Electron microscopies for magnetism

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Introduction on electron microscopes

SPLEEM SEMPA

Lorentz microscopy

Electron holography

Scanning Electron Microscope • Principle

- Modes
- Resolution and sample

Low Energy Electron Microscope

Transmission Electron Microscope

SEM

Principle





Low energy (few tenths of eV)

- Surface topographical image
- Can be manipulated by Faraday cage
- Production of SE is primarily independent of atomic number (however, since backscattered electrons do produce SE, heavier elements tend to create more SE)
- Not much information on the atomic composition of the sample



Secondary electron generated and able to escape (depends on the escape depth)

Secondary electron generated and unable to escap

SEM

Backscattered electrons

Depth image

- Large width of escape depth
- High energy (>50eV)
- Heavy elements produce more backscattered electrons
- · Provides information on the atomic composition of the sample

Electron beam

Few BS electron are generated

BS electrons generated and able to escape

BS electrons generated and unable to escape

EBSD : Electron Back Scattered Diffraction



Example

Secondary e-



Backscattered e-



Fungal hyphae with Ag preferentially deposited at polysaccharides

sample topography sample composition www.courses.vcu.edu/PHYS661/pdf/06TechMicroscopy041.ppt

Resolution and sample

SEM

Resolution

- Primarily determined by spot size
- Small spot size gives high resolution
- Spot size increases with working distance
- ~ few nanometers

Sample preparation

- Tedious for biological specimen (drying, ...)
- Carbon coating to make materials conductive



Scanning Electron Microscope

Low Energy Electron Microscope

- •Principle
- Modes
- Resolution and sample

Transmission Electron Microscope

LEEM

Principle





- Electron Energy is OeV
 - Electrons Return Before they Hit the Sample
- Contrast created by outer Potential
 - Workfunction
- Image appears Blurred





From F. Meyer zu Heringdorf's LEEM-Basics , http://www.leem-user.com/



Bright field



Bright field image: incident beam normal to surface reflected (0,0) beam normal to surface select (0,0) with contrast aperture





From <u>R. Tromp's LEEM-Basics</u>, http://www.leem-user.com/



Dark Field Imaging



Dark field image:

incident beam inclined to surface reflected (h,k) beam normal to surface select (h,k) with contrast aperture



From <u>R. Tromp's LEEM-Basics</u>, http://www.leem-user.com/



Resolution

Limited by the diffraction at the contrast aperture, the spherical and chromatic aberrations of the objective lens Lateral 5-10nm Depth subÅ

Samples

Mainly semi-conductive samples No special preparation required

Scanning Electron Microscope Low Energy Electron Microscope **Transmission Electron Microscope** • Principle Modes Resolution and sample



Different modes

Bright field mode

Dark field mode

High resolution mode



Link between magnetism and structure :

- Crystallographic structure of the elements (bcc, hcp, fcc)
 metastable phase
- Interface quality (roughness)
- Grain size and morphology of polycristalline materials
- Surface topography
- Strain in layers (magnetoelastic energy)
- •Chemical distribution of the elements

Example : NiO/Co

Epitaxial Co/NiO/MgO M along the NiO[110] axis, structure?





Structure of the cobalt layer?

hcp (easy axis along [0001] c axis) fcc (easy axis along the [111] axis)

Example : NiO/Co



hexagonal structure with 2 variants: Co[0002](1120) // NiO[200](001) or Co[0002](1120) // NiO[020](001)



c axis : easy axis c // [100]NiO or c//[010]NiO M // <110>

B. Warot et al., J. Appl. Phys. 2001 89 5414-5420



Example : Fe_3O_4



Ferrimagnetic Cubic inverse spinel structure

Structural defect : antiphase boundary ¼<110> shift of the stack

B sites: Fe³⁺, Fe²⁺
 A sites: Fe³⁺
 O²⁻

Example : Fe_3O_4

Fe₃O₄ /NiO/MgO



epitaxial growth interface quality observation of defects (APBs in Fe_3O_4)

NiO/Fe₃O₄ /Al₂O₃



C. Gatel et al., J. Mag. Mag. Mat. 2004 272–276 e823–e824

Electron energy loss spectroscopy – EELS

Energy loss due to inelastic interactions between the incoming electron beam and the target electron Loss characteristic of one element

