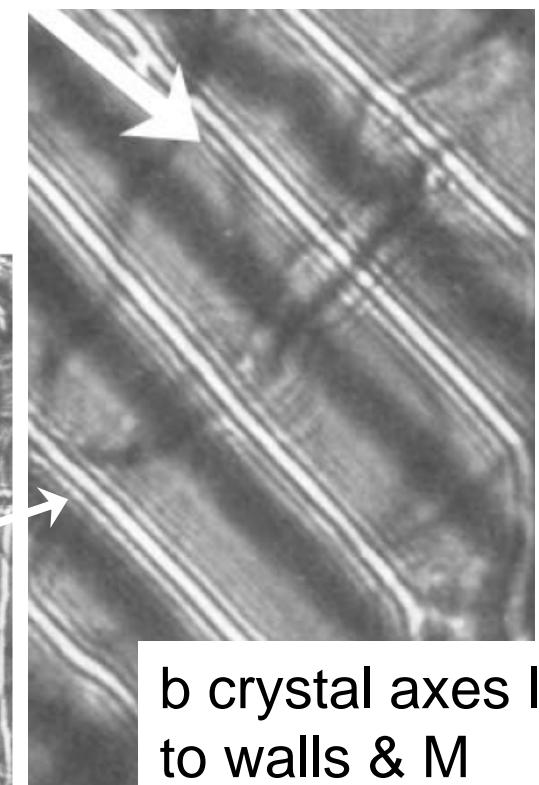


Domain walls in  $\text{SrRuO}_3$

A. F. Marshall et al.  
J. Appl. Phys. (1999)

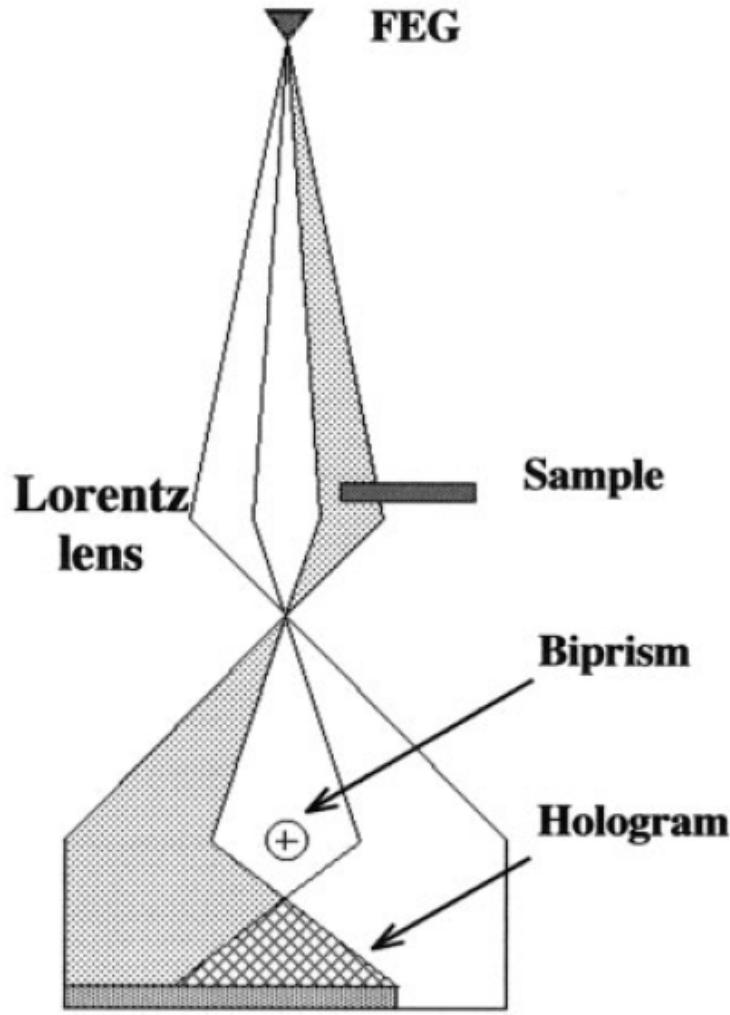


Fresnel Image



b crystal axes II  
to walls & M

# Off-axis Electron Holography



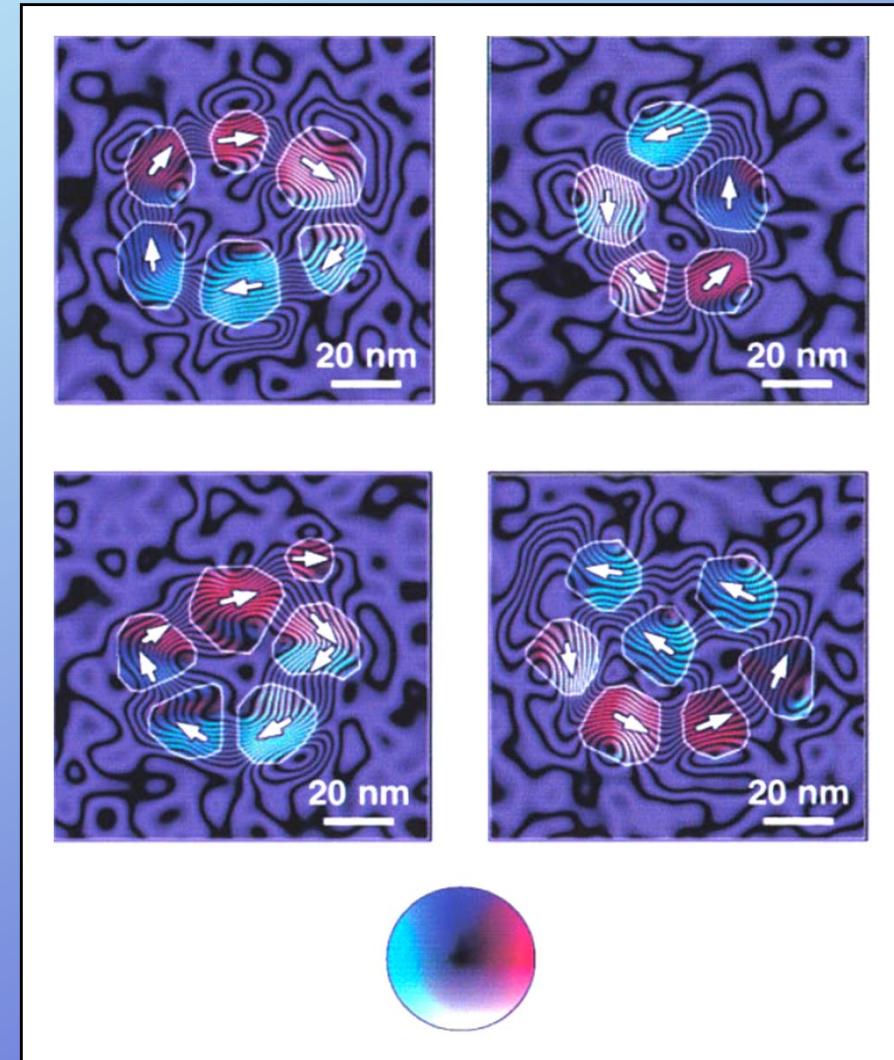
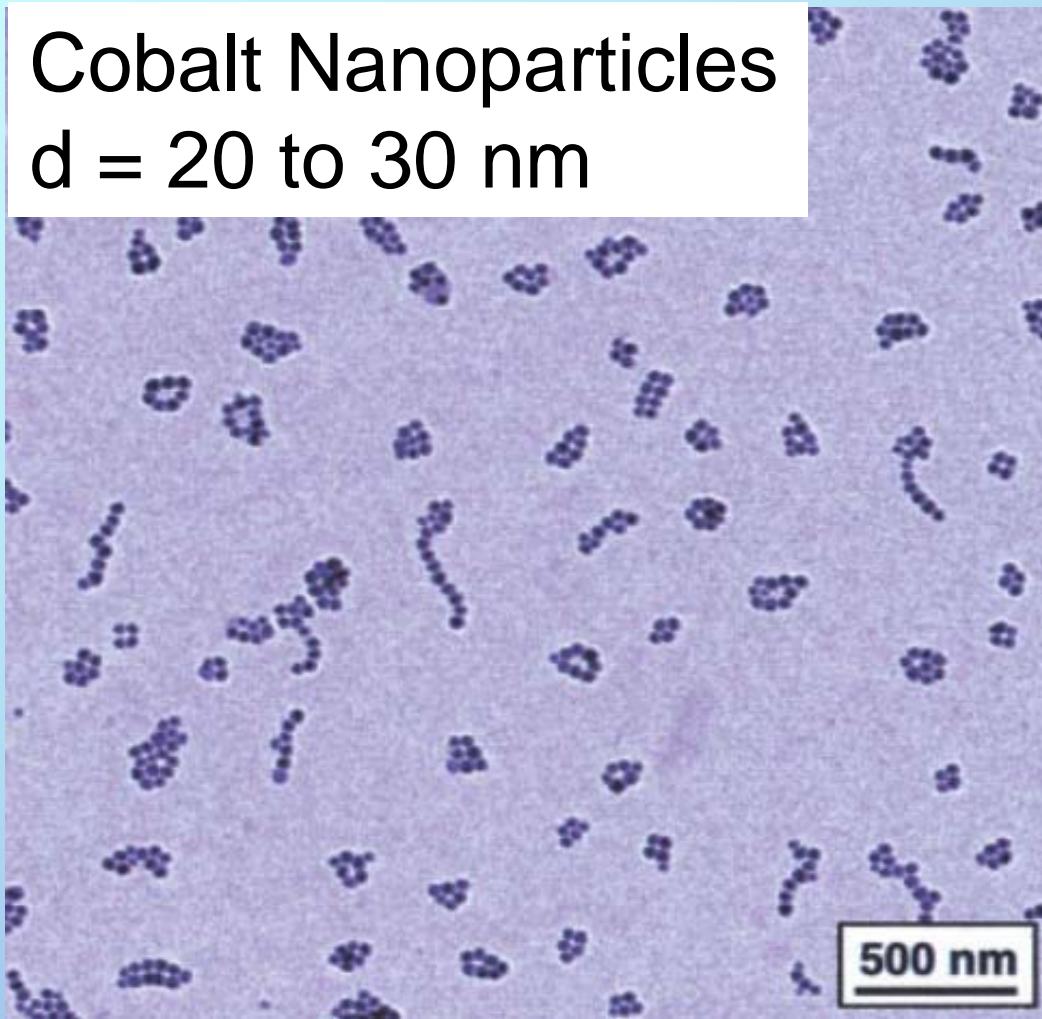
- apply a voltage to an electron biprism
- overlap of part of a coherent electron wave that has passed through a sample with the part that has passed through vacuum only.
- Analysis of resulting interference pattern allows the phase shift of the electron wave to be recovered and a quantitative map of the in-plane induction can be obtained.
- This provides a high spatial resolution information about domains, domain walls and stray field interactions.

R.E. Dunin-Borkowski, et al.  
Micr. Res. and Technique (2004)

# Off-axis Electron Holography

- High resolution ( $\approx 5$  nm)
- Magnetic induction: internal spin orientation + external stray field

Cobalt Nanoparticles  
 $d = 20$  to 30 nm



## Lorentz Microscopy Summary

- high spatial resolution (< 5 nm has been demonstrated)
- information on domain and domain wall structures
- straightforward image interpretation (usually)
- sensitive to induction (so contrast from sample magnetisation and stray fields)
- quantitative information on spatial distribution of integrated induction components
- suitable for real time studies involving field & temperature variation
- availability of complementary (perfectly registered) nanostructural information
- Sample must allow transmission of electrons

# Lorentz Microscopy Limitations

But.....

- phase shifts can be of magnetic or electrostatic origin, leading to severe problems when the latter contribution dominates
- no information about components of induction parallel to the direction of electron travel
- for multilayers, no way of separating contributions to images arising from individual layers
- no contrast from antiferromagnets
- time resolution is poor (typically ~1s, best ~20ms)

# Kerr Microscopy

# Kerr Effect



Michael Faraday  
(1791-1867)

1845

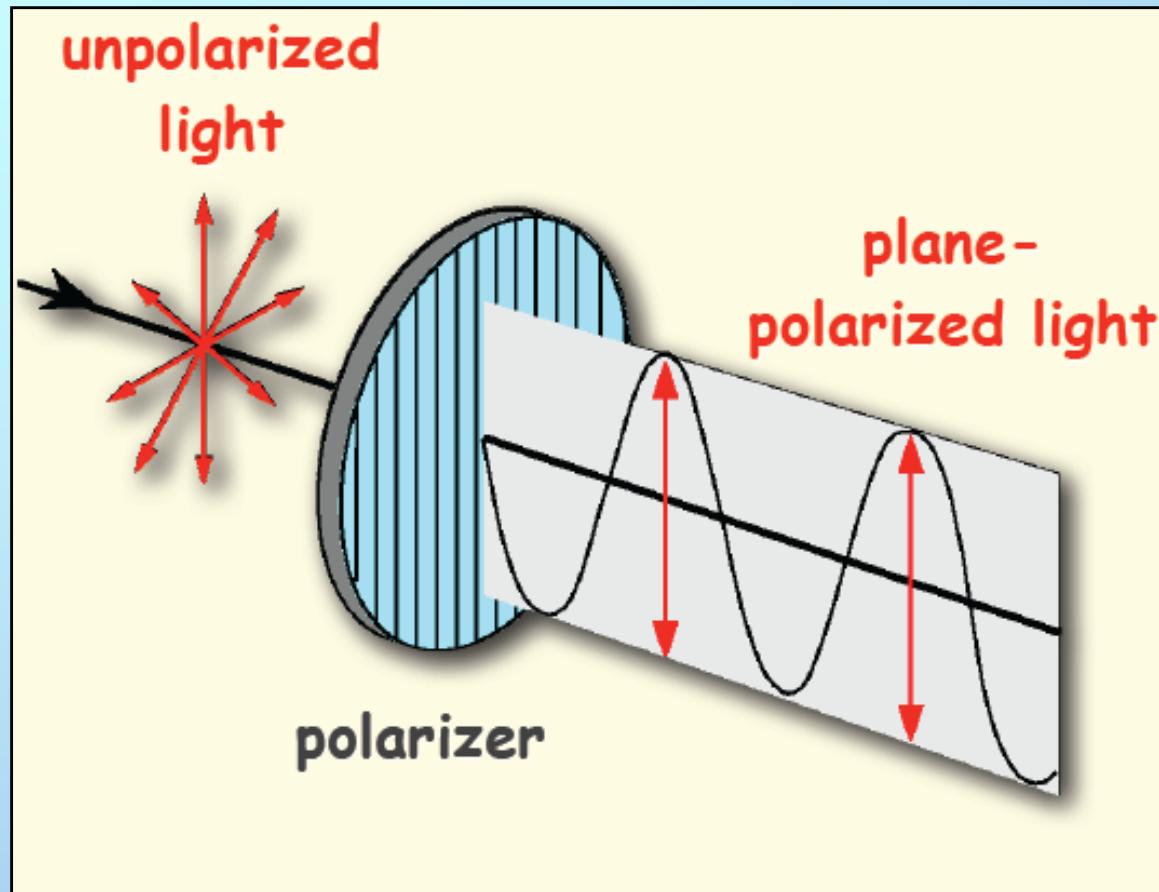
small change of polarization plane due to  
magnetooptic interaction in transmission,  
circular birefringence,  $\sim M$

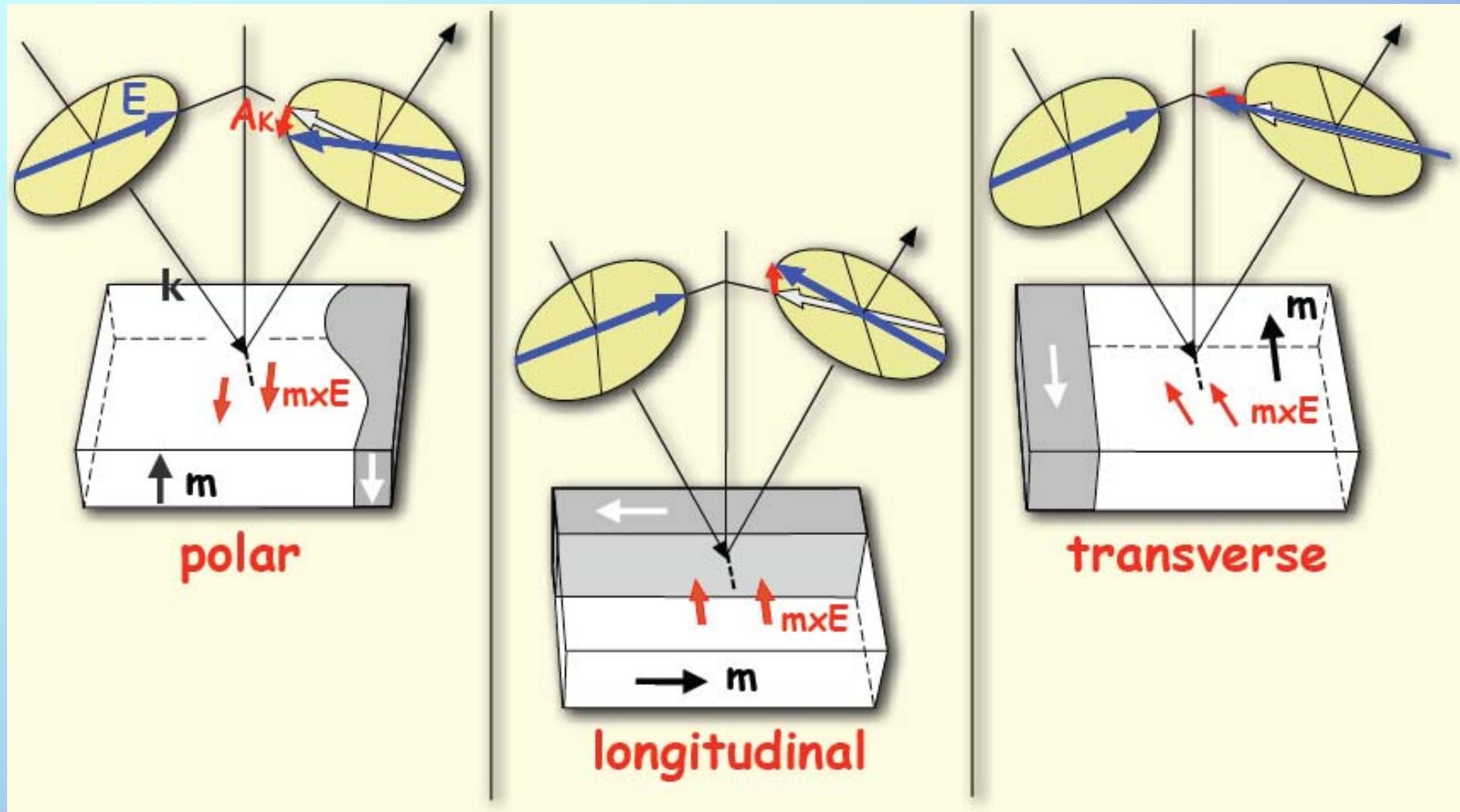


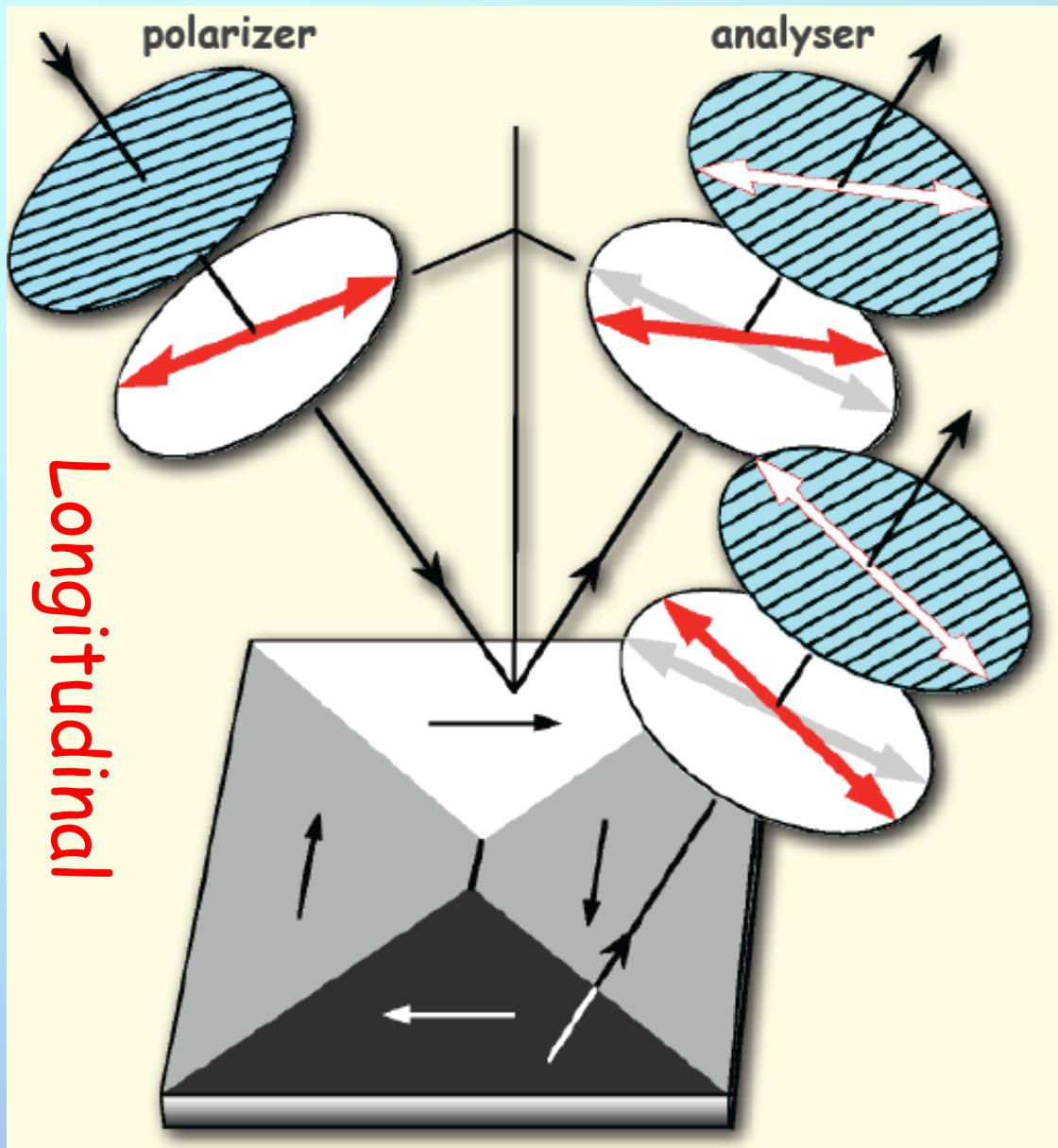
John Kerr  
(1824-1907)

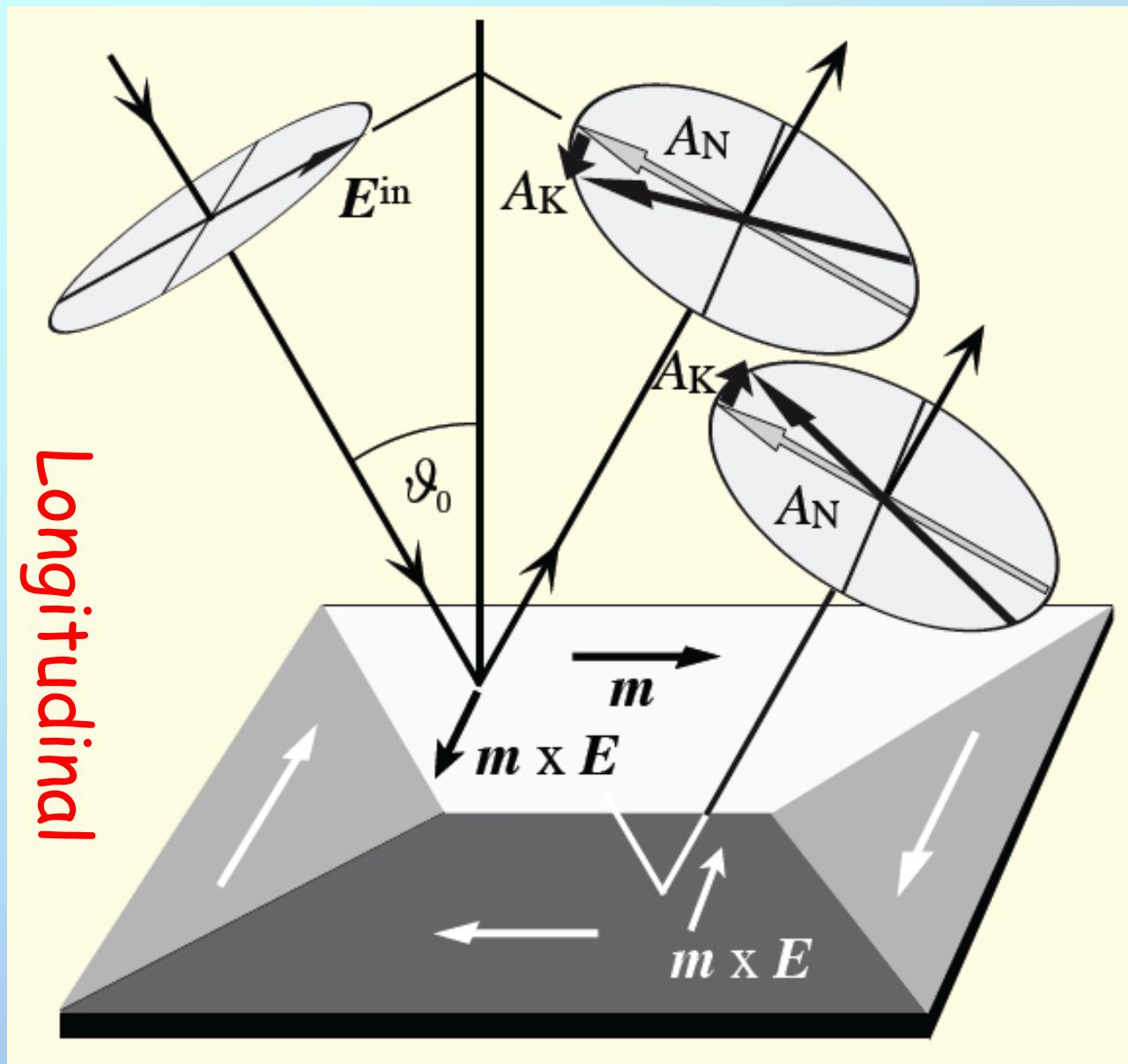
1877

small change of polarization plane due to  
magnetooptic interaction in reflection,  
circular birefringence,  $\sim M$









Linearly polarized light will induce electrons to oscillate parallel to its plane of polarization – the plane of the electric vector  $E$  of light

Secondary motion is proportional to  $-M \times E$ , and generates the Kerr amplitude,  $A_K$ , for reflection → rotation of polarisation

Opposite  $M$  direction → opposite Kerr rotation

# Kerr Microscopy

- Based on small rotations of the polarization plane of light
- Linearly polarized light will induce electrons to oscillate parallel to its plane of polarization – the plane of the electric vector  $E$  of light
- Regularly reflected light is polarised in the same plane as the incident light:  $A_N$  component
- Lorentz force induces a small component of vibrational motion perpendicular to original motion and to direction of magnetisation
- Secondary motion which is proportional to  $-M \times E$ , and generates the Kerr amplitude,  $A_K$ , for reflection

# Image Signal

If total signal amplitude:

$$A_{\text{TOT}} = A_N +/- A_K \text{ (normal & Kerr)}$$

Then Kerr Rotation:

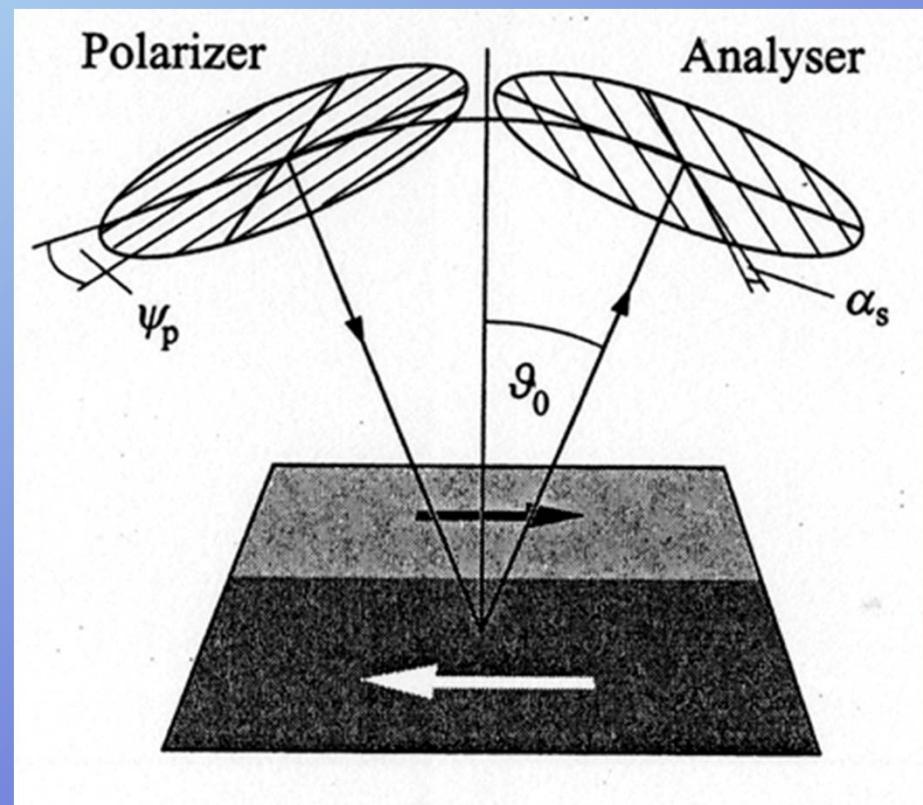
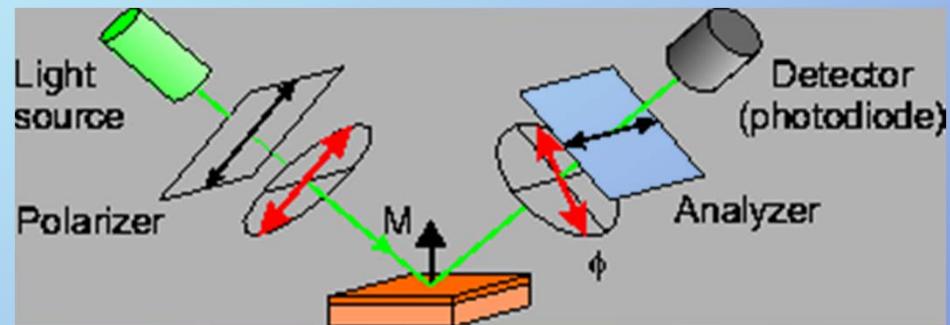
$$\phi_K = A_K / A_N$$

Start with  $\alpha_s = \phi_K$ ; extinguishes light from one domain; one domain appears dark and the other light

Actually better to rotate the analyser beyond the extinction point.

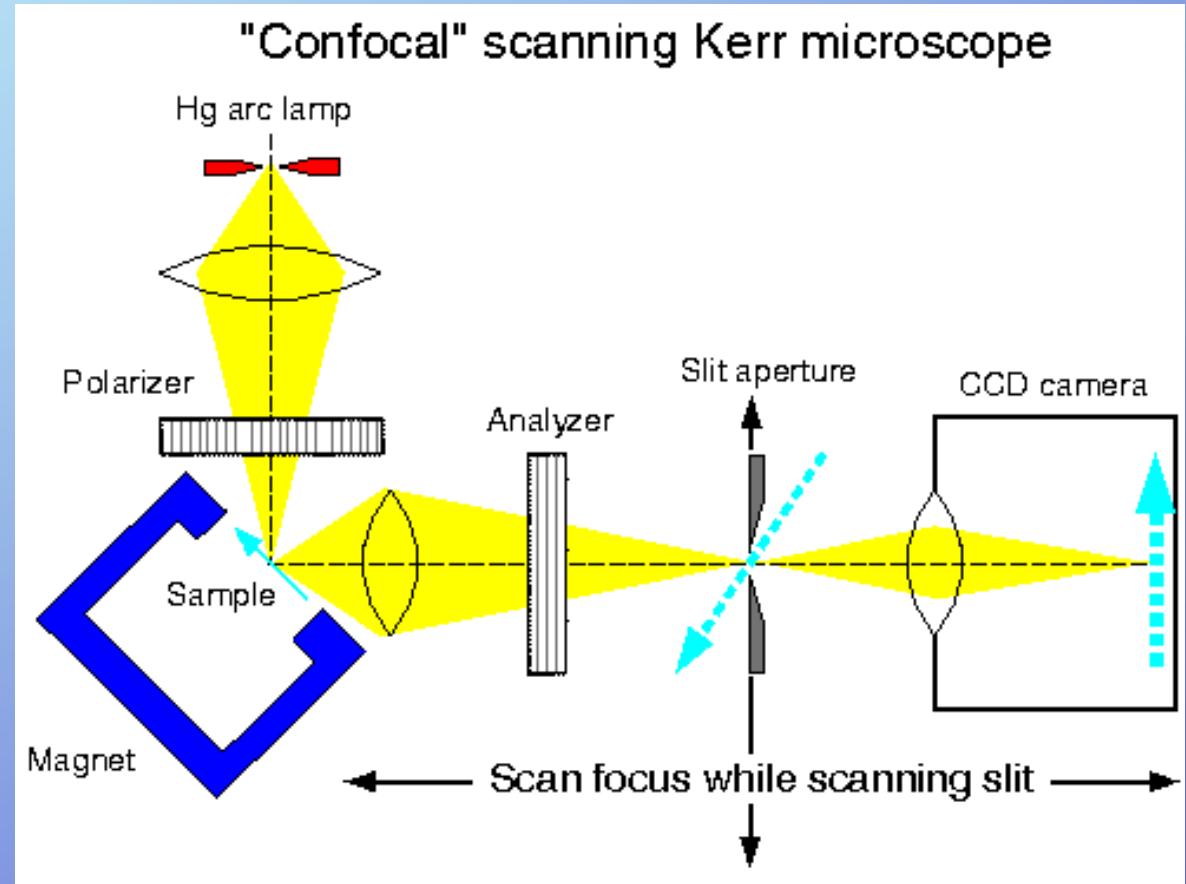
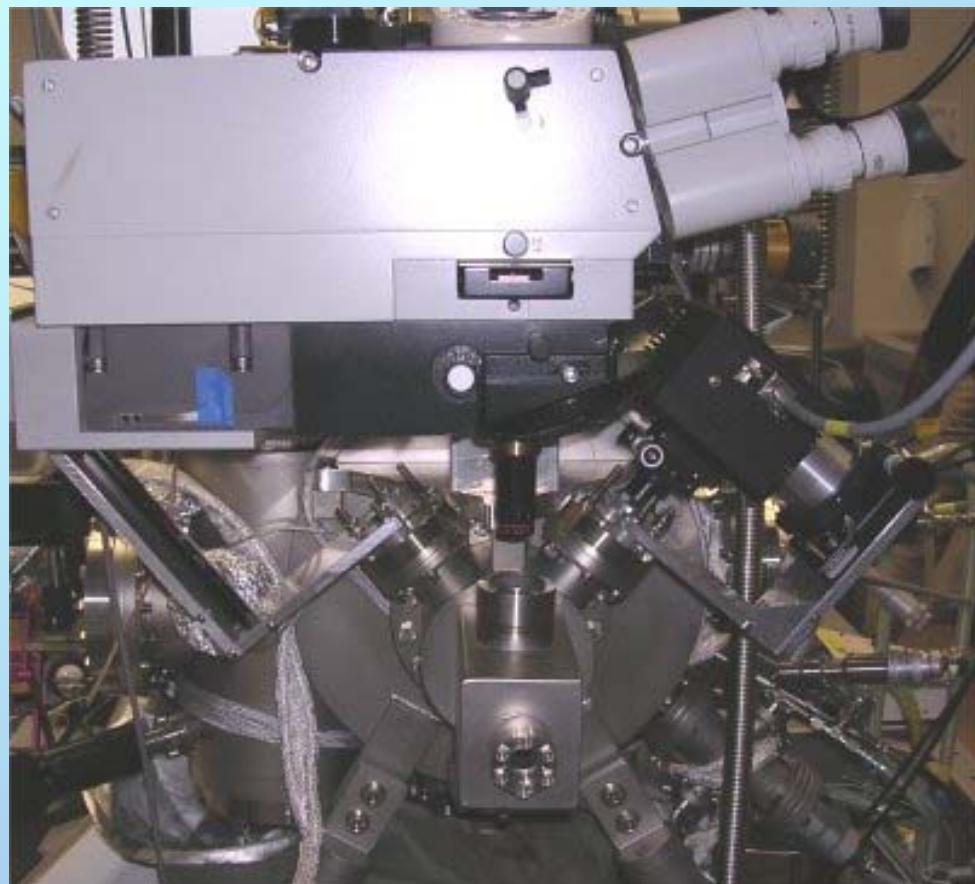
In practice, adjust polarizer and analyser until an image of satisfactory contrast and brightness is obtained.