Local probe (beam diameter size)

Quantitative chemical analysis







Chemical analysis

Composition profile in a Scanning TEM : a spectrum is recorded at each spot position



Composition maps using Electron Spectroscopic Imaging or Energy Filtered TEM: an image is recorded for each energy slit



Chemical analysis





Composition profile

C.Gatel, PhD thesis, Toulouse, 2004



Resolution and sample

Resolution

- Determined by the wavelenght of the incident electron
- Limited by lens aberration (spherical, chromatic)
 Few Å

Sample preparation : electron transparent sample (from 10nm

- 2 µm

to 300nm thick) Mechanical polishing Chemical polishing FIB system



— 3 µm

- 2 µm

specimen

. M. Langford et al., J. Vac. Sci. Technol. A 2001 19(5) 2186-2193

section specimen

Introduction on electron microscopes

SEMPA : Scanning Electron Microscopy with Polarization Analysis

SPLEEM Lorentz microscopy Electron holography

Theoretical principle

Experimental set-up

Examples

SEMPA

SEMPA measures the spin polarization of the secondary electrons that exit from a magnetic sample as the finely focused (unpolarized) beam of the scanning electron microscope rasters over the sample.



Interaction : Incident electron beam-specimen

Result from an incident electron passing "near" an atom in the specimen, near enough to impart some of its energy to a lower energy electron

Emission: Secondary electrons

Secondary electrons are predominantly produced by the interactions between energetic beam electrons and weakly bonded conduction-band electrons in metals

If the sample is ferromagnetic : the emitted electrons are spin-polarized (difference of occupancy of the up and down bands)



SEMPA

Emission

SEMPA measures the spin polarization of the secondary electrons that exit from a magnetic sample as the finely focused (unpolarized) beam of the scanning electron microscope rasters over the sample.



Physical principle : use of the spin-orbit interaction as a means of transforming a spin asymmetry into a spatial asymmetry.

Example

Mott detector : the electrons are accelerated to high energies (typically 50 to 100 keV) and scattered by a high-atomic-number target.

This scattering is spin-dependent because of the spin-orbit interaction.

Therefore, electrons with spin up and spin down with respect to the scattering plane are preferentially scattered into different directions.



SEMPA

SEMPA measures the spin polarization of the secondary electrons that exit from a magnetic sample as the finely focused (unpolarized) beam of the scanning electron microscope rasters over the sample.



The spin of the secondary electrons points preferentially in the opposite direction of the magnetisation vector : $M = -\mu_B(n_+ - n_-)$ as the electron spin magnetic moment and the electron spin are opposite





Measure of the topography $(n_+ + n_-)$ and the magnetisation at the same time $(n_+ - n_-)$

Surface technique : secondary electrons are emitted from 1 nm

Measure without applying a field (deviation of the electron beam)



Theoretical principle

Experimental set-up

Examples
Non polarized incident beam

Work at low incident energies:

High I

High P : the spin of the 2dary electron is dependent on the 2dary electron energy. The P at low energy is 2 or 3 times higher than expected due to preferential inelastic scattering of spin down electrons which leads to a higher escape probability for spin up electrons

BUT

Large beam diameter

Beam more susceptible to deflections and distortions

Typical work tension 10kV

Experimental set-up

Polarisation measurement



The anode is divided into quadrants so that the polarization components along both x and y may be measured simultaneously

$$P_x = \frac{1}{S} \frac{N_A - N_C}{N_A + N_C}$$

$$P_{y} = \frac{1}{S} \frac{N_{B} - N_{D}}{N_{B} + N_{D}}$$

- Schematic drawing of a low energy diffuse scattering spin polarization analyzer.
- Based on the scattering of 150eV electrons from an evaporated Au target.

Experimental set-up

Out of plane magnetisation

Spin rotator



Spatial **resolution** : determined largely by the electron beam diameter of the SEM.

Beam current \searrow as beam diameter \searrow

Compromise between resolution (beam diameter) and acquisition time

Limited by :

- sample drift
- deterioration of the sample surface
- operator patience

SEMPA

beam current of 1 nA acquisition in about 1 h resolution limits 50 nm for LaB6 and 10 nm for field emission SEM electron gun cathodes.

Samples : conductive (carbon coating)

Theoretical principle

Experimental set-up

Examples

Polycristalline iron

First experiments : Koike et al. – 1987

K.Koike, H.Matsuyama, H.Todokoro, K.Hayakzwa, Scanning microscopy 1987 **1** 31



Domain images n₊-n₋

Absorption current images n_++n_-

Fe/Si(100)

canning electron microscopy with polarization analysis (SEMPA)

M. R. Scheinfein, J. Unguris, M. H. Kelley, D. T. Pierce, and R. J. Celotta





Rev. Sci. Instrum. 1990 61 2501



Probe diameter ~ 50nm

50µm

Co/Cu

Observation of Antiparallel Magnetic Order in Weakly Coupled Co/Cu Multilayers

J. A. Borchers, J. A. Dura, J. Unguris, D. Tulchinsky, M. H. Kelley, and C. F. Majkrzak National Institute of Standards and Technology, Gaithersburg, Maryland 20899

S. Y. Hsu, R. Loloee, W. P. Pratt, Jr., and J. Bass Phys. R

Phys. Rev. Lett. 1999 82 2796

Department of Physics and Astronomy, Michigan State University, East Lansing, Michigan 48824



[Co(6nm)/Cu(6nm)]₂₀ multilayer

FIG. 3(color). SEMPA images of the topmost Co layer magnetization (a) and topography (b) and second Co layer magnetization (c) in the $[Co(6 \text{ nm})|Cu(6 \text{ nm})]_{20}$ sample. The magnetization direction is mapped into color as indicated by the color wheel in the center. A histogram of the difference in the magnetization direction between the two layers, $\Delta\phi$, is shown in (d).

Antiferromagnetic coupling between two adjacent Co layers due to the Cu spacer

In situ ion sputtering using 2 keV Ar⁺ ions was used to clean and depth profile the sample.

Co/Au(111)

Perpendicular magnetisation of a Co/Au(111) sample M.Speckmann, H.P.Oepen, H.Ibach, Phys. Rev. Lett. 1995 **75** 2035



Out-of-plane mag : white In-plane mag : black 3 ML

Wedge shaped Co layer



After annealing 240°C 10 min

Domain enlargement Increase of t_c

20µm

SEMPA

X

Zigzag elements

Zigzag-shaped magnetic sensors

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A. B. Kos, S. Schima, and J. Aumentado National Institute of Standards and Technology, Boulder, Colorado 80305

Appl. Phys. Lett. 2004 85 6022

Aim : design of an element in which a bias is maintained between the current and the magnetisation

Zigzag element Ta(5nm)/**NiFe**(30nm)/Ta(5nm) J. Unguris National Institute of Standards and Technology, Gaithersburg, Maryland 20899

D. P. Pappas^{a)} National Institute of Standards and Technology, Boulder, Colorado 80305



FIG. 2. (Color) (a) Experimental SEMPA image of the eight-block zigzag structure. (b) Simulation performed on the same geometry using OOMMF. The magnetization direction maps onto the angle color map ring.



Ferromagnetic rings

Two magnetic states:

• the flux-closure vortex state

 the 'onion' state, accessible reversibly from saturation and characterized by the presence of two opposite head-to-head walls.



SEMPA





"Onion" state

"Vortex" state

Fig. 1. (a) Hysteresis loop measured on an array of rings (outer diameter D = 1200 nm, inner diameter d = 900 nm and thickness t = 15 nm polycrystalline Co). The magnetization configurations of the onion and the vortex states are shown schematically. Red arrows indicate the field path used to obtain the rings in the states imaged with PEEM in (b). (b) PEEM image of four polycrystalline Co rings. The top ring is in the clockwise vortex state, the bottom and right rings are in the counter-clockwise vortex state and the left ring is in the onion state pointing along the direction of the applied field H. The field of view is $\approx 10 \ \mu m$ across and the blue arrow points in the direction of the photon beam. The colour scale indicates the direction of the magnetization with reference to the direction of the incoming photon beam (magnetization parallel to the incoming photon beam (red) to anti-parallel (purple)). For colour, please see on-line version.