[http://www.physik.fu-berlin.de/~bauer/habil\\_online/node9.html](http://www.physik.fu-berlin.de/~bauer/habil_online/node9.html)



# Polarising Microscope

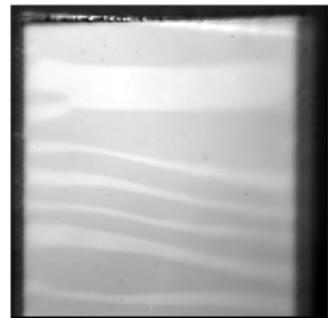
<http://physics.nist.gov/Divisions/Div841/Gp3/Facilities/kerr.html>



<http://www.fkf.mpg.de/kern/facilities/kerr/kerr.html>

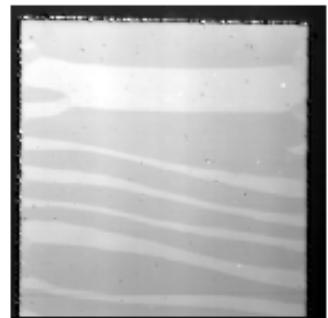
# Digital Image Enhancement

Digital difference technique:  
non-magnetic background image is digitally subtracted



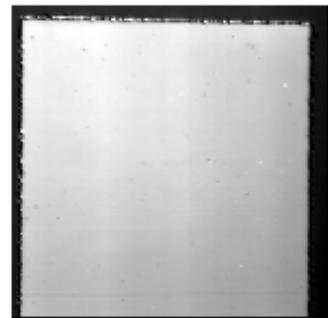
Images of 500  $\mu\text{m}$  wide NiFe square

0 Oe field, no scanned slit



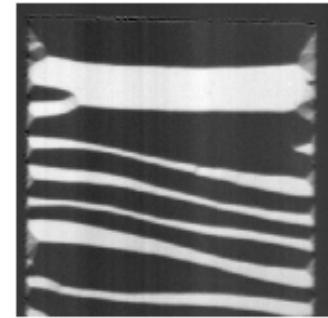
with confocal scanned slit

-



100 Oe field

=

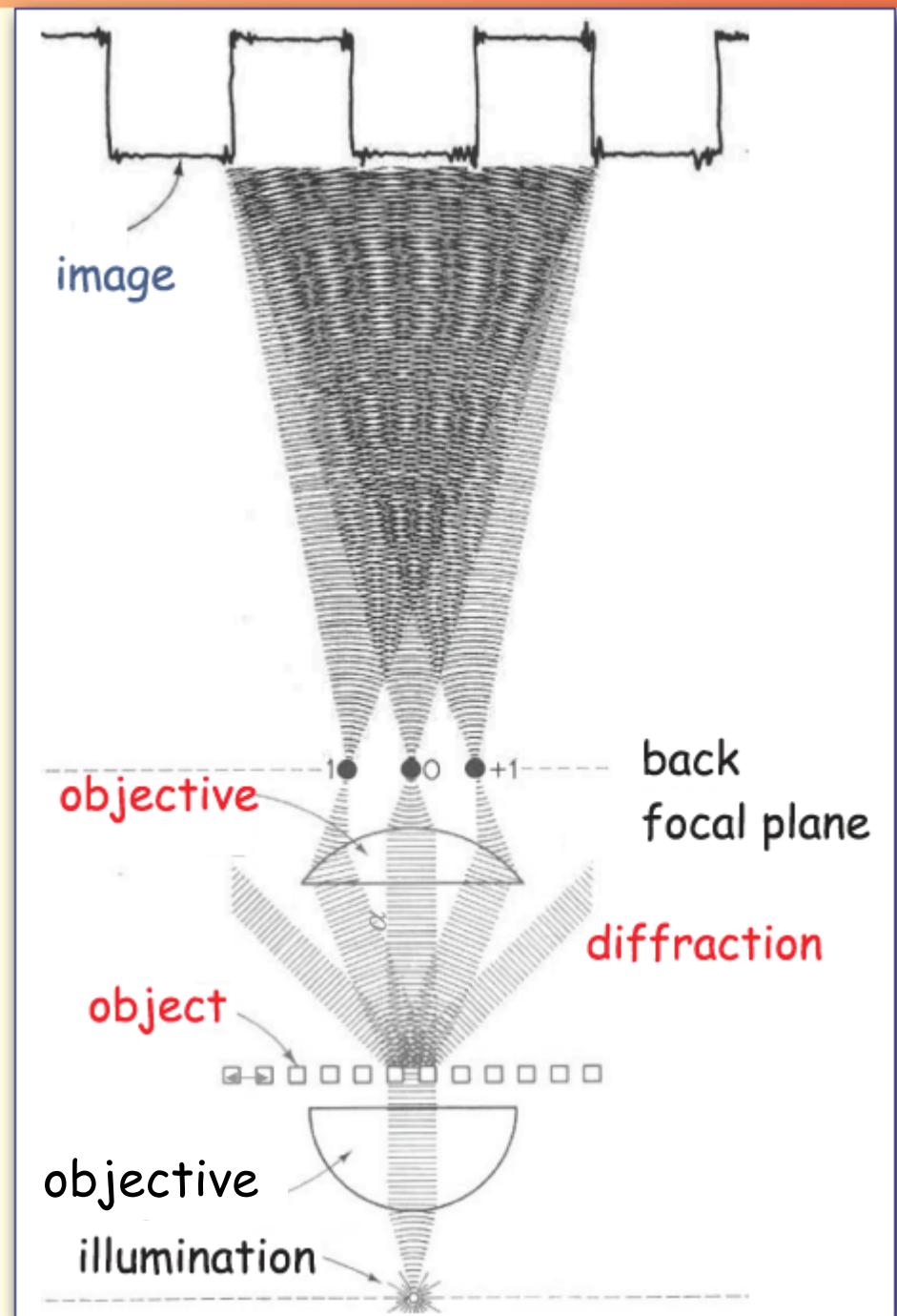


Difference

Resolution  
approx.  
200 nm

# Resolution of optical microscopy (E. Abbe 1840 - 1905)

11



resolution determined by  
constructive interference

diffraction limited image formation

$$\text{Rayleigh equation: } d = \frac{0.5 \lambda}{\text{NA}}$$

$d$  = separation between particles,  
still allowing to see them

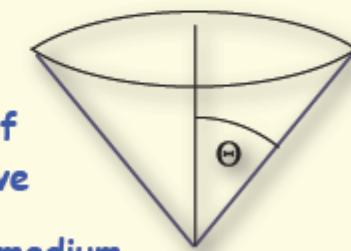
$\lambda$  = wavelength

NA = numerical aperture of objective

$$\text{NA} = n \sin \theta$$

$\theta$  = half the cone angle of  
light accepted by objective

$n$  = refraction index of medium  
between sample and objective

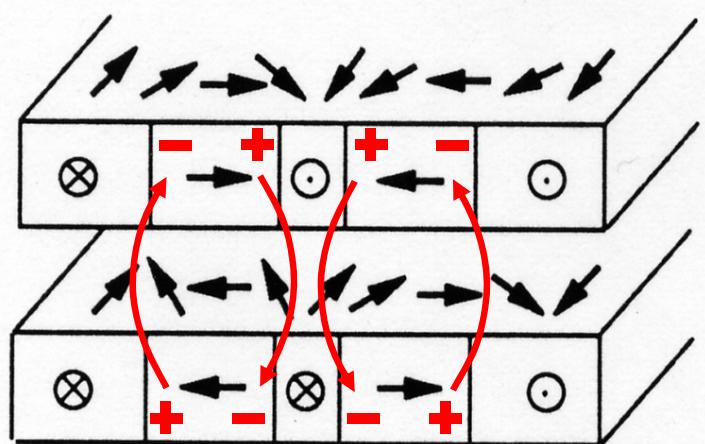


best around 200 nm

Rudolf Schäfer  
IFW Dresden

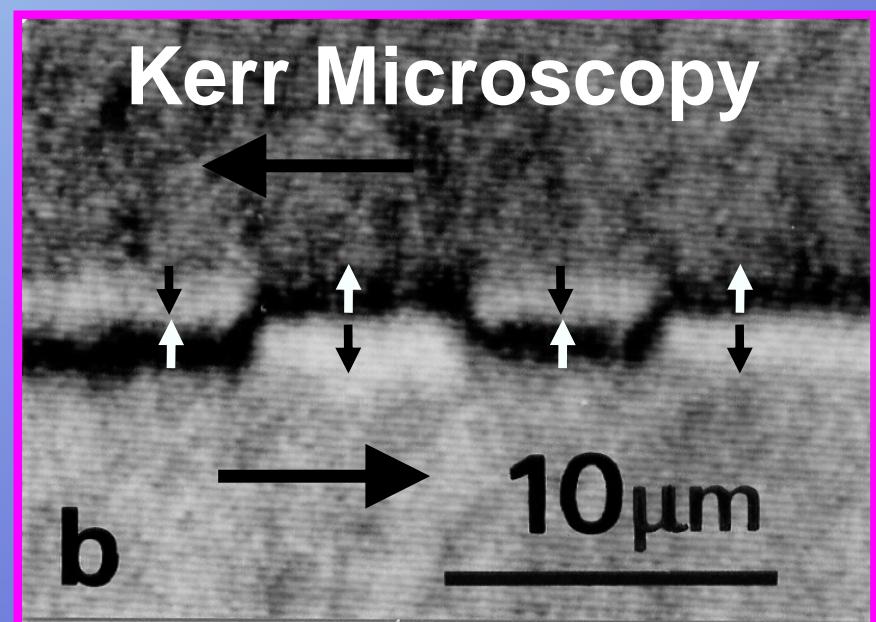
# Observation of Quasi-domain Walls

## Twin Walls



Ferromagnetic Thin Film  
Non-magnetic Spacer Layer  
Ferromagnetic Thin Film

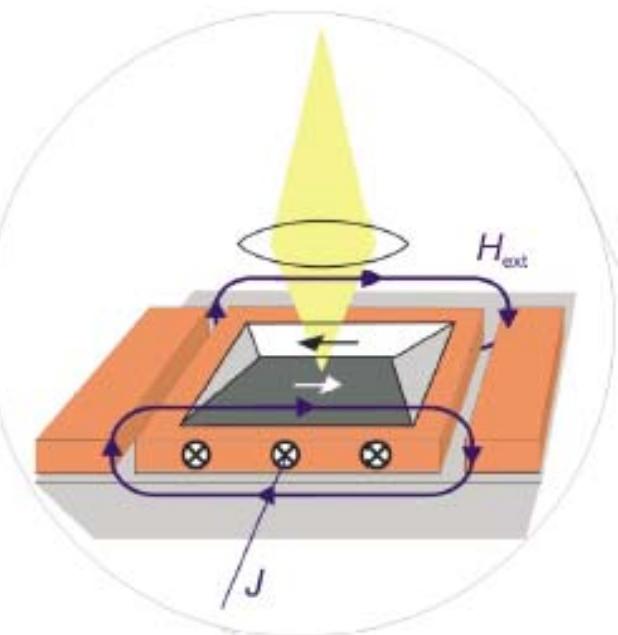
Image shows domain wall (black line) and quasi-domain wall (white line) in top layer of sandwich film



# Time-Resolved Kerr Microscopy

## Stroboscopic technique:

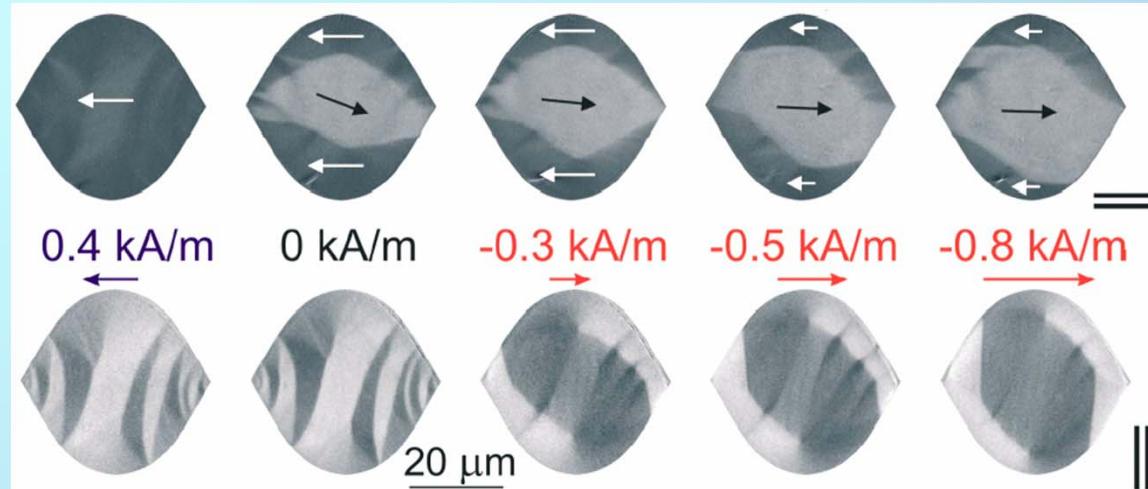
- Short field pulses (e.g. 20 ns, 10 kA/m) with copper microstrip line
- With a defined time delay, the magnetization is probed
- Reasonable signal-to-noise ratio by integrating the optical signal
- Accumulation over repeatable magnetization processes is required.
- Gated & intensified CCD camera providing temporal resolution down to 200 ps



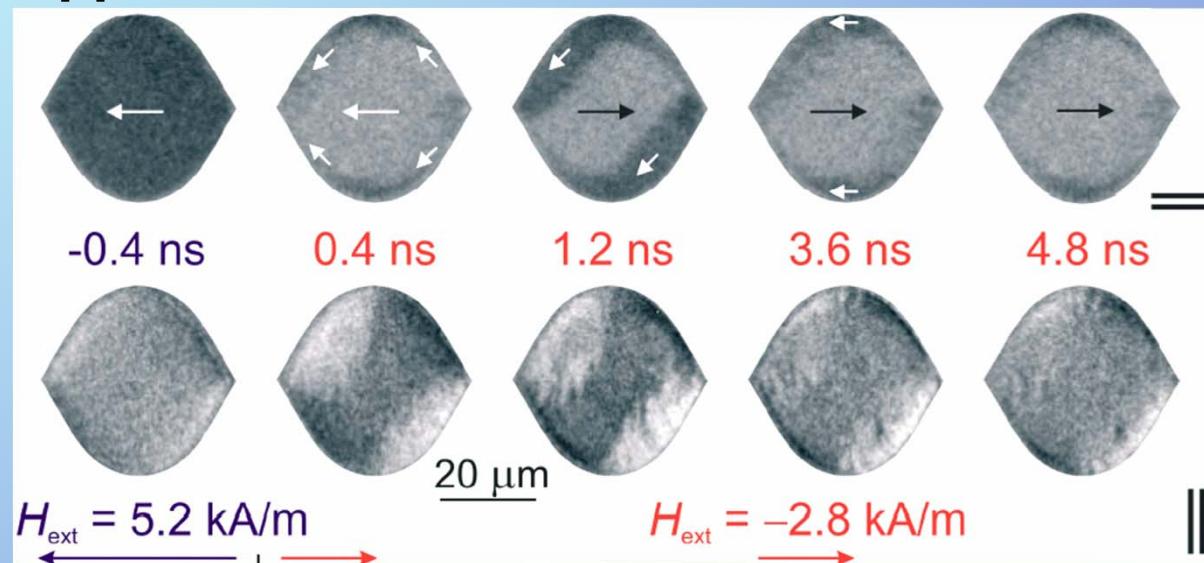
Chumakov, McCord, Schäfer, Schultz

# Time-Resolved Kerr Microscopy

Applied Field: quasi-static reversal



Applied Field Pulse: time-resolved



Applying a field pulse:  
magnetization not able to  
instantaneously follow  
magnetic field.

To reach new magnetization  
direction, will have to spin  
about the field axis:  
precessional motion that is  
gradually opposed by  
damping.

Reversal looks very  
different!

# Kerr Microscopy Summary

- Flat and smooth surface
- Resolution approx. 200 nm
- Magnetization can be observed directly
- Quantitative measurements possible but need to take care with calibration
- Observation does not influence magnetization
- Dynamic processes can be observed at high speed
- Sample may be easily manipulated: fields, high or low temperature, mechanical stress
- Surface magnetization: penetration depth of 10-20 nm
- Can look at back and front of sample

# X-Ray Microscopy Techniques



The Swiss Light Source, Paul Scherrer Institut

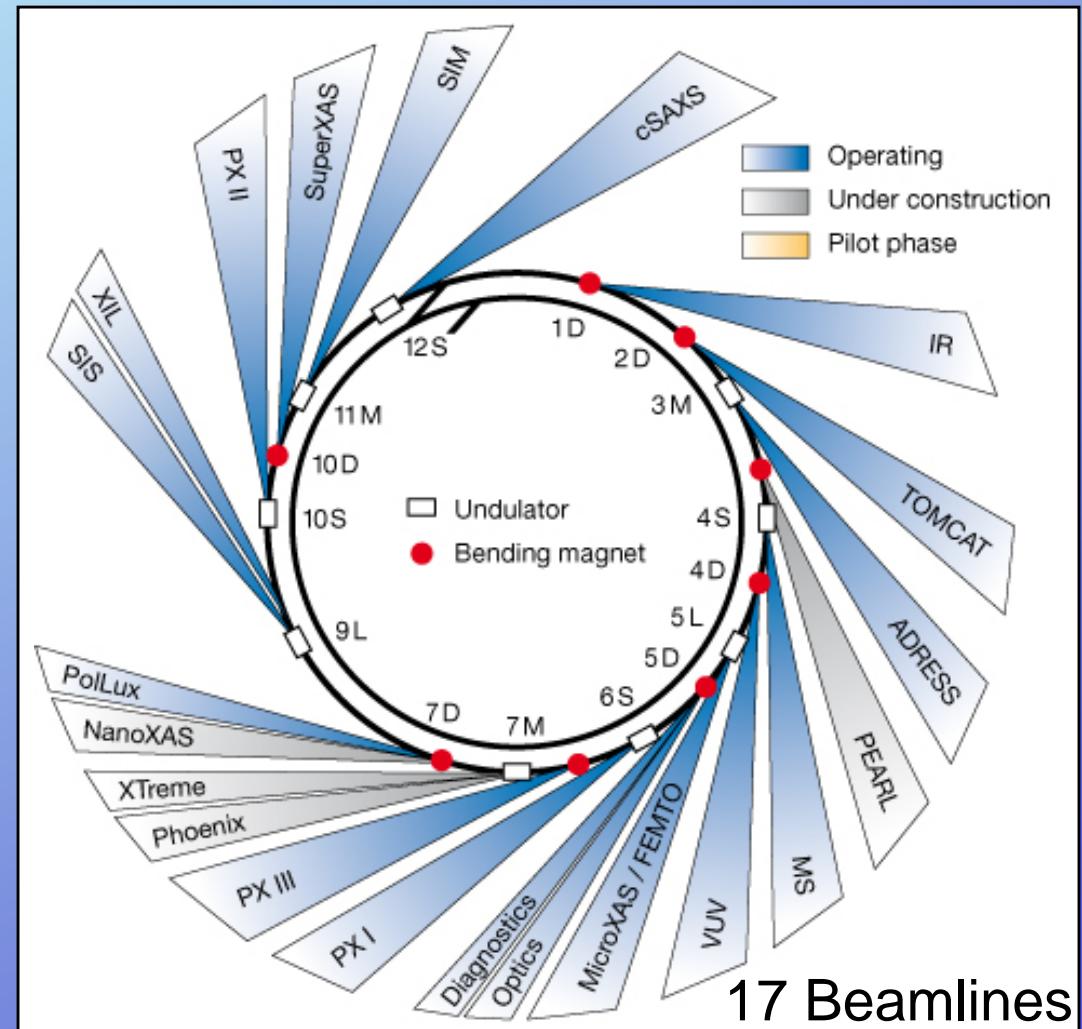
- The electrons are accelerated close to the speed of light in a linear accelerator and injected into the storage ring
- Bending magnets or insertion devices (wigglers or undulators) cause electrons to bend or wobble through the section and emit light.



**Reference energy:** 2.4 GeV

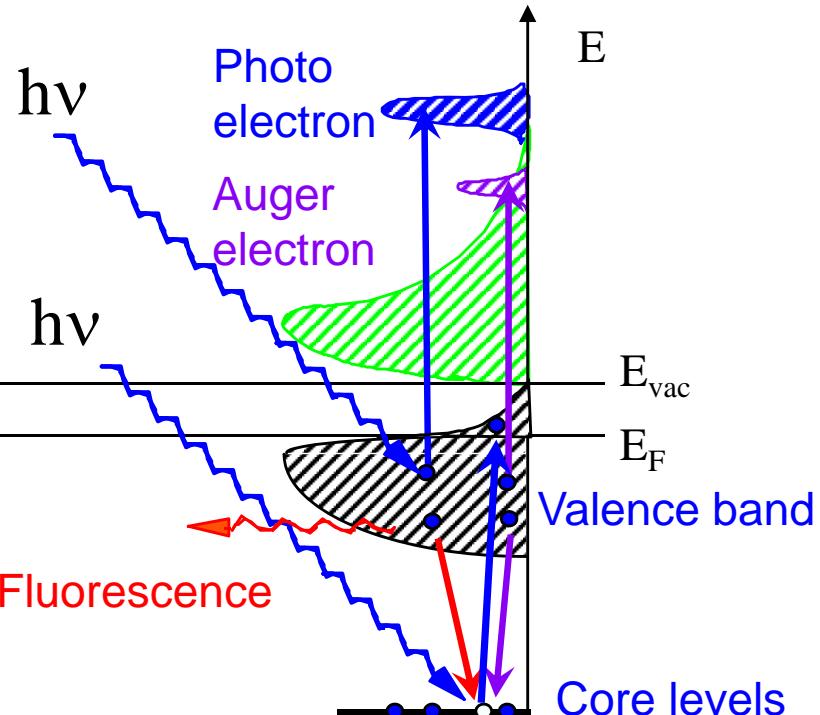
**Circumference:** 288 m

**Current:** 350 mA (400 mA)

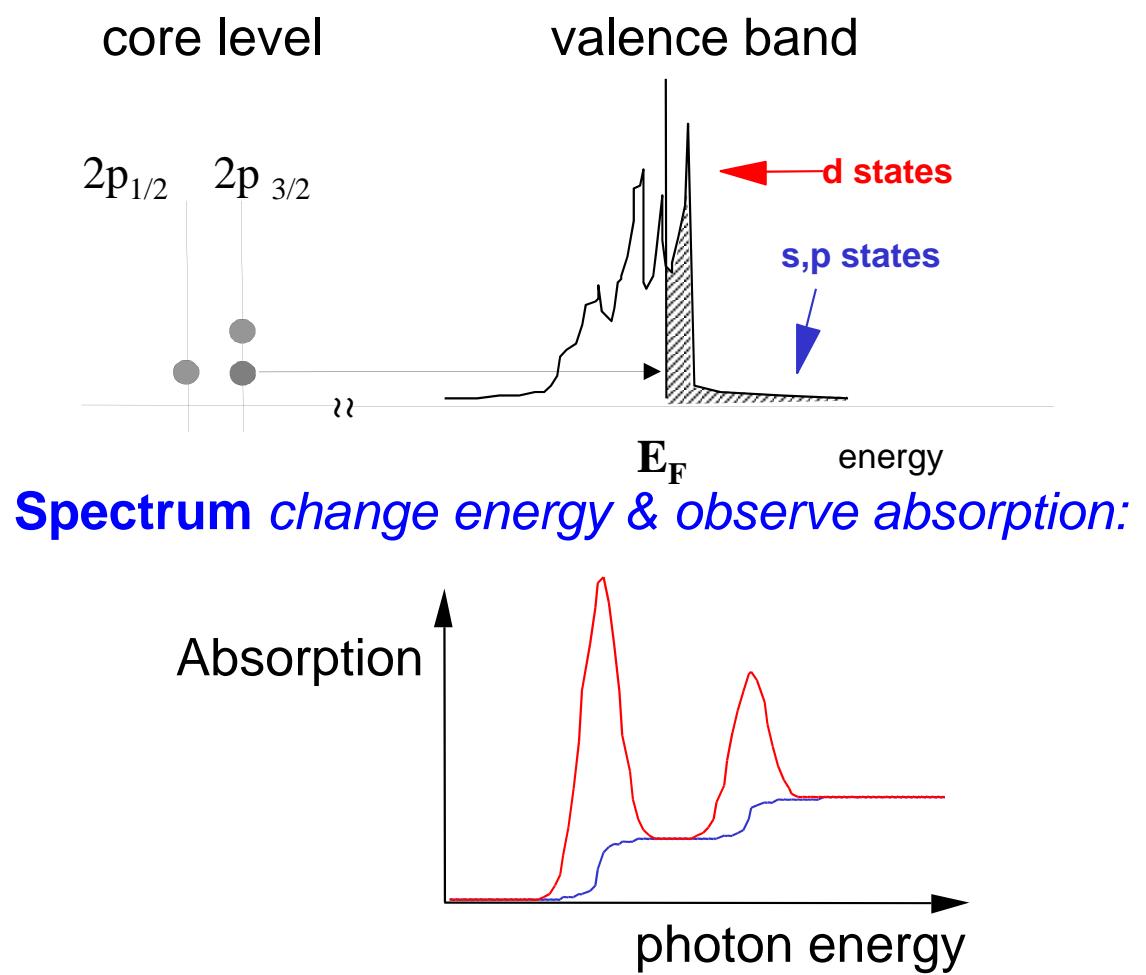


17 Beamlines

# X-ray Absorption Spectroscopy

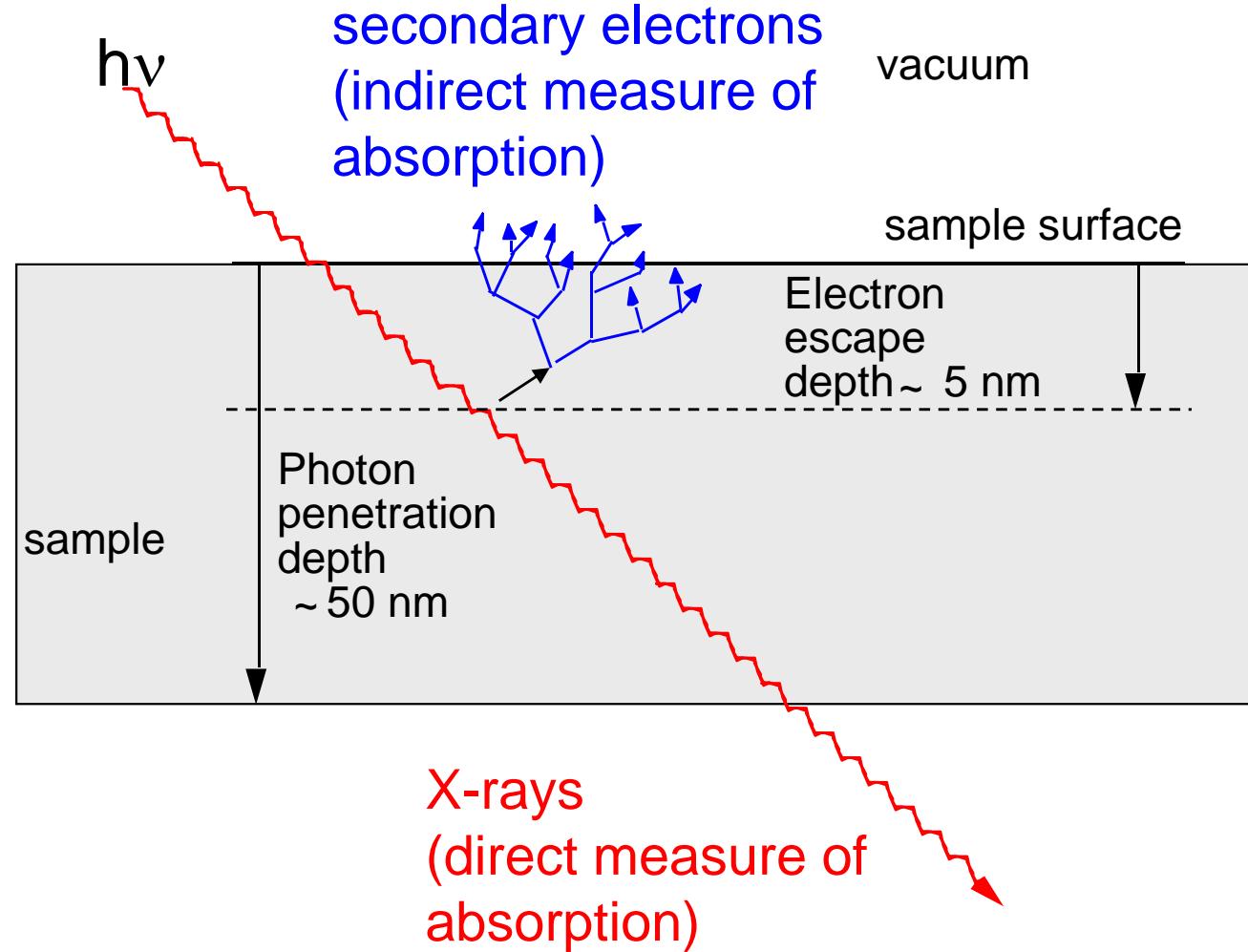


- X-rays excite an electron from the core to the valence band
- Relaxation of the electron from the valence to the core level gives:
  - Soft x-rays: more Auger es
  - Hard x-rays: more fluorescence
- Therefore different interactions: more than just imaging



- Peak corresponds to (set of) transition(s) from core level to valence band
- Density of unoccupied states above Fermi level
- Each element: own characteristic peaks

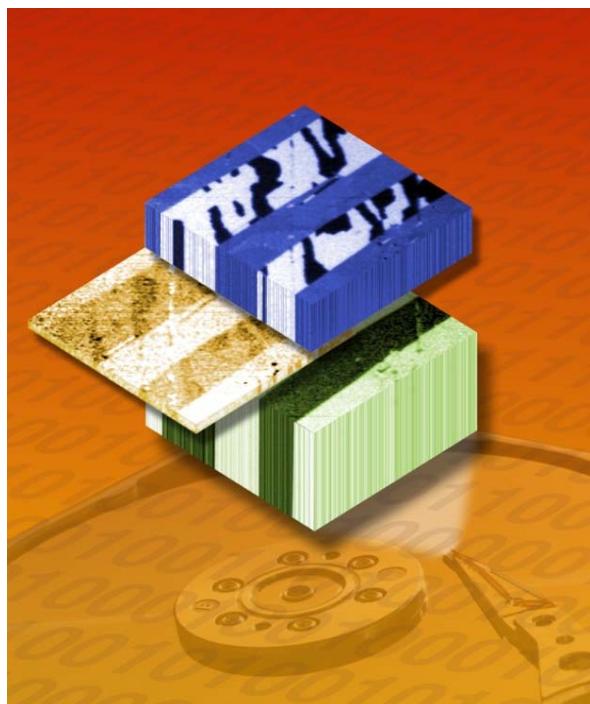
# PEEM & TXM



**Photoemission  
Electron  
Microscopy  
(PEEM) to probe  
surface / interfaces**

**Transmission  
X-ray Microscopy  
(TXM)**

# Photoemission Electron Microscopy

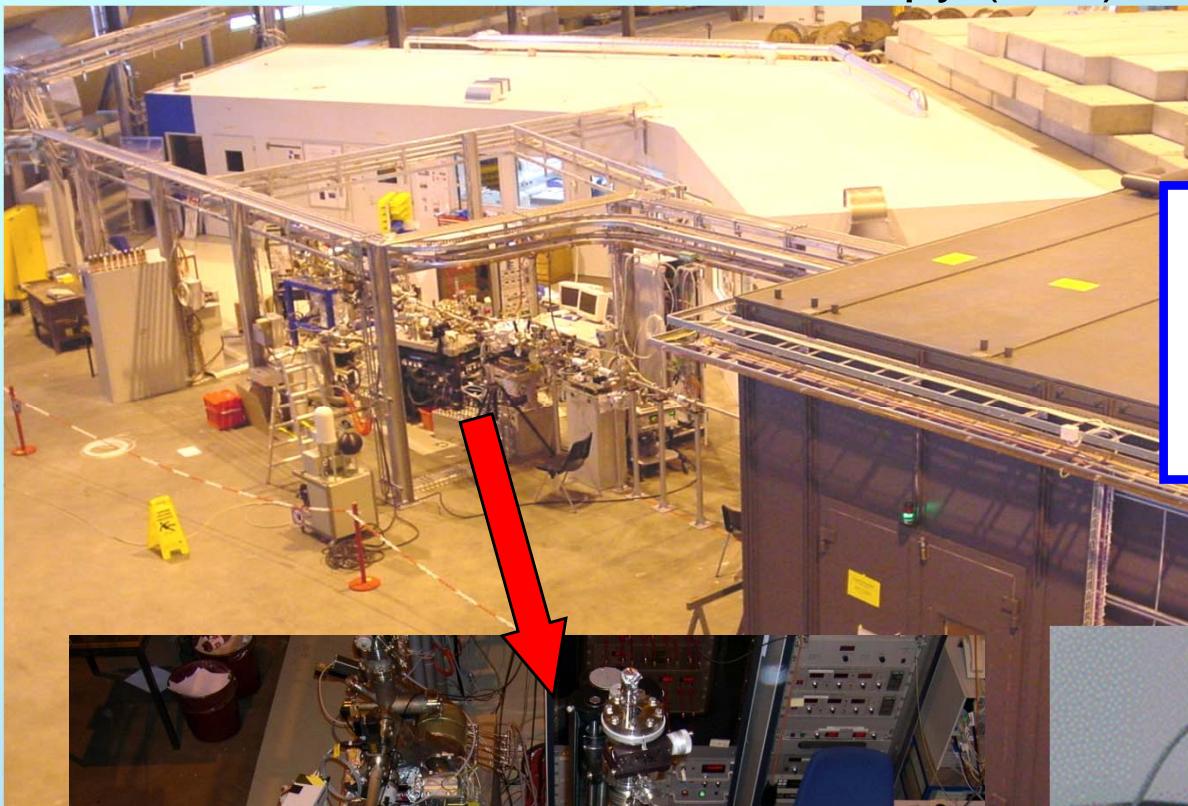


Slow electrons: mean free path is submono to several monolayers (few nm's)

Surfaces, thin films and interfaces

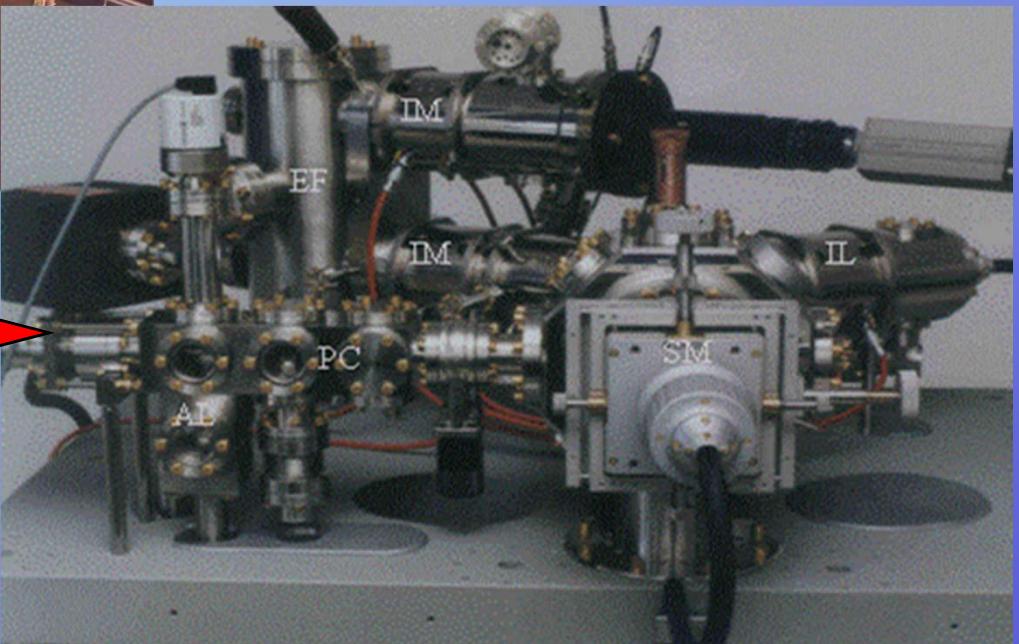
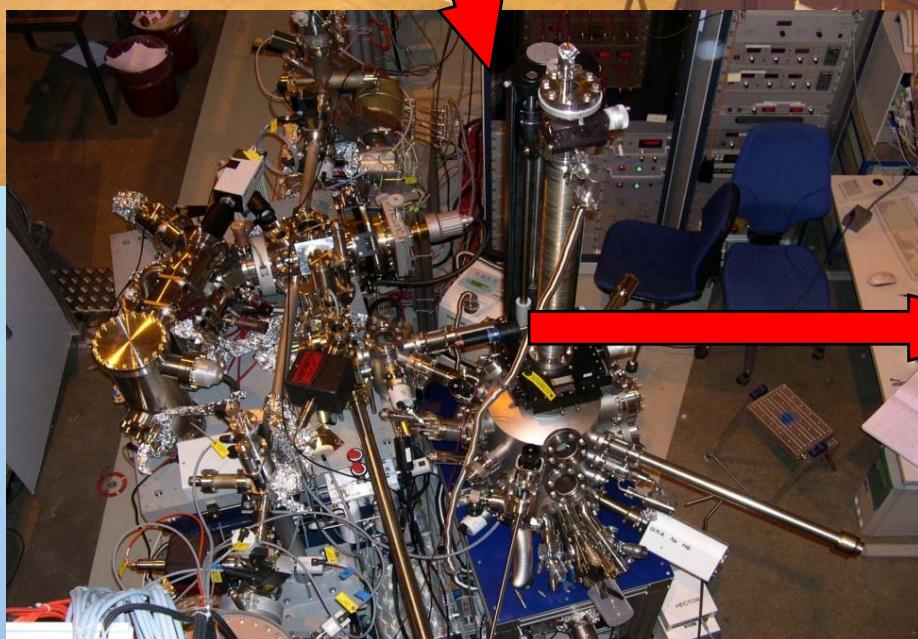
....consequences for electron optics.