Ferromagnetic rings

Spin configurations and classification of switching processes in ferromagnetic rings down to sub-100 nm dimensions M. Klaui, C.A.F. Vaz, T.L. Monchesky, J. Unguris, E. Bauer, S. Cherifi, S. Heun, A. Locatelli, L.J. Heyderman, Z. Cui, J.A.C. Bland, J. Mag. Magn. Mat. 2004 **272-276** 1631



500nm

Onion state in a wide (inner diameter=900nm) and in a narrow (inner diameter=1200nm) epitaxial 34-nm fcc Co ring, outer diameters 1.7 μ m.

SEMPA images of the wide (a) and the narrow (c) ring. Corresponding micromagnetic simulations (OOMF) of the wide (b) and the narrow (d) ring, showing the vortex- and transverse-type domain walls.

Groups

- NIST Gaithersburg (USA) http://physics.nist.gov/Divisions/Div841/Gp3/Facilities/sempa.html
- ETH Zürich (Switzerland) http://www.solid.phys.ethz.ch/pescia/sempa.htm
- University of Hamburg (Germany) http://www.physnet.uni-hamburg.de/iap/group_g/index.htm
- University of Seoul (South Korea) http://csns.snu.ac.kr/lab/systems/sempa/
- University of California, Irvine (USA) http://www.physics.uci.edu/NEW/cmexpt.shtml

Introduction on electron microscopes

SEMPA

SPLEEM : Spin Polarized Low Energy Electron Microscopy

> Lorentz microscopy Electron holography

Theoretical principle

Experimental set-up

Examples

Exchange interaction between the incident electrons with spin \mathbf{s}_i and the target electrons with spin \mathbf{s}_i

$$Vex = \sum_{ij} J(\mathbf{r}_i - \mathbf{r}_j) \mathbf{s}_i \bullet \mathbf{s}_j$$

J being the exchange coupling strength



E. Bauer, Rep. Prog. Phys. 1994 57 895

In crystalline materials, the contrast can also be understood in terms of spin-dependent band structure



 E_{beam} <1eV

reflection because no energy state is available

 $1eV < E_{beam} < 2eV$ low reflectivity for spin up and large for spin down $E_{beam} > 2eV$

difference in reflectivity due to the difference in the densities of states (up and down)

Combination of two effects explaining the contrast:

- The exchange interaction
- The different inelastic mean free paths of electrons with spin parallel and antiparallel to the spin of the electrons in the ferromagnet

Best magnetic contrast for low electron energy, typically 10eV

Most of the contrast is determined by the microstructure

Theoretical principle

Experimental set-up

Examples

Requirements

• Spin Polarized incident beam with high intensity

Possibility to rotate the polarisation of the incident beam to optimize the orientation between
P and M : acquisition with antiparallel P directions

• No magnetic lenses in the system Electrostatic condenser and objective lenses

 Rapid and flexible image accumulation and processing so that the difference (up-down) can be obtained rapidly

Suitable acquisition system

T.Duden et al., Jour. Elec. Micr. 1998 **47** 379

GaAs band-gap photoexcitation



http://nvl.nist.gov/pub/nistpubs/sp958-lide/203-208.pdf

SPLEEM

Reduction of the vacuum level of the GaAs (4eV) by a surface treatment using CsO



Spatial **resolution** : 20 nm

Samples : ferromagnetic or ferrimagnetic conductive



Theoretical principle

Experimental set-up

Examples

atomically flat ultrathin (110)-oriented Fe films on a W(110) surface



blue and red colours ⇔ regions with opposite magnetization intensity ⇔ magnitude of the asymmetry.

R. Zdy et al., Applied Surface Science 249 (2005) 38-44

University of Arizona, USA http://phy.asu.edu/homepages/bauer/spleem.htm

University of California, Berkeley, USA http://ncem.lbl.gov/frames/spleem.htm

http://www.leem-user.com/

Introduction on electron microscopes

SPLEEM

SEMPA

Lorentz microscopy

Electron holography

Theoretical principle

Experimental set-up

Examples

Electron moving through a region of space with an electrostatic field and a magnetic field B experiences the Lorentz force F_L : $F_L = -e(E+v_{\wedge}B)$

If E=0, F_{L} acts normal to the travel direction of the electron, a deflection will occur. Only the in-plane magnetic B_{L} induction will contribute to the deflection



Theoretical principle

Interaction



Introduction to conventional electron microscopy M. De Graef, 2003, Cambridge University Press

Schematic of a magnetic thin foil and the resulting deflection of an incident electron beam

Find
$$\mathbf{t}$$
 with $\mathbf{F}_{\mathrm{L}} = \mathbf{ev}_{\mathbf{z}}\mathbf{B}_{\perp}$ $p_{y} = e \int_{0}^{t} B_{\perp} dz = e B_{\perp} t$

 $C_L(E)$ depends on the acceleration voltage of the microscope

$$\mathbf{q}_{L} = \frac{p_{y}}{p_{z}} = \frac{eB_{\perp}t}{mv} = C_{L}(E)B_{\perp}t$$

 $\boldsymbol{\theta}_{\!\! L} \, depends$ on $\boldsymbol{B}_{\!\! L}$ and t

 θ_L ~ tens of µrad

 θ_{B} (electron diffraction angles) ~ tens of mrad

Theoretical principle

Experimental set-up

Examples





Fresnel mode



Experimental set-up

Foucault mode



Experimental set-up

Foucault mode



0 Oe



19 Oe





38 Oe



45 Oe







51 Oe

0 Oe

-10 Oe

-40 Oe

Problem : objective lens magnetic field : 2T How to get rid of this field?

Turn of the objective lens -> no magnification Lorentz lens


Lorentz TEM useful to observe the domain evolution under an applied field

Applying a field :

- Use the objective lens low excitation, rotate the sample holder in the fixed vertical field : the inplane field depends on the tilt angle
- Build a dedicated sample holder with coils to apply a field



Tilt angle

B_{max} set to a low value by adjusting the current in the objective lens

•Use the objective lens low excitation, rotate the sample holder in the fixed vertical field : the in-plane field depends on the tilt angle

•Build a dedicated sample holder with coils to apply a field

Uhlig at al. Ultramicroscopy. 2003 **94** (3-4) 193-6







C K Lim et al J. Phys. D: Appl. Phys. 2003 36 3099-3102

Figure 1. Schematic of the new magnetizing stage.

Samples

- Plane view
- •On transparent windows (example : Si3N4) polycrystalline samples
- Sample deposition dedicated to Lorentz experiments



Resolution 20nm

Experimental set-up

Examples

Fresnel mode

Determination of the magnetization axis



Magnetisation ripple visible in Fresnel images of polycrystalline specimens as a result of small fluctuations in the magnetisation direction.

Series of Fresnel images showing the magnetization loop of NiFe/FeMn exchange coupled layers.

Applied field direction and values together with easy-axis direction are indicated.

X.Portier et al. J. Appl. Phys., 2000 87 6412

Interaction

Kirk et al., Appl. Phys. Lett. 1999 75 3683

Co elements



FIG. 2. Array of Co elements $300 \times 80 \text{ nm}^2$, with elements in opposite magnetization states as indicated by light and dark shading. (a) Foucault image with magnetic induction mapped parallel to the double-headed arrow, and (b) schematic diagram showing magnetization directions.



NiFe elements

DPC analysis Liu et al., J. Appl. Phys 2004 **96** 5173

Tunnel junctions



Two reversal modes are observed :

- domain nucleation and propagation (for 2μm wide elements)
- single domain reversal (for $0.7\mu m$ and $1\mu m$ wide elements)

Tunnel junctions



Oe



aspect ratio

Asymmetry of the reversal

AP–P reversal : the reversal field increases as the aspect ratio increases

P–AP reversal : no strong variation in reversal field as a function of aspect ratio

B.Warot et al., J. Appl. Phys., 2003 93 7287-7289

Calculations



Reversal simulation of NiFeCo single layers (LLG software)

Comparaison with experimental data : discrepancy explained by microstructural defects (grain boundaries, composition inhomogeneities...)

J.Imrie, Part II, Oxford (2003)

Active devices

A current passes through a spin valve element during simultaneous application of a magnetic field



Fig. 1. (a) Scheme of the device developed for in situ LTEM experiments. (b) TEM image of a $10 \ \mu m \times 10 \ \mu m$ SV element with the Au contacts at each end of the element, allowing the magnetoresistance to be measured for various current values in both directions.