## The Surface and Interface Microscopy (SIM) Beamline



**SIM Beamline,  
Swiss Light Source**

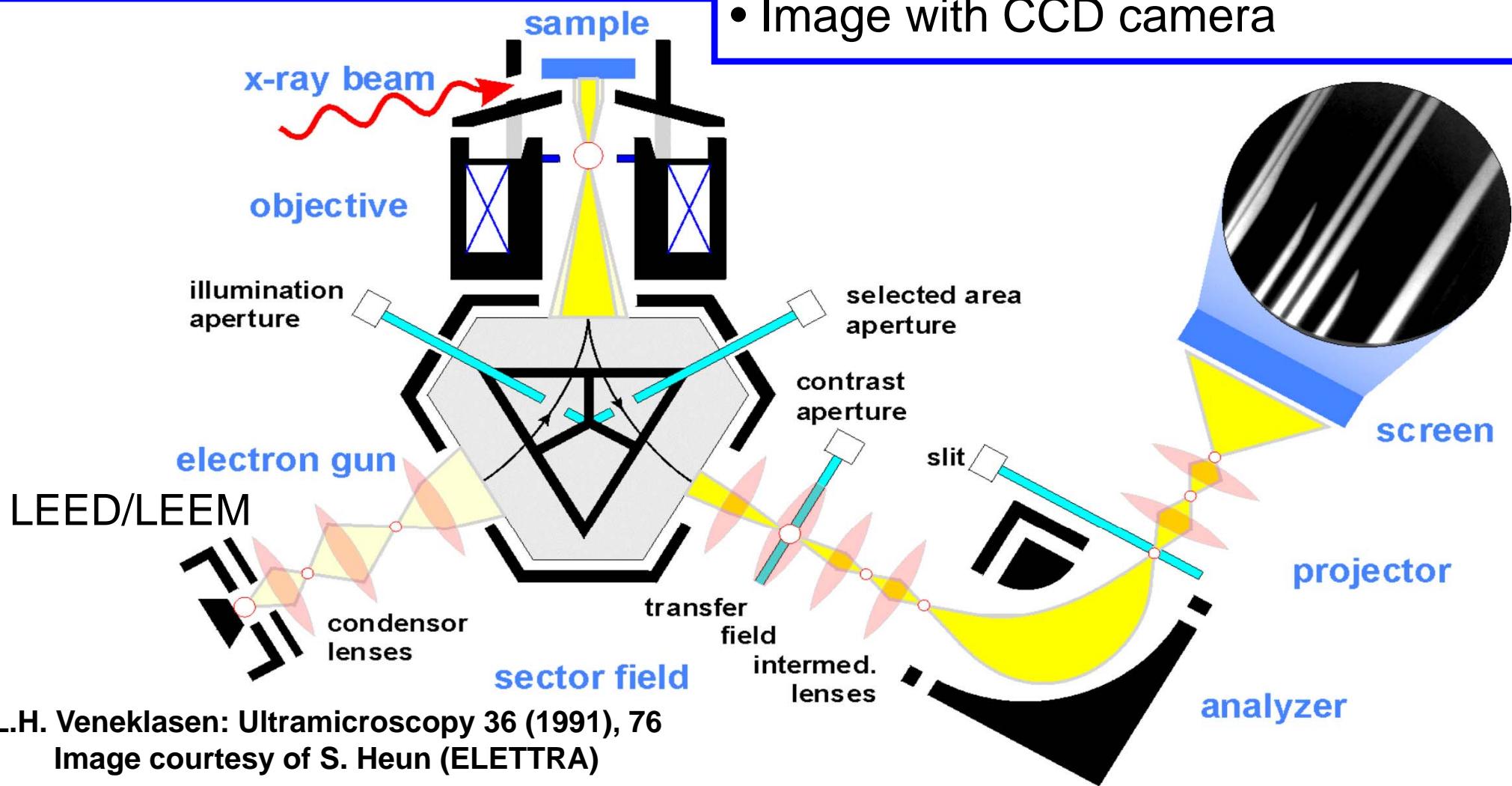
Close-up of the PEEM



The Photoemission Electron Microscope (PEEM)

# Spectromicroscope

- Channel plate: amplifies electrons
- Phosphor screen: converts to light
- Image with CCD camera



L.H. Veneklasen: Ultramicroscopy 36 (1991), 76  
Image courtesy of S. Heun (ELETTRA)

Elmitec Elektronenmikroskopie GmbH

Clausthal-Zellerfeld, Germany

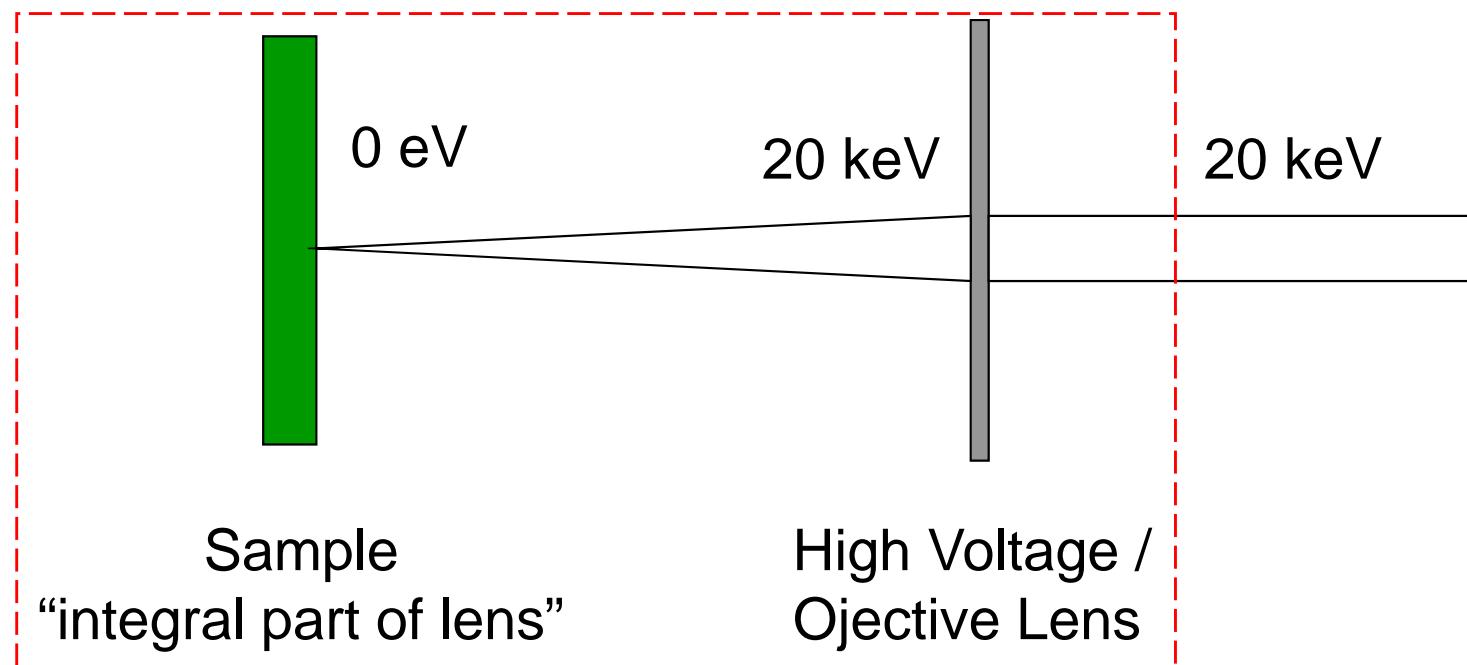
# Slow Electrons

Probe : slow electrons

Imaging: high energy electrons

(more stable and maintain spatial information)

Lens Equivalent has two functions: accelerating field due to potential & focussing function



## High voltage:

- reduced sensitivity to external magnetic fields
- reduced energy spread and smaller electron beam diameters

Immersion lens:

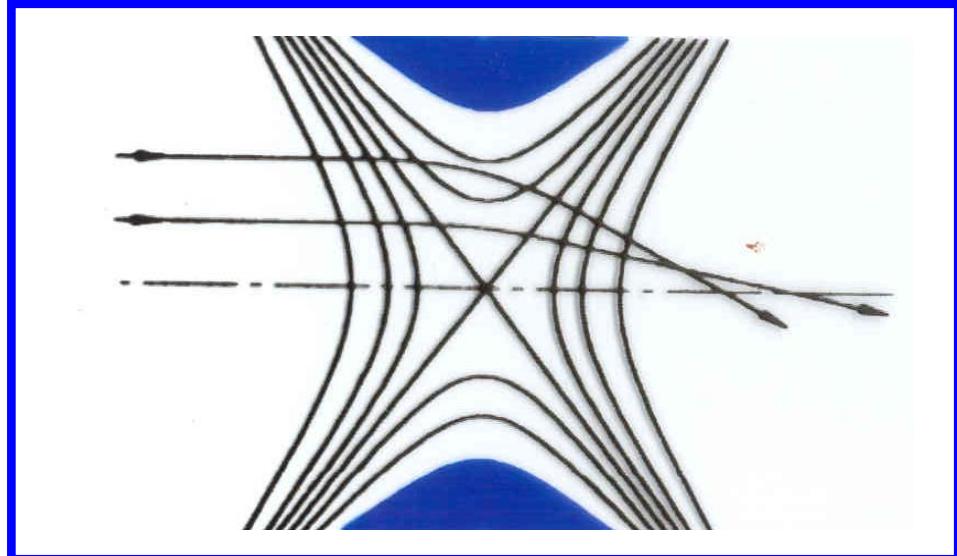
electrons have before and after the lens different velocity (different wavelength)

Cathode lens:

Sample is cathode  
electron microscope is anode

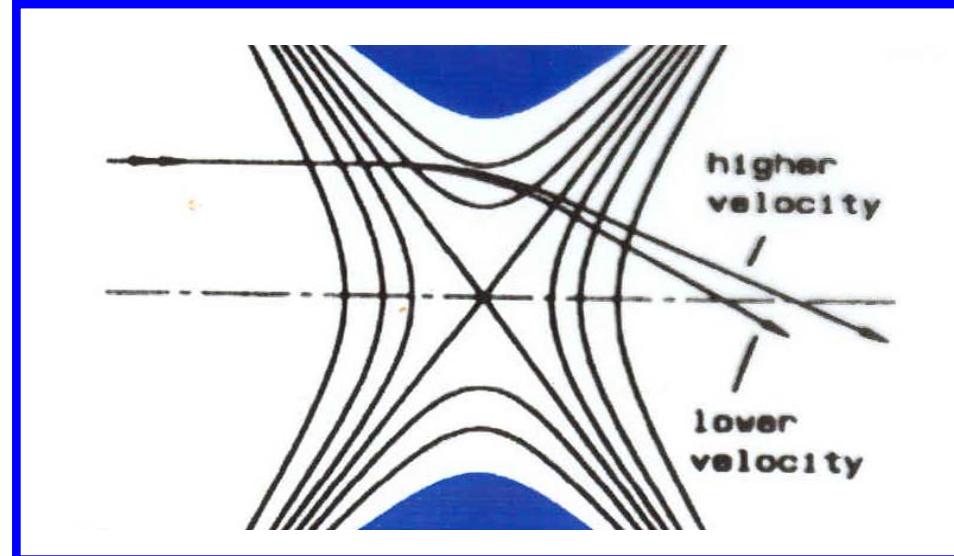
# Two Kinds of Aberrations

## Spherical



Beams parallel away from the lens axis are focused in a slightly different place than beams close to the axis and therefore a blurring of the image.

## Chromatic

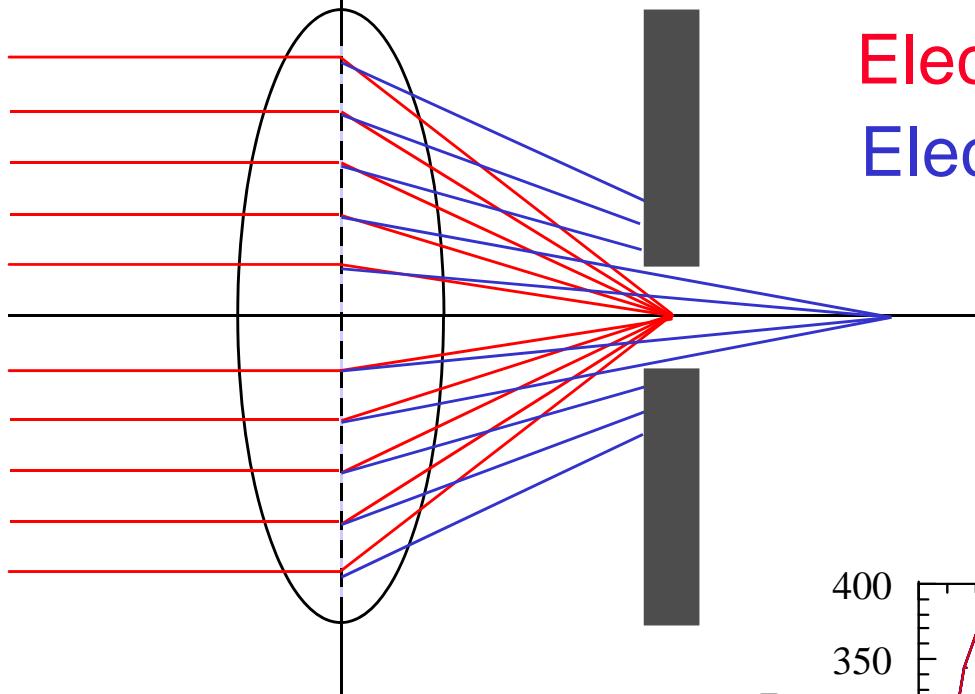


Different wavelengths of light are focused to different positions.

Light → electrons  
Glass → electrostatic/magnetic lenses

# Energy Filter

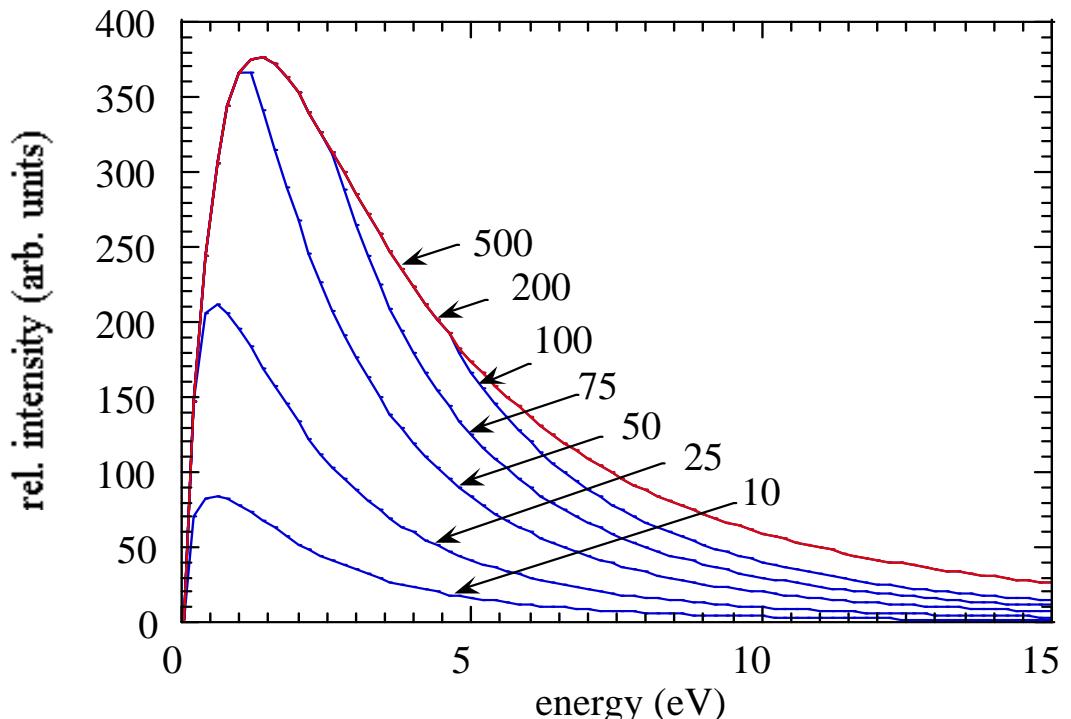
To remove chromatic aberrations:



Electron energy  $E$   
Electron energy  $E + \Delta E$

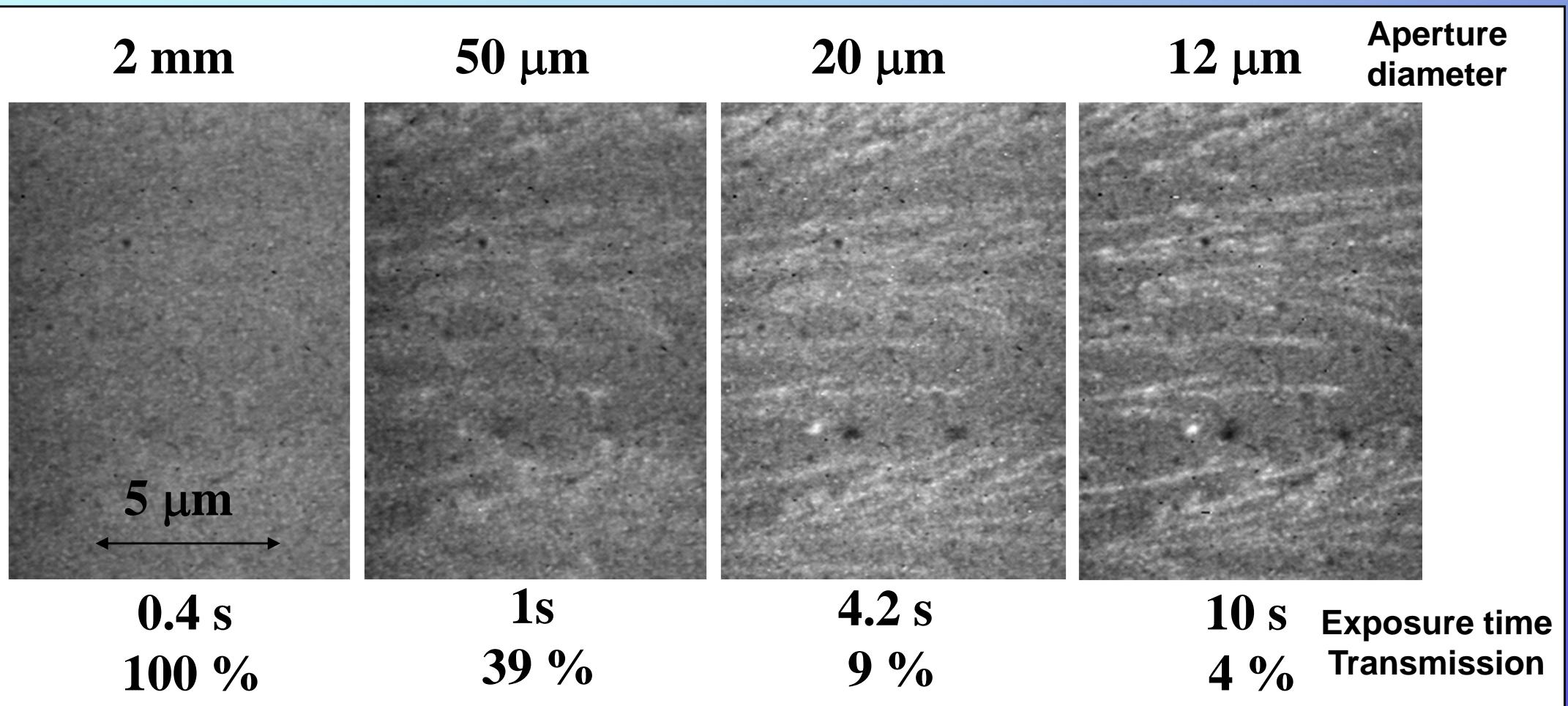
Aperture cuts off  
transmission of electrons  
with higher energy

Energy distribution is narrowed but transmission (intensity) is reduced. Therefore need to find compromise.



## Effect of aperture size on resolution

- Spatial resolution depends on aperture size - limits pencil angle of transmitted electrons and transmission
- Highest resolution is achieved with 12 µm aperture for PEEM2



# Spatial Resolution for Magnetic Imaging

PEEM with X-rays: **50-20 nm** spatial resolution

**Aberration-corrected instruments using an electron mirror:**

SMART (spectromicroscope for all relevant techniques)  
at BESSY II, Berlin, Germany  
collaboration of seven Universities in Germany

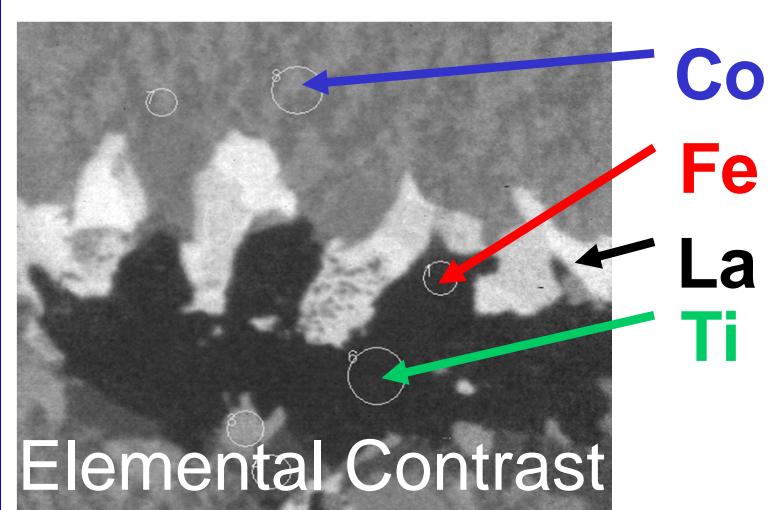
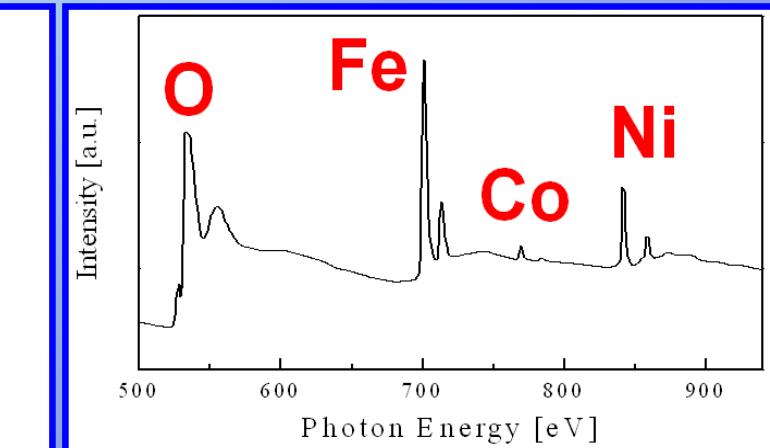
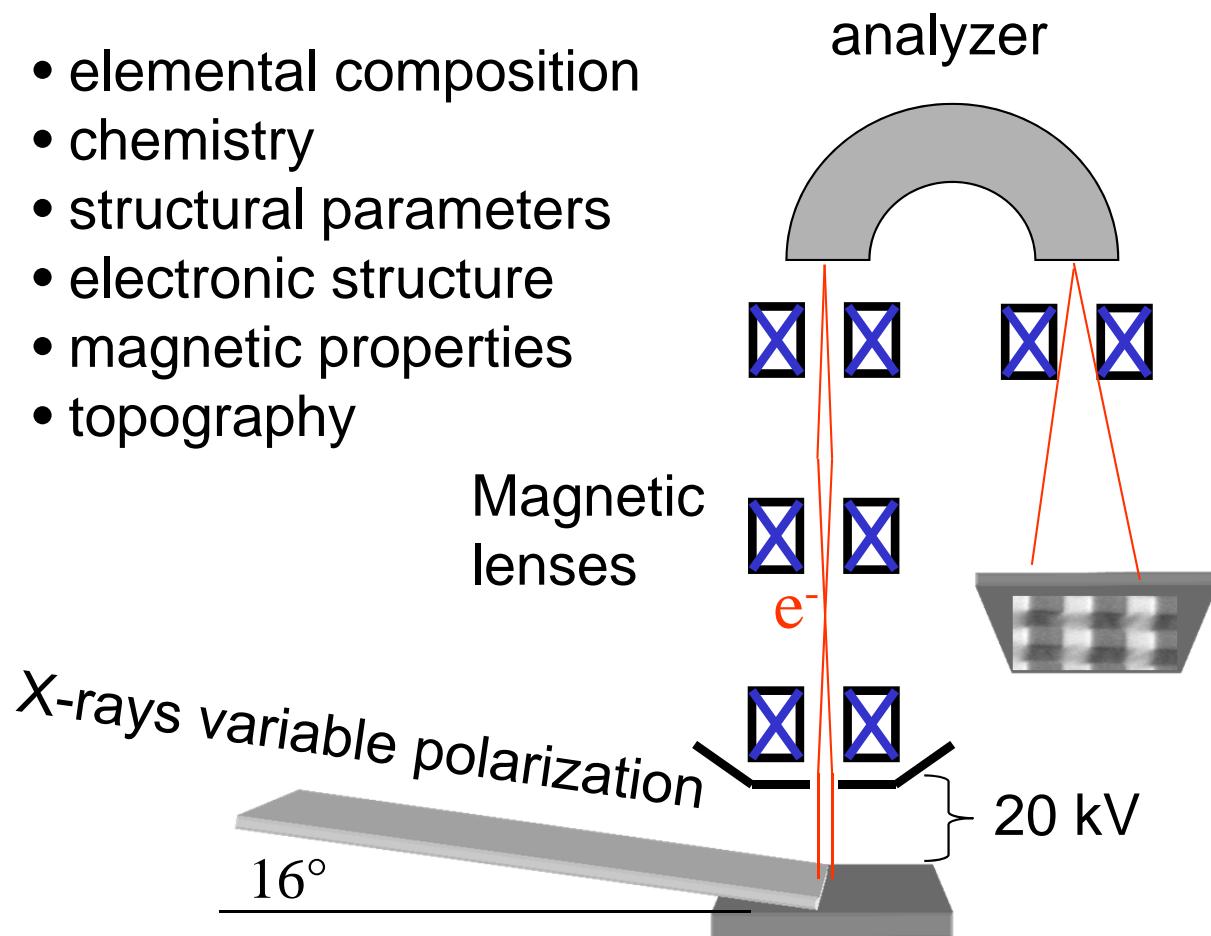
PEEM III  
at ALS, Berkeley, USA  
mainly ALS

**down to a few nm** spatial resolution

# Photoemission Electron Microscope SIM beamline (SLS)

Arantxa Fraile-Rodriguez  
Frithjof Nolting

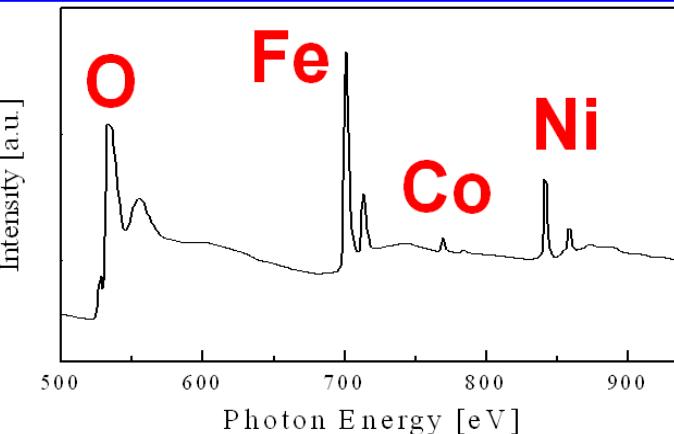
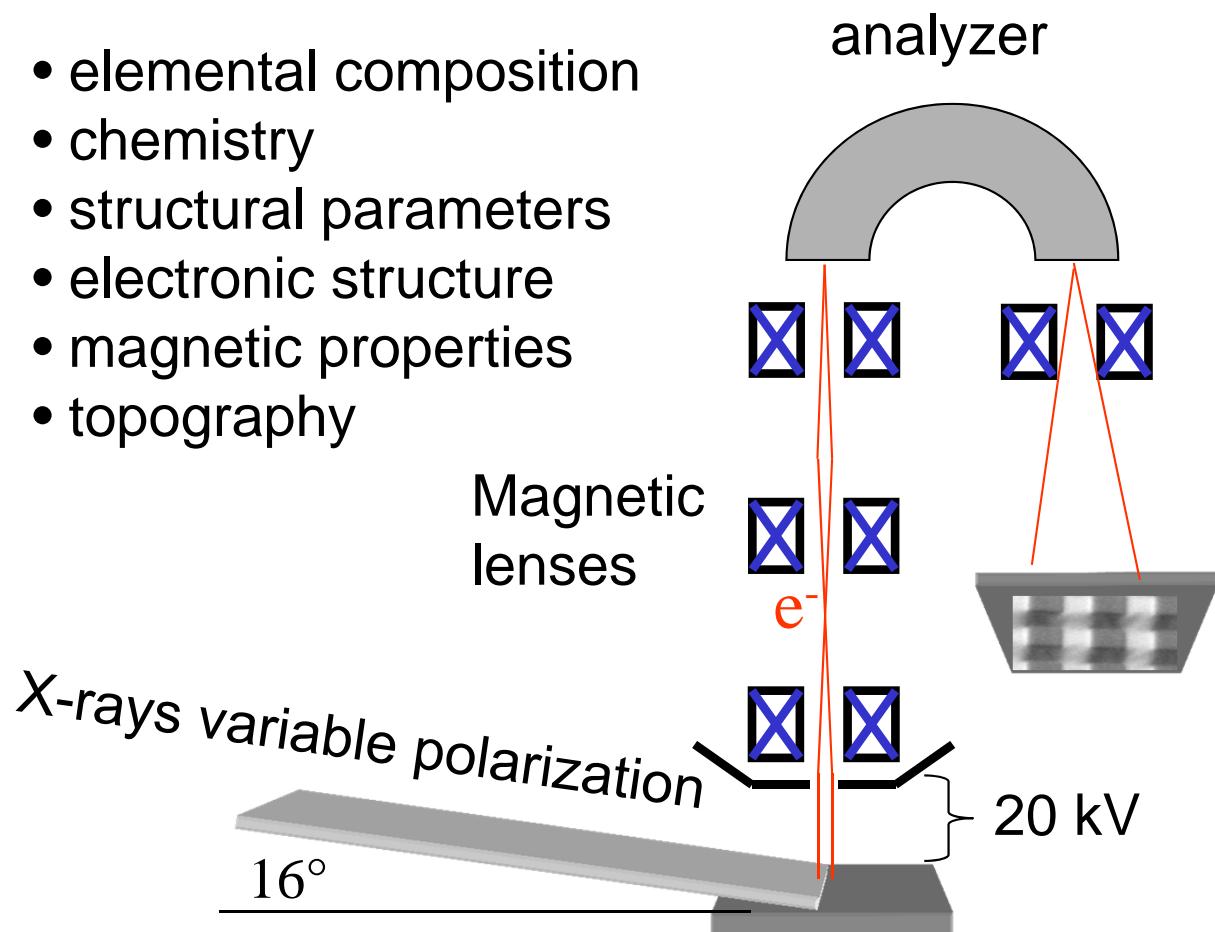
- elemental composition
- chemistry
- structural parameters
- electronic structure
- magnetic properties
- topography



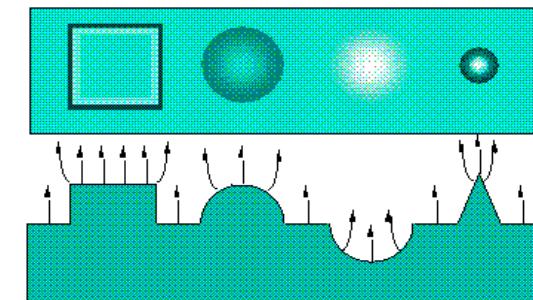
# Photoemission Electron Microscope SIM beamline (SLS)

Arantxa Fraile-Rodriguez  
Frithjof Nolting

- elemental composition
- chemistry
- structural parameters
- electronic structure
- magnetic properties
- topography



## Topographical Contrast

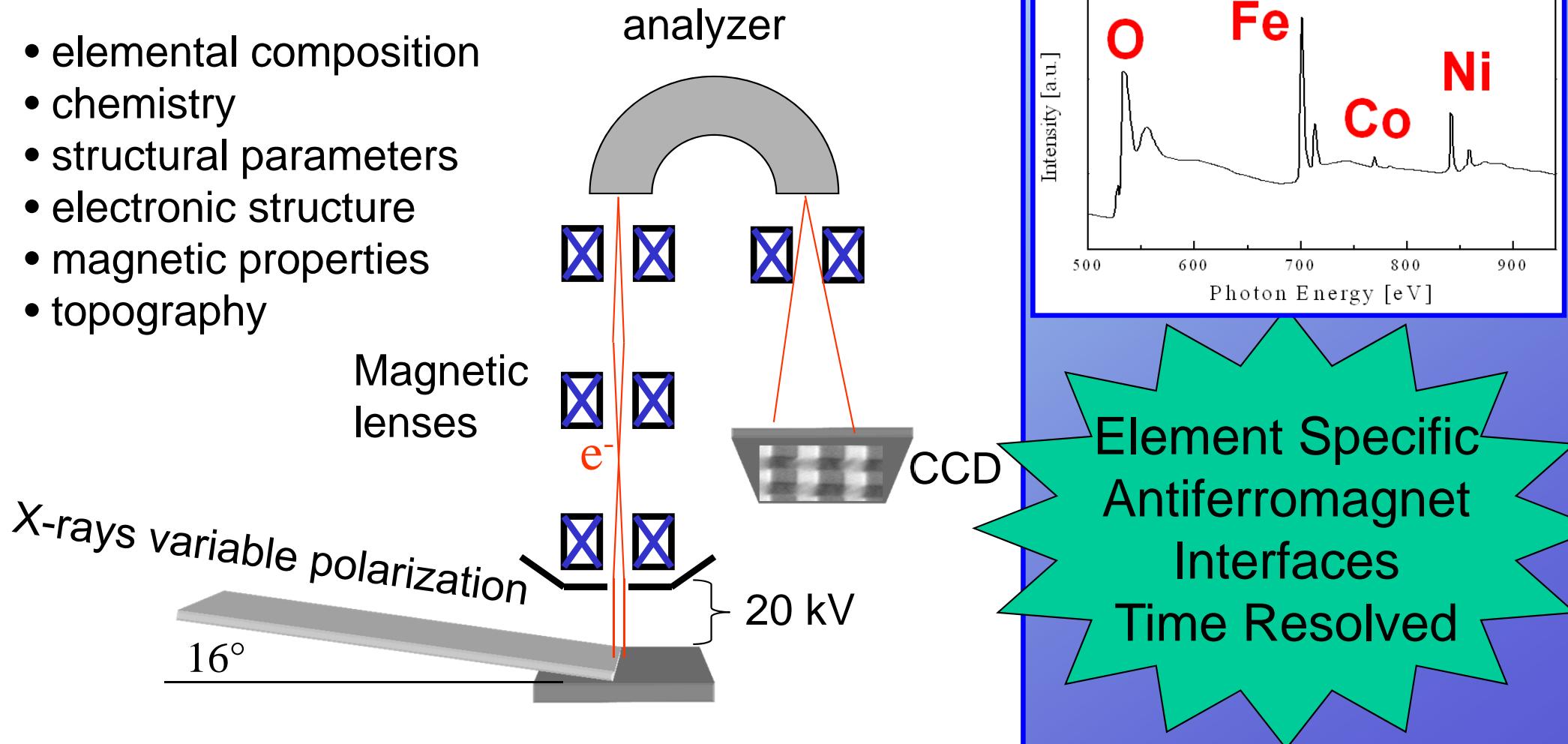


Microfocussing due to distortion  
of the local electric fields

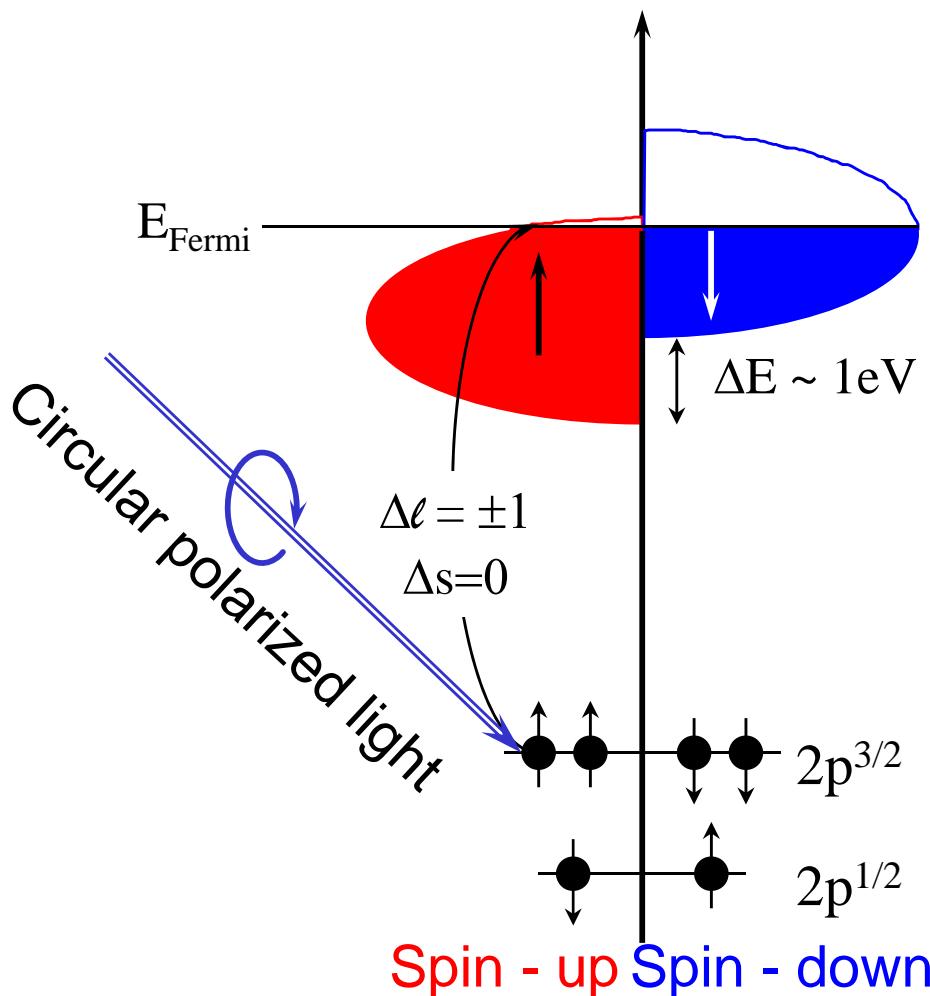
# Photoemission Electron Microscope SIM beamline (SLS)