Fig. 3. (a) Plot of the GMR curves for a current value of 6 mA corresponding to the (a i) and (a ii) configurations of Fig. 2. Numbers refer to the Foucault images shown in 3b. (b) Sequence of Foucault images showing the P to AP transitions for both current directions H_a , H_b and H_{FM} are the applied field, the field induced by the current and the field induced by the magnetostatic coupling between the FM layers, respectively.

Portier et al., Appl. Phys. Lett., 1997 **71** 2042 Portier et al., J. Mag. Mag. Mat., 1998 **187** 145



To get quantitative information out of Lorentz images

- the differential phase contrast method (DPC)
- the non interferometric phase retrieval method



The DPC method

Measure of the deflection angle in a STEM by noting the differences between the current falling on opposite segments of a quadrant deflector





Magnetic signal : A-C, D-B Structural signal : E-G, F-H



specimen in DPC microscopy.



JOURNAL OF APPLIED PHYSICS

VOLUME 95, NUMBER 1

1 JANUARY 2004

DPC

Observation of magnetic structures in Fe granular films by differential phase contrast scanning transmission electron microscopy

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FIG. 4. Vector map of magnetization of sample I. The STEM bright-fi image is superimposed on the vector map.



a)Differentation along the x axisb)Differentation along the y axisc)STEM bright field image

Summed Image DPC

DPC in a TEM : Summed Image Differential Phase Contrast

Daykin et al., Ultramicroscopy 1995 **58** 365 A.K.Petford-Long et al., IEEE Trans. On Mag. 1999 **35** 788



Schematic illustrating the SIDPC technique for obtaining quantitative magnetisation maps using Lorentz microscopy $B_{sum}-A_{sum} = y$ component $C_{sum}-D_{sum} = x$ component



By component



FIG. 1. SIDPC images showing the x and y components of the plane magnetization in the 10 μ m × 30 μ m SV element in an plied field of -10.26 Oe. The applied field as well as the x axis e parallel to the long axis of the element. The y axis is vertical.

Portier et al., Phys. Rev. B 1998 58 R591

Ta/NiFe/Cu/Co/NiFe/MnNi/Ta (5/8/3/2/6/25/5 nm)

Non interferometric phase retrieval

Aharonov–Bohm phase shift : Relation between the phase of the electron wave to a trajectory integral of the magnetic and electrostatic potentials along the electron path

Phase?

Mathematic formalism to extract the phase out of Lorentz images the transport-of-intensity equation (TIE)

$$\nabla \cdot \left[I(r_{\perp}, 0) \nabla \phi(r_{\perp}) \right] = -\frac{2\pi}{\lambda} \left. \frac{\partial I(r_{\perp}, z)}{\partial z} \right|_{z=0}$$

Intensity of the in-focus image



Calculated from a derivative of the image intensity along the optic axi z (from out-of-focus images, in Fresnel mode)



Magnetic?

If the electrostatic contribution is neglected:

$$\nabla \boldsymbol{f}(\boldsymbol{r}_{\perp}) = -\frac{e}{\hbar} (B_{\perp} \times \boldsymbol{n}_{z})t$$

$$\phi_e = \frac{2\pi}{\lambda} \frac{E_0 + E}{E(2E_0 + E)} V_{ip} t \equiv \sigma(E) V_{ip} t.$$

Only the electrostatic component is energy dependent

Images recorded at various electron energy

Only some magnetic materials have suitable properties (domain wall thickness, value of B) to be studied with this method

Kohn et al., Phys. Rev. B 2005 72 0144444

Non interferometric phase retrieval

Induction maps in Co islands (7µm wide)

tB_v iΒ

H = 0

H=280e



tB(r) Phase contours (electrostatic contribution neglected)