Arantxa Fraile-Rodriguez  
Frithjof Nolting

- elemental composition
- chemistry
- structural parameters
- electronic structure
- magnetic properties
- topography



# X-Ray Magnetic Circular Dichroism (XMCD)

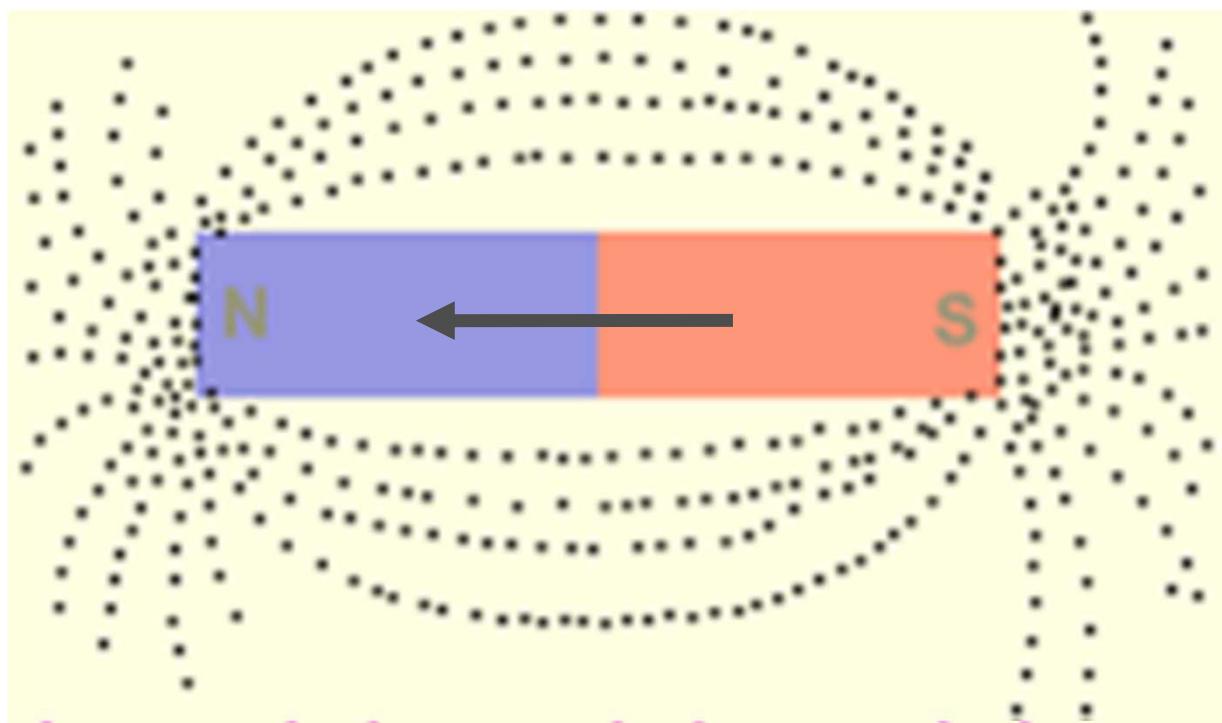


G. Schütz et al. PRL (1987)

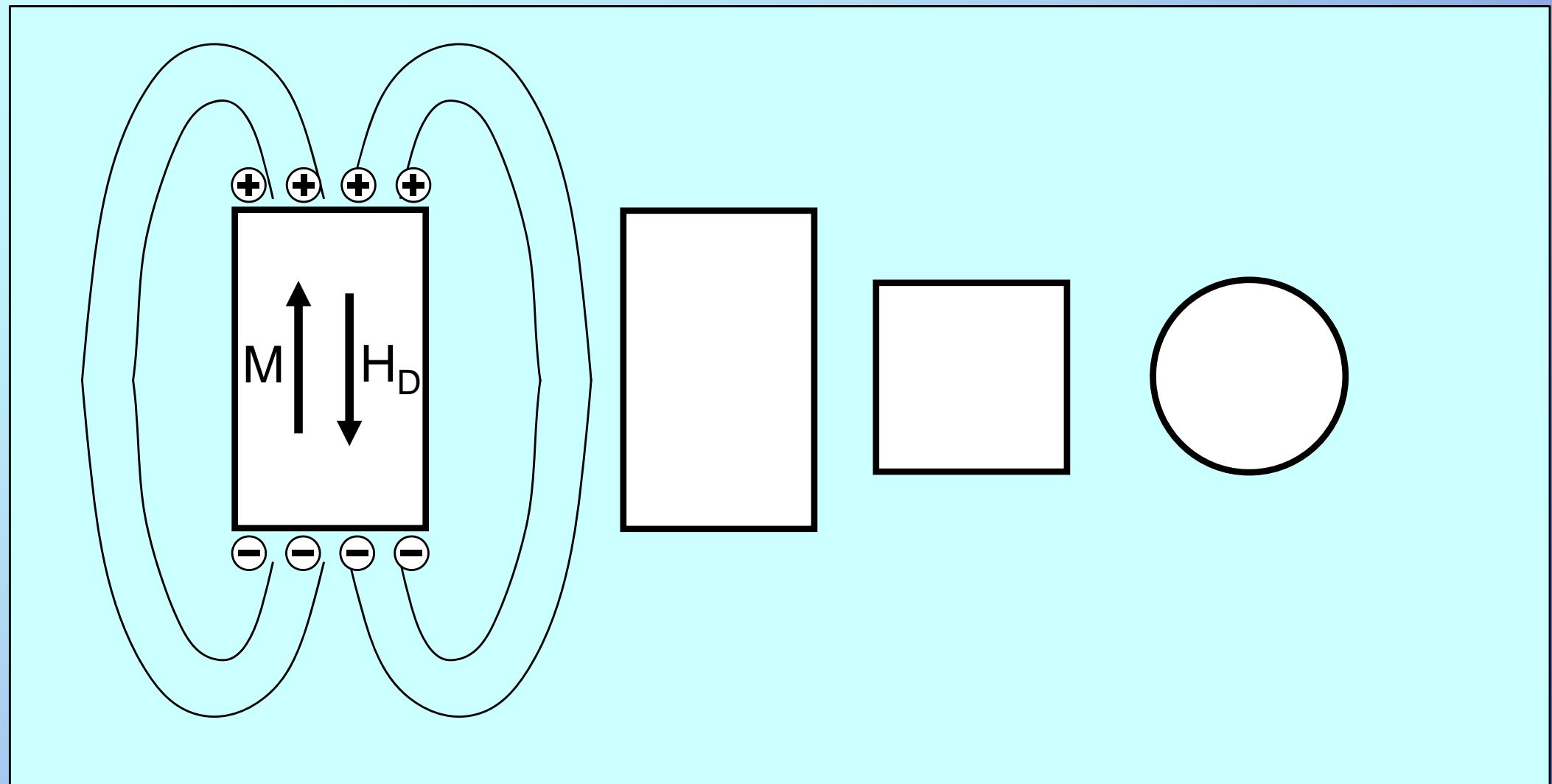
- L-edge absorption in d band transition metal
- Magnetic metal: d valence band split into spin-up and spin-down with different occupation
- Absorption of right/left circular polarisation: light mainly excites spin-up/down photoelectrons
- Spin flips forbidden: measured resonance intensity reflects number of empty d-band states of a given spin
- Can determine sizes and directions of atomic magnetic moment

Time for a  
game. . . .

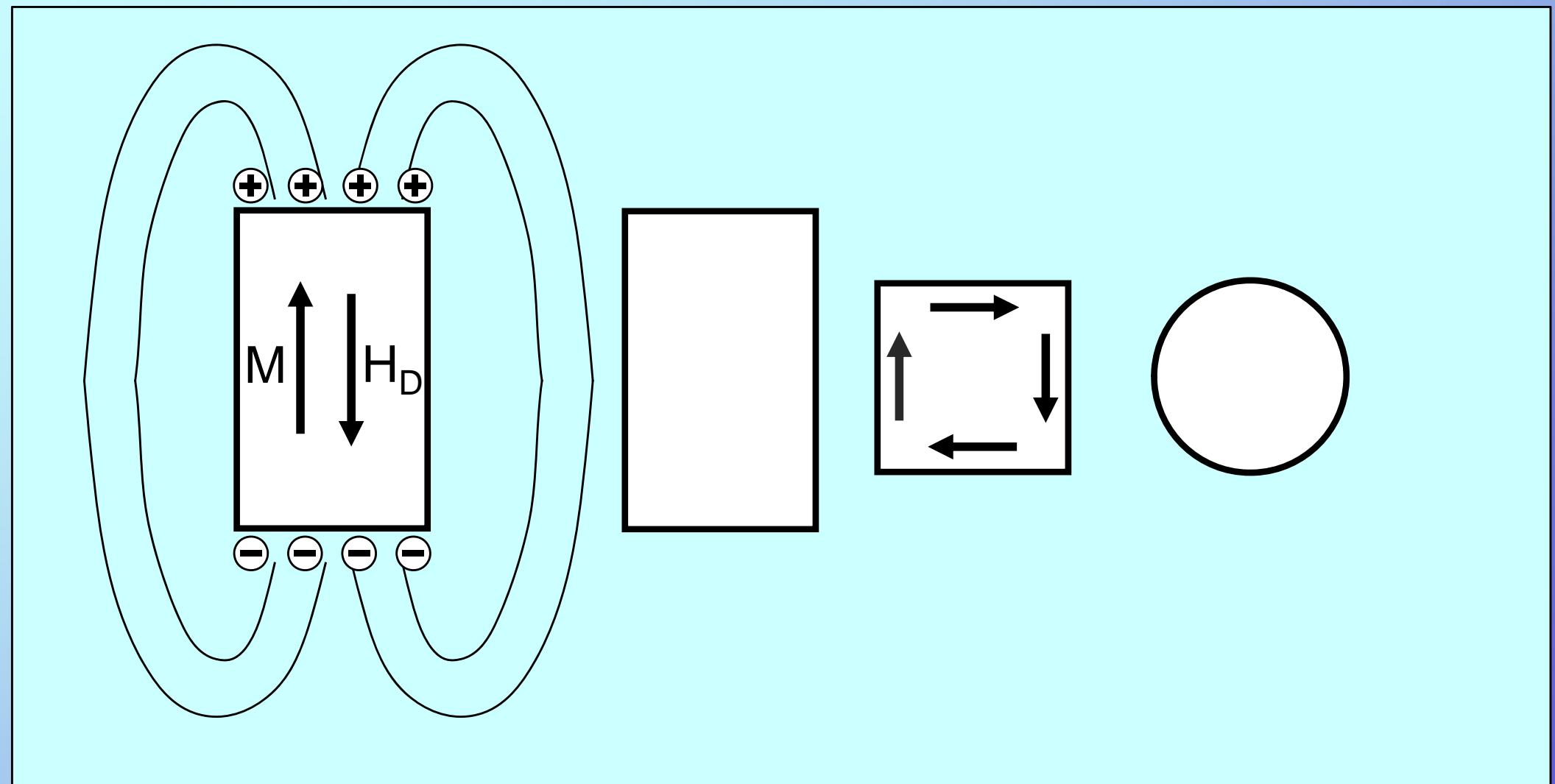
# Magnetostatic or Stray Field Energy



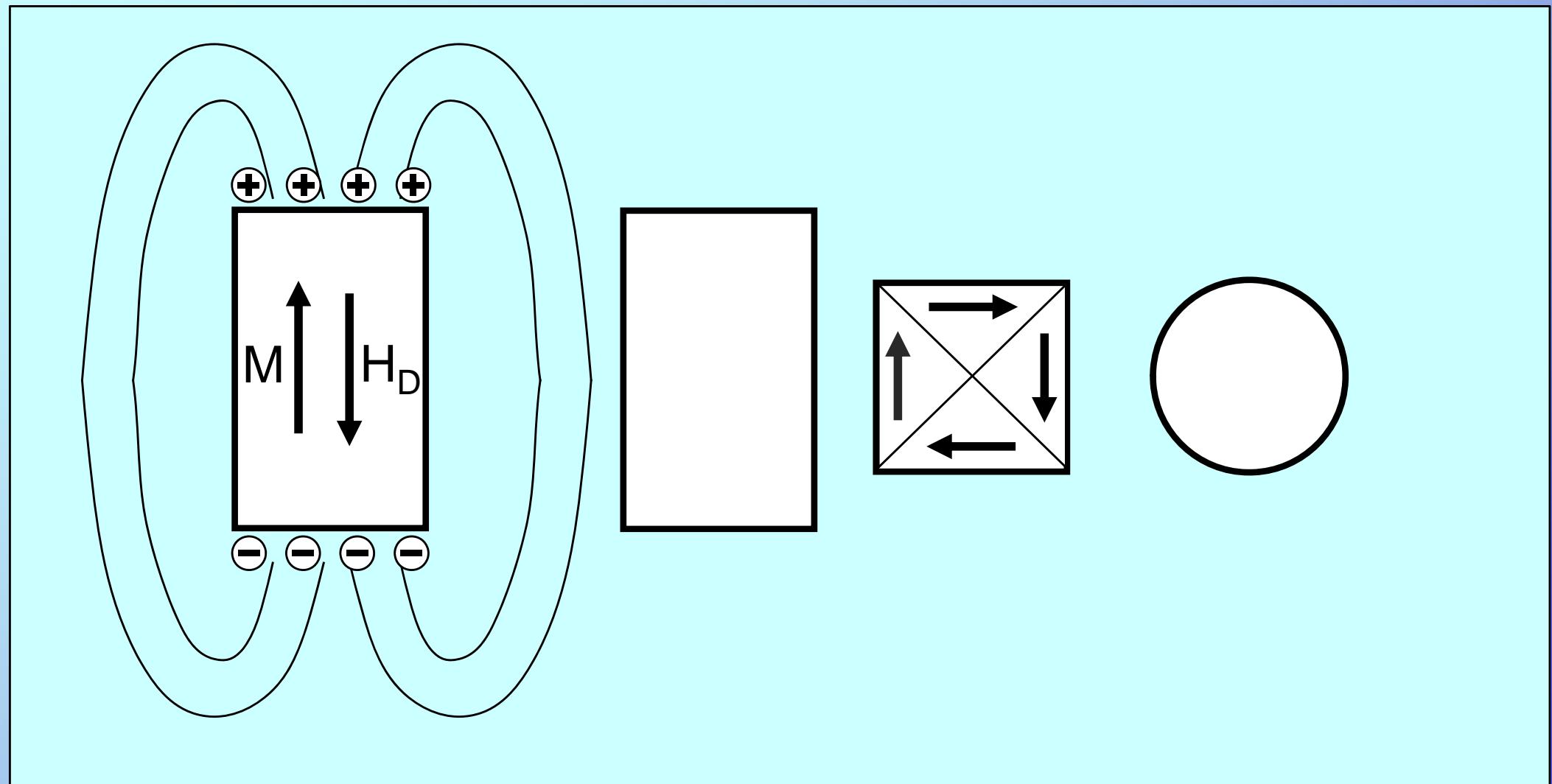
# Rectangle, Square, Disk



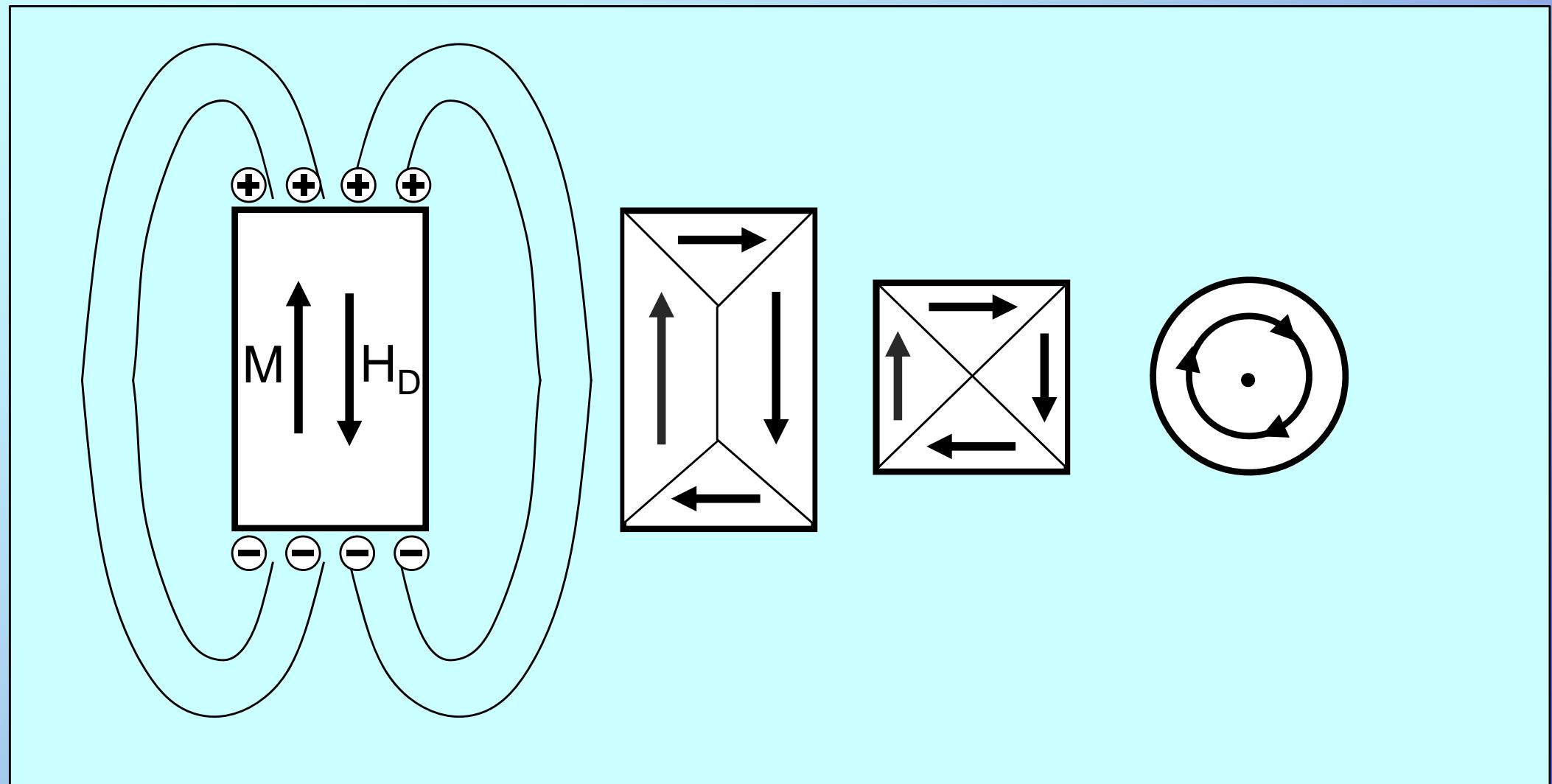
# Rectangle, Square, Disk



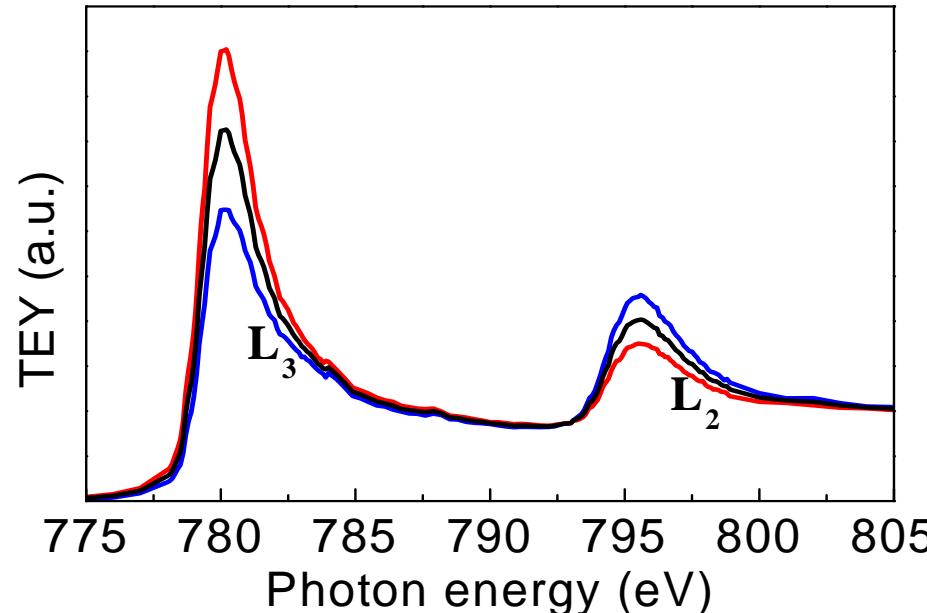
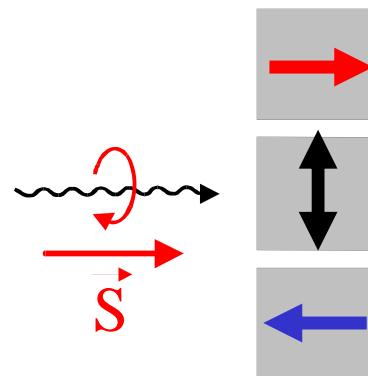
# Rectangle, Square, Disk



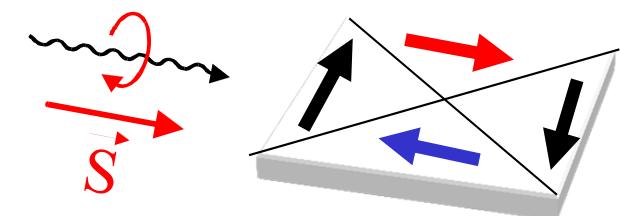
# Rectangle, Square, Disk



# X-ray Magnetic Circular Dichroism (XMCD)

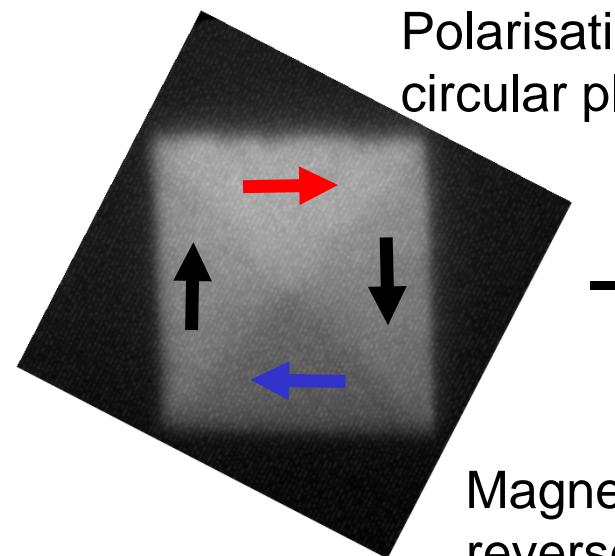


$$\text{XMCD} \sim \mathbf{M} \cos(\mathbf{M}, \mathbf{S})$$

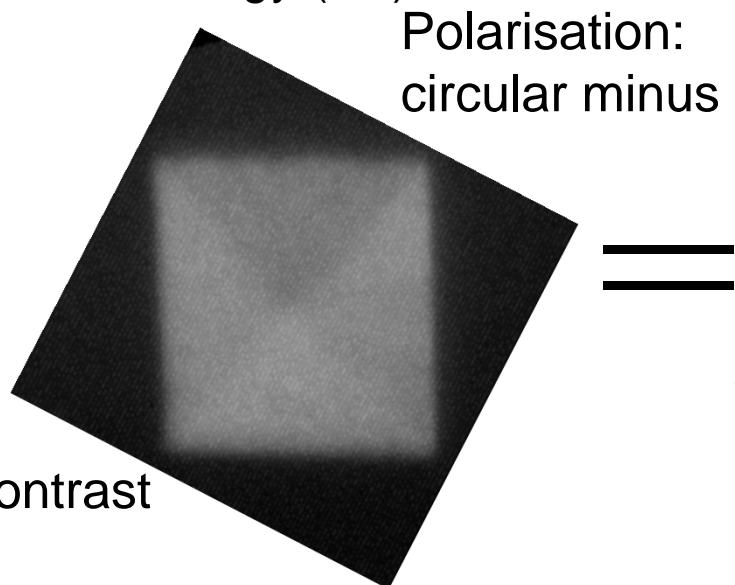


Square Ferromagnetic Element: Landau Domain Pattern

Polarisation:  
circular plus

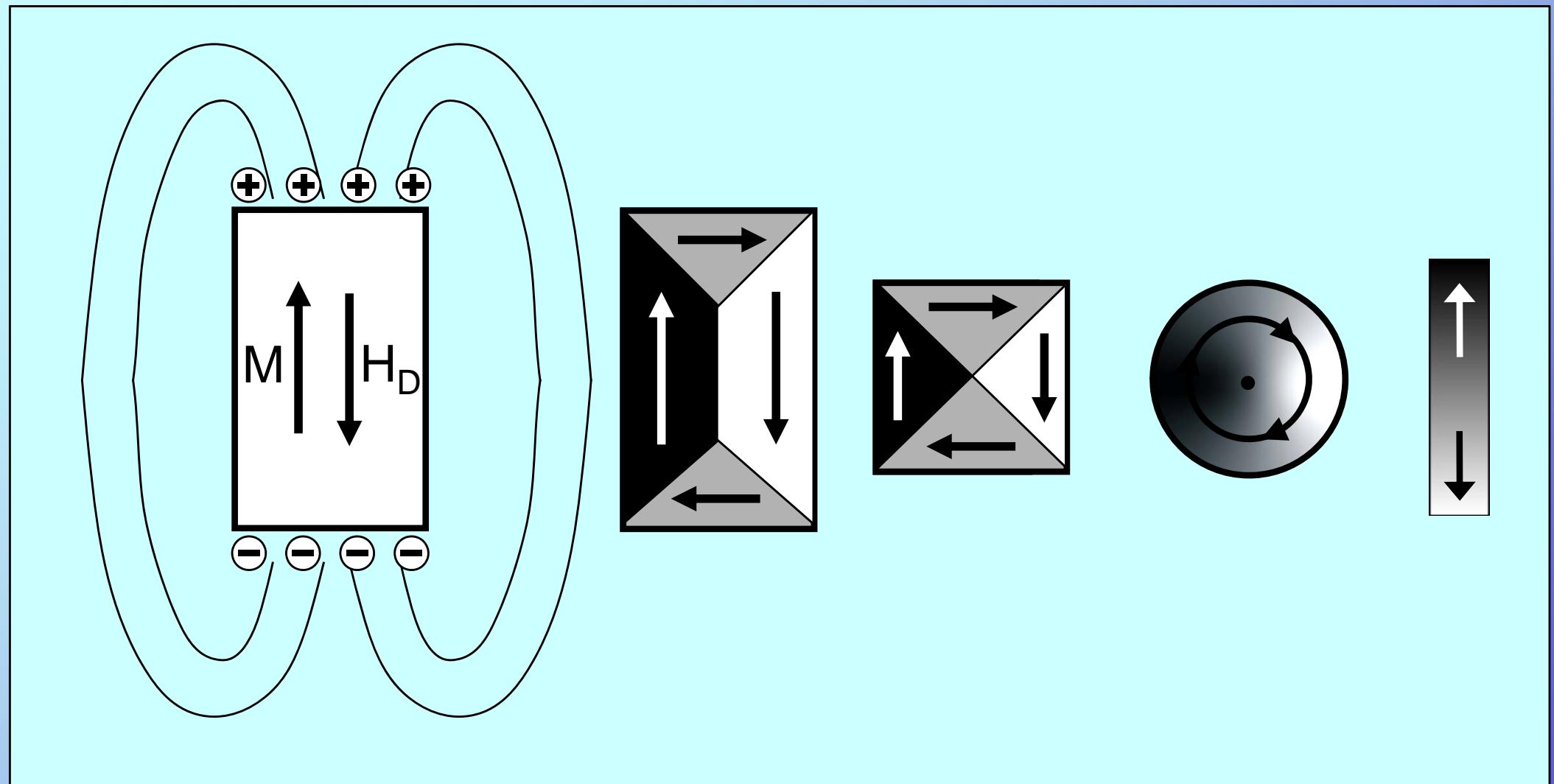


Polarisation:  
circular minus

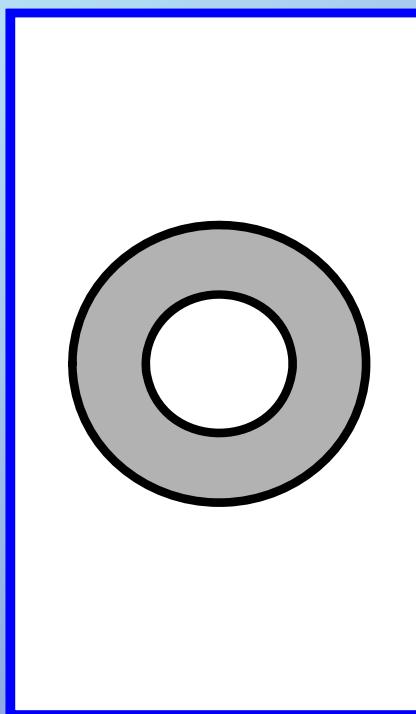


Magnetic contrast  
reverses

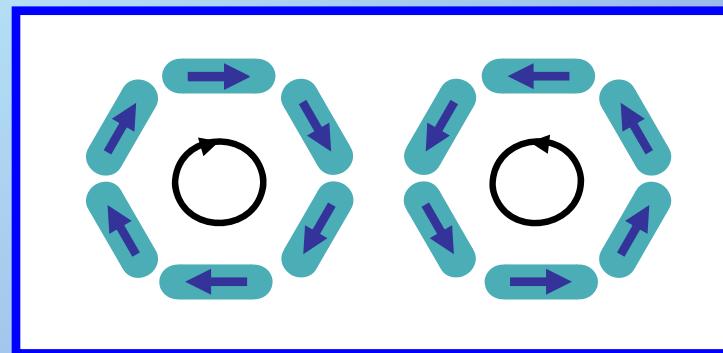
# Rectangle, Square, Disk



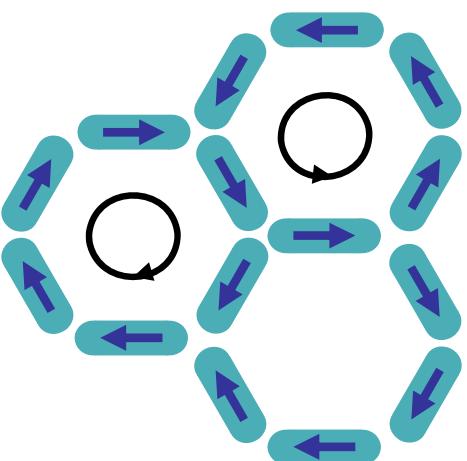
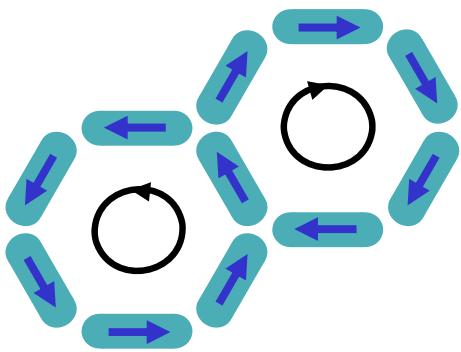
Ring



# Ring of Nanomagnets



# Rings of Nanomagnets

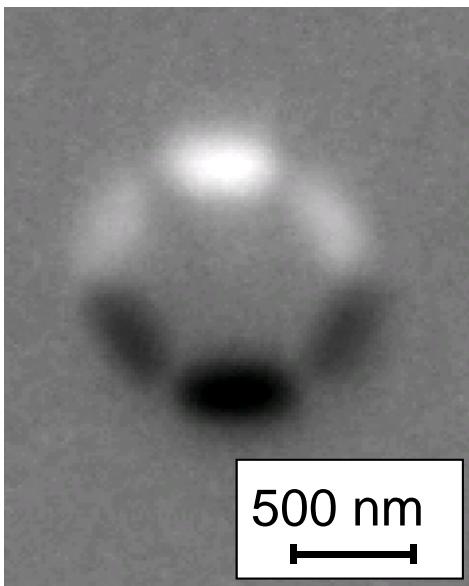


?

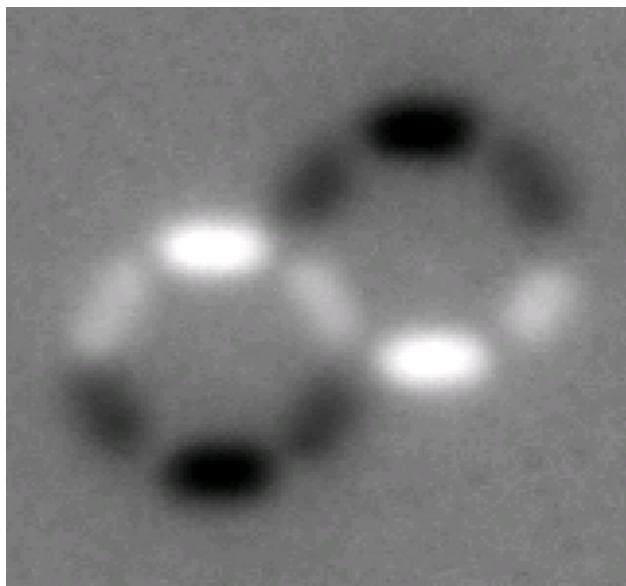


# Artificial Spin Ice in PEEM

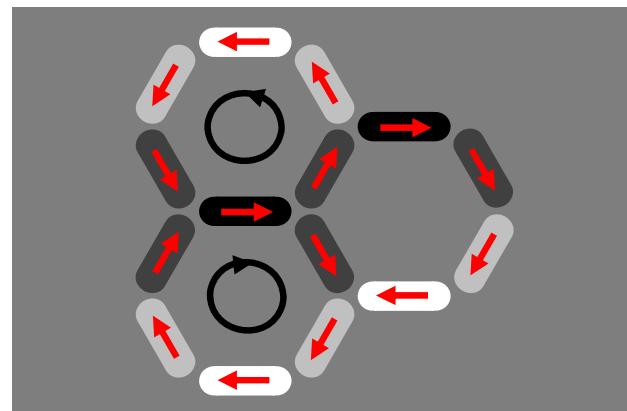
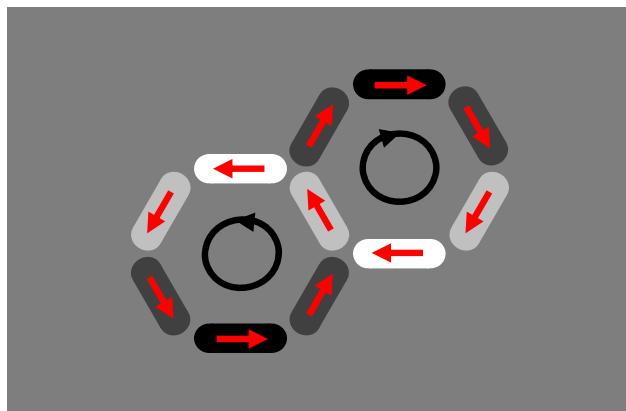
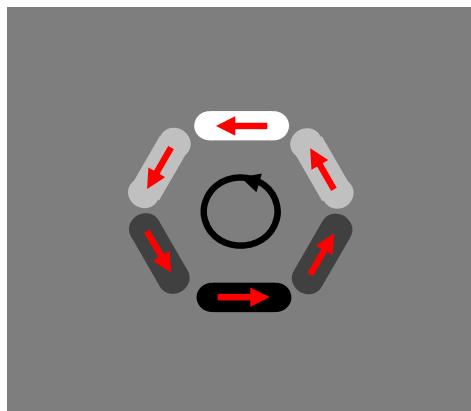
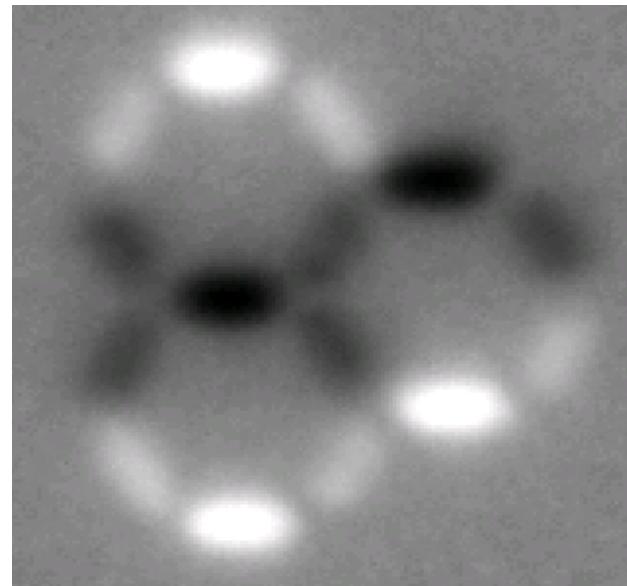
One ring



Two rings



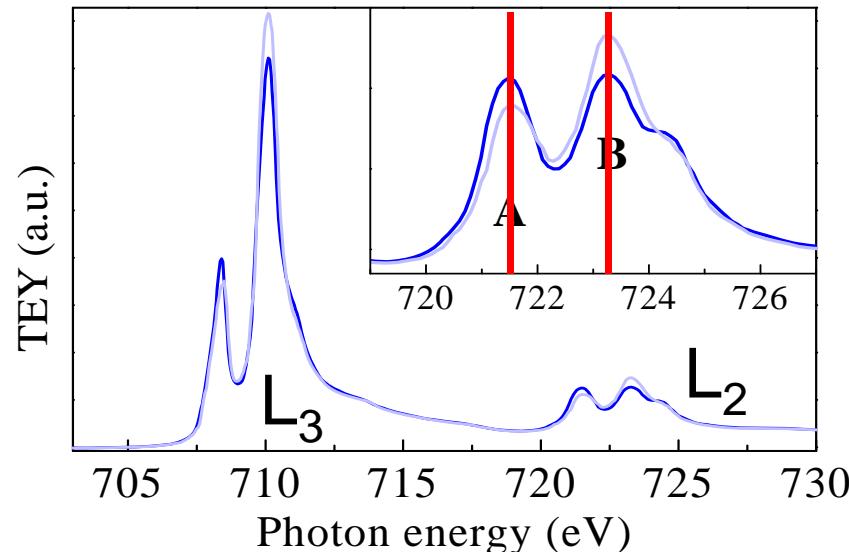
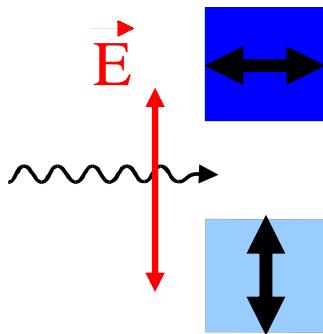
Three rings



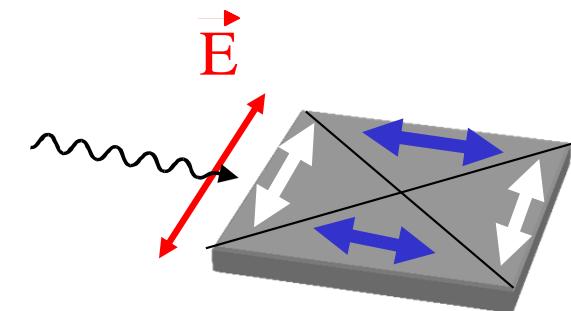
X-ray direction



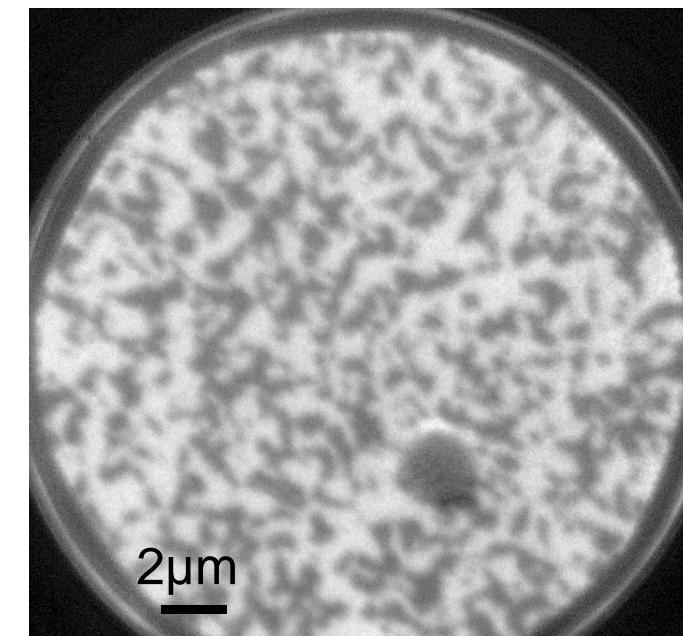
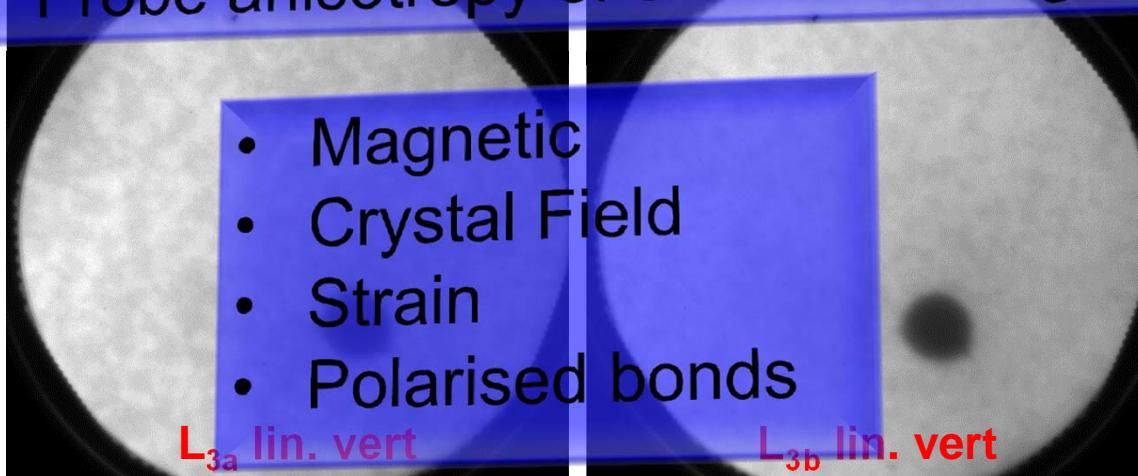
# X-ray Magnetic Linear Dichroism (XMLD)



$$\text{XMLD} \sim \langle \mathbf{M}^2 \rangle \cos^2(\mathbf{M}, \mathbf{E})$$

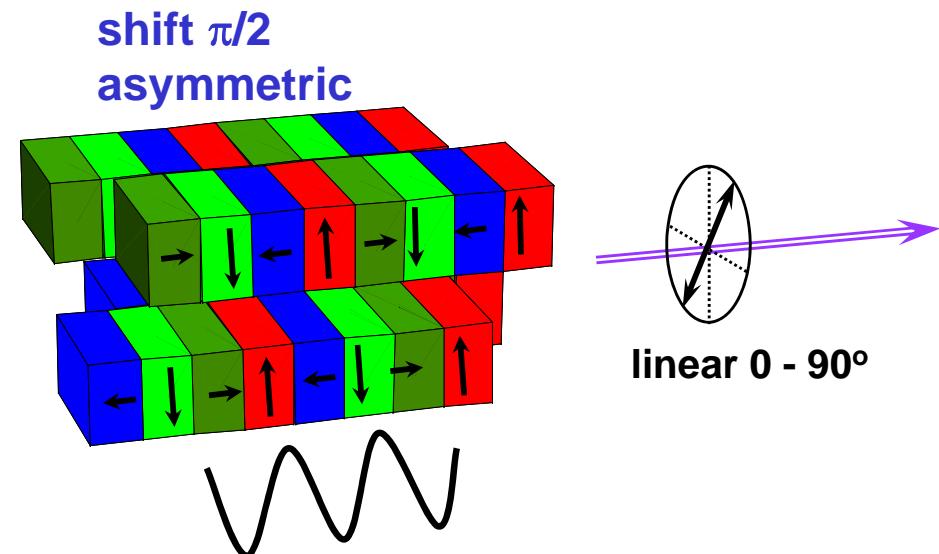
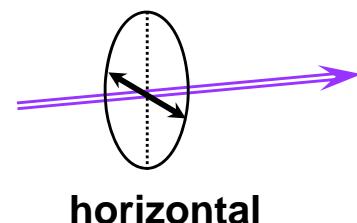
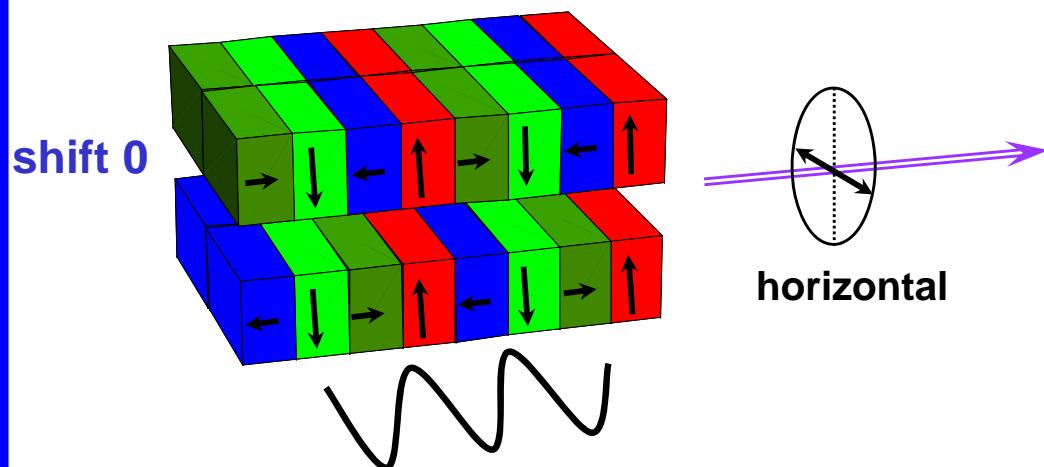
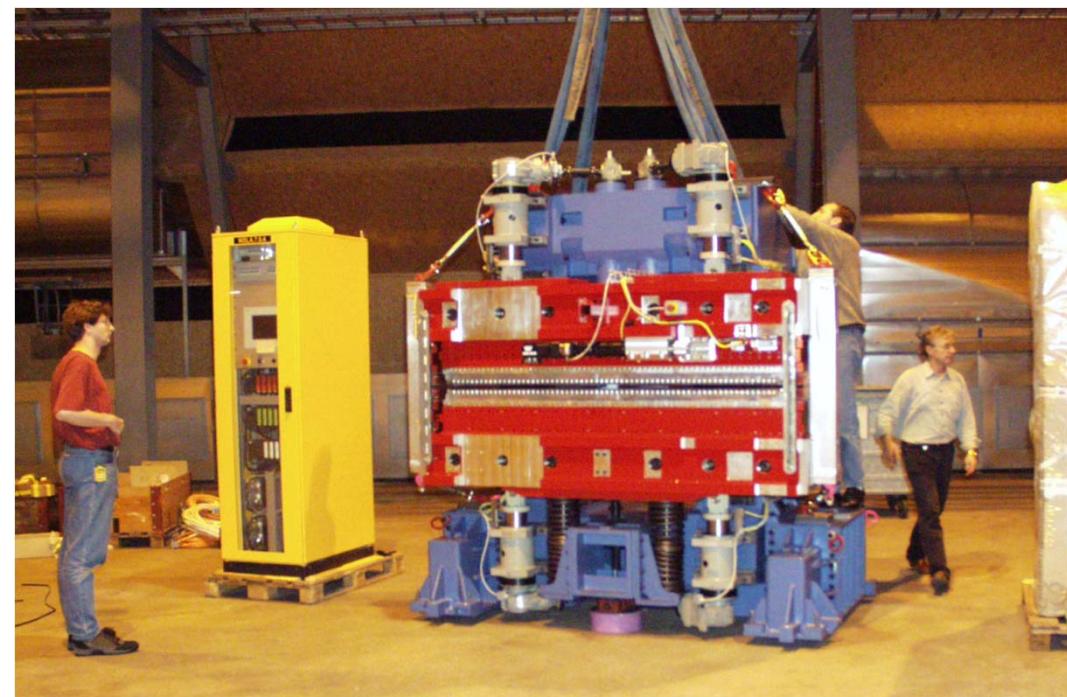
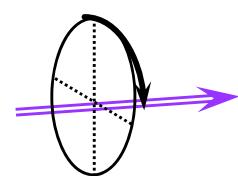
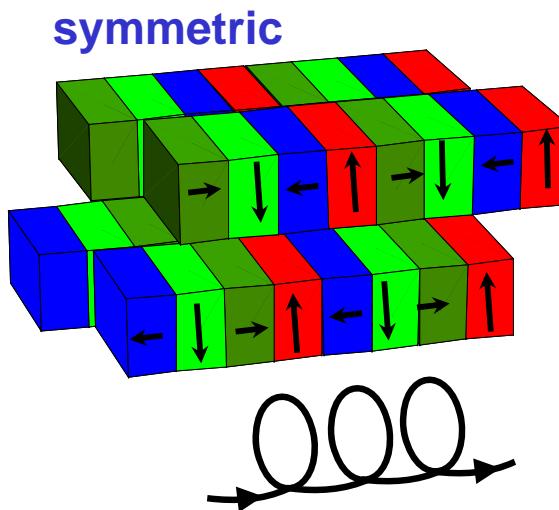


"Probe anisotropy of electron charge"

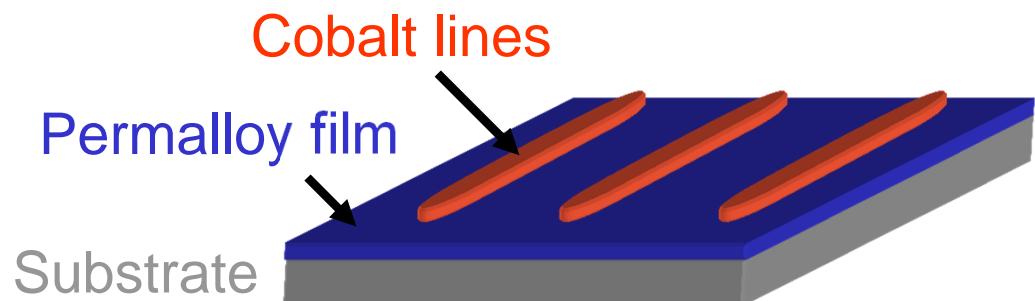
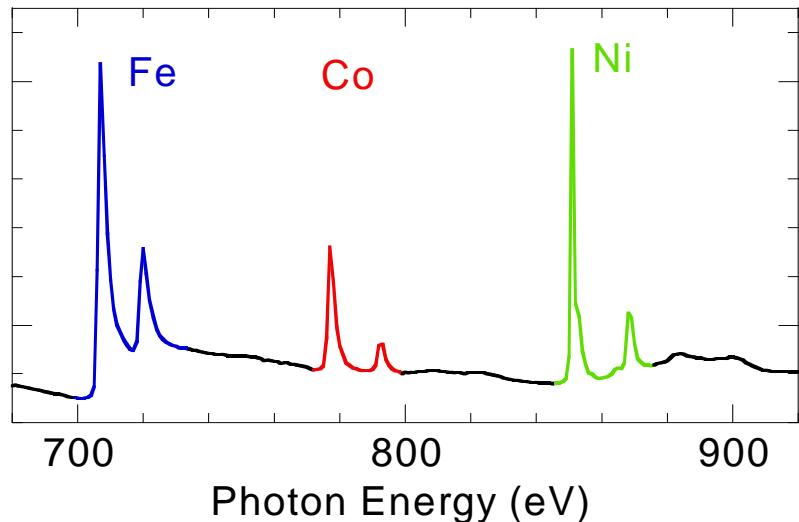


# Undulator

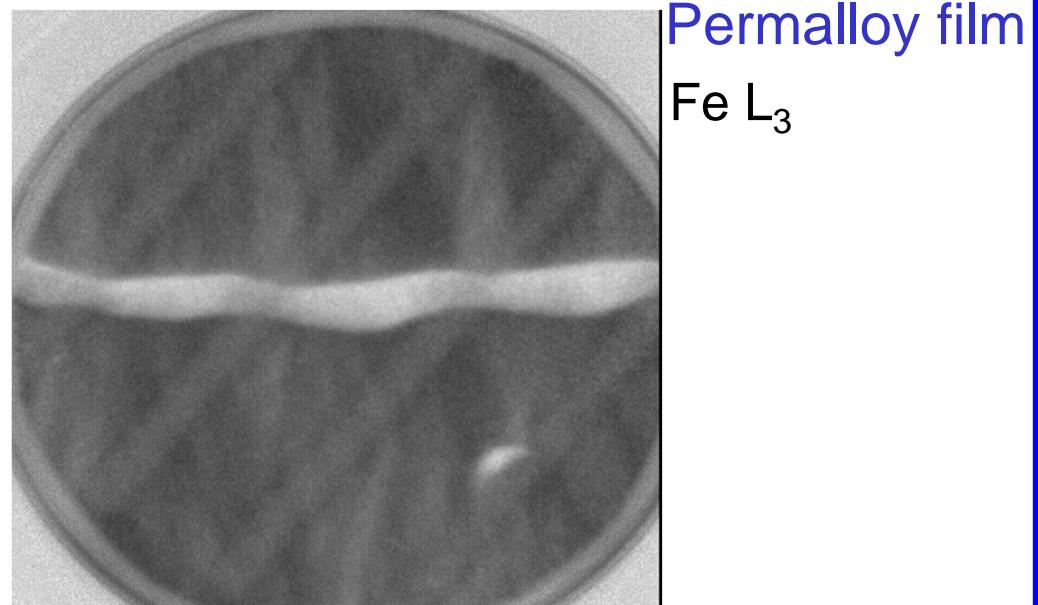
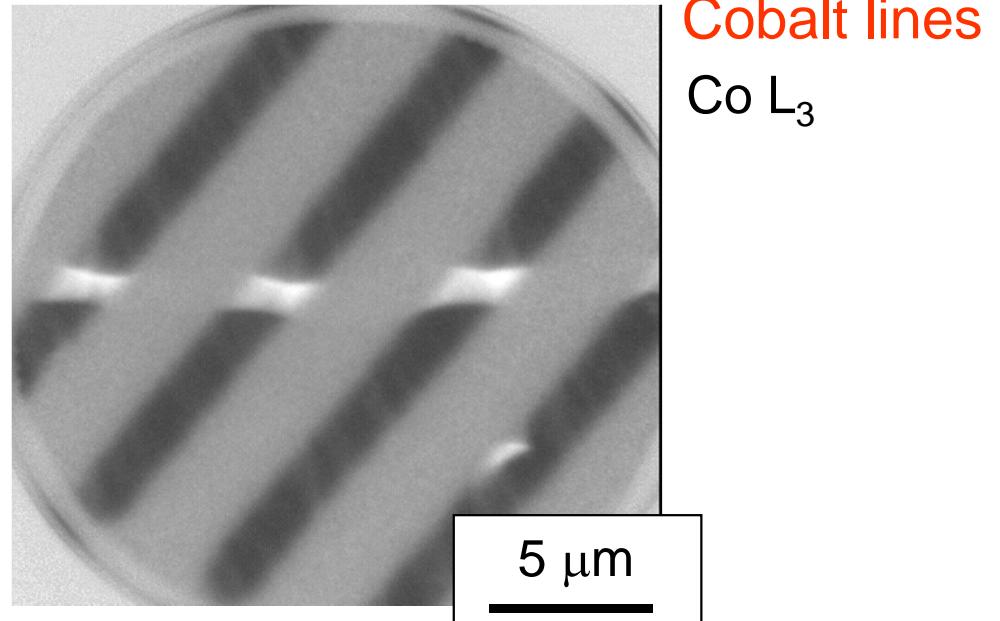
Magnetic Structure:  
changing phase, changes  
polarisation



# Element specific contrast



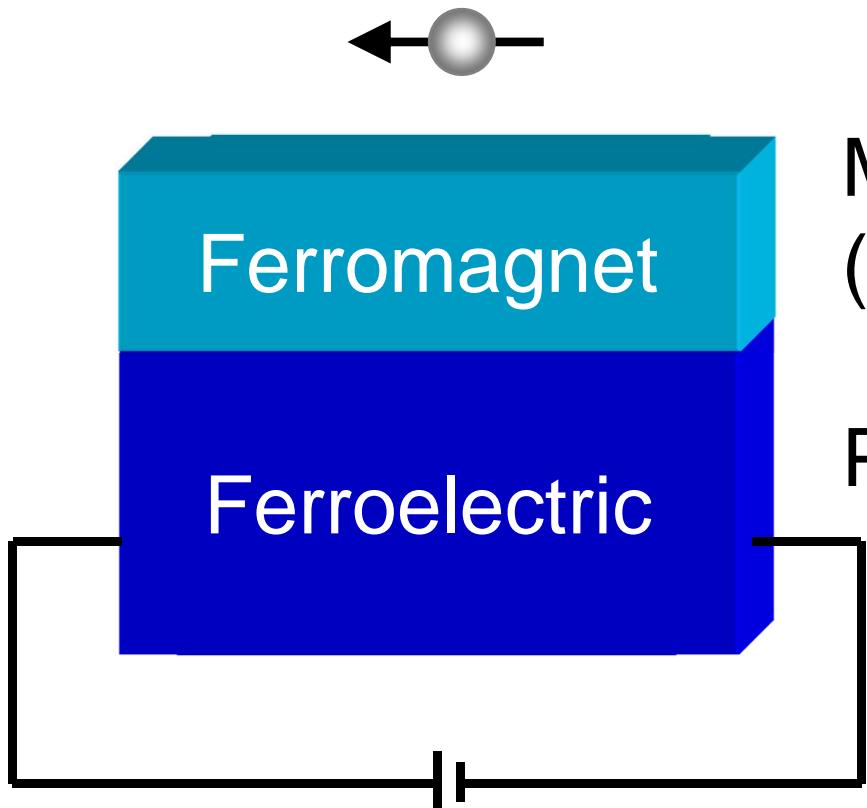
Coupling of hard and soft magnetic layer:  
L. Heyderman, A. Fraile-Rodriguez, A. Hoffmann



Cobalt lines  
 $\text{Co L}_3$

Permalloy film  
 $\text{Fe L}_3$

# Multiferroic Composites



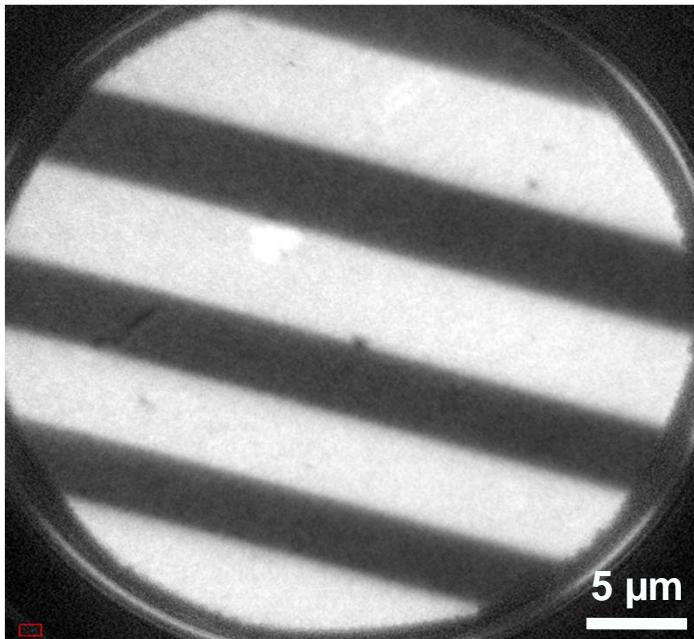
Magnetostrictive:  $\text{CoFe}_2\text{O}_4$   
(Ferrimagnetic Spinel)

Piezoelectric:  $\text{BaTiO}_3$

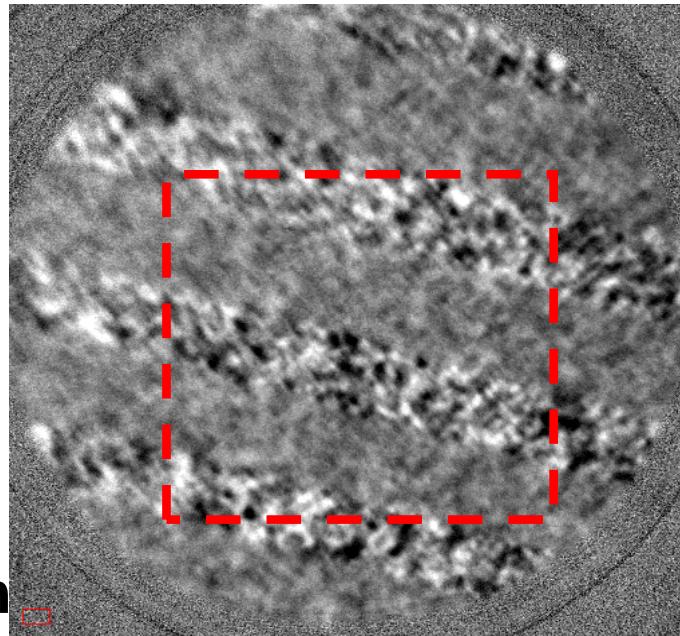
→ Coupling via strain

# Strain and Magnetic Domains

37



Fe X-ray Linear Dichroism



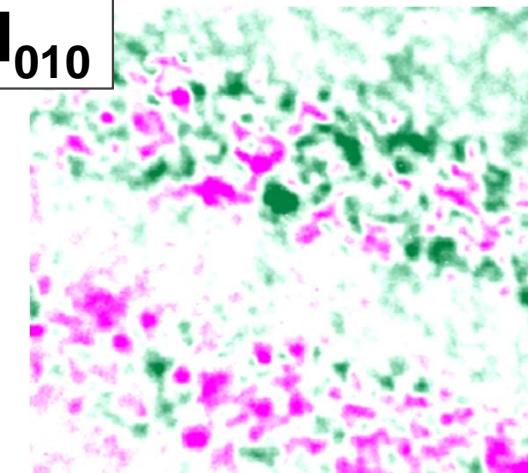
Fe X-ray Magnetic Circular Dichroism



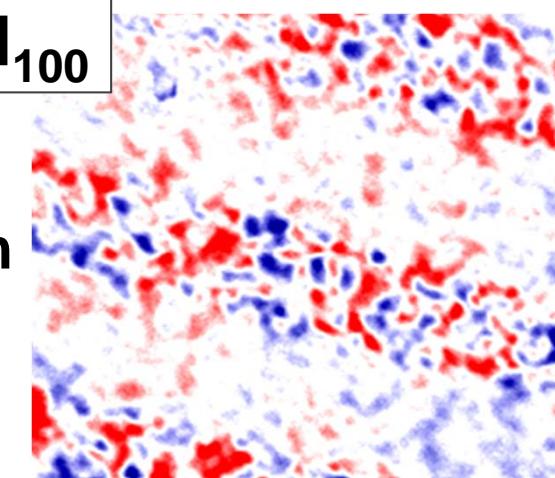
- Fe XLD: strain domains
- Fe XMCD: magnetic domains
- Substrate-induced strain strongly modifies magnetic anisotropy

RV Chopdekar et al. PRB (2012)

$\mathbf{M}_{010}$

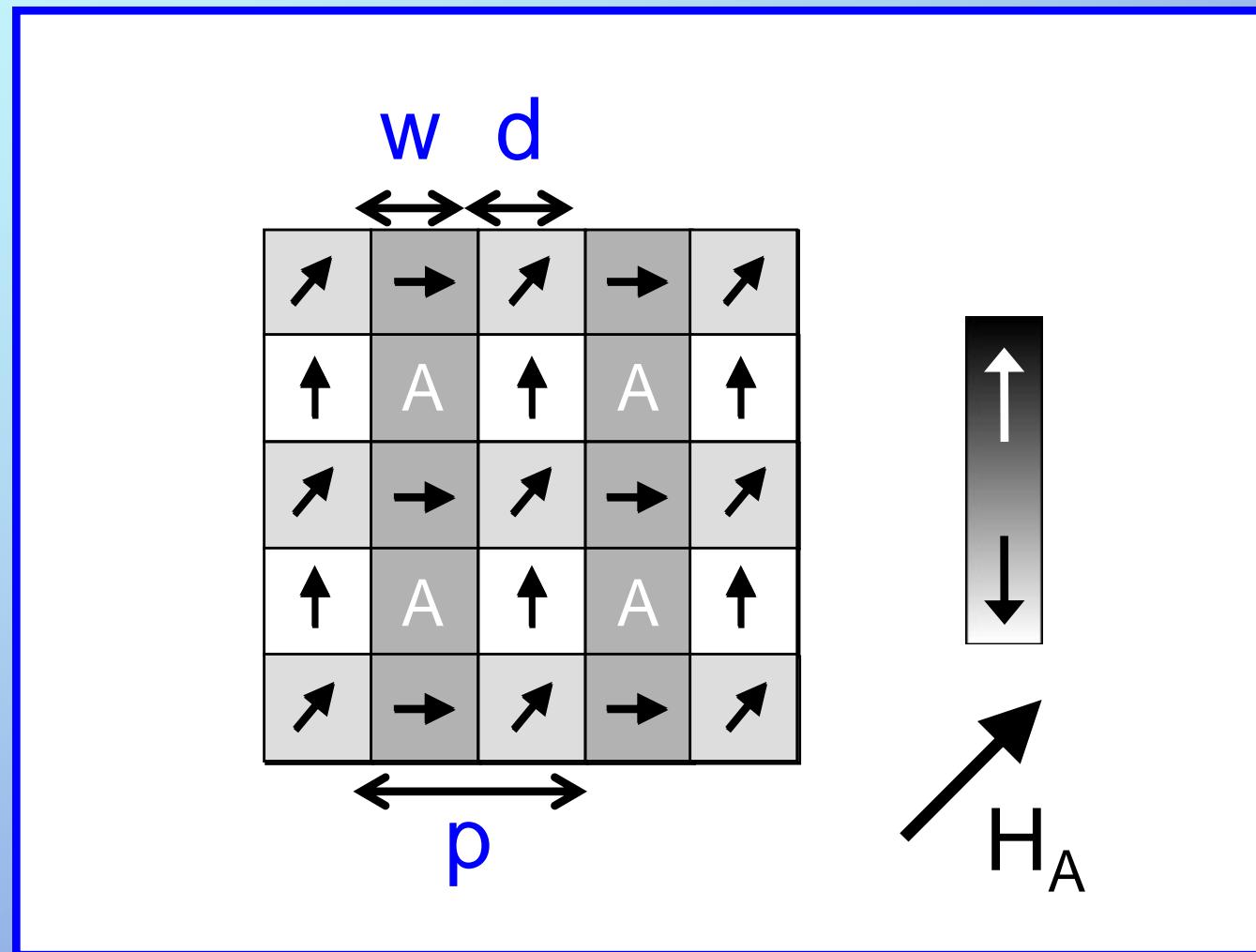


$\mathbf{M}_{100}$



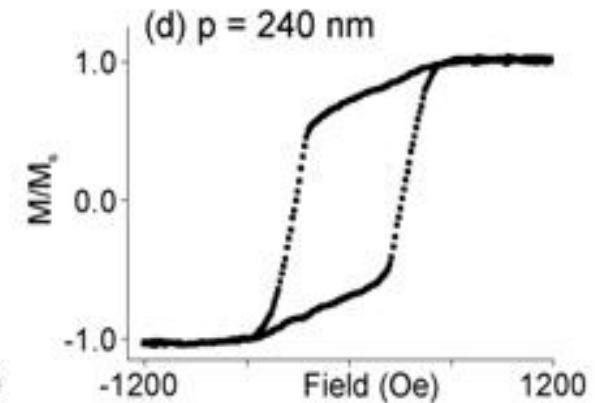
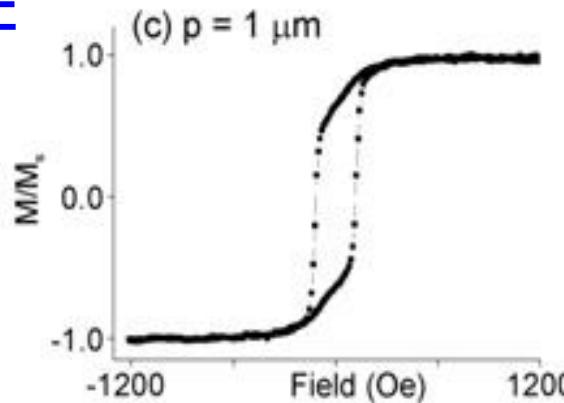
$\mathbf{M}_{001} \sim 0$

# Antidot Arrays – Basic Domain Configuration



# Remanent Hysteresis Loop in Antidot Arrays

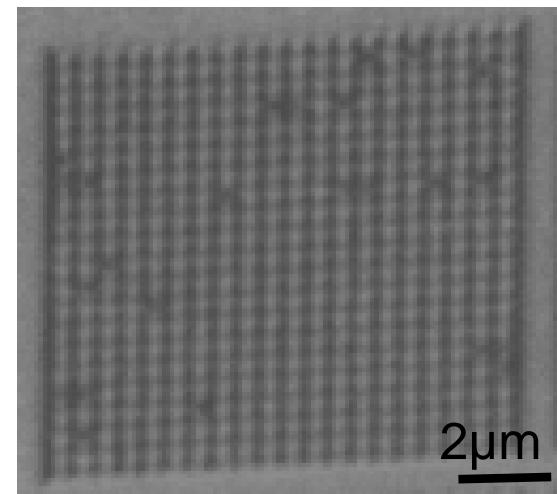
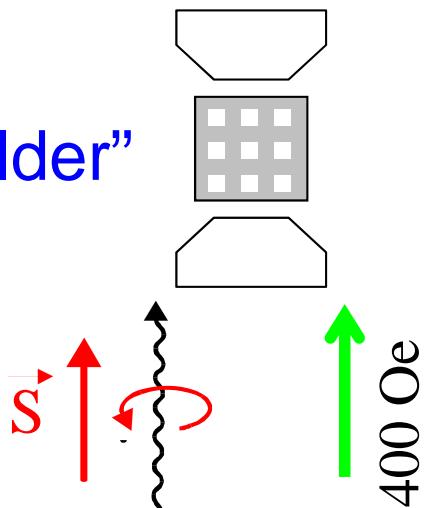
MOKE



Observe magnetisation reversal in applied magnetic field:

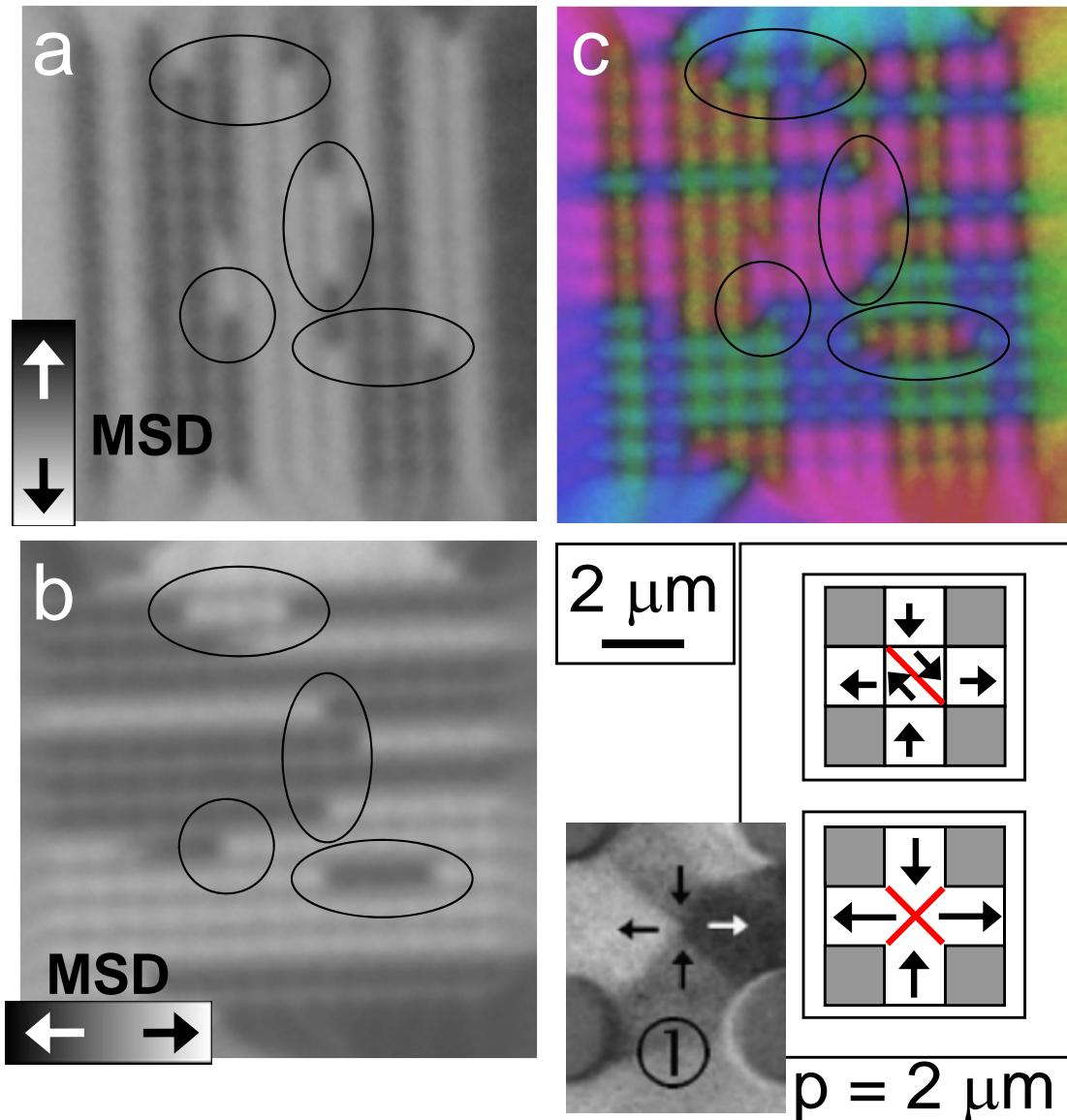
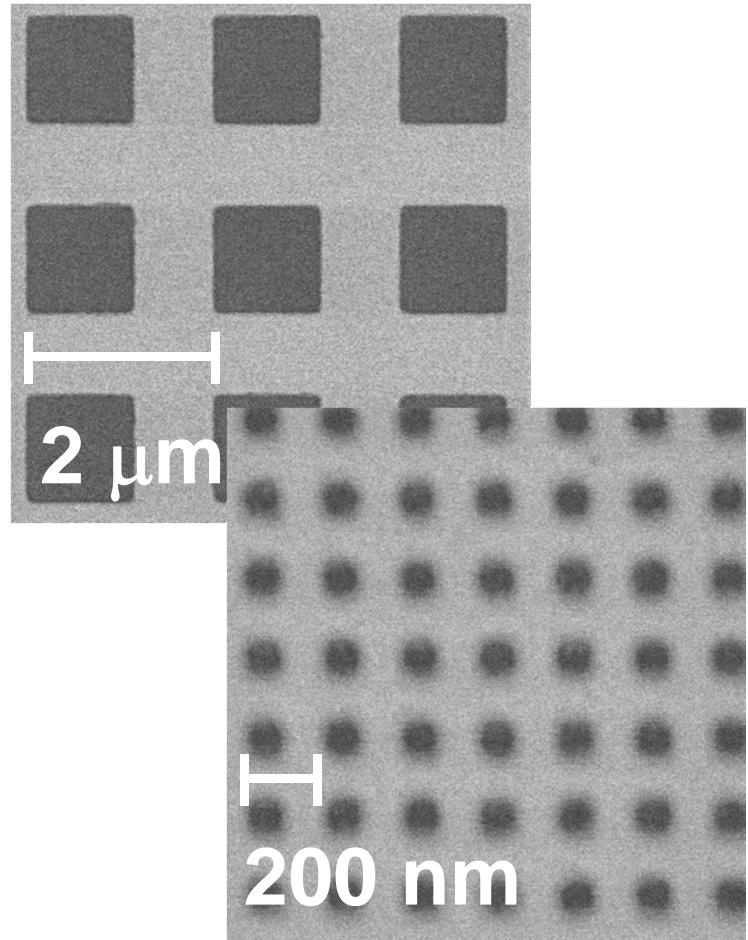
PEEM