Volkov et al., Ultramicroscopy 2004 98 271–281

Groups

University of Glasgow, UK http://www.ssp.gla.ac.uk/ResSum/SSPSummary.htm

University of Oxford, UK http://www-magnetics.materials.ox.ac.uk/

University of Cambridge, UK http://www-hrem.msm.cam.ac.uk/research/index.shtml

Arizona state university, USA http://www.asu.edu/clas/csss/chrem/main.html

National Center for Electron Microscopy, Berkeley, USA http://ncem.lbl.gov/

Carnegie Mellon University, Pittsburgh, USA http://neon.mems.cmu.edu/

Brookhaven national laboratory, New York, USA http://www.bnl.gov/tem/

Introduction on electron microscopes

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Experimental set-up

Examples

Conventional TEM





Phase contributions







Phase shift due to the electrostatic potential

$$\boldsymbol{j}_{elect}(\mathbf{x}) = \mathbf{C}_{E} \int V(x, z) dz$$

Phase shift due to the magnetic induction

$$\boldsymbol{j}_{mag}(\mathbf{x}) = -\frac{e}{\hbar} \iint B_n(\mathbf{x}, \mathbf{z}) d\mathbf{x} d\mathbf{z}$$

Total phase shift

$$\boldsymbol{j}_{Tot}(\mathbf{x}) = \boldsymbol{j}_{elect}(\mathbf{x}) + \boldsymbol{j}_{mag}(\mathbf{x})$$

Phase gradient



$$\Delta \boldsymbol{j} = \boldsymbol{j}(x_2, y) - \boldsymbol{j}(x_1, y) = -\frac{e}{\hbar} \int_{\boldsymbol{x}=x_2}^{x_1} \int B_n(\boldsymbol{x}, y, z) d\boldsymbol{x} dz$$

$$\frac{\partial \boldsymbol{j}(x)}{\partial x} = \frac{e}{\hbar} B_n(x) t$$

The phase gradient is proportional to the inplane induction component B_n and the equiphase lines give the direction of B

Magnetic samples







Magnetic samples

If B only varies in the (x,y) plane $:B_x, B_y$



Experimental set-up

Examples

How to get rid of the magnetic field due to the objective lens?

How to measure the phase shift?

How to separate the electrostatic and magnetic contributions?

Electron Holography

Problem : objective lens magnetic field : 2T How to get rid of this field?

Turn of the objective lens -> no magnification Lorentz lens





Biprism

1

Sactem FEI – FEG Cs corrected + biprism + Tridiem (Gatan)







Elec. and magn. contributions



Phase shift due to the electrostatic potential

$$\boldsymbol{j}_{elect}(\mathbf{x}) = \mathbf{C}_{E} \int V(x, z) dz$$

Phase shift due to the magnetic induction

$$\boldsymbol{j}_{mag}(\mathbf{x}) = -\frac{e}{\hbar} \iint B_n(\mathbf{x}, \mathbf{z}) d\mathbf{x} d\mathbf{z}$$

Total phase shift

$$\boldsymbol{j}_{Tot}(\mathbf{x}) = \boldsymbol{j}_{elect}(\mathbf{x}) + \boldsymbol{j}_{mag}(\mathbf{x})$$

Elec. and magn. contributions

$$\varphi(x) = C_E \int V(x,z) dz - \frac{e}{\hbar} \iint B_n(x,z) dx dz$$

1/ Taking two holograms (4 with the reference holo) with the TEM sample switched upside down

=> the sign of the « B » contribution changes not the electrostatic one.



Problem : finding back the same area after returning the sample

Elec. and magn. contributions

$$\varphi(x) = C_E \int V(x,z) dz - \frac{e}{\hbar} \iint B_n(x,z) dx dz$$

I Taking two holograms changing the TEM accelerating high voltag

$$C_{E} = \left(\frac{2p}{?}\right) \left(\frac{E + E_{0}}{E(E + 2E_{0})}\right)$$

$$\Delta \phi_1 = C_{E_1} V_i \cdot t - \frac{e}{\hbar} \iint B_n ds$$

$$\Delta \phi_2 = C_{E_2} V_i \cdot t - \frac{e}{\hbar} \iint B_n ds$$

$$V_{i}t = \frac{\Delta \varphi_{2} - \Delta \varphi_{1}}{(C_{E_{2}} - C_{E_{1}})}$$

Problem : keeping the same acquisition conditions when changing the high voltage

Elec. and magn. contributions

-Θ

$$\varphi(x) = C_E \int V(x,z) dz - \frac{e}{\hbar} \iint B_n(x,z) dx dz$$

3/ Switching the magnetization of the sample with the objective lens field Hobjective



 $+\Theta$

$$\phi_{1} = C_{E}V_{i.}t - \frac{e}{\hbar} \cdot t \cdot \int B_{n}(x) \cdot dx$$
$$\frac{e}{\hbar} \cdot t \cdot \int B_{n}(x) \cdot dx = \frac{\phi_{2} - \phi_{1}}{2}$$