"Magnetising Holder"



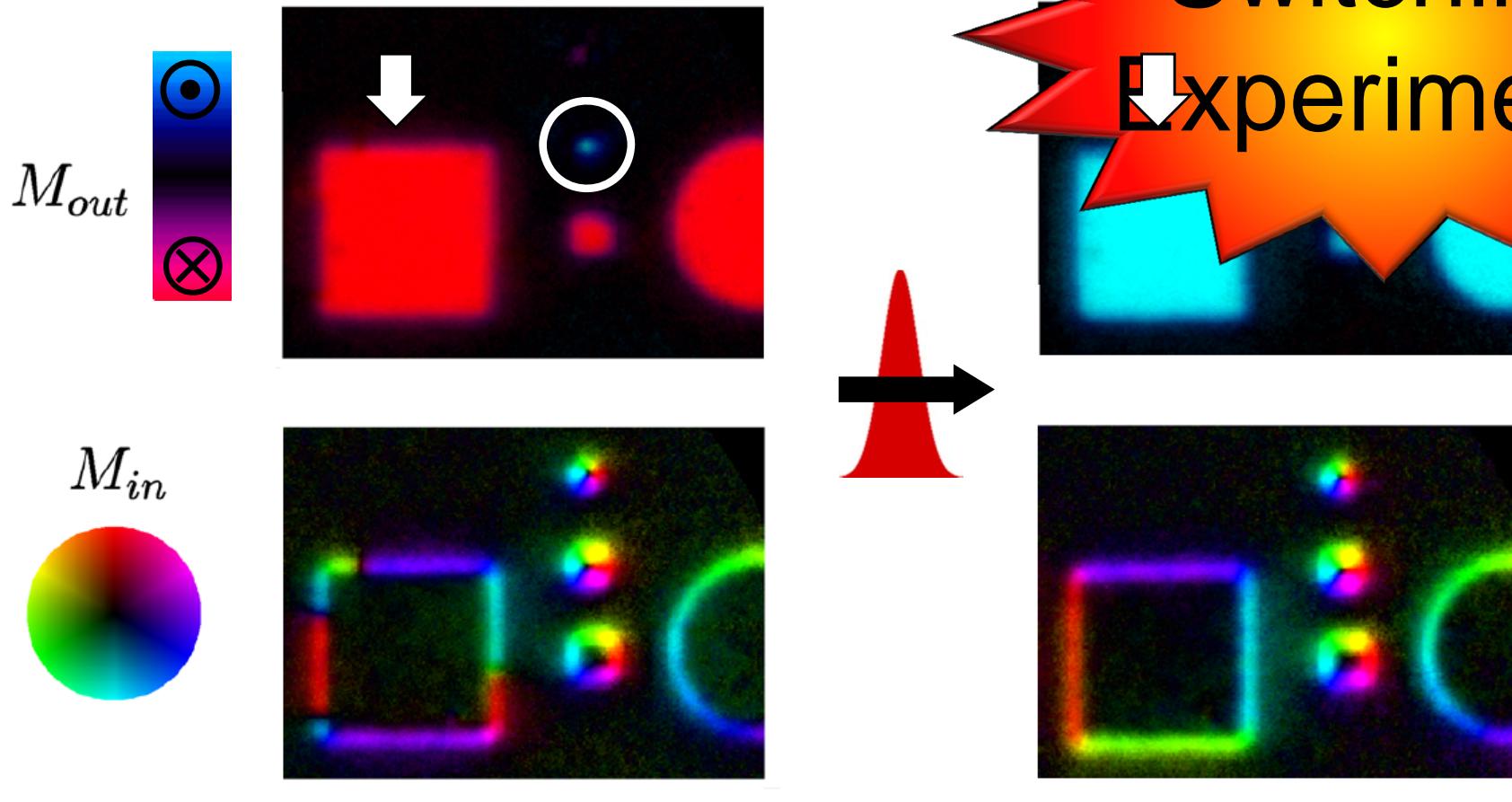
Spins  $\uparrow$   
to  
Spins  $\downarrow$

# Cobalt Antidot Arrays



L. J. Heyderman et al., APL (2003), JAP (2004), PRB (2006), JMMM (2007)  
Mengotti et al., JAP (2007)

- ❖ 200 nm domains in 400 nm GdFeCo nano



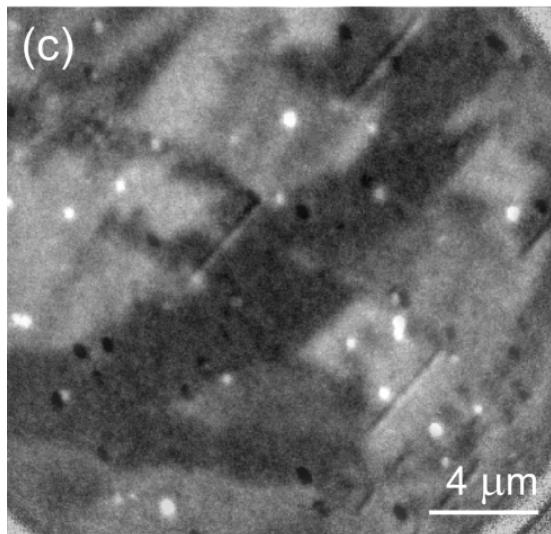
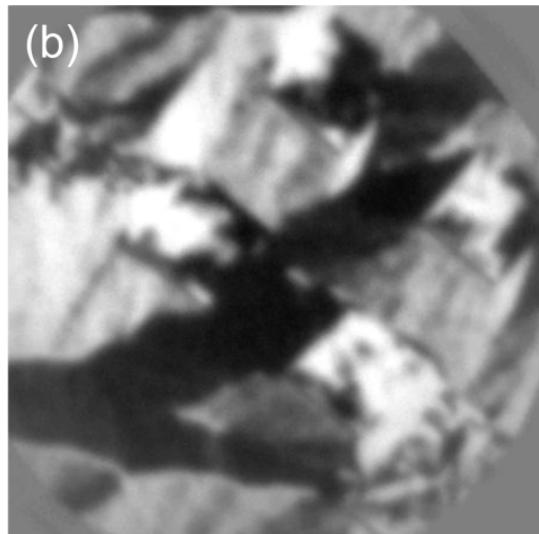
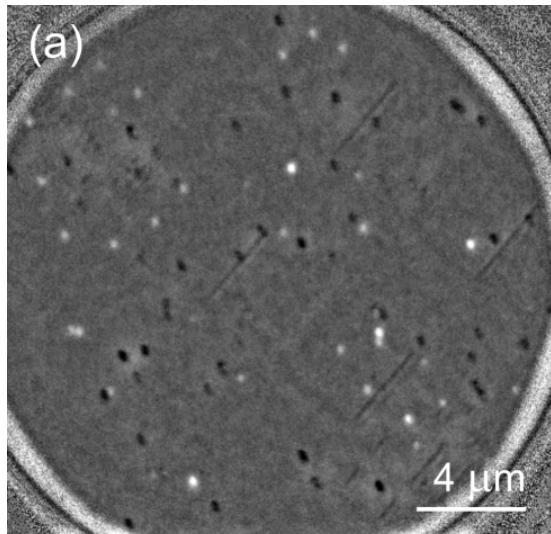
L. Le Guyader et al., APL (Accepted 2012)

T. A. Ostler et al., Nature Communications (2012)

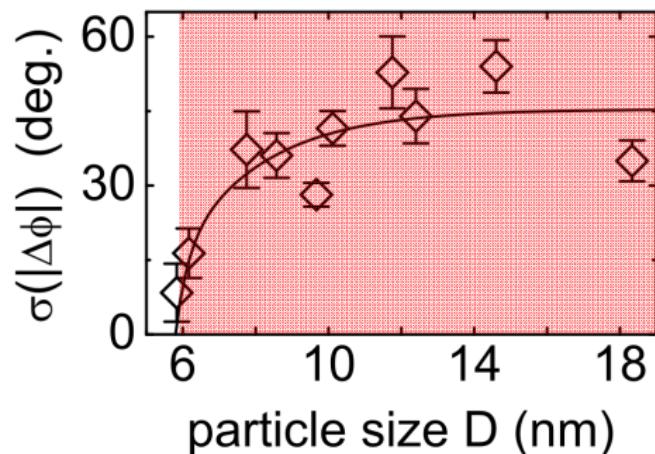


# Iron Nanoparticles Coupled to Cobalt Thin Film

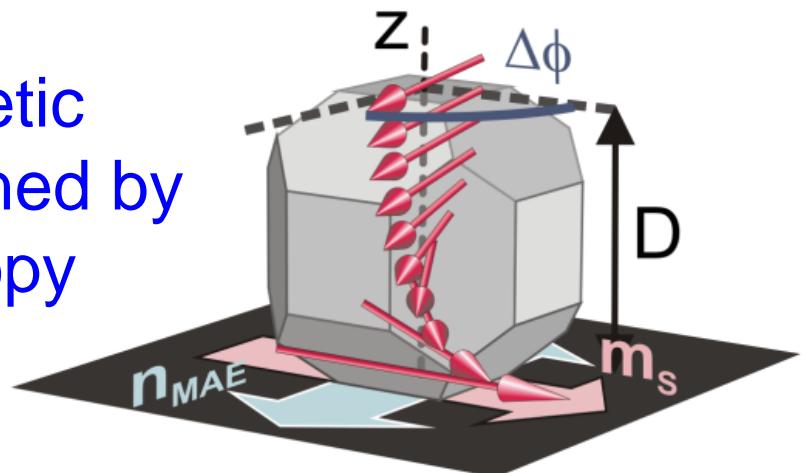
5-25 nm Fe particles/Co thin Film



Noncollinear alignment  
for particles  $> 6$  nm



Spin-spiral magnetic  
structure determined by  
magnetic anisotropy  
energy



A. Fraile Rodríguez, A. Kleibert, J. Bansmann, A. Voitkans, L. J. Heyderman,  
and F. Nolting, PRL (2010)

# Time Resolved Imaging

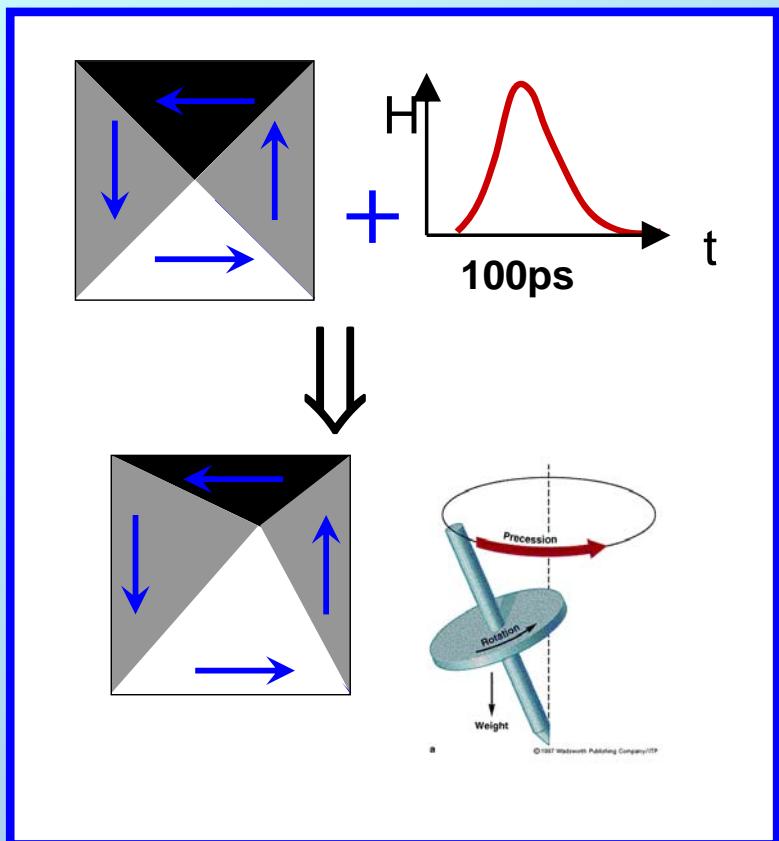
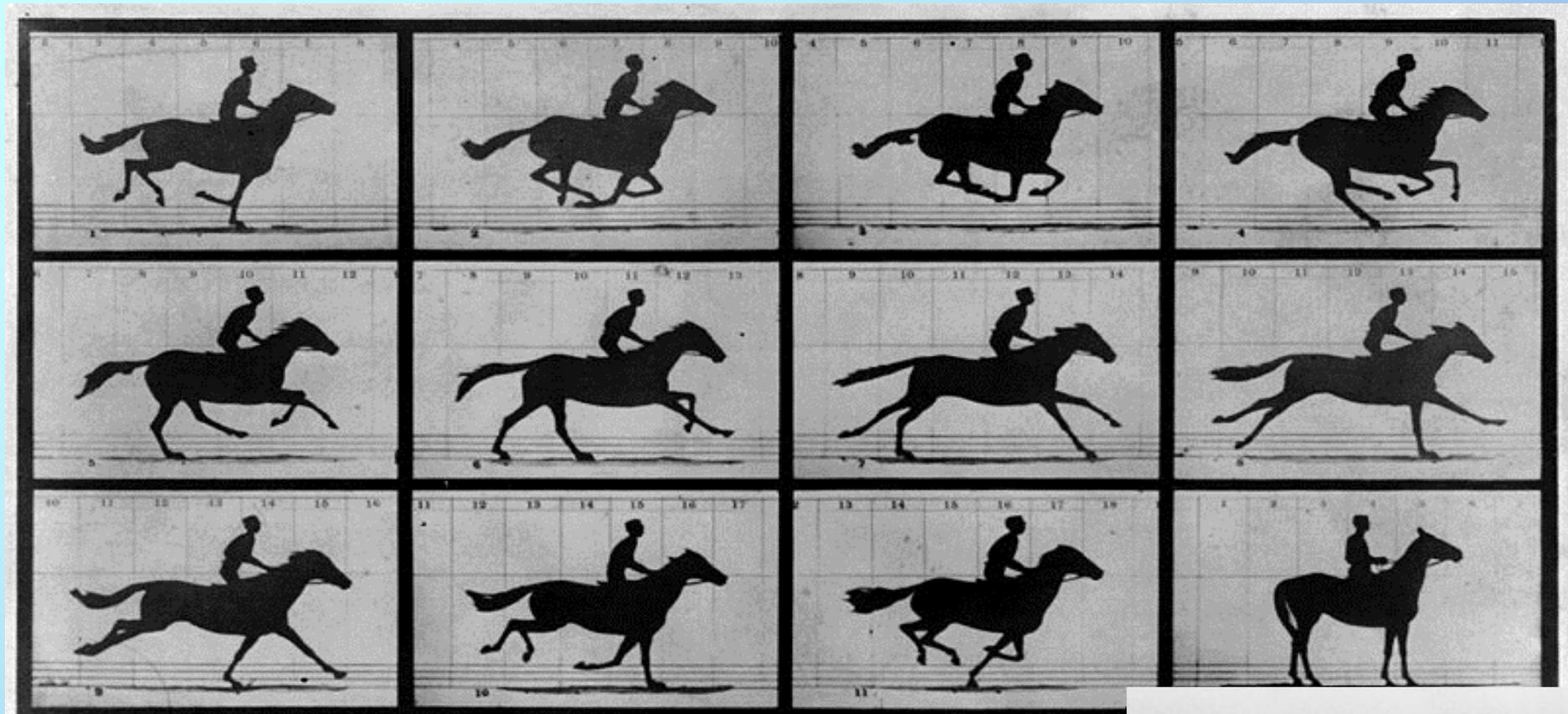


Image excitations in magnetic nanostructures  
Precession frequency & damping  
Pump-probe experiment  
SLS: X-ray stroboscope

J. Raabe et al., Phys. Rev. Lett. 94, 217204 (2005)

# Why perform time-resolved imaging?



Copyright, 1878, by MUYBRIDGE.

## THE HORSE IN MOTION.

Illustrated by  
MUYBRIDGE.

"SALLIE GARDNER," owned by LELAND STANFORD; running at a 1.40 gait over the Pal-

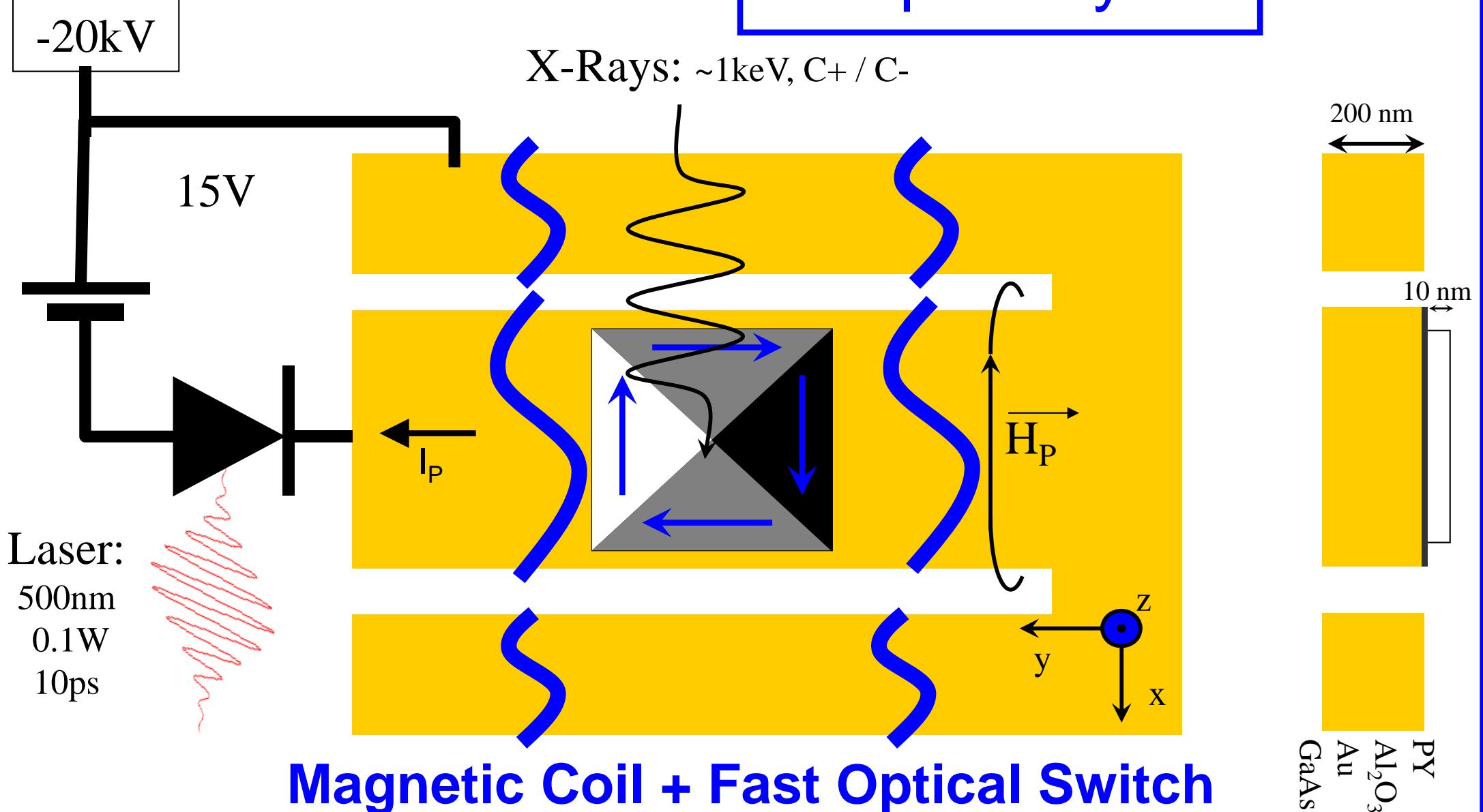
The negatives of these photographs were made at intervals of twenty-seven inches of distance, and about the twenty-fifth part of a second of time assumed in each twenty-seven inches of progress during a single stride of the mare. The vertical lines were twenty-seven inches apart, and represent elevations of four inches each. The exposure of each negative was less than the two-thousandth part of a second.

Are all four feet of a horse off the ground at the same time during a gallop.

Galloping horse, animated in 2006, using photos by Eadward Muybridge, Wikipedia



# Sample Layout



## Magnetic Coil + Fast Optical Switch

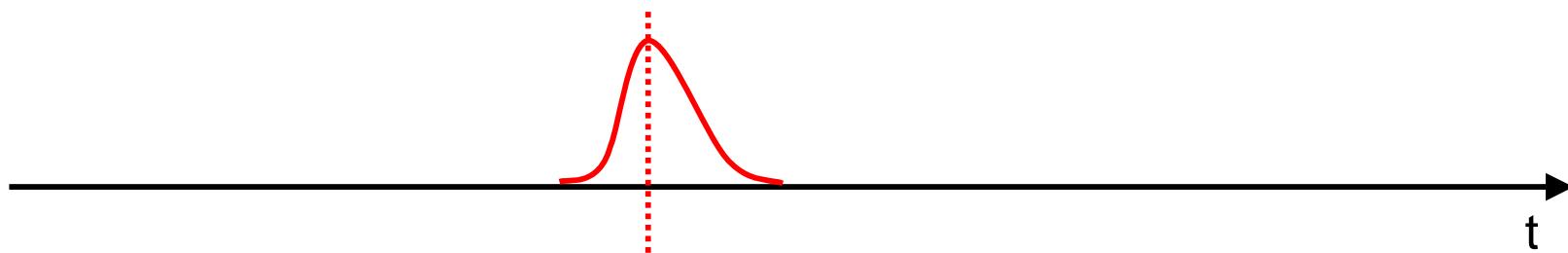
Pulsed laser illuminates photodiode to give a current pulse  
This creates a magnetic field pulse exciting the magnetization.

# Magnetic Pump - X-ray Probe

- Excite system with magnetic pulse
- Time later: measure with an x-ray pulse

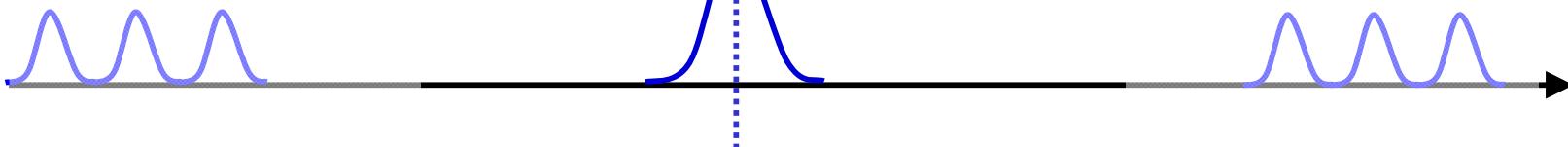
## Pump

Magnetic  
pulse,  
laser  
pulse etc



## Probe

X-ray  
pulse



## Gate

detector  
voltage

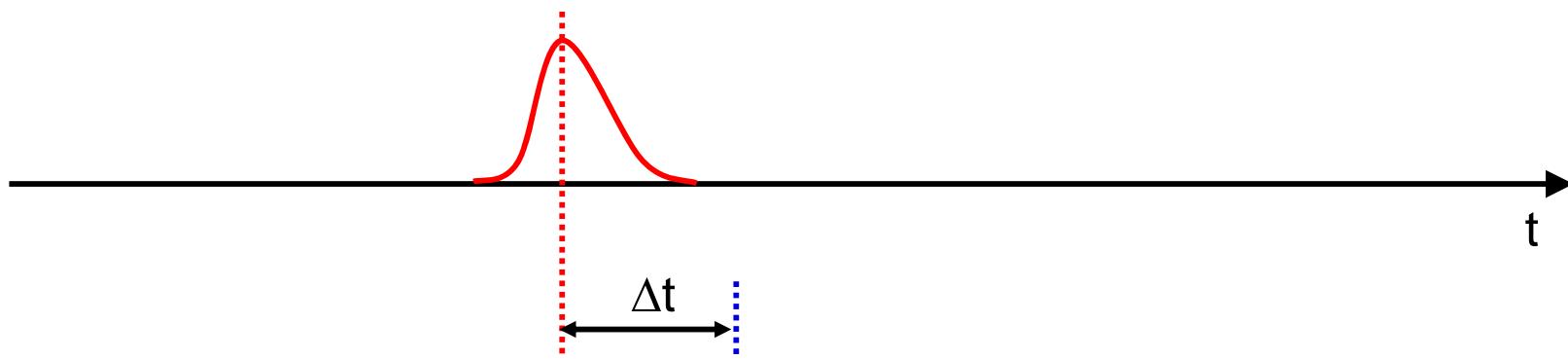


# Magnetic Pump - X-ray Probe

- Excite system with magnetic pulse
- Time later: measure with an x-ray pulse

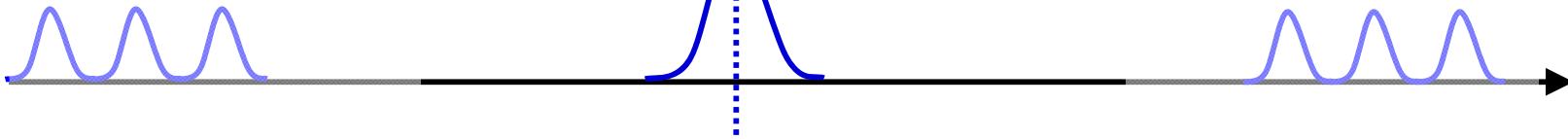
## Pump

Magnetic  
pulse,  
laser  
pulse etc



## Probe

X-ray  
pulse



## Gate

detector  
voltage

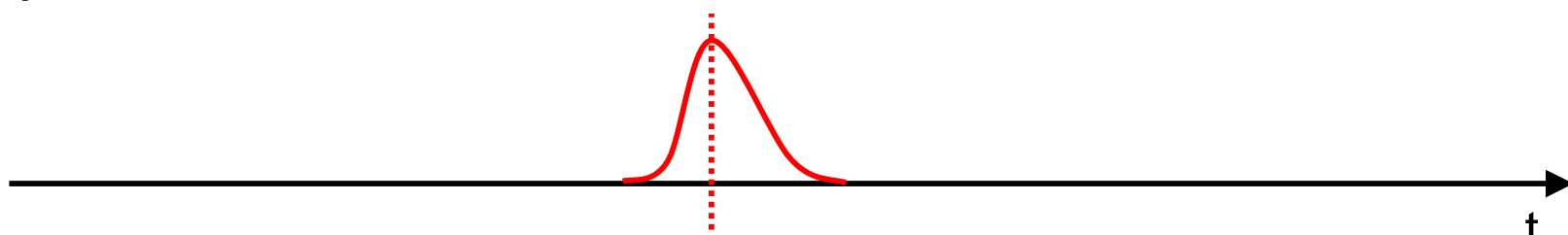


# Magnetic Pump - X-ray Probe

- Not enough intensity in each shot so repeat several times:  
pump-probe

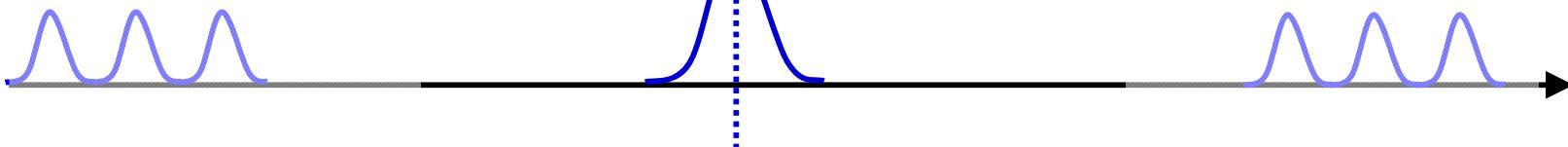
## Pump

Magnetic  
pulse,  
laser  
pulse etc



## Probe

X-ray  
pulse



## Gate

detector  
voltage



**Pump:**

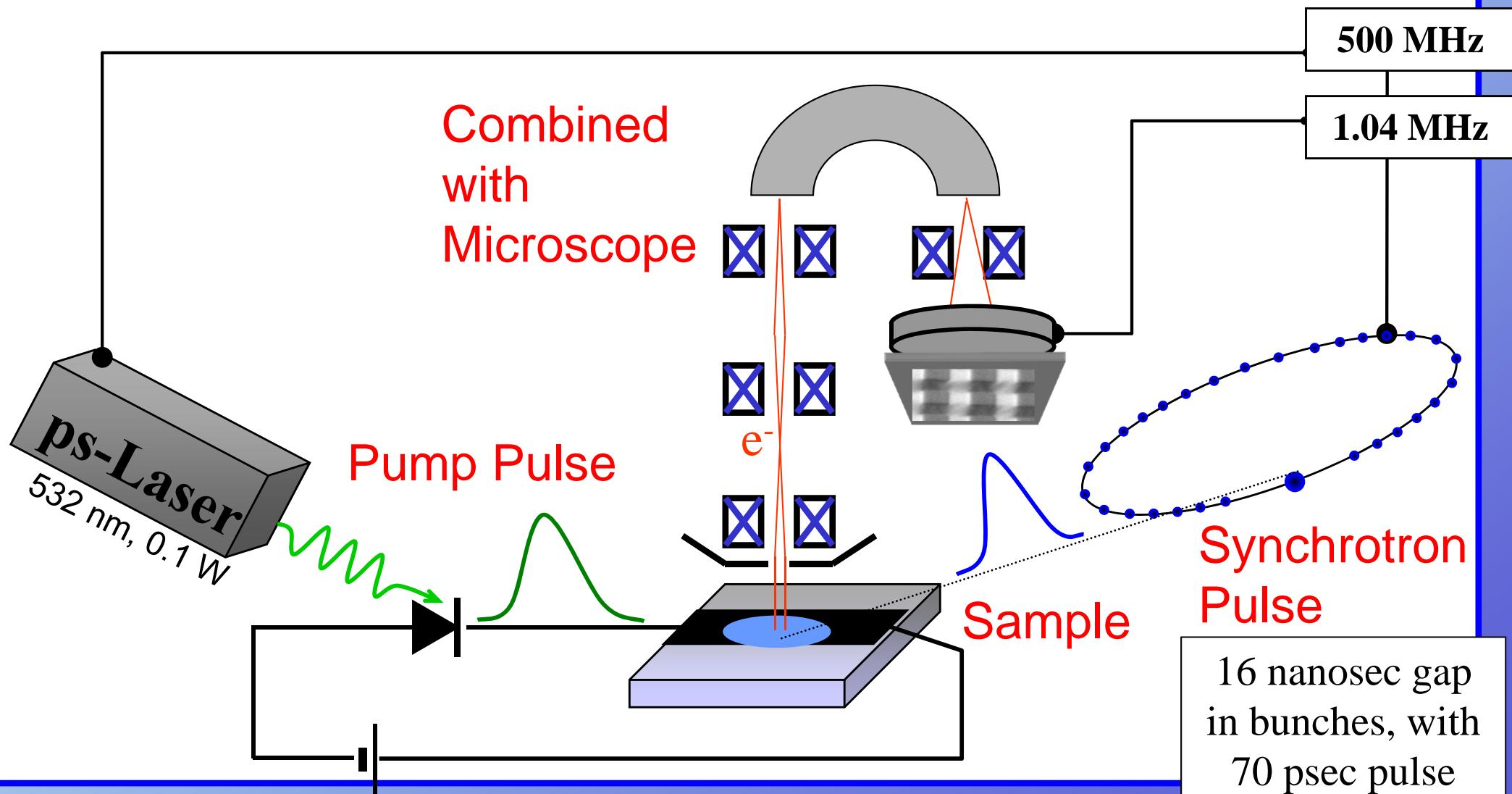
stripline / coil

Pulse:  $H < 100 \text{ G}$ ,  $10^2 \text{ ps}$ **Detect:**

gated PEEM

 $\Delta x \sim 100 \text{ nm}$ ,  $\sim 1 \text{ ML}$ **Probe:**

X-ray stroboscope

 $\sim 1 \text{ keV}$ ,  $\Delta t = 70 \text{ ps}$ 

# Py Square: Excitation



Permalloy ( $Ni_{81}Fe_{19}$ )