Magnetic contribution

Electron Holography

$$\phi_2 = C_E V_{i.}t + \frac{e}{\hbar} \cdot t \cdot \int B_n(x) \cdot dx$$
$$C_E V_i t = \frac{\phi_2 + \phi_1}{2}$$

Mean inner potentiel (MIP)

Measurement of the phase shift

$$\psi_{O} = \exp(i\mathbf{K}.\mathbf{r})$$
 $\psi_{S} = A_{S}(\mathbf{r})\exp(i\mathbf{K}.\mathbf{r} + \varphi_{S}(\mathbf{r}))$

$$I_{\text{Holo}} = \mathcal{W}_{o} \mathbf{Y}_{s}^{*} | + \text{background} = 1 + A_{s}^{2}(\mathbf{x}, \mathbf{y}) + 2A_{s}(\mathbf{x}, \mathbf{y}) \cos[2\pi R_{0} \cdot \mathbf{x} + \varphi_{s}(\mathbf{x}, \mathbf{y})] + I_{\text{inelast}}(\mathbf{x}, \mathbf{y})$$



Interference fringes of period : $\Lambda = 2\pi/R_o$

- R₀ depends on
 - the incident beam wavelength
 - the biprism polarisation
 - the beam convergence

How to extract $\varphi(r)$?
Experimental set-up

Measurement of the phase shift





Phase image $\phi_{S}(x, y)$

Amplitude image $A_{s}(x, y)$

Experimental set-up

Measurement of the phase shift



Samples

Transparent to the electron beam Mainly particles deposited on carbon grids Deposition on transparent membranes



Resolution 5nm

What are good holograms?

Large energy coherence of the electron beam



Elliptic illumination (large condenser astigmatism)



Experimental set-up

- Reducing dynamical effects by being far from a zone axis
- Optimising fringe contrast

$$\frac{I_{max}-I_{min}}{I_{max}+I_{min}} > 15\%$$

Tricks

• Setting the biprism polarisation to get optimum fringe period to detect the phase shift



V = 0 Volts V = 60 Volts

Theoretical principle

Experimental set-up

Examples

Baryum ferrite

Observation of the magnetisation and the ferromagnetic domains in baryum ferrite particles

T. Hirayama, A. Tonomura et al. Appl. Phys. Lett. 1993 63 418



TEM micrograph of a particle

Phase image of the stray field



Image showing magnetic domains

Co nanowires



Magnetic contribution magnified 128 times





Magnetic contribution of 280 nanowires uniformly magnetized with a saturation magnetisation of B = 1.7 T

E. Snoeck, R. E. Dunin-Borkowski, et al. Appl. Phys. Lett. **82**, p 88 (2003).

Isolated Co nanowire





Cosine of 256 times the magnetic contribution

Isolated Co nanowire



Co single nanowire



100% ±0.19 of the Co nanowire is fully magnetized

Co nanoparticle ring





Image of self-assembled Co nanoparticle rings and chains deposited onto an amorphous carbon film.

Co diameter ~ 20-30 nm

Magnetic phase contours (0.049 radian spacing), formed from the magnetic contribution to the measured phase shift, in four different nanoparticle rings.

R E Dunin-Borkowski et al., Microsc. Res. Techn. 2004 **64** 390-402

Titanomagnetite

lanar array of Titanomagnetite natural system (magnetite rich locks separated by non-magnetic materials)



R E Dunin-Borkowski et al., Microsc. Res. Techn. 2004 64 390-402

C nanotubes filled with Fe



Groups

CEMES, Toulouse, France http://www.cemes.fr/

University of Cambridge, UK http://www-hrem.msm.cam.ac.uk/research/index.shtml

Institute of Structure Physics (ISP), Dresden University, Germany http://www.physik.tu-dresden.de/isp/member/wl/TBG/start/lichteS.htm

University of Arizona, USA http://www.asu.edu/clas/csss/chrem/main.html

EPFL, Lausanne, Switzerland http://cime.epfl.ch/

Hitachi, Japan http://www.hqrd.hitachi.co.jp/global/fellow_tonomura.cfm