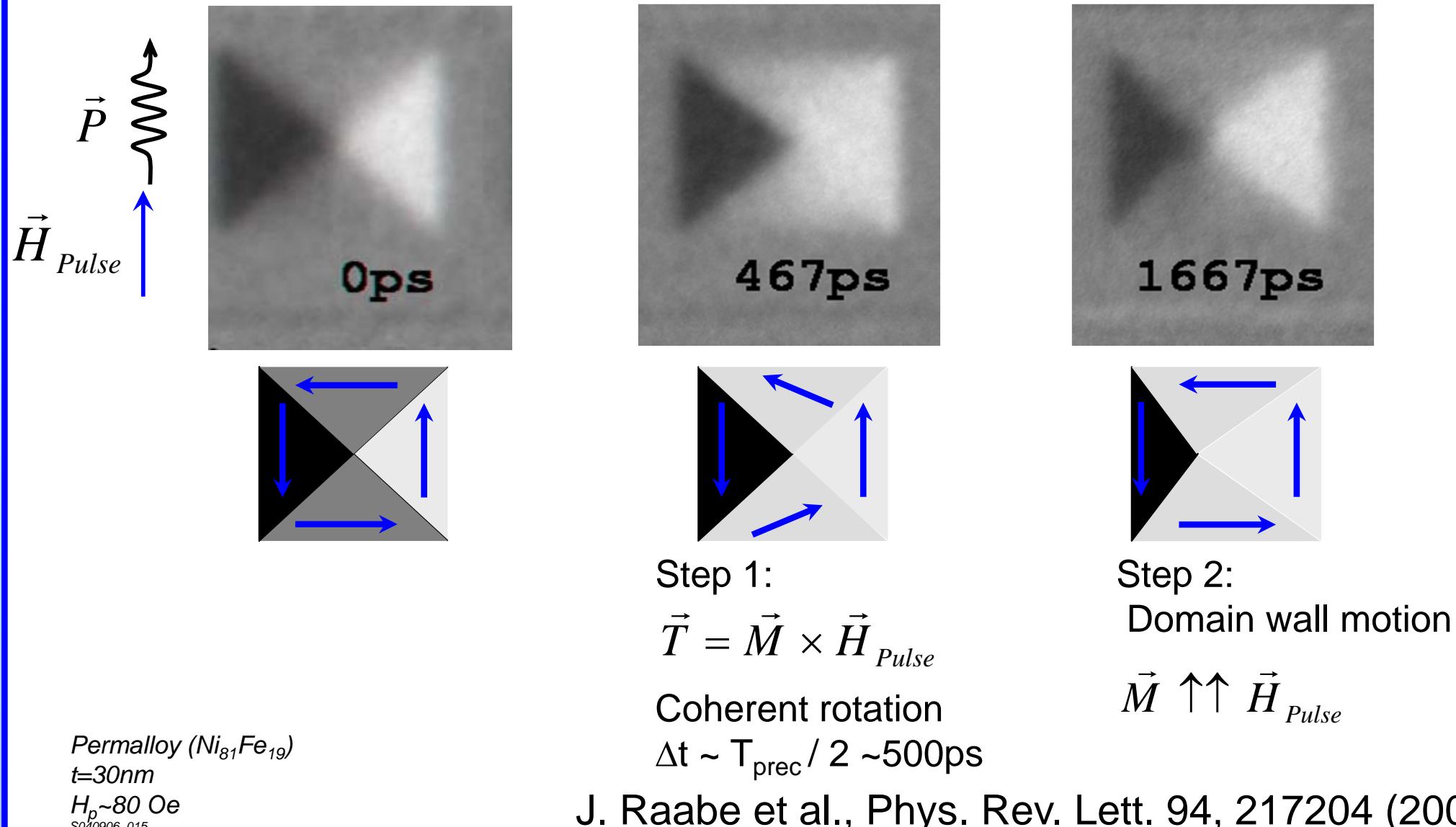
$t=30nm$

$H_p \sim 80$  Oe

S040906\_015

J. Raabe et al., Phys. Rev. Lett. 94, 217204 (2005)

# Py Square: Excitation



# Summary

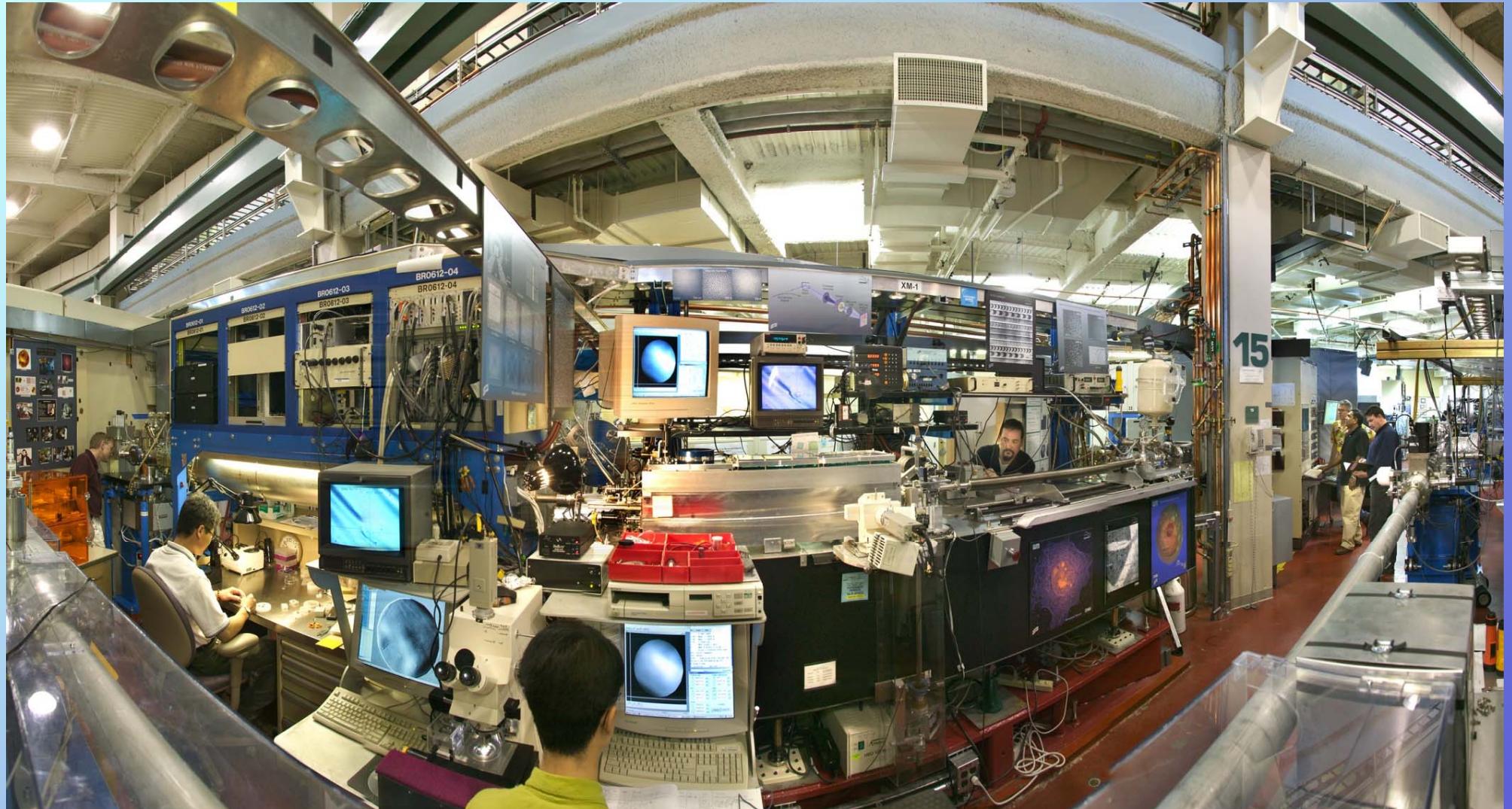
- Element selective (multilayer, coupled systems)
- Surface/interface sensitive (sampling depth a few nm)
- Antiferromagnetic and Ferromagnetic domains
- Spatial resolution: 50-20 nm, future aberration corrected: few nm's
- Time resolved measurements
- Temperature 120 K – 1000 K
- Submonolayer sensitivity
- Combination with other analytical techniques: LEEM & LEED
- In-situ and ex-situ sample preparation
- Sample size 3 to 15 mm diameter, 0.2 mm - 2 mm thick

## Challenges (limitations):

- UHV compatible ( $<10^{-7}$  mbar)
- Smooth surface ( $< 1 \mu\text{m}$ , hard to say)
- X-ray damage
- Image in applied magnetic field below 50 Oe
- High voltage often leads to discharges (20 keV, at 2 mm distance)
- Charging effects due to electrical insulating sample (can get around this)

# Transmission X-Ray Microscopy

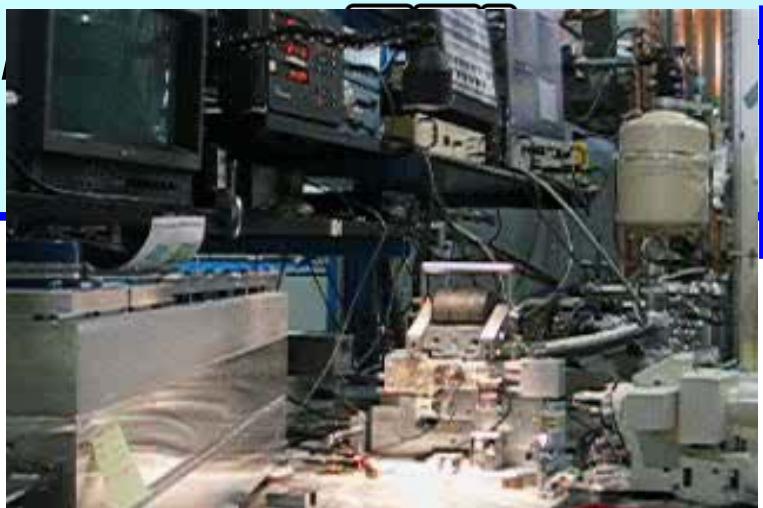
# Magnetic Soft X-ray microscopy, ALS



<http://www.cxro.lbl.gov/BL612/>

Peter Fischer, ALS

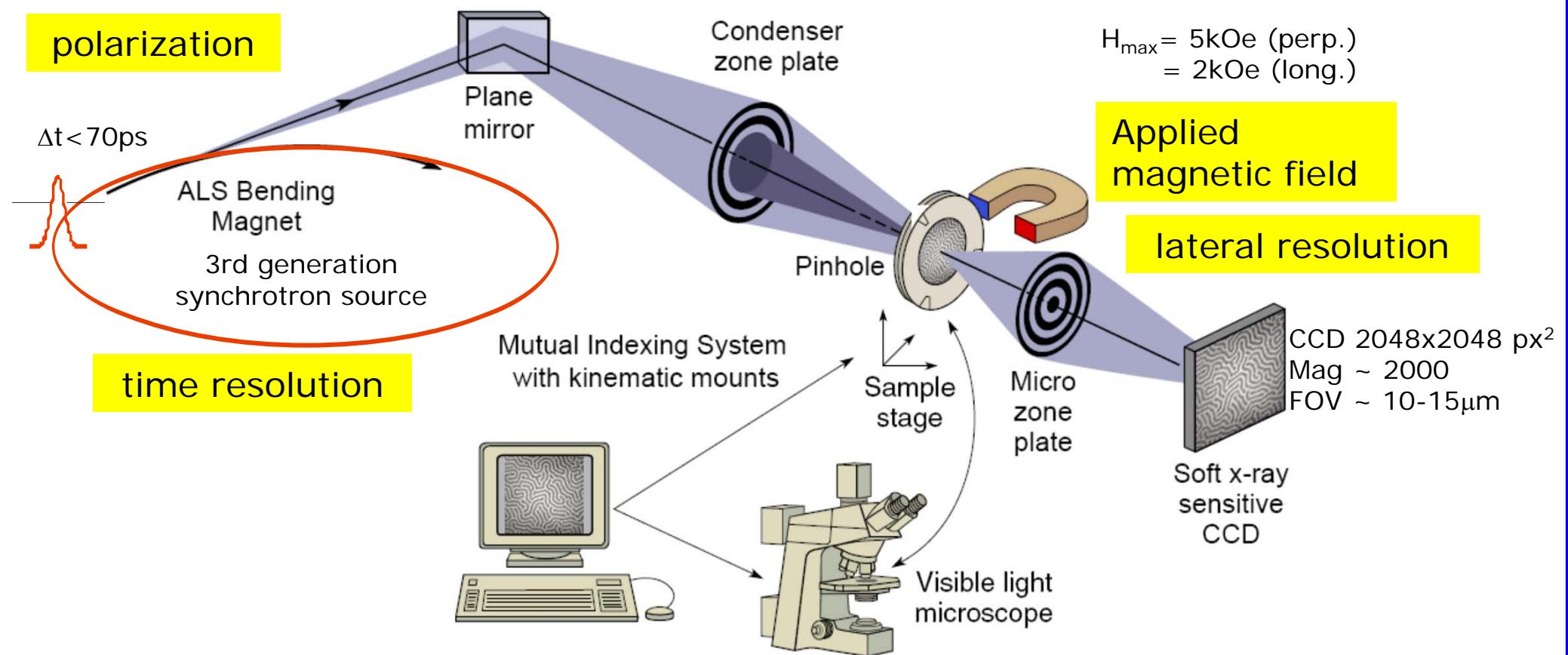
# Soft X-ray microscope XM-1, Advanced Light Source



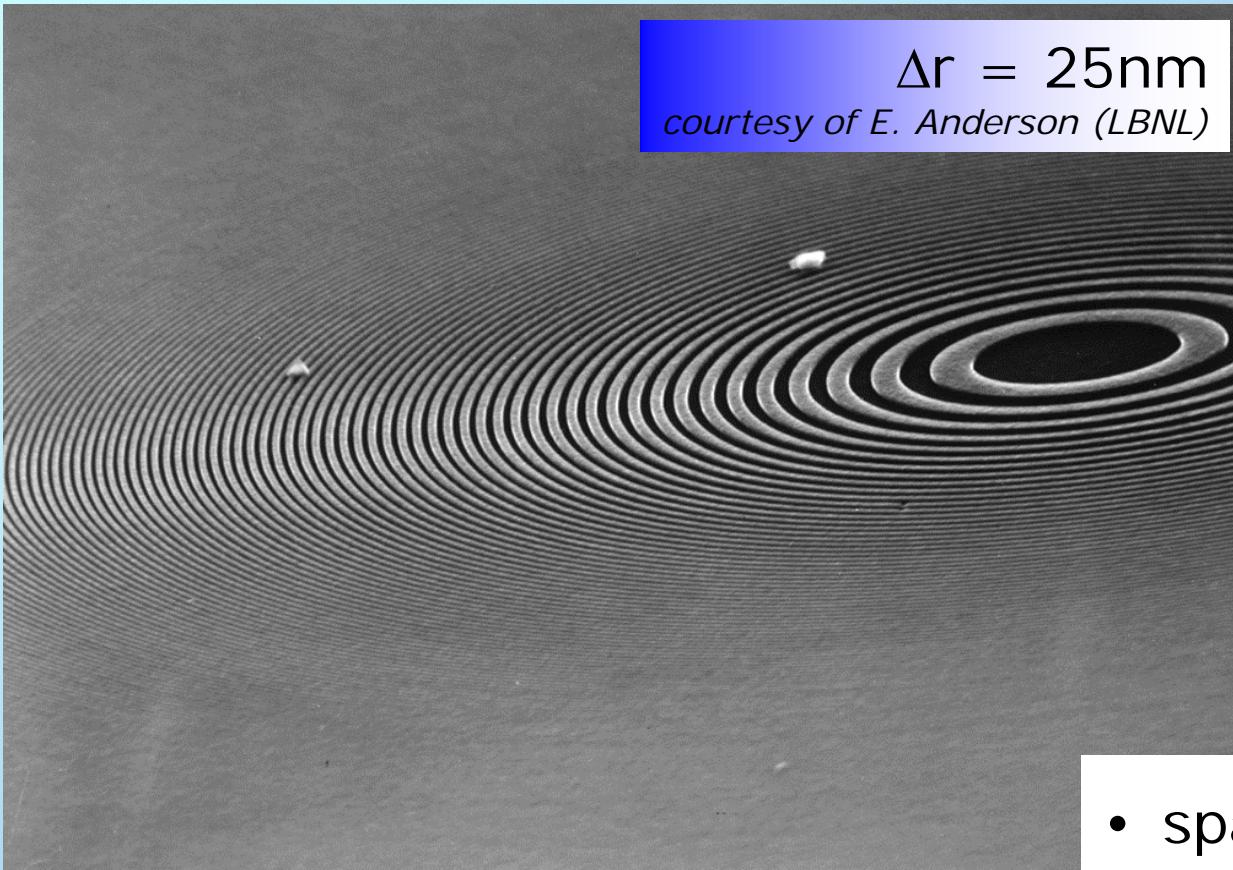
element specificity

$E = 250\text{eV} - 1.8 \text{ keV}$   
 $\lambda = 0.7 \text{ nm} - 5 \text{ nm}$   
 $\Delta E/E = 500$

$H_{\max} = 5\text{kOe}$  (perp.)  
=  $2\text{kOe}$  (long.)



# Fresnel Zone Plate Lenses



diffractive optics:  
concentric rings

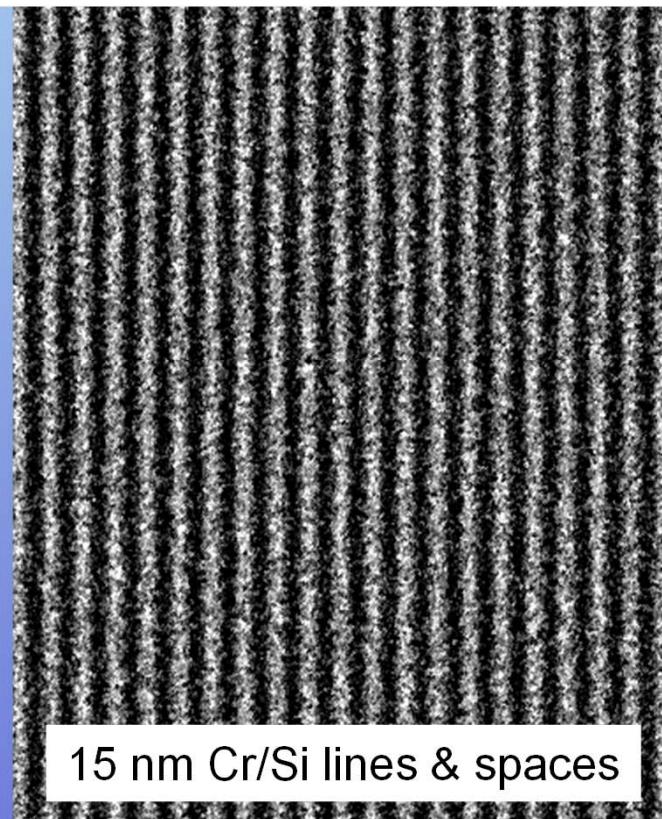
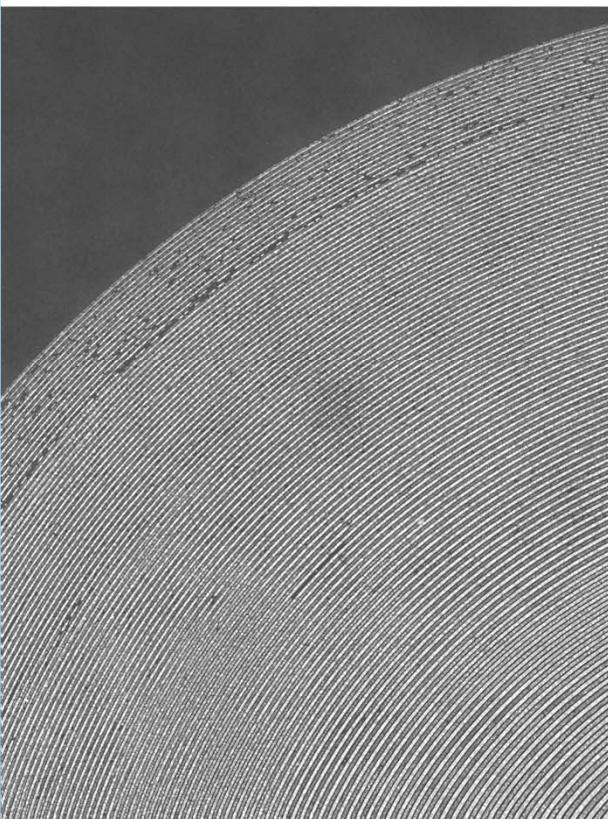
- spatial resolution  $\sim \Delta r$
- focal length  $\sim N(\Delta r)^2/\lambda$
- spectral bandwidth  $\Delta\lambda/\lambda \sim 1/N$

nature

Vol 435 | 30 June 2005 | doi:10.1038/nature03719

## LETTERS

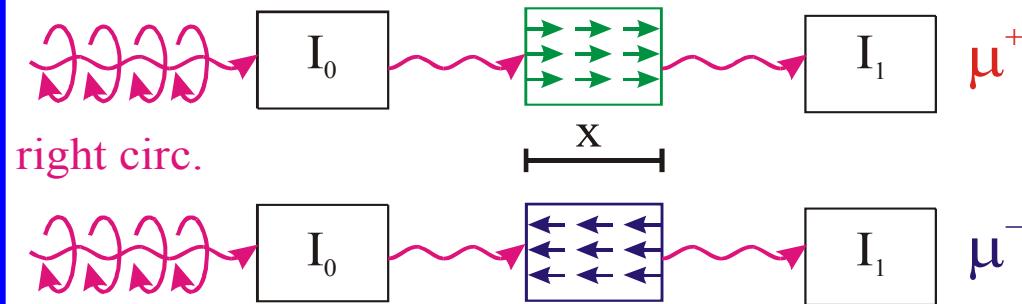
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**Soft X-ray microscopy at a spatial resolution better than 15 nm**Weilun Chao<sup>1,2</sup>, Bruce D. Harteneck<sup>1</sup>, J. Alexander Liddle<sup>1</sup>, Erik H. Anderson<sup>1</sup> & David T. Attwood<sup>1,2</sup>

# Magnetic absorption contrast

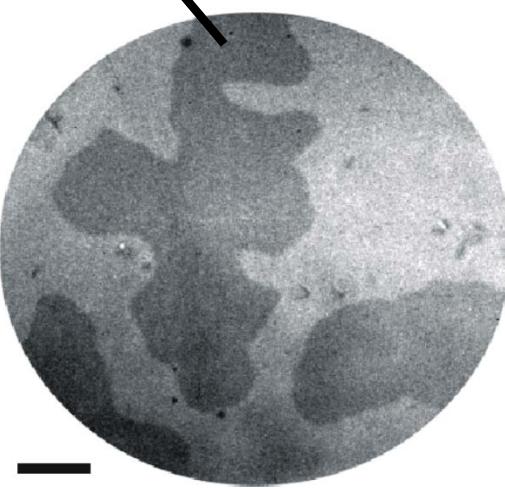
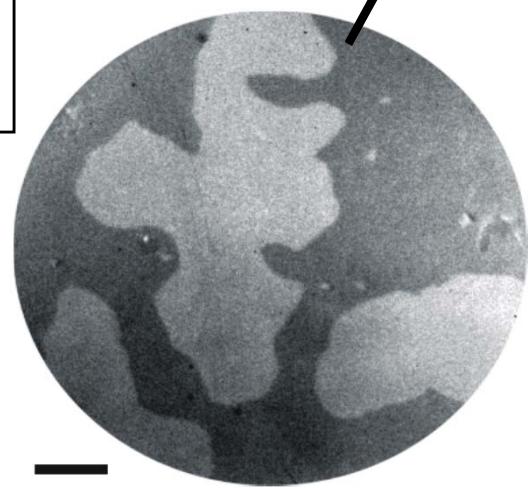
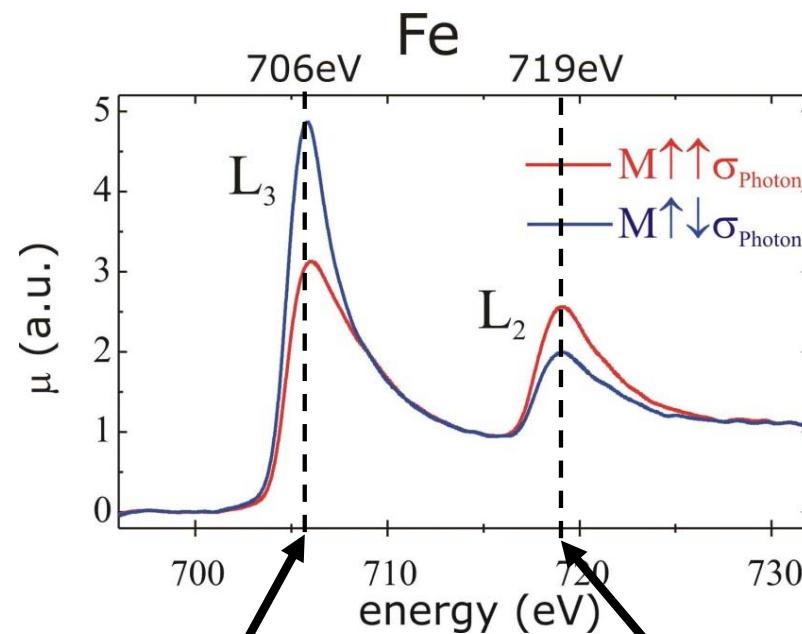
# X-ray magnetic circular dichroism

### absorber



Absorption coefficient  
 $(I_1/I_0)^{\pm} = \exp(-\mu^{\pm}x)$   
 intensity thickness

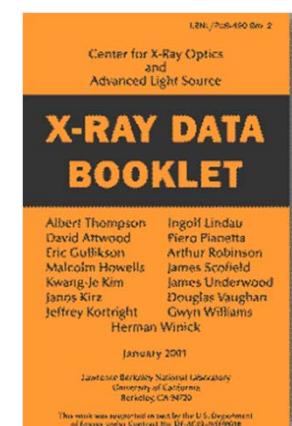
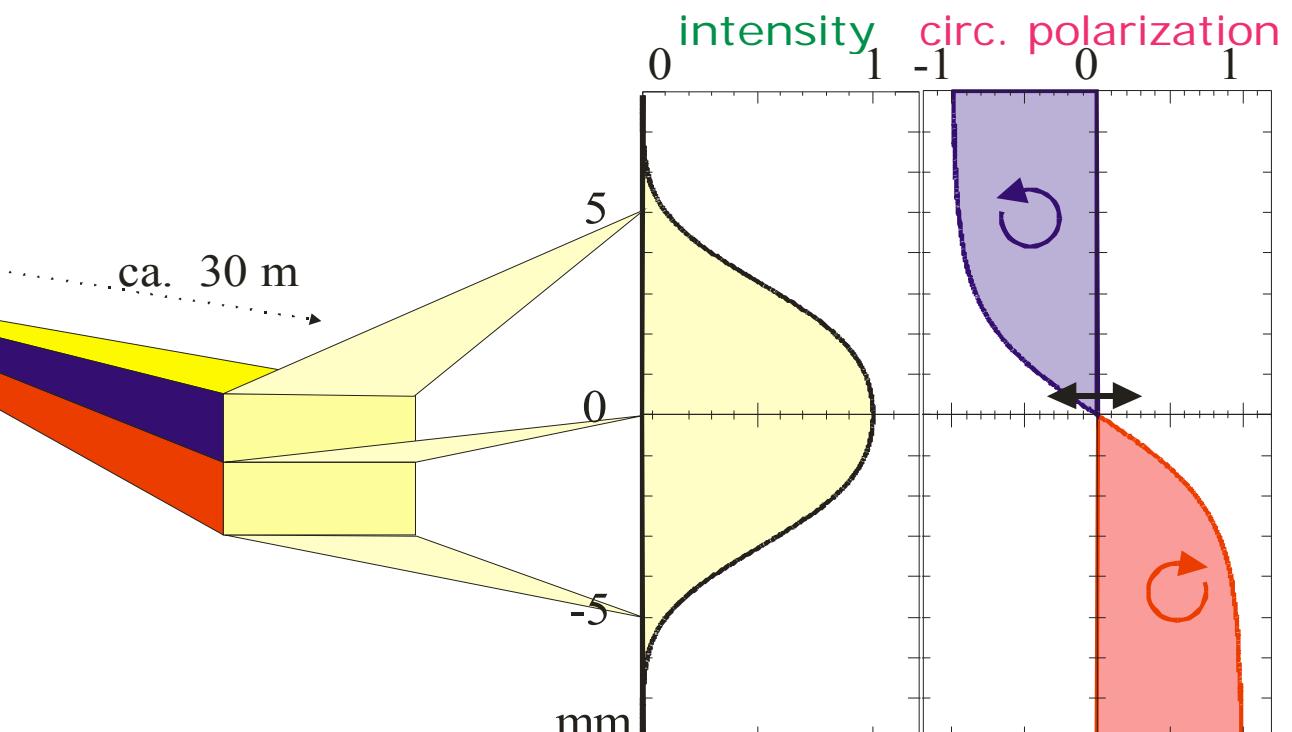
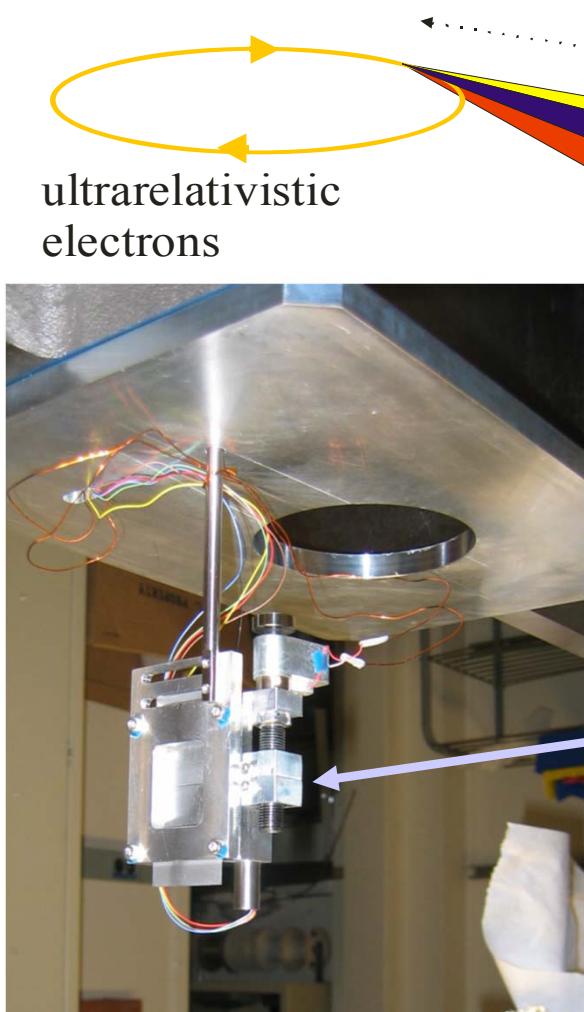
- ❖ element-specific
  - ❖ huge magnetic contrast
  - ❖  $\underline{M} \cdot \underline{\sigma}_{\text{Photon}}$  (polarisation vector of photons)
  - ❖ spin-orbit information



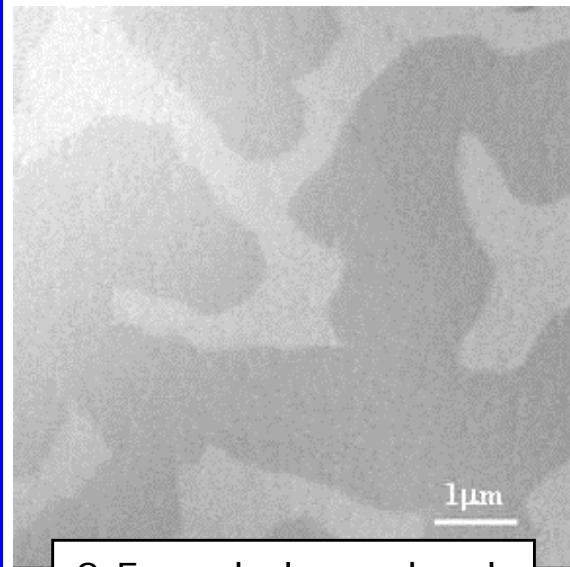
1 μm

# Peter Fischer, ALS

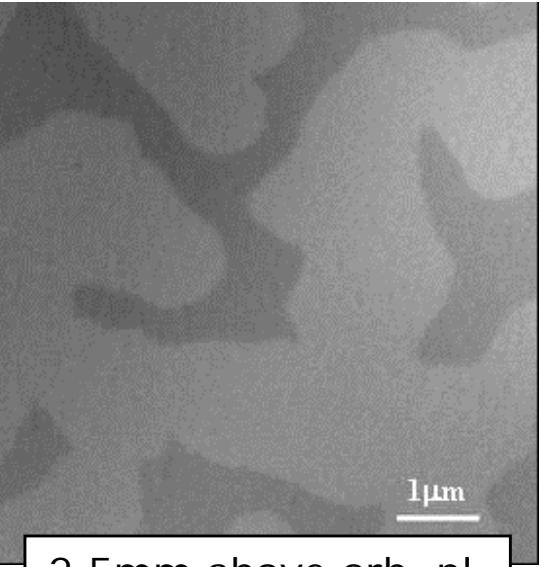
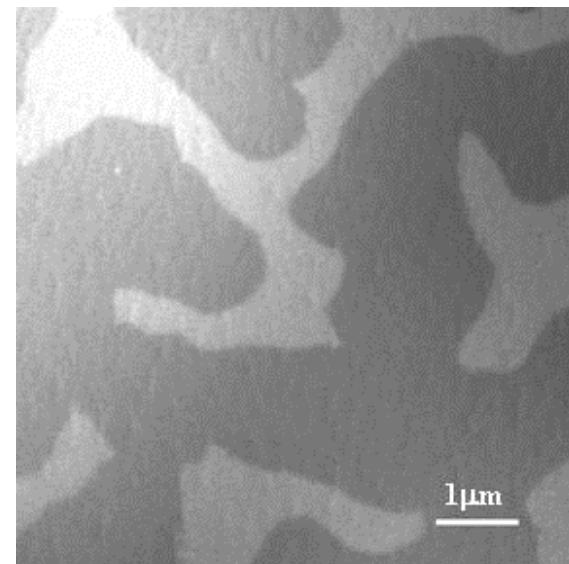
# Polarization properties at bending magnet



# Modulation of circ. polarization



2.5mm below orb. pl.



2.5mm above orb. pl.

- ❖ reducing non-magnetic background
- ❖ increase magnetic contrast

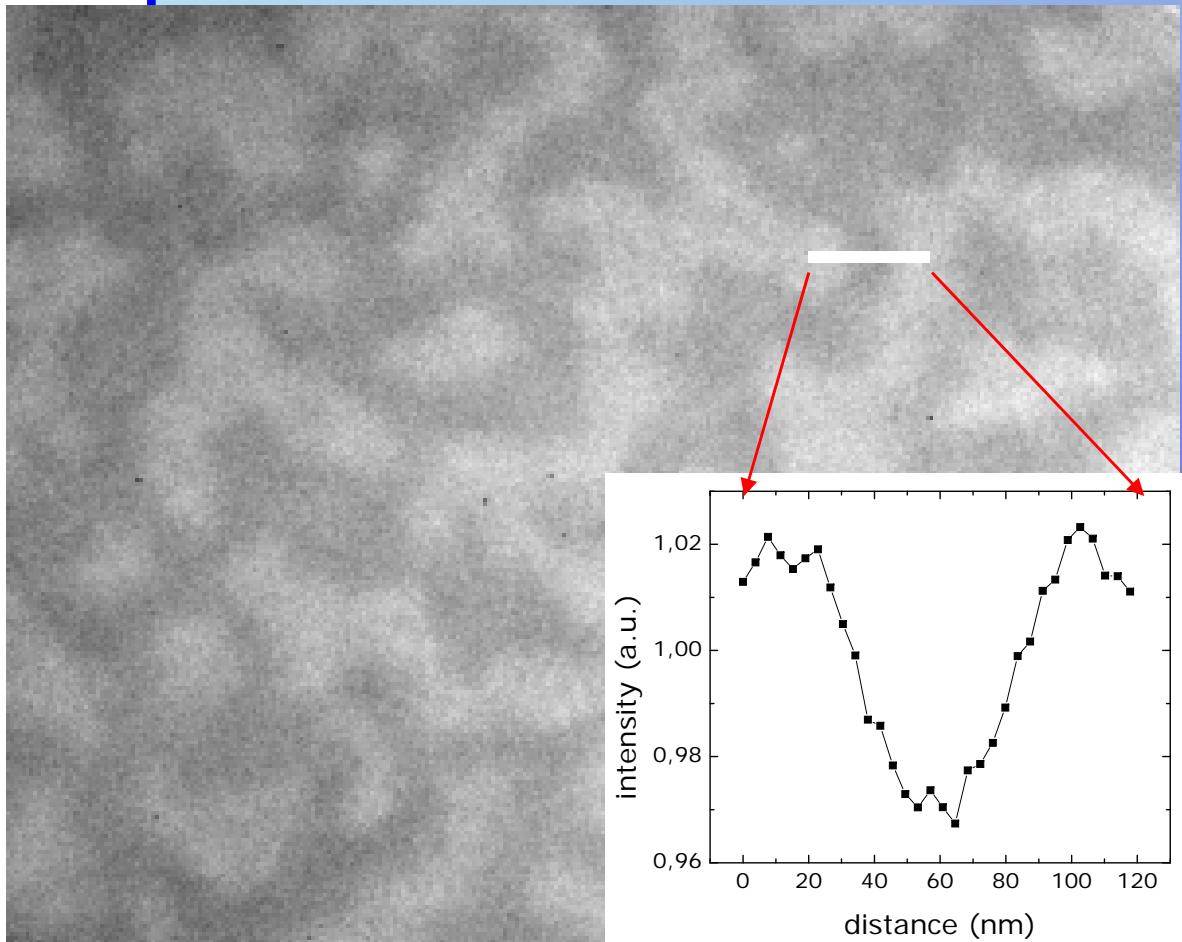
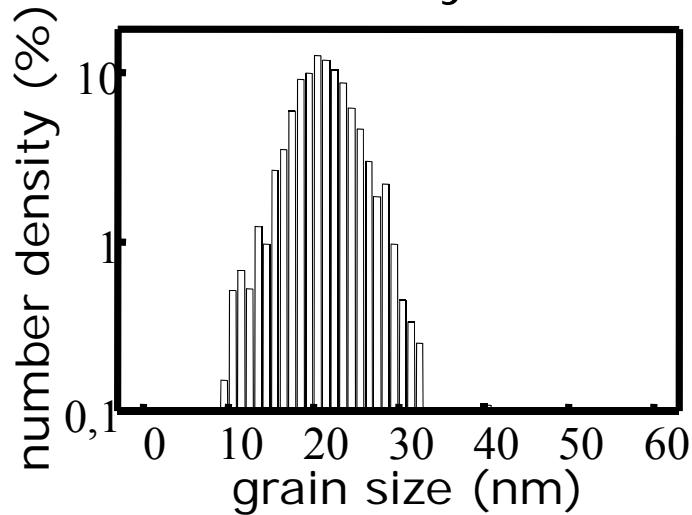
B.-S. Kang et al., J Appl. Phys **98** (2005) 093907

# Magnetic imaging at 15 nm resolution

sample:

$(\text{Co}_{84}\text{Cr}_{16})_{87}\text{Pt}_{13}$  (50nm)

TEM analysis

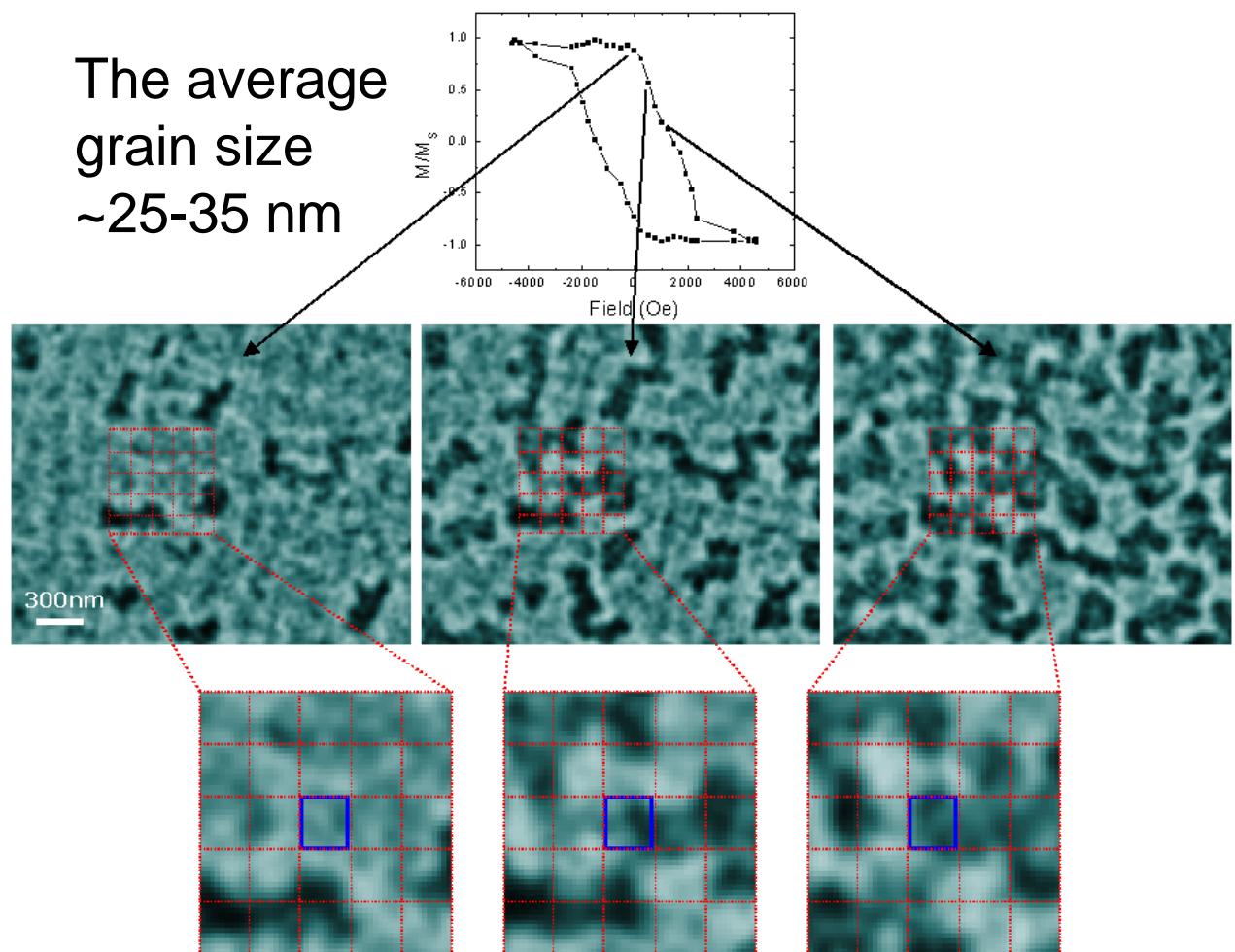


► imaging at fundamental  
magnetic length scales

Co L<sub>3</sub> edge @777eV

- CoCrPt alloy films: possible high-density magnetic recording media:
  - strong perp. anisotropy
  - low media noise: exchange decoupled at grain boundaries
- Dark & white areas correspond to regions where the Co magnetization is pointing in and out.

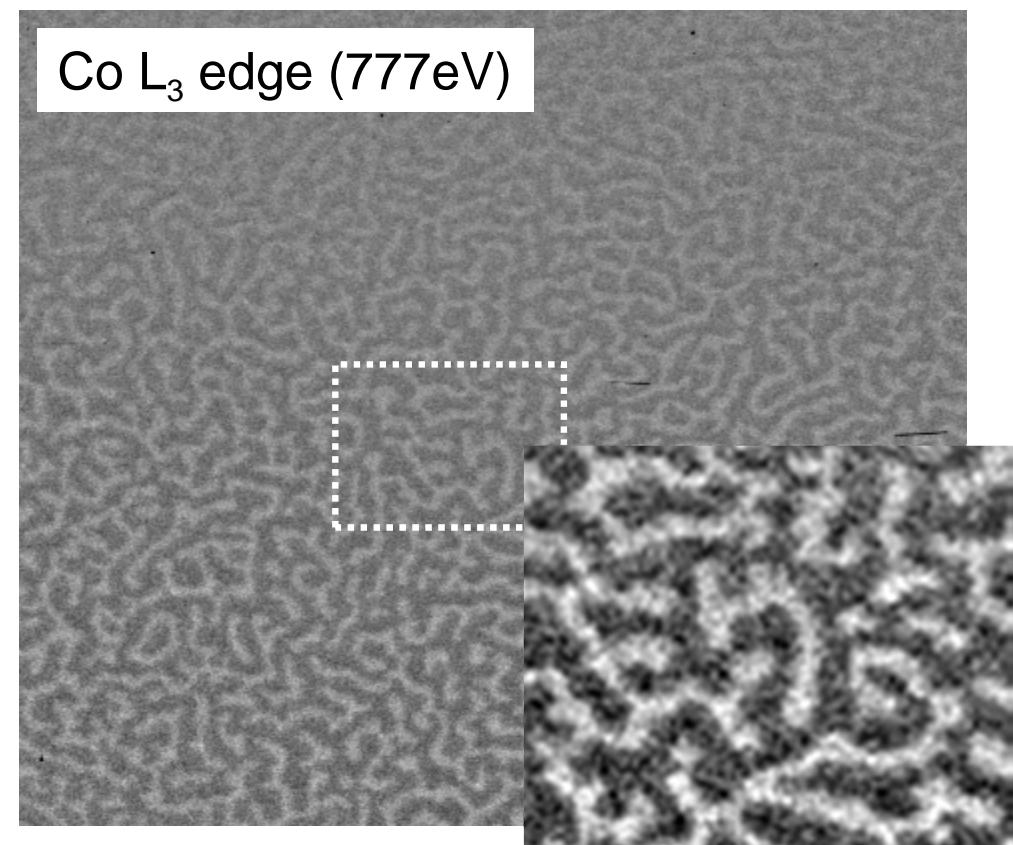
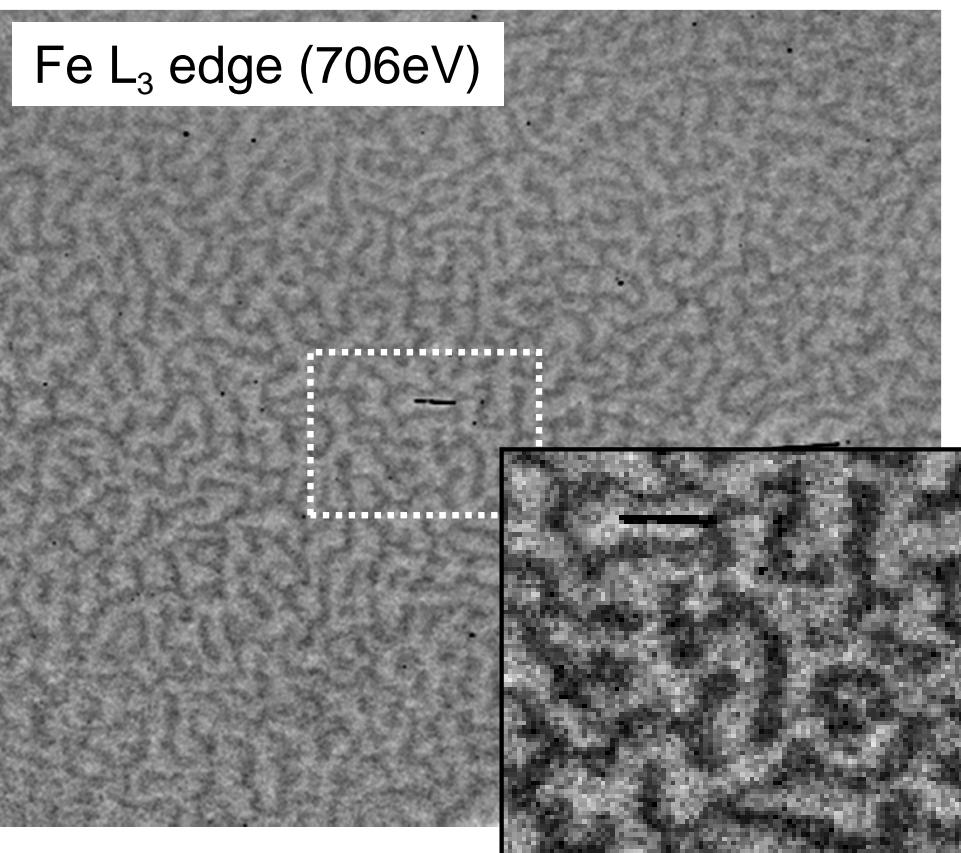
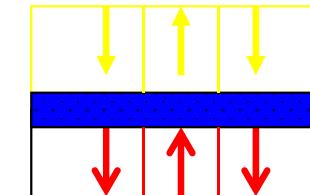
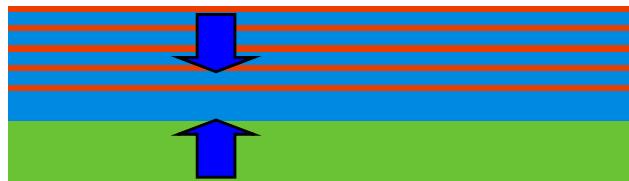
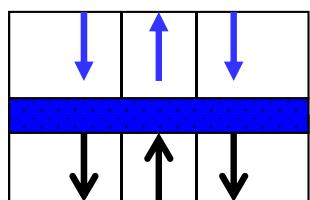
The average grain size  
~25-35 nm



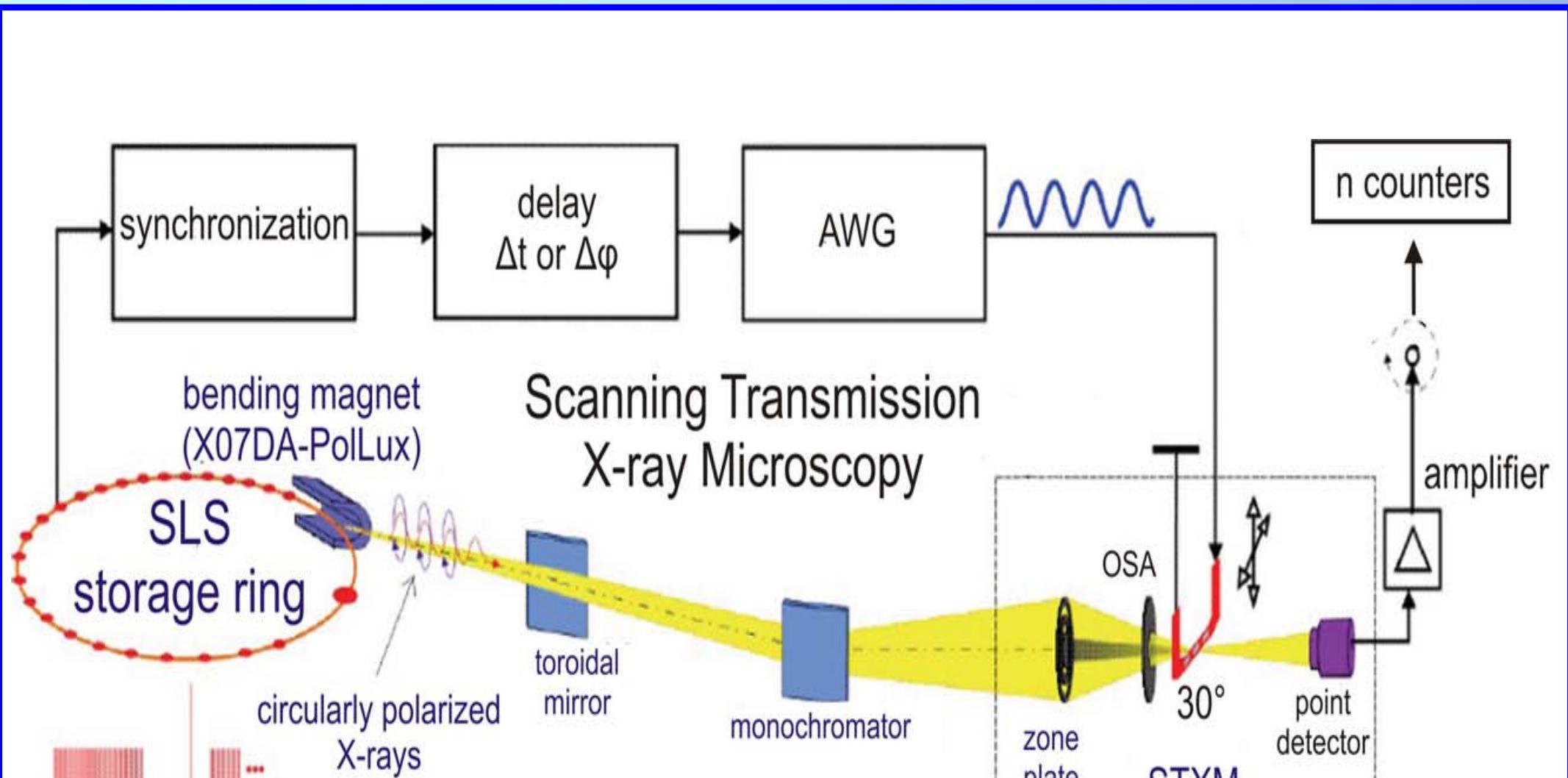
*Details of magnetization reversal for each grain: closely related to the size, irregularity, and stability of written domains.*

# Layer resolution via element specificity

Pt/[Pt 0.75nm/**Co 0.35nm**<sub>50</sub>/Pt 3nm/Tb<sub>45</sub>**Fe<sub>55</sub>** 25nm)/Pt 5nm

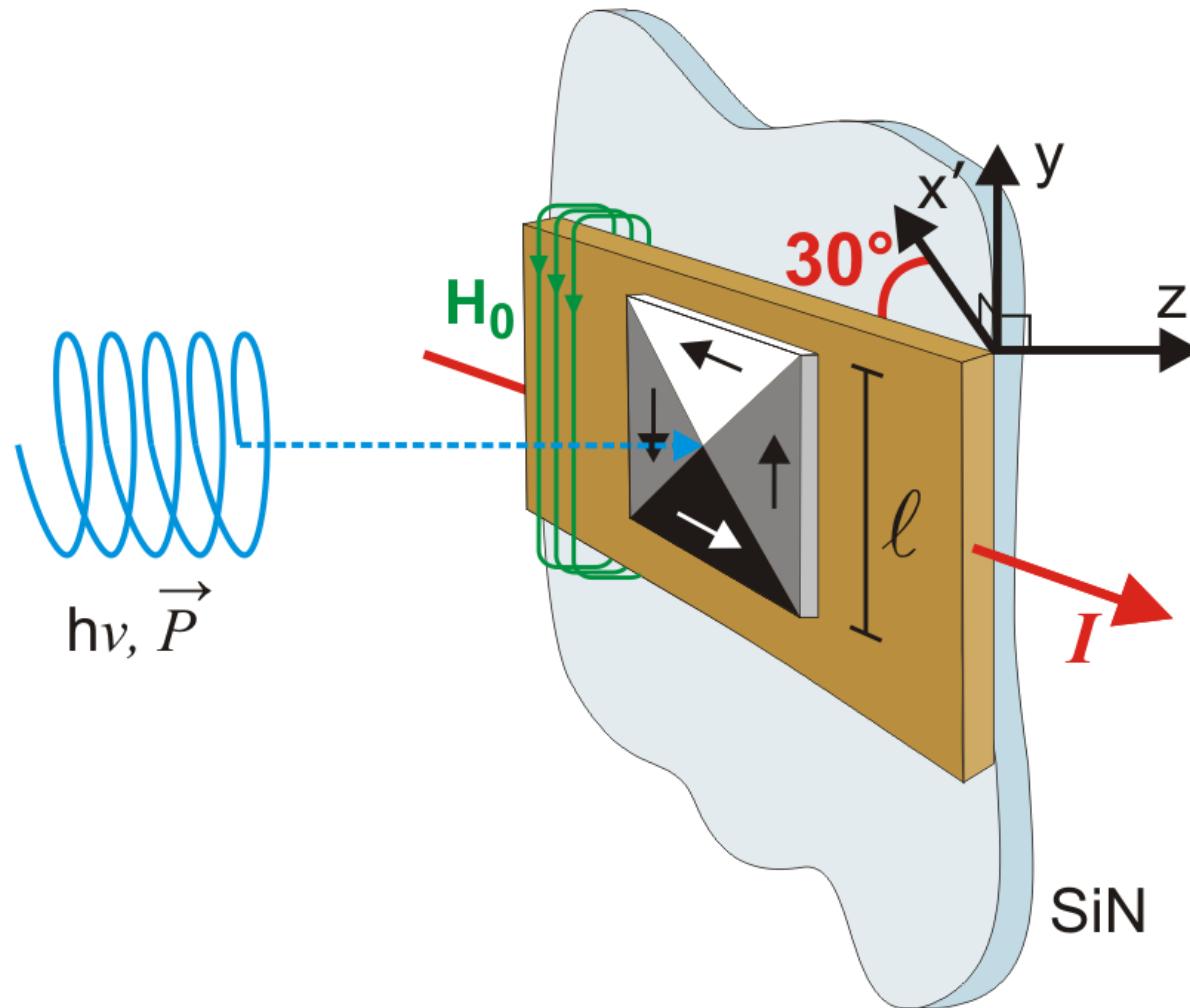


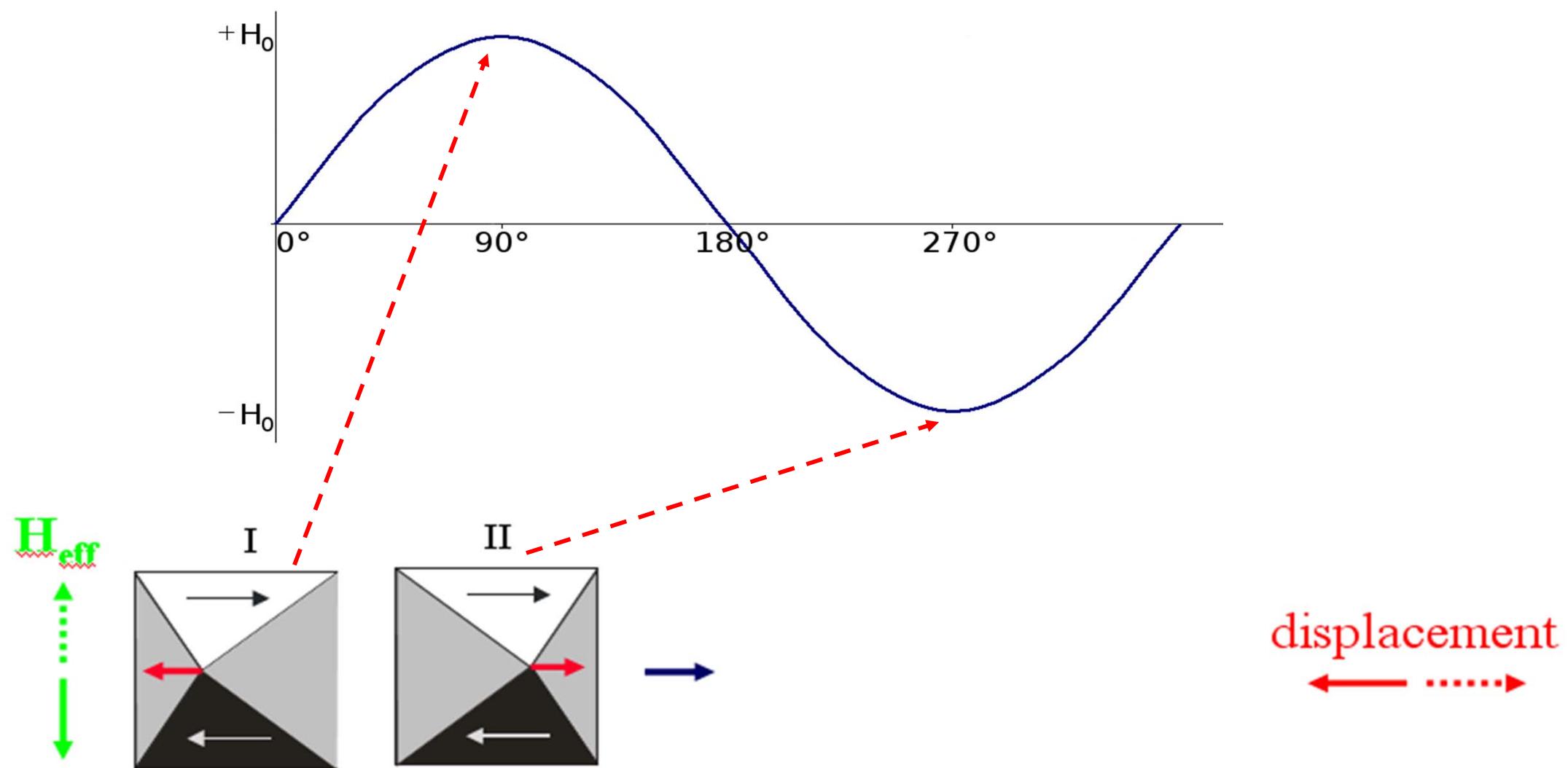
w/ S. Mangin, A. Berger, E. Fullerton (HITACHI/Almaden) (2005)



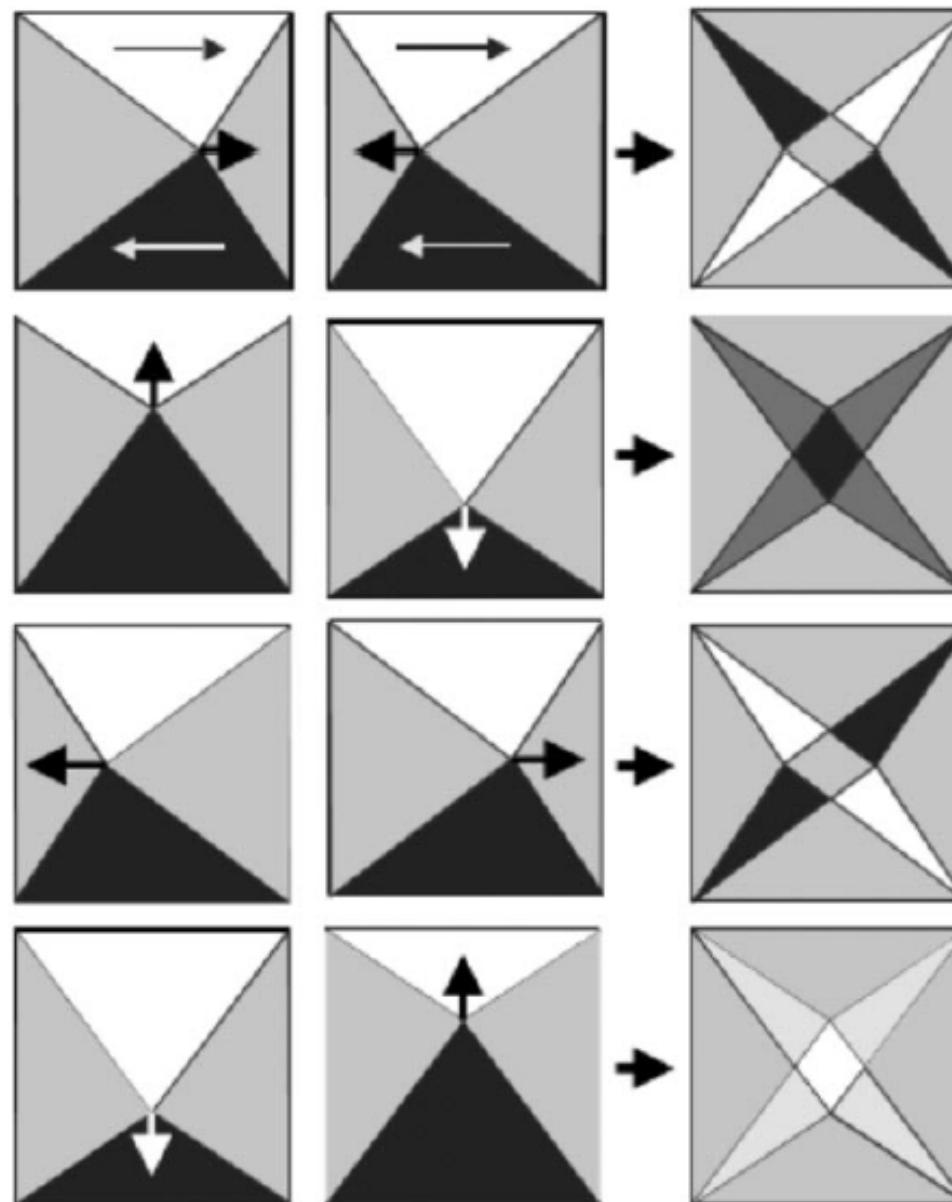
*PolLux Beamlime, Swiss Light Source*

# Towards fundamental time scales



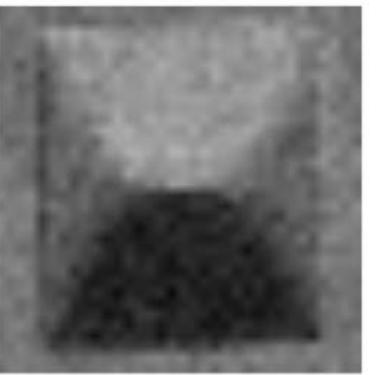


# Towards fundamental time scales

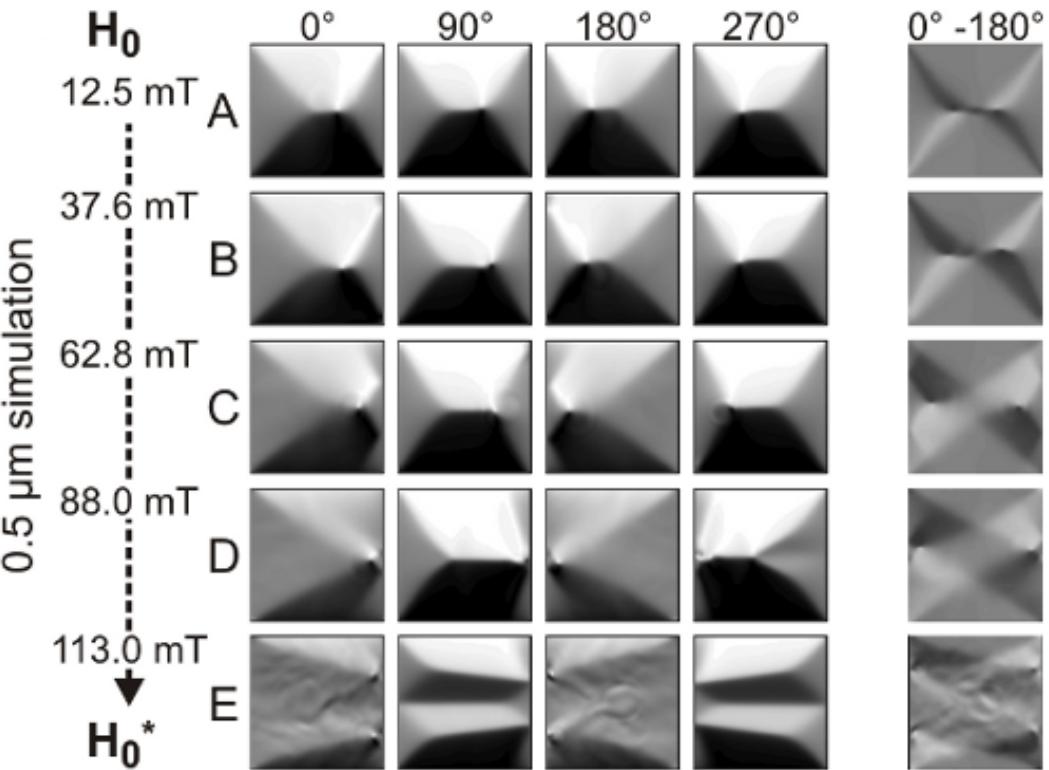
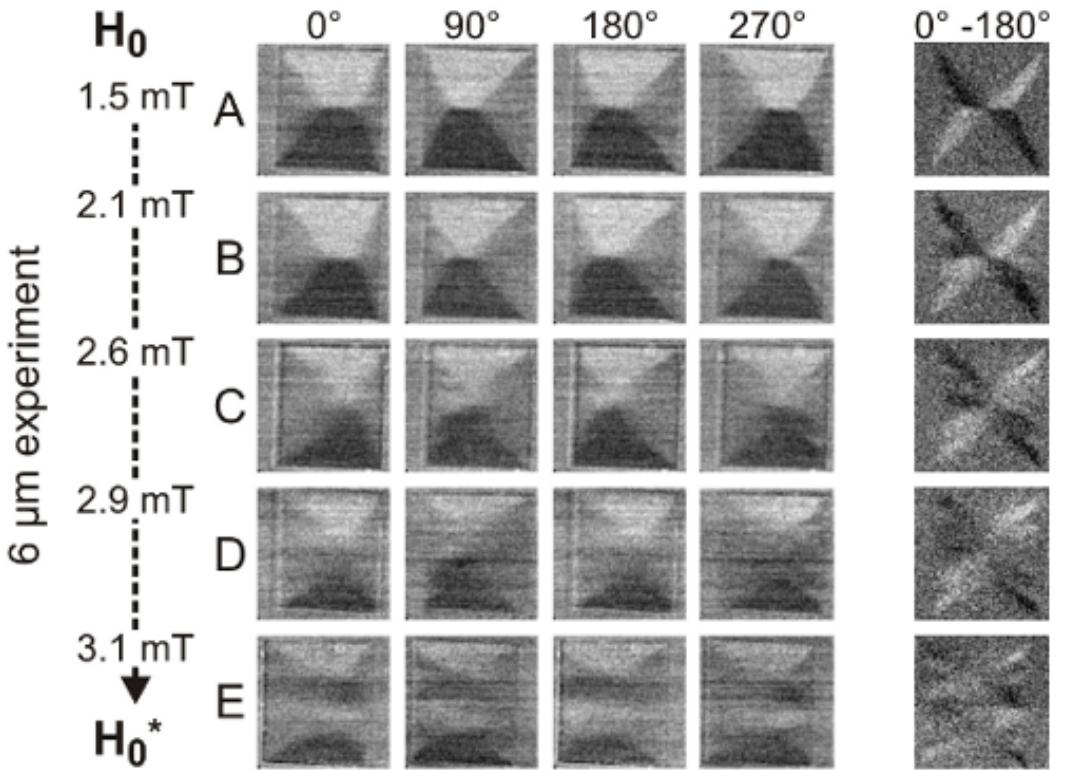
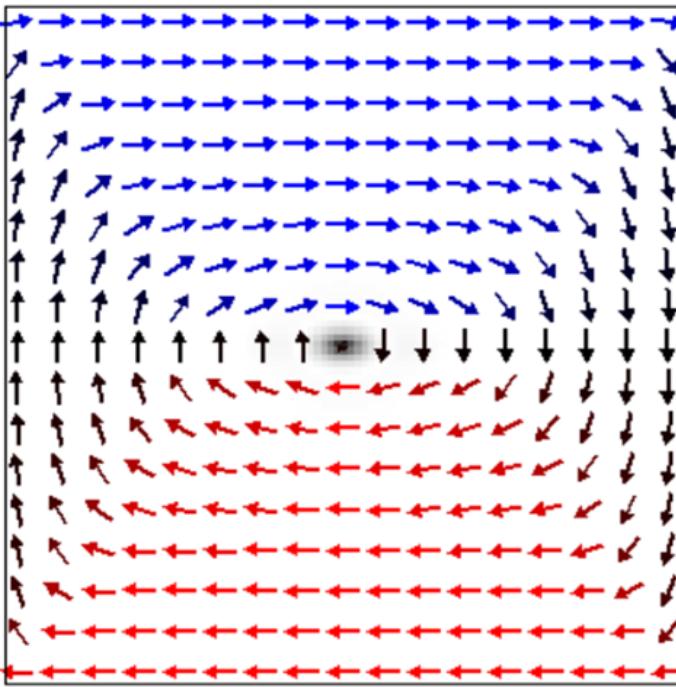


# Dynamic Differential Imaging

S. Stevenson et al.  
PRB 2013



6  $\mu\text{m}$



# Summary

- Element selective (multilayer, coupled systems)
- Sample can be in air or HV
- Spatial resolution currently down to below 15 nm
- Sensitivity to out-of-plane magnetisation, but sample can be tilted
- Image in applied fields
- Time resolved measurements

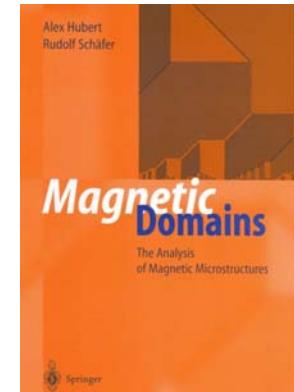
## Some Challenges (limitations):

- X-ray damage
- Sample must allow transmission of x-rays

# Conclusion

# Information on Different Methods

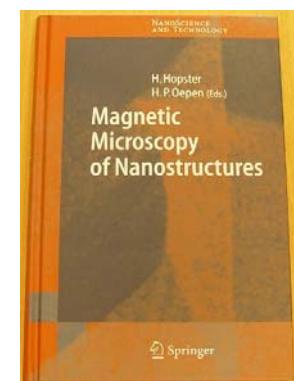
**A. Hubert and R. Schäfer, Magnetic Domains  
The Analysis of Magnetic Microstructures**



**Magnetic Microscopy of Nanostructures**

**An overview of techniques to image the magnetic structure on the nano-scale**

**H. Hopster and H. P. Oepen**



**Internet, for example:**

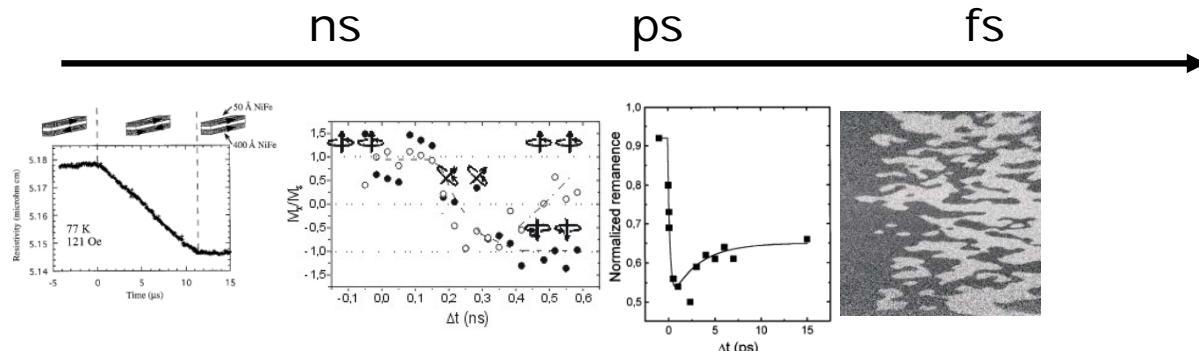
**Techniques to Measure Magnetic Domain Structures, R.J. Celotta, J. Unguris, M.H. Kelley, and D.T. Pierce, Methods in Materials Research (2000)**

# Comparison Between Different Techniques

- Contrast Origin: B, M,  $H_{ext}$
- In-Plane or Out-of-Plane components
- Quantitative or Qualitative
- Best Resolution, but better Typical Resolution
- Information depth
- Sensitivity, Acquisition Time
- Vacuum Equipment: none, HV, UHV
- Sample requirements: thickness, surface roughness, clean surface, insulators ?
- In-situ experiments: maximum field, heating, stress
- Additional information: crystallography, topography, chemical, electronic
- Commercial Availability, Cost & Complexity - Manpower

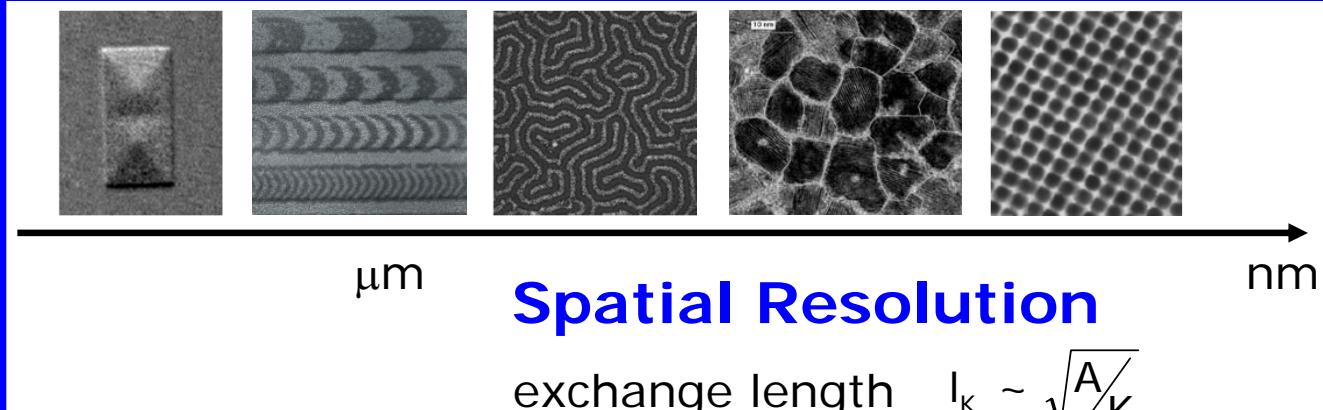
## Time Resolution

- Currently sub100ps ( $\approx 10$ ps): precession relaxation dynamics (LLG).
- Limited flux of photons: repeatable phenomena (stroboscopic pump-probe).



$$\text{exchange interactions} \quad t(\text{fs}) \sim \frac{4}{E(\text{eV})}$$

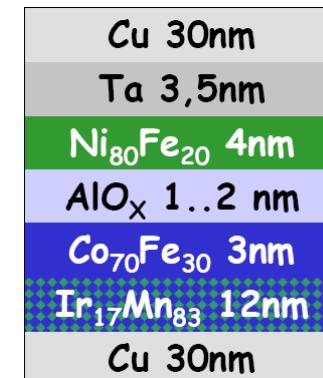
- Future challenge: fs time scale (times associated with exchange interactions, spin fluctuation rates)
- nm spatial resolution in single shot experiment.
- Need high flux ( $10^{12}$ ph/s ) X-ray source
- Lensless imaging and Full-field X-ray microscopy



## Spatial Resolution

$$\text{exchange length} \quad l_K \sim \sqrt{A/K}$$

## Elemental Sensitivity



magnetic tunnel junction

Peter Fischer, ALS